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Additional Information

1 **Isolation and characterisation of microcrystalline cellulose and cellulose**
2 **nanocrystals from coffee husk and comparative study with rice husk**

3
4 Sofía Collazo-Bigliardi^{a*}, Rodrigo Ortega-Toro^b and Amparo Chiralt Boix^a

5
6 ^aInstituto de Ingeniería de Alimentos para el Desarrollo, Universidad Politécnica de
7 Valencia, Camino de Vera s/n, 46022 Valencia, Spain. socol@alumni.upv.es;
8 dchiralt@tal.upv.es.

9 ^bPrograma de Ingeniería de Alimentos, Grupo IFCRA, Facultad de Ingeniería,
10 Universidad de Cartagena, Carrera 6 # 36-100, Cartagena de Indias D.T y C, Colombia.
11 rortegap1@unicartagena.edu.co

12
13 *Corresponding author: Sofía Collazo-Bigliardi. Instituto de Ingeniería de Alimentos
14 para el Desarrollo, Universidad Politécnica de Valencia, Camino de Vera s/n, 46022
15 Valencia, Spain. Phone: 34-963877924. Fax: 34-963877926. socol@alumni.upv.es.

16
17 **Abstract**

18 Cellulosic material from coffee husk has not been previously studied despite being a
19 potential source of reinforcing agents for different applications. This material has been
20 extracted and characterised from coffee husk, in parallel with previously studied rice
21 husk. Samples have been analysed as to their ability to obtain cellulosic fibres and
22 cellulose nanocrystals (CNC) by applying alkali and bleaching treatments and final
23 sulphuric acid hydrolysis. Microstructural changes were analysed after treatments, and
24 the size and aspect ratio of CNCs were determined. Crystallinity and thermal stability of
25 both materials progressed in line with the enrichment in cellulosic compounds. The

26 CNC aspect ratio was higher than 10, which confers good reinforcing properties. These
27 were tested in thermoplastic starch films, whose elastic modulus increased by 186 and
28 121 % when 1 wt% of CNCs from rice and coffee husks, respectively, was incorporated
29 into the matrix. Coffee husk represents an interesting source of cellulosic reinforcing
30 materials.

31 Keywords: Agro-wastes; Cellulose fibres; Cellulose nanocrystals; Biocomposite;
32 Tensile properties.

33

34 **1. Introduction**

35 A growing concern for environmental conservation has encouraged research into the
36 development of new types of green bio-based and biodegradable materials from natural
37 sources for different engineering applications (Balaji, Pakalapati, Khalid, Walvekar, &
38 Siddiqui, 2017; Bassas-Galia, Follonier, Pusnik, & Zinn, 2017; Jiang & Zhang, 2017;
39 Brigham, 2018; Ng et al., 2015), such as biodegradable packaging materials for the food
40 industry (Fabra, López-Rubio, & Lagarón, 2014; Talegaonkar, Sharma, Pandey, Mishra,
41 & Wimmer, 2017). Many studies have focused on the use of agro-wastes to obtain more
42 valuable materials in a sustainable and environmentally-friendly way, which can be
43 applied in different industrial applications, as an alternative to conventional petroleum-
44 derived plastics (Mondal, 2017; Reis et al., 2015). The use of residual biomass can give
45 added value to a waste product while also offering new renewable materials. This way,
46 lignocellulosic materials are one of the most important natural sources of renewable
47 polymers, which are attractive because of their biodegradability, low density and
48 excellent mechanical properties, such as great stiffness and strength (Hake, Mondal,
49 Khan, Usmani, Bhat, & Gazal, 2017; Patel & Parsania, 2018). Several authors have
50 studied the potential use of lignocellulosic materials as sources of reinforcing fillers for

51 thermoplastic biopolymers. In this sense, different agro wastes, such as sunflower stalks
52 (Fortunati et al., 2016), hemp fibres (Luzi et al., 2016), bamboo pulp (Borkotoky, Dhar,
53 & Katiyar, 2018), cotton (Ludueña, Vázquez, & Alvarez, 2012), spruce bark (Le
54 Normand, Moriana, & Ek, 2014), sisal fibres (Santos, Rodrigues, Ramires, Ruvolo-
55 Filho, & Frollini, 2015), pineapple leaf fibres (Shih et al., 2014), garlic skin (Reddy &
56 Rhim, 2014), soy hull (Flauzino Neto, Silvério, Dantas, & Pasquini, 2013), rice straw
57 (Boonterm et al., 2015; Kargarzadeh, Johar, & Ahmad, 2017), coconut husk fibres
58 (Rosa et al., 2010), mango seeds (Henrique, Silvério, Neto, & Pasquini, 2013), red algae
59 waste (El Achaby, Kassab, Aboulkas, Gaillard, & Barakat, 2018) or banana peel waste
60 (Hossain, Ibrahim, & AlEissa, 2016) have been evaluated, although no previous studies
61 into the extraction and characterization of cellulosic materials from coffee husk have
62 been reported. However, coffee husks, obtained after de-hulling the coffee cherries
63 during dry processing, also constitute a source of lignocellulosic materials, containing
64 ~57% of cellulosic and ~22% lignin components (Moreno-Contreras, Serrano-Rico, &
65 Palacios-Restrepo, 2009). Likewise, the coffee (*Coffea sp.*) crop generates a significant
66 amount of waste, which could be used as a source of different valuable products (Alves,
67 Rodrigues, Nunes, Vinha, & Oliveira, 2017), with a positive impact on the economy of
68 producing countries, such as Brazil, Vietnam, Indonesia, Colombia, Ethiopia, India or
69 Mexico (Oliveira & Franca, 2015).

70 On the other hand, rice husk, with a similar composition in lignocellulosic compounds
71 to coffee husk (~55% cellulose and ~35% lignin, Brinchi, Cotana, Fortunati, & Kenny,
72 2013), has been previously studied as a source of these materials. The previously
73 described methods to obtain and characterize cellulose fibres from rice husk (Johar,
74 Ahmad, & Dufresne, 2012; Kargarzadeh et al., 2017) could be applied to evaluate the

75 coffee husk's potential as a source of cellulosic fractions, useful as filling compounds in
76 composite materials.

77 Natural cellulose obtained from these kinds of biomasses can be transformed into
78 micro- and nano-scale materials, yielding products, such as microcrystalline cellulose,
79 microfibrillar cellulose or cellulose nanocrystals (CNC) (Azeredo, Rosa, & Mattoso,
80 2017; Brinchi et al., 2013; Sanjay et al., 2018) with very good properties as reinforces,
81 at the same time that this allows for a better exploitation of lignocellulosic residues.
82 The isolated cellulose fibres submitted to different treatments yield different crystalline
83 fractions such as CNCs. These are a good option as reinforcing agent due to their
84 abundant hydroxyl groups, which allows for obtaining different derivative materials
85 with adequate compatibility, high degree of crystallinity, excellent mechanical
86 properties, large specific surface area, high aspect ratio, and high thermal stability
87 (Azeredo et al., 2017; Ng et al., 2015). The final yield of the isolation process depends
88 on the type of initial raw material and process conditions.

89 The aim of this study was to isolate and characterise the cellulose fibres and cellulose
90 nanocrystals from coffee husks, in parallel with rice husk, for comparison purposes.
91 Likewise, the reinforcing capacity of the isolated cellulose fibres and CNCs, from both
92 coffee and rice husk, have been analysed in corn starch films through their effect on
93 tensile behaviour of the films.

94

95 **2. Materials and methods**

96 **2.1. Materials**

97 Rice husk was obtained from Dacsa (Almàssera, Valencia, Spain) and coffee husk was
98 provided by Centro Surcolombiano de Investigaciones en Café de la Universidad
99 Surcolombiana (Neiva, Colombia). Corn starch was purchased from Roquette (Roquette

100 Laisa, Benifaió, Spain). Glycerol and sodium hydroxide was purchased from Panreac
101 Química, S.A (Castellar del Vallès, Barcelona, Spain). Sodium chlorite and acetate
102 buffer for bleaching treatment and sulphuric acid (purity 98%) used for acid hydrolysis
103 were provided by Sigma Aldrich Química S.L (Madrid, Spain). For sample
104 conditioning, phosphorus pentoxide (P_2O_5) and magnesium nitrate-6-hydrate
105 ($Mg(NO_3)_2$) were supplied by Panreac Química, S.A. (Castellar del Vallès, Barcelona,
106 Spain). All other chemicals used were reagent grade and underwent no further
107 purification.

108

109 **2.2. Extraction/purification process applied to rice and coffee husks**

110 The same process conditions for both raw materials were applied, adapted from Johar et
111 al. (2012). Ground rice or coffee husks, with a mean particle size of 2-3 mm, were alkali
112 treated and afterwards submitted to bleaching treatment and acid hydrolysis.

113

114 **2.2.1. Alkali treatment**

115 Alkali treatment was carried out with a 4 wt% NaOH solution, with a solid solution
116 mass ratio of 1:20, at reflux temperature for 3 h, under continuous stirring. Then, the
117 solid was filtered and washed with distilled water several times until the alkali solution
118 was removed. This treatment was repeated twice.

119

120 **2.2.2. Bleaching treatment**

121 For bleaching treatment, equal parts of acetate buffer solution, sodium chlorite (1.7
122 wt%) and water were mixed with the alkali treated solid (1:20 mass ratio) and submitted
123 at reflux temperature for 4 h. This process was repeated as many times as necessary (4
124 in rice samples and 6 in coffee samples) until the samples were completely white. After

125 that, the samples were filtered and washed with distilled water several times until the
126 solution was removed.

127

128 **2.2.3. Acid hydrolysis**

129 Cellulose nanocrystals were prepared by acid hydrolysis of the obtained bleached fibres
130 as described Johar et al. (2012) and Cano et al. (2015). Fibre treatment with sulphuric
131 acid, (64 %, wt/wt) at 50 °C for 40 min, was carried out under continuous stirring, using
132 7.5 wt% of fibre content. The hydrolysed cellulose sample was washed several times
133 with distilled water by centrifugation at 14.000 rpm for 30 min to concentrate cellulose
134 material and to remove acid excess. The suspension was then dialysed against distilled
135 water until a constant pH was reached and then neutralised with 10 wt % ion resin
136 (Dowex Marathon MR-3) for 24 h. The resin was separated by vacuum filtration
137 through a Whatman 541 filter, and the filtrate CNC suspension was sonicated for 30
138 min, using a tip sonicator (Vibra-Cell™ VCX 750, Sonics & Materials, Inc., Newton,
139 USA) in an ice bath and kept refrigerated for further analyses.

140

141 **2.3. Structural characterization of fibres and cellulose nanocrystals (CNC).**

142 **2.3.1. Thermal properties**

143 The thermal stability of the different samples was analysed using a Thermogravimetric
144 Analyzer TGA 1 Star^e System analyser (Mettler-Toledo, Inc., Switzerland) under
145 nitrogen atmosphere (gas flow: 10 mL min⁻¹). Samples (about 8 mg) were heated from
146 25 to 900 °C at 10 °C/min (Johar et al., 2012). At least two replicates for each sample
147 were obtained. Initial degradation temperature (Onset) and peak temperature (Peak)
148 were registered from the first derivative of the resulting weight loss curves using the
149 STAR^e Evaluation Software (Mettler-Toledo, Inc., Switzerland).

150 **2.3.2. X-ray diffraction**

151 A Diffractometer (XRD, Bruker AXS/D8 Advance) was used to obtain X-Ray
152 diffraction patterns of the different samples conditioned at 25 °C and 53% RH, between
153 2θ : 5° and 40° using $K\alpha$ Cu radiation (λ : 1.542 Å), 40 kV and 40 mA with a step size of
154 0.05°. For this analysis, the samples were milled and spread until covering the sample
155 holder. The degree of crystallinity (X_c) of the samples was estimated from the ratio of
156 crystalline peak areas and the integrated area of XRD diffractograms, using OriginPro
157 8.5 software, assuming Gaussian profiles for crystalline and amorphous peaks, as
158 described by Ortega-Toro et al. (2016a).

159

160 **2.3.3. Optical microscopy**

161 Light microscopy (Optika Microscopes B-350 connected to an Optikam B2 camera,
162 Italy) was used to measure the length range of the fibres after alkali and bleaching
163 treatments. A drop (30 μ L of the 1% solid dispersions was spread over the porta and
164 observed at 10X magnification level. The bleached fibres were stained with gentian
165 violet to obtain the adequate contrast.

166

167 **2.3.4. Scanning Electron Microscopy (SEM)**

168 A Scanning Electron Microscope (JEOL JSM-5410, Japan) was used to analyse the
169 microstructure of the material after different treatments: untreated (C), alkali treated
170 (At) and bleached (Bt) fibres. The fibres were maintained in desiccators with P_2O_5 for 2
171 weeks at 25 °C and the SEM observations were carried out in duplicate for each
172 material. Samples were gold coated and observed, using an accelerating voltage of 15
173 kV.

174

175 **2.3.5. Morphological analysis in CNCs**

176 The particle size distribution of rice and coffee cellulose nanocrystals was analysed by
177 using 2 mL of CNC suspensions in Zetasizer equipment (Zetasizer Nano ZS, Malvern
178 Instruments, U.K). Five measurements were taken for each sample.

179 Transmission Electron Microscopy (JEOL JEM-1010, Japan) was also used to analyse
180 morphology and the size of the CNCs. A drop of diluted CNC dispersion was deposited
181 on the carbon film supported by the copper grid. TEM analysis was performed at an
182 accelerating voltage of 80 kV. The aspect ratio (Ar) (L/d , L mean length and d mean
183 diameter) was measured in 100 individualised crystals of each sample, **from the digital
184 images recorded with an AMT V600 camera, using the ImageJ software.**

185

186 **2.3.6. Chemical composition**

187 The contents of cellulose, hemicellulose, lignin and ashes of the rice and coffee husk
188 samples submitted to the different treatments were determined according to the NREL
189 standard method for biomass (NREL/TP-510-42618, 2011), applying sulphuric acid
190 hydrolysis to the samples and determining the cellulose and hemicellulose content from
191 the glucose, xylose and arabinose amounts determined by HPLC (Hitachi LaChrom,
192 Japan) in the sample hydrolysate. The Biorad Aminex HPX-87H column and an IR
193 detector (Agilent 1200 series RID) were used. The injection volume of all samples was
194 60 μ L, and 5 mM sulphuric acid has been used as mobile phase at 0.5 mL/min flow rate
195 and 65 °C. Data acquisition was carried out by using Galaxie Chromatography Data
196 System. Raw husk powders were first water extracted, according to the method
197 recommendation (NREL/TP-510-42619, 2008) to avoid interferences in the HPLC
198 analysis. Lignin and ashes were also determined from the acid insoluble fraction by

199 filtration, desiccation and incineration. Total ashes were analysed in the whole sample
200 by sample incineration at 575 °C for 24 h.

201

202 **2.3.7. Reinforcing capacity of the cellulose fibres and CNCs**

203 The reinforcing capacity of 1 wt % of cellulose fibres and CNCs was evaluated in corn
204 starch films containing 30 % glycerol, obtained by melt blending (using Model LRM-
205 M-100, Labtech Engineering, Thailand roller) and compression moulding (using Model
206 LP20, Labtech Engineering, Thailand press). Pellets of thermoplastic starch (with or
207 without filler) were obtained at 160 °C in the roller-mill using a prepared dispersion of
208 the starch-glycerol-filler-water (1:0.3:0.01:0.5). The pellets conditioned at 53% relative
209 humidity (RH) were compressed at 130 bars and 160°C for 8 min, and cooled down to
210 50°C for 3 min.

211 The obtained films were conditioned at 53% RH and 25 °C for 1 week and analysed as
212 to the tensile behaviour (ASTM standard method D882; ASTM, 2001), using a
213 universal test machine (TA.XTplus model, Stable Micro Systems, Haslemere, England).
214 Rectangular film samples (2.5 x 10 cm) were submitted to the tensile test at 50 mm/min
215 till break. Tensile strength-Henky strain curves were obtained and elastic modulus,
216 tensile strength and deformation at break were determined for each sample. Ten
217 replicates were carried out for each film sample.

218

219 **2.3.8. Statistical analysis**

220 Statgraphics Plus for Windows 5.1 (Manugistics Corp., Rockville, MD) was used for
221 statistical analyses of data through analysis of variance (ANOVA). Fisher's least
222 significant difference (LSD) was used at the 95% confidence level.

223

224 **3. Results and discussion**

225 **3.1. Morphological changes and yield in alkali and bleaching treatments of rice** 226 **and coffee husk.**

227 The effect of the different treatments on the appearance of the raw material can be
228 observed in Fig. 1a, 1b and 1c. Alkali treatment provoked a colour change in both rice
229 and coffee husk samples, the former turning more brownish whereas the latter became
230 brownish-orange, which could be associated with the different pigmentation of the
231 lignin fraction remaining in the material. This treatment was effective at purifying the
232 cellulose fibres, removing non-cellulosic components, such as a part of the lignin
233 fraction, hemicellulose, pectin and wax (Kallel et al., 2016). After the bleaching
234 treatment, the sample colour change was more evident and the samples exhibited the
235 characteristic whiteness of cellulose fibres, where extensive extraction of cementing
236 compounds, such as lignin, occurred. Table 1 shows the sample composition of the
237 products submitted to the different treatments, where the contents in lignin,
238 hemicellulose, cellulose and ashes are shown. The progressive enrichment in cellulose
239 of the husk products can be observed after the successive treatments. Cellulosic fibres
240 obtained after bleaching still contained a fraction of hemicellulose in both rice and
241 coffee products and the cellulose enrichment of the rice sample was slightly lower than
242 the 96% reported by Johar et al. (2012), applying similar treatments to rice husks.

243 The cellulose and lignin content of the husk samples ranged between 34-35% and 22-23
244 %, respectively, showing lower values than those previously reported by other authors.
245 Brinchi et al. (2013) reported about 55% and 36% for cellulose and lignin content in
246 rice husk, respectively, while Moreno-Contreras et al. (2009) quantified about 57%
247 cellulose and 22% lignin in coffee husk. Both coffee and rice husks contained a similar

248 amount of cellulosic compounds, which agrees with the similar solid yield obtained for
 249 both products after the alkali and bleaching processes.

250

251 **Table 1.** Chemical composition (wt. percentage) of the rice (R) and Coffee (C) husk
 252 samples at the different treatment step (untreated: C, alkali treated: At, bleaching
 253 treated: Bt)

254

Samples	Water extractables (%)	Cellulose (%)	Hemicellulose (%)	Lignin (%)	Ashes (%)
R-C	14.5 ± 0.9	33.8 ± 0.5	17.1 ± 0.2	21.5 ± 0.3	16.5 ± 0.4
R-At	-	55.9 ± 0.3	15.8 ± 0.4	19.9 ± 0.8	0.59 ± 0.05
R-Bt	-	73.8 ± 0.3	19.2 ± 0.7	1.6 ± 0.5	0.14 ± 0.03
C-C	17.8 ± 1.4	35.4 ± 0.9	18.2 ± 1.3	23.2 ± 0.5	1.4 ± 0.3
C-At	-	52.6 ± 1.1	19.0 ± 0.2	20.4 ± 0.6	0.74 ± 0.15
C-Bt	-	61.8 ± 2.6	27.2 ± 0.9	2.6 ± 0.4	0.49 ± 0.19

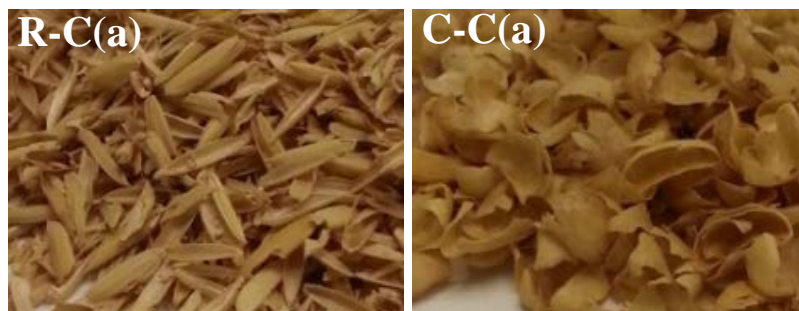
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256 Alkali treatment yielded 86 and 88 % with respect to the initial dry powder of rice and
 257 coffee husks, respectively, while this yield was 48 and 60 %, respectively, for the
 258 bleaching treatment, which produced a cellulose-rich sample (41 and 53 %, respectively
 259 with respect to the initial mass). Moreover, taking the final cellulose
 260 content of each bleached product into account, a better purification of the rice husk
 261 cellulosic fraction than that of the coffee husk was reached. The cellulose content found
 262 in different lignocellulosic wastes varies between 40-80 %. Jonoobi, Ahmad &
 263 Dufresne (2015) reported cellulose and lignin contents of different raw materials, such
 264 as kenaf stem (~58%, ~17.5%), wheat straw (~43%, ~22%), pineapple leaf (~81%,
 265 ~3.5%) and banana rachis (~48%, ~12%).

266 After treatments, the particle size of the ground raw materials (2-3 mm) was drastically
 267 reduced, as reflected in the light microscopy images (Fig. 1c and 1e), in which the fibre
 268 length ranges are shown after the alkali and bleaching treatments. Particle size reduction

269 provoked by bleaching treatment was similar for both husk materials and reflects the
270 effective attack of the chemical agents disrupting the internal structure of the material
271 while removing the non-cellulosic components. Other authors reported different fibre
272 sizes depending on the raw material. As examples, bleached water hyacinth fibre was
273 25–50 μm in size (Sundari & Ramesh, 2012); bleached garlic straw filaments were
274 about 14-17 μm length (Kallel et al., 2016) and pineapple leaf fibres were in the range
275 of 50-12 μm (Cherian et al., 2011). The main difference between cellulosic rice and
276 coffee fibres are the trend to aggregate in the water dispersion. Whereas coffee fibres
277 appear as scattered elements, rice fibres are flocculated in the aqueous medium (Fig. 1,
278 R-Atc). This suggests the action of more attractive forces between particles despite the
279 similarity in size and composition, which could be associated to different surface
280 bonded molecules.

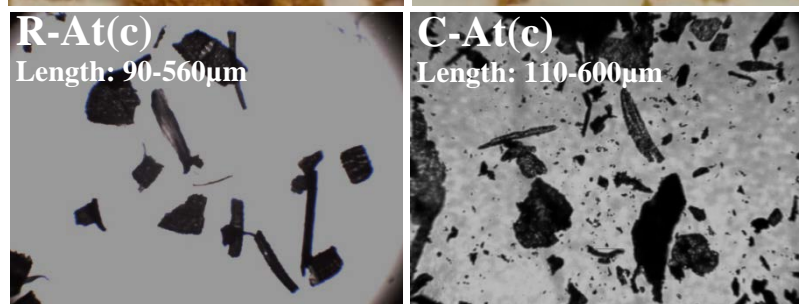
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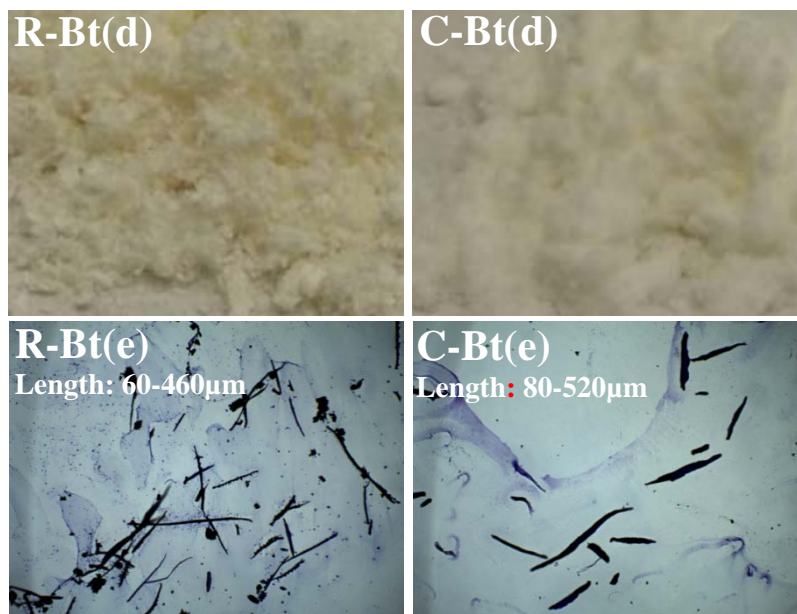


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Fig. 1. Appearance of rice (R) and coffee (C) husk samples after each treatment: untreated sample (control: C-a), alkali treatment (At-b), bleaching treatment (Bt-d). Light micrographs of rice and coffee husk fibres after alkali (At-c) and bleaching treatment (Bt-e).

3.2. Micro- and Nano-structural analysis

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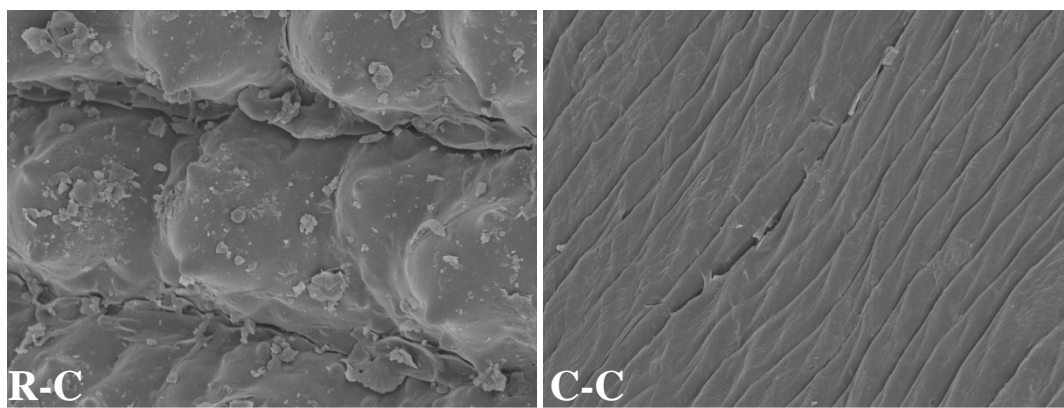
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Fig. 2 shows SEM micrographs of the rice and coffee fibres, either untreated or submitted to the different treatments, where the structural changes produced by the successive processes can be observed. In the raw material, an ordered and homogeneous organisation could be distinguished evidencing the natural arrangement of the constituents. In both samples, the fibres are arranged in parallel and an ordered assembly can be observed, while pectin and intercellular junctional material fill the cell unions. Wax crystals covering the sample surface can also be seen, mainly in rice husk, which showed a rougher appearance. After the alkali treatment, the fibre surface becomes rougher due to the losses in the outer non-cellulosic compounds (hemicellulose, lignin, pectin and wax). This change was more appreciable in rice husk.

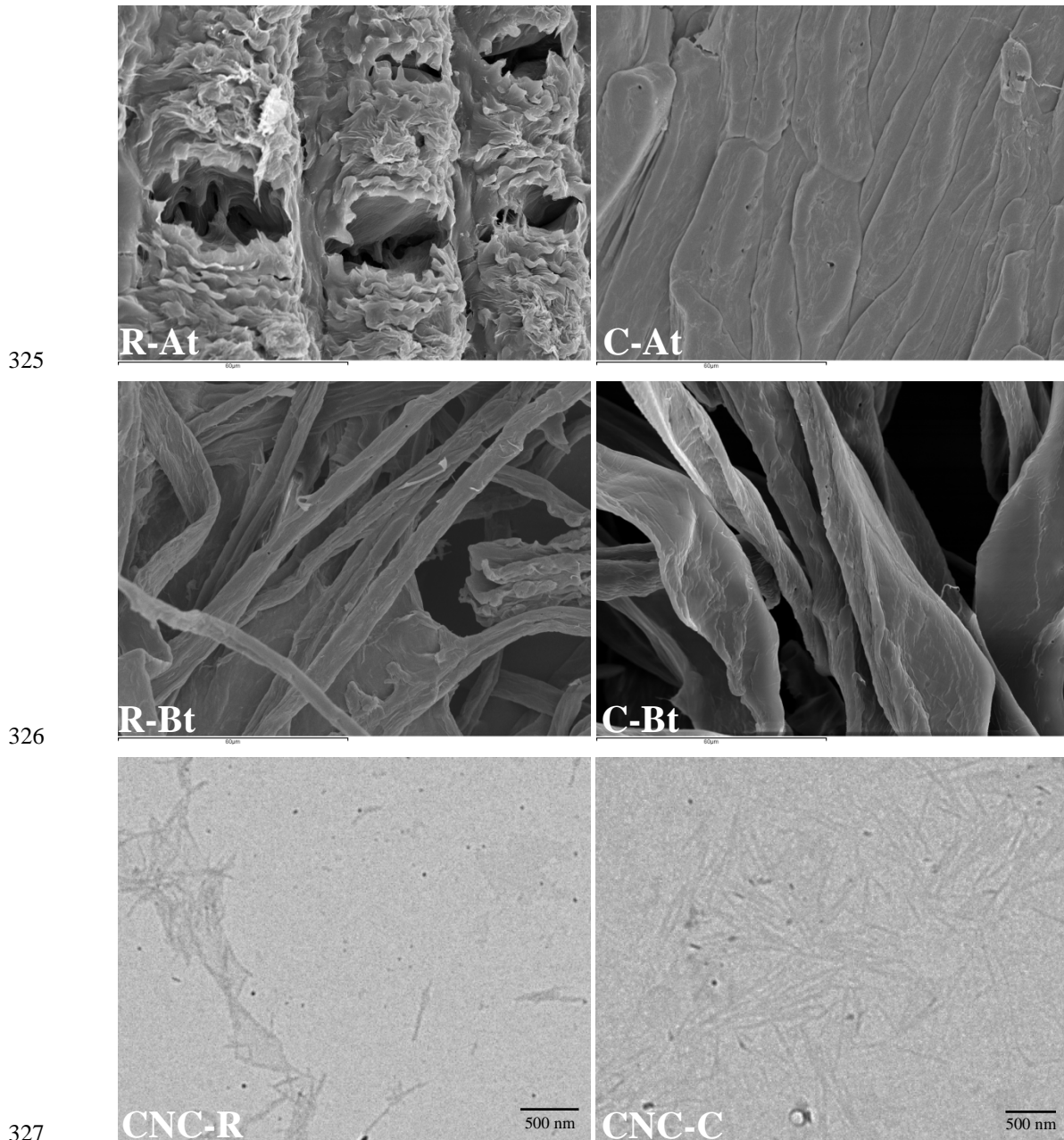
306 Both wax and pectin are known to surround the surface of natural fibres as a protective
307 layer. In both cases, fibre bundles remain after alkali treatment, which indicates the
308 retention of the cementing lignin material, which acts as a binder in the fibre
309 components and preserves the bundle shape during the alkali treatment. Nevertheless, a
310 great part of the pectin and hemicellulose content was removed from the fibres during
311 this treatment, according to previous studies (Batra, 1981; Johar et al., 2012). The
312 appearance of coffee husk samples reveals a less aggressive effect of the alkali
313 treatment, which can be explained by its different composition. In fact, the product yield
314 in the alkali treatment was higher in coffee than in rice husk, which could be due to a
315 lower extraction of components in this product, as reflected in sample composition
316 (Table 1).

317 The alkali process is more effective when the biomass has low lignin content and the
318 process yield depends on the raw material and treatment conditions. Treatment with
319 sodium hydroxide decreases the degree of polymerization and crystallinity in the
320 material, disrupts the lignin structure and increases the internal surface of cellulose
321 fibres (Singh, Shukla, Tiwari, & Srivastava, 2014). All of this favours the penetration of
322 the bleaching agents into the structure (Kallel et al., 2016).

323



324



328 **Fig. 2.** SEM micrographs of the rice and coffee husks untreated (R-C or C-C) and after
 329 the alkali (R-At or C-At) and bleaching treatments (R-Bt or C-Bt). **TEM micrographs of**
 330 **nanocrystals obtained from rice (CNC-R) and coffee (CNC-C) bleached fibres are also**
 331 **shown.**

332

333 In both husk materials, the subsequent bleaching treatment was very effective at
 334 separating isolated fibres, as observed in Fig. 2 (R-Bt and C-Bt). The rice and coffee
 335 husk fibre bundles separate into very thin individual fibres with different diameters, thus

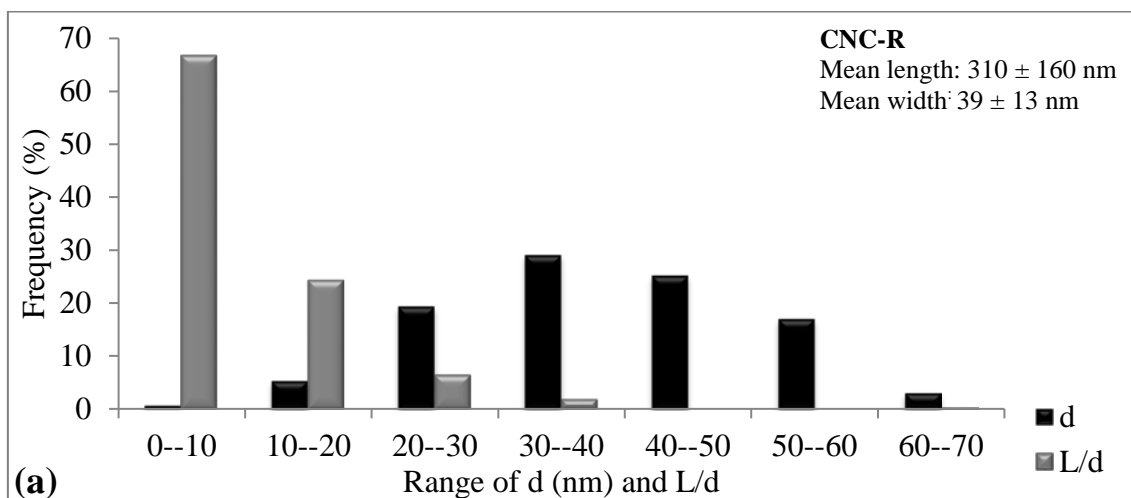
336 indicating the great efficacy of the treatment to eliminate the non-cellulosic
337 components, as previously reported (Chen et al., 2011; Espino et al., 2014; Savadekar &
338 Mhaske, 2012). In the treatment, sodium chlorite becomes chlorine dioxide in acetate
339 buffer solution, provoking the oxidation of the lignin aromatic ring (Ng et al., 2015; Ni,
340 Kubes, & Van HelnIngen, 1993; Zainuddin, Ahmad, Kargarzadeh, Abdullah, &
341 Dufresne, 2013). Coffee fibres were flatter and mainly helically folded, while rice fibres
342 were similar to those previously reported by Johar et al. (2012), more cylindrical of less
343 than 10 μm in diameter. The fibres diameters were in the range of previously reported
344 for other plant wastes such as bleached garlic straw (Kallel et al., 2016). The obtained
345 cellulose fibres showed a very high aspect ratio (L/d , L being the length and d the
346 diameter), which provides them with adequate properties as reinforcing materials for
347 composite applications.

348 The Van der Waals forces and hydrogen bonds that act in the crystalline zones of the
349 cellulose provide high resistance to acid attack, while the amorphous regions are
350 disordered and prone to sulfuric acid hydrolysis (Khalil et al., 2016; Qiao, Chen, Zhang,
351 & Yao, 2016). Thus, the sulphuric acid treatment removes the cellulose fibres of the
352 amorphous zones and reduces the fibre size to nanometric scale (Azizi Samir, Alloin, &
353 Dufresne, 2005). Therefore, observations of the isolated CNCs were carried out by
354 TEM at higher magnification. Fig. 2 also shows micrographs of the CNCs with a short,
355 rod-like structure and many aggregates. These aggregates emerged in the drying step for
356 sample preparation due to strong intermolecular hydrogen bonds between the particles
357 (Sung et al., 2017). Likewise, the particle morphology and aggregation is highly
358 affected by the TEM sample preparation (Chauve, Fraschini, & Jean, 2013; Kaushik,
359 Fraschini, Chauve, Putaux, & Moores, 2015). However, some specimens remained
360 isolated, which allows for the characterisation of their length and diameter and so, their

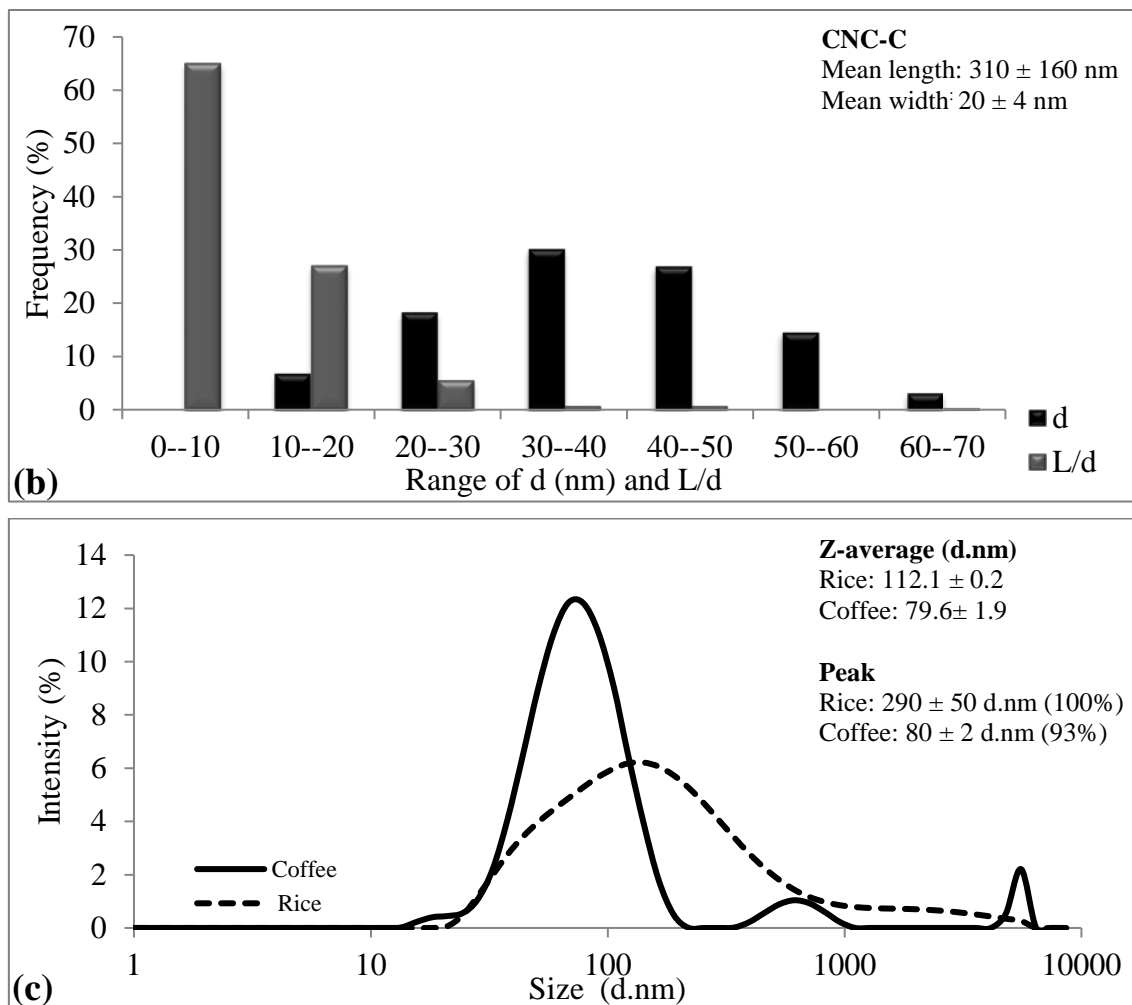
361 aspect ratio, from digital images recorded with the camera and the use of the image
362 analysis software. Fig. 3a and 3b show the distribution of diameter and aspect ratio of
363 CNCs extracted from rice and coffee husk fibres. Length (L) and width (d) of rice CNC
364 were 310 ± 160 and 39 ± 13 nm, and similar values were obtained for coffee CNC (310
365 ± 160 and 20 ± 4 nm, respectively), the latter being slightly thinner. Most of the CNCs
366 exhibited an aspect ratio (L/d) in the range of 10-20 for both husk materials. These were
367 similar both to those obtained by Johar et al. (2012) for rice husk and to the values
368 reported for different lignocellulosic materials, such as ramie fibres (Habibi &
369 Dufresne, 2008), bagasse pulp (Bras et al., 2010), sugarcane bagasse (Hassan M.L.,
370 Mathew, Hassan E. A, El-Wakil, & Oksman, 2012), kenaf fibres (Kargarzadeh et al.,
371 2012), banana peel (Khawas & Deka, 2016) or *Alfa tenassissima* (BenMabrouk, Wim,
372 Dufresne, & Boufi, 2009). Kallel et al. (2016) report that the aspect ratio of CNC
373 extracted from different materials usually vary from 10 to 70, although the highest
374 values (L/d = 80) were obtained in CNC extracted from garlic straw (Kallel et al., 2016)
375 and tunicate (*Polycarpa aurata*), a sea animal with external cellulosic microfibrils, (L/d
376 = 50-200) (Anglès & Dufresne, 2000). Silvério et al. (2013) reported that CNC could be
377 considered as a good reinforcing material if their aspect ratio exceeds a value of 10. The
378 CNCs dimensions ranges from 2 to 20 nm in diameter and from 100 to 600 nm in length
379 (Morian, Vilaplana, & Ek, 2016), depending on the nature of the lignocellulosic raw
380 material, purification, mechanical process, pre-treatment and conditions of acid
381 hydrolysis (Chauve et al., 2013).

382 Fig. 3c also shows the hydrodynamic diameter distribution of the isolated CNCs for
383 water suspensions of CNC from rice and coffee husks obtained from dynamic light
384 scattering analysis. Hydrodynamic diameter corresponds to that of an equivalent
385 spherical specimen that would have the same diffusion coefficient and, for rod-like

386 particles, such as CNCs, are only used for qualitative comparisons. The obtained
 387 hydrodynamic diameter distributions were similar to that reported by Kallel et al. (2016)
 388 for CNCs obtained from garlic straw. The z-average values and peak values for each
 389 sample are also shown in Fig. 3c. This average size, related with the hydrodynamic
 390 volume of the crystals, and the wide distribution reflects the aggregation tendency of
 391 CNC in the aqueous media. CNC from coffee husk were smaller and with lower
 392 tendency to aggregate as revealed by the smaller z-average and narrow distribution.
 393 This agrees with the behaviour deduced from the microscopic observations.
 394 Given the crystalline nature of cellulose, contrary to the amorphous nature of
 395 hemicellulose and lignin, the crystallinity of the natural fibres rises as the extraction
 396 progresses with the different chemical treatments. Indeed, the alkali treatment increases
 397 the stiffness of the fibre as it removes different amorphous fractions.



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Fig. 3. Diameter distribution and aspect ratio of the CNC extracted from rice (a) and coffee (b) bleached fibres. Particle size distribution from dynamic light scattering of the CNC aqueous dispersion from rice and coffee samples (c).

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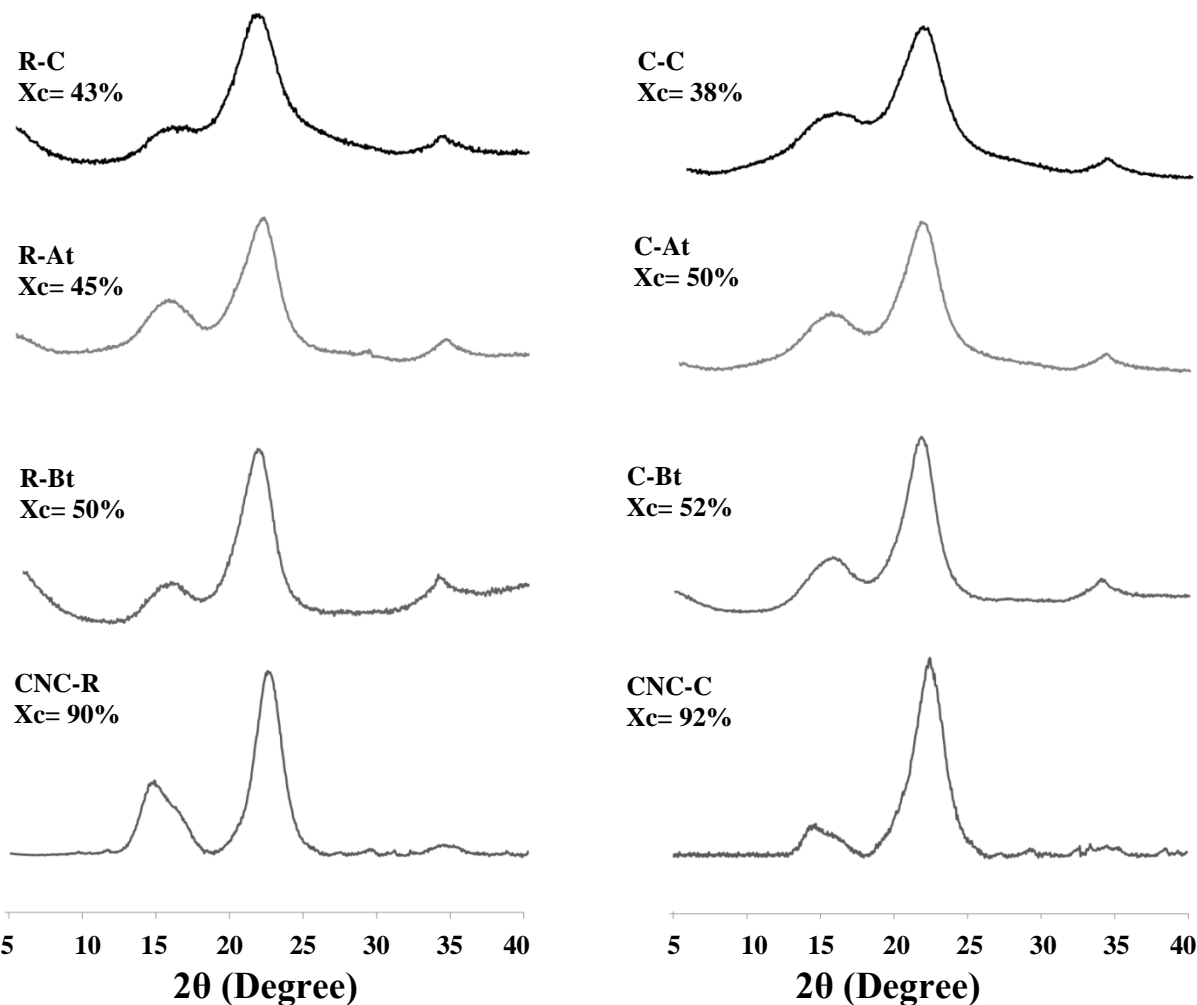
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Acid treatment does not affect the crystalline domains, but destroys the amorphous region of the fibre (Fengel & Wegener, 1984). Then, the X-ray diffraction spectra of the different products obtained after each treatment were analysed to control the progressive enrichment in crystalline cellulose.

Fig. 4 shows the X-ray diffraction patterns of the untreated, alkali treated and bleached fibres as well as of the isolated CNC of rice and coffee fibres. In all samples, the typical crystalline peaks of type I cellulose (2θ : $15-16^\circ$ [110], 22° [200] and 34° [004]) were observed as reported by other authors (Johar et al., 2012; Kallel et al., 2016; Savadekar

415 & Mhaske, 2012). As expected, these peaks become more defined upon each chemical
416 treatment. Likewise, the degree of crystallinity (X_c) in the different samples was
417 estimated from the ratio between the area of the peaks and the total area under spectra
418 which include the amorphous response (Ortega-Toro et al., 2016a). The crystallinity
419 degree increased after each chemical treatment, according to the progressive removal of
420 components of the amorphous fraction (Bettaieb et al., 2015; Le Normand et al., 2014;
421 Sheltami, Abdullah, Ahmad, Dufresne, & Kargarzadeh, 2012). This increase was more
422 noticeable in the last acid treatment when the amorphous regions of cellulose were also
423 eliminated and CNC were purified. During the hydrolysis process, hydronium ions can
424 penetrate the more accessible amorphous regions of cellulose provoking the cleavage of
425 glycosidic bonds, releasing individual crystallites (Lima & Borsali, 2004), which can
426 grow or realign in parallel, thus increasing the cellulose crystallinity (Li et al., 2009).
427 CNC exhibited high crystallinity due to the chemical cellulose structure, where each
428 monomer has three hydroxyl groups with the ability to form intra- and inter-molecular
429 hydrogen bonding between cellulose chains giving rise to a highly compact system (Ng
430 et al., 2015; Siqueira, Bras, & Dufresne, 2010). The high degree of crystallinity confers
431 stiffness on the CNC and so a reinforcing capability when included in other polymer
432 materials.



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Fig. 4. X-Ray diffraction patterns of rice and coffee husks untreated (C), alkali treated (At), bleaching treated (Bt) and acid hydrolysed (CNC).

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438 3.3. Thermal analysis

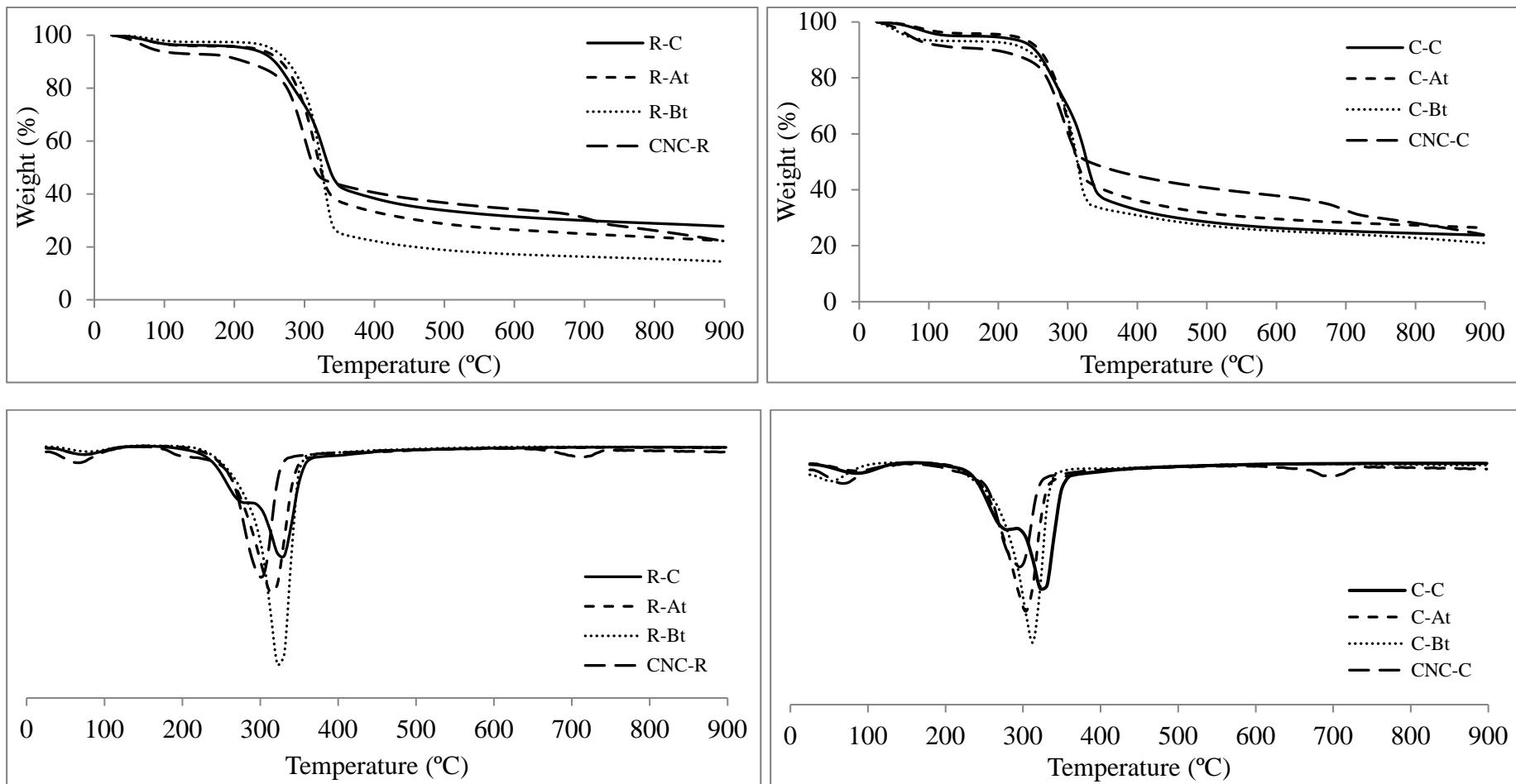
439 Thermogravimetric analysis allows us to obtain information about how the thermal
440 stability of the material changes in line with the different treatments. The progressive
441 extraction of low molecular weight compounds leads to the enhancement of the thermal
442 stability.

443 Fig. 5 shows the TGA and DTG curves of the rice and coffee husk samples, untreated
444 and submitted to the different treatments. The peaks associated with the different weight
445 losses caused by thermal degradation can be observed in DGTA curves. All samples

446 exhibit a small weight loss below 100°C, which can be attributed to the elimination of
447 physically adsorbed water in the samples. The other peaks are associated with the
448 thermal degradation of the lignocellulosic compounds. Table 2 summarises thermal
449 degradation temperatures (onset and peak) for both rice and coffee samples submitted to
450 the different chemical treatments. The first degradation step in untreated samples,
451 around 271-279 °C, can be attributed to the decomposition of hemicelluloses and a part
452 of the lignin fraction (Shebani, Van Reenen, & Maincken, 2008) and was very similar in
453 both husk samples. In all samples, the peak around 300 °C is attributable to cellulose
454 and lignin decomposition. Mensaray & Ghaly (1998) reported that lignocellulosic
455 materials decompose in the temperature range of 150-500 °C. Specifically,
456 hemicellulose decomposes mainly from 150 to 350 °C, cellulose at between 275 and
457 350 °C and lignin undergoes gradual decomposition in the range of 250-500 °C. As
458 shown in Table 2, the rice and coffee samples exhibited similar thermal behaviour,
459 similarly affected by chemical treatments.

460 Two decomposition peaks were observed in both untreated material associated to the
461 successive degradation of the different lignocellulosic fractions. However, only one
462 peak appeared after after alkali treatment in both cases, which can be attributed to the
463 loosening of the low molecular weight lignin fragments, which degrade at lower
464 temperatures (Johar et al., 2012). After the bleaching treatment, a significant increase in
465 degradation temperature was observed due to the removal of hemicellulose and lignin
466 fractions and the higher content of cellulose.

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Fig. 5. TGA and DTG curves of rice and coffee husks untreated (C), alkali treated (At), bleaching treated (Bt) and acid hydrolysed (CNC).

473 The residual mass at temperature around 400–700 °C in raw rice husks fibre were
474 remarkably high (more than 30%), which is due to the high content of ash and lignin
475 (Fahma, Iwamoto, Hori, Iwata, & Takemura, 2011) as well as to the high silica content
476 of the rice husk, all of these contributing to the high char content for raw fibres. Similar
477 thermogram patterns have also been reported on raw rice husk fibres (Johar et al.,
478 2012). For coffee husk no remarkable differences on the mass residues were observed for
479 untreated and treated fibres, all these being lower than 30%, according to the lower ash
480 content of the material. The isolated, more crystalline cellulose fraction from the acid
481 treatment had a lower degradation temperature in both husk products, which has been
482 attributed to the surface sulphation resulting from the sulphuric acid treatment (Johar et
483 al., 2012) by the replacement of hydroxyl groups of the cellulose structure by sulphate
484 groups throughout the hydrolysis process (Jonoobi et al., 2015; Siqueira et al., 2010).
485 Moreover, this treatment induces an increase in the char fraction compared to the
486 bleached fibres, also due to the introduction of sulphated groups which act as flame
487 retardants (Roman & Winter, 2004). However, at about 700 °C, a final mass loss step
488 was observed in both acid hydrolysates, reflecting the action of a final degradation
489 mechanism, which led to a similar residual mass to the bleached fibres and which was
490 not observed in previous studies on rice husk cellulosic materials (Johar et al., 2012).
491 The small differences in the thermal behaviour of rice and coffee husk can be explained
492 by the compositional differences, since thermal degradation of cellulose-based fibres is
493 significantly influenced by their structure and composition.

494

495 **Table 2.** Mean values and standard deviation of onset and peak temperatures for
496 thermal degradation of rice and coffee husks of each stage of treatment (untreated: C,
497 alkali treated: At, bleaching treated: Bt).

Samples	[245-280]°C		[248-330]°C	
	Onset (°C)	Peak (°C)	Onset (°C)	Peak (°C)
R-C	247.6 ± 0.5 ^b	271.1 ± 0.7 ^a	301.8 ± 0.2 ^f	329.8 ± 0.7 ^f
R-At	-	-	263 ± 4 ^c	317.3 ± 1.4 ^d
R-Bt	-	-	292.4 ± 0.6 ^e	326.8 ± 0.2 ^e
CNC-R			261.8 ± 0.4 ^c	301.2 ± 0.3 ^b
C-C	245.0 ± 0.4 ^a	279.16 ± 1.12 ^b	302.6 ± 1.2 ^f	327.8 ± 0.7 ^{ef}
C-At	-	-	254.1 ± 0.4 ^b	304.8 ± 1.2 ^c
C-Bt	-	-	280 ± 3 ^d	315.6 ± 2 ^d
CNC-C			248.9 ± 0.4 ^a	295.6 ± 1.1 ^a

498 Different superscript letters within the same column indicate significant differences among formulations
499 (p < 0.05).
500

501

502 **3.4. Reinforcing capacity of the cellulose fibres and CNCs**

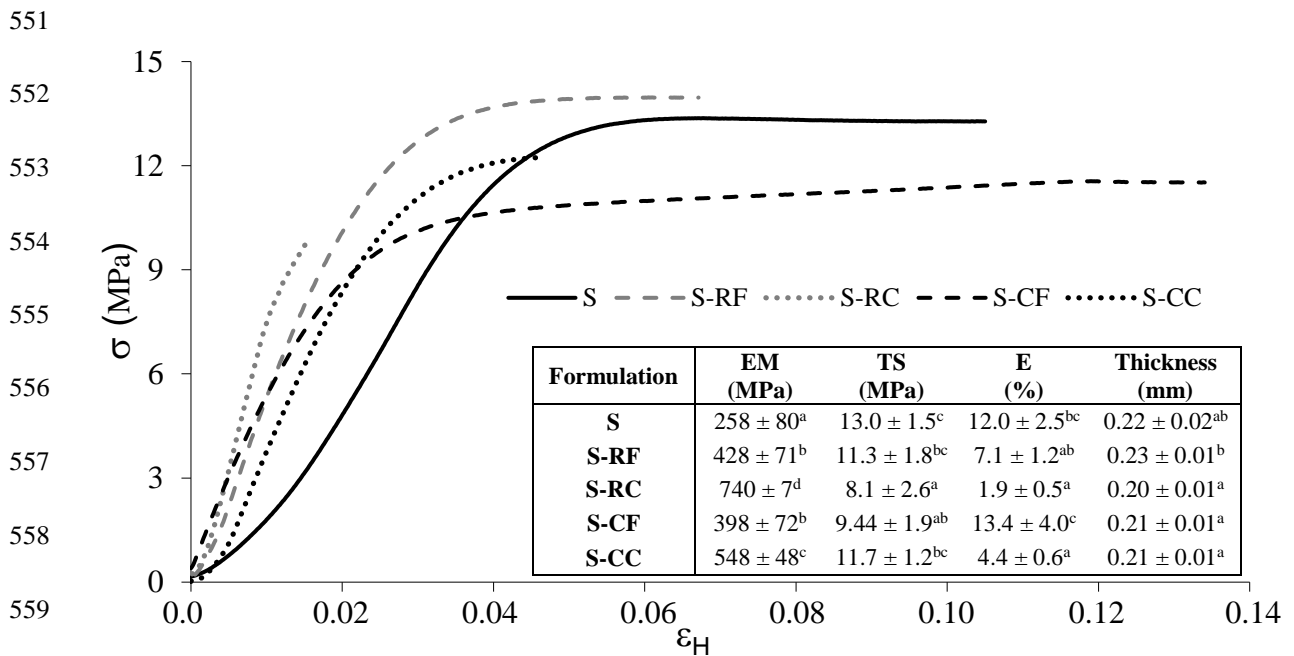
503 Starch is one of the most widely studied thermoplastic biopolymers because it is an
504 abundant material, cheap, renewable and biodegradable (Ortega-Toro, Collazo-
505 Bigliardi, Talens, & Chiralt, 2016b). Thermoplastic starch (TPS) film matrices exhibit
506 high barrier properties to oxygen, carbon dioxide and lipids (Ghanbarzadeh, Almasi, &
507 Entezami, 2011), although they have limited mechanical properties and are highly water
508 sensitive. Different studies revealed the improvement in mechanical properties of starch
509 films by the incorporation of cellulosic materials, especially CNCs from different
510 sources (Kargarzadeh et al. 2017; Zainuddin et al, 2013). Likewise, CNCs also
511 increased the film hydrophobicity, reducing the water vapour permeability (Slavutsky &
512 Bertuzzi, 2014) and water uptake capacity (Kargarzadeh et al., 2017). In this sense,
513 cellulosic materials (isolated fibres and CNCs) obtained from rice or coffee husks were
514 incorporated into TPS films in order to analyse their reinforcing ability in this
515 biopolymer matrix.

516 Fig. 6 shows the tensile behaviour of corn starch films containing, or not, cellulosic
517 reinforcing agents, where the values of the elastic modulus (EM), tensile strength (TS)

518 and percentage deformation (E) at break are also shown. Both cellulosic fibres and
519 CNCs enhanced the elastic modulus of the films, reflecting the reinforcing capacity of
520 both cellulosic fractions. Nanocrystals provoked a higher increase in the film elastic
521 modulus than fibres, while rice husk crystals were more effective at reinforcing the TPS
522 network (186 and 121% higher elastic modulus, respectively for rice and coffee husk
523 CNCs). However, no significant differences were observed for the effect of cellulosic
524 fibres, which increased the elastic modulus of TPS films by 60 %. Other authors also
525 observed better reinforcing properties for nanocrystals than for fibres, by incorporating
526 multiscale kenaf fibres (Zainuddin et al., 2013) or rice husk (Kargarzadeh et al. 2017)
527 fibres in cassava starch films, due to the higher surface area and aspect ratio of the
528 nanocrystals. As previously reported, the mechanical impact of CNCs on TPS films can
529 be, in part, explained by the formation of a percolating network within the polymer
530 matrix, where the stress is assumed to be transferred through crystal-crystal interactions
531 and crystal-polymer matrix interactions (Fortunati et al., 2013). According to the
532 statistical percolation theory for cylindrically shaped particles (Favier, Cavaille,
533 Canova, & Shrivastavas, 1997), the critical percolation volume fraction (percolation
534 threshold: X_P) can be estimated from the particle aspect ratio (Ar) as $0.7/Ar$. For the
535 composite films with CNCs, X_P could be estimated as 0.09 and 0.045, respectively for
536 rice and coffee crystals, considering the respective Ar values (7.9 and 15.6). So, the
537 percolation threshold was reached in the nanocomposites and percolation network
538 formation can explain the enhancement of the mechanical behaviour of the films
539 containing CNCs.

540 As concerns resistance and elongation at break, all cellulosic materials provoked a
541 decrease in the film extensibility, except coffee husk fibres which did not significantly
542 affect the stretchability and resistance of the TPS films. Reduction of the elongation

543 capacity at break implied slightly lower values of the corresponding resistance. Other
 544 authors also observed the reinforcing effect of starch films by CNCs, through the
 545 enhancement of the elastic modulus, although the film stretchability became more
 546 limited (González K, Retegi, González A, Eceiza, & Gabilondo, 2015). This was
 547 attributed to the stronger affinity between glycerol and the nanocrystal phase, as
 548 deduced from the dynamic mechanical analyses (DMA), which show an increase in the
 549 relaxation temperature of the material with the addition of nanocrystals (González et al.
 550 2015).



560 **Fig. 6.** Tensile behaviour of corn starch films without (S) and with 1wt% cellulose
 561 fillers (S-CF: with coffee husk fibres, S-RF: with rice husk fibres) and CNCs (S-CC:
 562 with CNCs from coffee husk, S-RC: with CNCs from rice husk).

563

564 4. Conclusions

565 The cellulose contents of rice and coffee husks were similar in the range of 34-35 %,
 566 which make them a good source of cellulosic material for different industrial uses.

567 Purification of cellulose, using the classical alkali and bleaching treatments was

568 effective at removing hemicellulose and lignin, providing white fibres with about 50%
569 crystallinity degree and 60-500 µm length, with good thermal stability (thermo-
570 degradation T peak: 315-327°C). Coffee fibres were flatter and more helically folded
571 than rice fibres. Acid hydrolysis of both fibres gave rise to CNCs, with small
572 morphological differences. CNCs from coffee husk were slightly thinner (20 against 39
573 nm, in average diameter), but with similar aspect ratio (higher than 10). The CNCs
574 exhibited slightly lower thermal stability than the cellulose fibres (thermo-degradation T
575 peak: 249-362°C). The properties of cellulosic fractions from rice and coffee husk make
576 them very adequate as reinforcing materials in biopolymer composites, especially nano-
577 sized reinforcement (CNCs from rice and coffee husks) which increased the elastic
578 modulus of TPS films by 186 % and 121 %, respectively. So, coffee husk represents an
579 interesting source of cellulosic reinforcing material, whose use in different applications,
580 such as packaging materials, could boost the value of this waste.

581

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