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Additional Information

Release kinetics and antimicrobial properties of carvacrol encapsulated in electrospun poly-(ϵ -caprolactone) nanofibres. Application in starch multilayer films.

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4 Abstract

5 Electrospun poly-(ϵ -caprolactone) (PCL) fibre mats encapsulating Carvacrol (CA) were
6 obtained with good encapsulation efficiency (85%) and CA load (11 % in the fibre). These mats
7 were effective at controlling the growth of *Escherichia coli*, when the surface density of CA
8 loaded fibres was 1.2 or 1.8 mg/cm², in line with the CA released into the culture medium that
9 exceeded the MIC of the bacteria. However, they were not effective at controlling the growth of
10 *Listeria innocua*, since a greater release of CA was necessary to achieve the MIC of this
11 bacterium. It was not only the CA load in the fibres, but also its release capacity in the media
12 that determined the antimicrobial effect. The fibre showed higher release rate and ratio in less
13 polar simulants, D1 (50% ethanol) and D2 (isooctane) (representing fatty foodstuff), where
14 practically the total amount of CA was released; whereas in more polar systems (simulants A
15 (10% ethanol) and B (3% acetic acid)) a more limited CA delivery (60-75 %) occurred, at a
16 slower rate. The antimicrobial action of the active PCL mats was reproduced in multilayer
17 starch films containing the CA-loaded electrospun PCL fibres between two starch sheets, with a
18 slightly delayed response. In the multilayer films, a great reduction in the water vapour
19 permeability was also observed with respect to that of starch films, without relevant changes in
20 other functional properties of the films for packaging purposes.

21

22 **Keywords:** PCL, carvacrol, release kinetics, antimicrobial action, starch multilayer films

23 1. Introduction

24 The electrohydrodynamic process known as electrospinning is an efficient and straightforward
25 method with a simple working principle (Bhardwaj & Kundu, 2010) that allows micro- and
26 nanoscale polymer structures to be obtained. It can generate continuous (fibres) or discrete
27 (particles) polymer delivery systems able to encapsulate compounds of specific interest for
28 applications in many fields, such as that of medicine (Hamori et al., 2014; Sill & von Recum,
29 2008), optoelectronics (Hernández-Martínez, Nicho, Hu, León-Silva & Arenas-Arrocena, 2017;
30 Xue et al, 2017), sensor technology (Mercante, Scagion, Migliorini, Mattoso & Correa, 2017;

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31 Macagnano & De Cesare, 2017) or food packaging (Fabra, López-Rubio & Lagaron, 2016).
32 Incorporating active natural compounds into biodegradable food packaging materials is an
33 innovative trend that focuses on the enhancement of the quality and shelf-life lengthening of the
34 packaged foods (Majid, Nayik, Dar & Nanda, 2016; Padgett & Han, 1998), while these
35 biodegradable materials reduce the environmental impact of the packaging waste. The
36 electrospinning technique could be a feasible way to encapsulate the active compound, while
37 providing a controlled delivery system (electrospun layer) when applied to such packaging
38 materials. Electrospun layers provide structures with a large specific surface area for the
39 compound diffusion, while the losses of active compounds during the process are minimized
40 due to the use of room temperatures.

41 Of the active agents that are currently used in the development of active food packaging, plant
42 essential oils (and their constituents) are a prominent group of interest. Plant-derived phenolic
43 terpenoids have been successfully incorporated in different polymers with food packaging
44 potential, as reported by Requena, Vargas & Chiralt, (2017) for eugenol encapsulated in
45 poly(hydroxybutyrate-co-hydroxyvalerate), Fernandez-Pan, Maté, Gardrat & Coma, (2015) for
46 carvacrol in chitosan, Rieger & Schiffman, (2014) for cinnamaldehyde in
47 chitosan/poly(ethylene-oxide) or Ramos, Beltrán, Peltzer, Valente & Garrigós, (2014) for
48 carvacrol and thymol in polypropylene. Carvacrol is one of the most widely used active
49 compounds (Higuera, López- Carballo, Hernández-Muñoz, Catalá & Gavara, 2014) recognized
50 as a food additive (Joint FAO/WHO, 2001) and as a flavouring substance (EFSA, 2012). It is a
51 phenolic terpenoid present at great concentration in several essential oils, such as oregano or
52 thyme oil (Burt, 2004), which exhibits high antimicrobial activity against both Gram positive
53 and Gram negative bacteria (Ben Arfa, Combes, Preziosi-Belloy, Gontard & Chalier, 2006;
54 Ultee, Gorris & Smid, 1998), as well as different fungi (Tunc, Chollet, Chalier, Preziosi-Belloy
55 & Gontard, 2007). Its incorporation in food-grade polymers would offer alternative active
56 packaging materials, replacing some of the synthetic antimicrobials currently in use such as
57 weak organic acids or salts (Fu, Sarkar, Bhunia & Yao, 2016). Its controlled delivery, until the
58 minimal inhibitory concentration of the microorganisms is reached at the target point, is
59 required to ensure its antimicrobial activity. This controlled release should also prevent any
60 overdose in the food system in order to limit the dilution effects by diffusion and the food
61 sensory impact.

62 A previous study (Tampau, González-Martínez & Chiralt, 2017) reported a good encapsulation
63 efficiency (EE) of carvacrol in electrospun poly- ϵ -caprolactone (PCL) mats, using 15 % glacial
64 acetic acid solution of PCL containing 0.15 g carvacrol / g polymer. These mats exhibited a
65 fibrous structure, which could adequately coat biodegradable packaging films to obtain active
66 materials for food applications. 85% of the carvacrol content of the solution could be
67 encapsulated in the PCL fibres, which is highly efficient when taking into account the volatile
68 nature of the active and the high losses incurred by other techniques. Despite the high EE, the

69 antimicrobial effect of the encapsulated compound will be affected by its initial load in the
70 active material and its active release capacity into the applied medium. All of this defines the
71 active effective concentration on the target point, which must exceed the minimal inhibitory
72 concentration of the contaminating bacteria.

73 Likewise, electrospun PCL fibres carrying carvacrol could be applied to obtain multilayer starch
74 films including carvacrol loaded fibres between the starch sheets, thus contributing to the
75 improvement of the functional properties of the packaging material. The multilayer assembly of
76 biopolymers with complementary properties allows packaging material to be obtained that
77 better meets the specific requirements of different kinds of foods. In fact, more and more of the
78 food available in the stores comes in high-tech plastic packaging multilayer films, ensuring the
79 food is preserved for longer than when using a monolayer structure. Starch is a good candidate
80 since it is widely available and cheap, while its films are extensible with very good oxygen
81 barrier properties (López, Zaritzky, Grossmann, & García, 2013). However, starch films exhibit
82 poor water vapour barrier capacity, being water sensitive (Pushpadass et al., 2009). In this
83 sense, PCL exhibits good barrier capacity to water vapour (Ortega-Toro, Morey, Talens &
84 Chiralt, 2015) and its inclusion in the starch multilayer assembly has been demonstrated to
85 enhance the film functionality (Ortega-Toro et al., 2015). Likewise, the incorporation of
86 carvacrol into the PCL electrospun layer can confer antimicrobial properties on the
87 starch/PCL/starch multilayers.

88 The aim of the study was to develop active electrospun layers of poly- ϵ -caprolactone with
89 encapsulated carvacrol with a high enough load of the active to be applied on food packaging
90 films, by analysing the release kinetics of carvacrol in different food simulants (solvents with
91 different polarity) and verifying their antibacterial activity. Likewise, the improvement in the
92 functional properties of the starch multilayer films, containing electrospun PCL fibres between
93 the starch sheets, has been analysed in terms of the barrier and tensile properties and
94 antimicrobial activity.

95 **2. Materials and methods**

96 **2.1. Materials and reagents**

97 Poly-(ϵ -caprolactone) (PCL) pellets (average Mn 80,000) and carvacrol (CA) were acquired
98 from Sigma-Aldrich (Sigma-Aldrich Chemie, Steinheim, Germany). All UV-grade solvents
99 used (ethanol, glacial acetic acid and isooctane) were from Panreac AppliChem (Panreac
100 Química S.L.U, Barcelona, Spain).

101 **2.2. Obtaining and characterizing the electrospun fibre of CA-loaded PCL**

102 On the basis of previous studies (Tampau et al., 2017), nanofibres were obtained from PCL in
103 glacial acetic acid (GAA) with and without CA. Briefly, PCL (15 wt %) and CA (15 wt % with
104 respect to the polymer) were dissolved in GAA under stirring for 24 h at room temperature. The
105 solutions were electrospun using Fluidnatek equipment (BioInicia S.L., Valencia, Spain), at a

106 flow rate of 1.2 mL/h through the syringe needle (internal diameter=0.6 mm), by applying an
107 electric field of 12.0 kV. The electrospun fibre material was collected on aluminium foil disks
108 placed on the collector at 20 cm from the tip of the needle. In these conditions, an encapsulation
109 efficiency (EE) of CA in the fibres of about 85 % was expected, which would suppose 12 g
110 CA/100 g fibres in the obtained mat.

111 Nanofibres were obtained for different electrospinning (ES) times on aluminium foil to
112 determine the process yield (g fibre/cm²) as a function of the process time. Carvacrol content in
113 the fibre (mg/g fibre) was analysed at the longest process time (90 min) and the expected EE
114 from the previous study (Tampau et al. 2017) was verified. The CA surface density (mg
115 CA/cm²) vs the ES time was estimated from this analysis, in order to determine the time
116 required to reach enough CA load in the mat. This is for the purposes of ensuring that the
117 minimum inhibitory concentration (MIC) of CA for different bacteria can be reached when the
118 active mat is applied on a target product surface. Likewise, CA content was also determined in
119 different zones of the circular electrospun surface, considering radial distance and angle with
120 respect to the centre (6 zones), to analyse the CA distribution homogeneity on the mat surface
121 electrospun for 90 min.

122 CA quantification was carried out by extraction of fibre (with a determined surface and weight)
123 with UV grade absolute ethanol and spectrophotometric determination at 275 nm using a
124 UV/Vis spectrophotometer (Evolution 201 UV-Vis, Thermo Fisher Scientific Inc.), as
125 previously described by Tampau et al. (2017). The obtained results were expressed as µg
126 CA/cm² or µg CA/g fibre.

127 Structural characterization of the fibres was performed using Field Emission Scanning Electron
128 Microscopy (FESEM Ultra 55, Zeiss, Germany). Samples were mounted on support stubs and,
129 after platinum coating, were observed using an accelerating voltage of 2 kV. ImageJ software
130 (National Institutes of Health, U.S.A.) was employed for the image analysis, in order to assess
131 the morphological differences between the two types of matrices with or without the active.

132 **2.3. Release kinetics of CA in different food simulants**

133 The kinetic study of CA release from the electrospun PCL fibres was carried out in four types of
134 solvents, acting as food simulants of different polarities or pH (Regulation 10/2011/EC). As
135 described by Requena et al. (2017), ethanol 10% (v/v) (simulant A) and acetic acid 3% (w/v)
136 (simulant B) were chosen to emulate aqueous foodstuffs neutral and acidic (pH<4.5) in
137 character, respectively. Ethanol 50% (v/v) (simulant D1) was used to imitate foods with alcohol
138 content higher than 20% or oil-in-water emulsions, whereas isooctane (simulant D2) was used
139 for foods with a highly lipophilic surface. For this purpose, fibre samples of 50 mg were placed
140 in glass bottles containing 100 mL of the corresponding simulant, and kept under stirring at 22 ±
141 2 °C. Aliquots of the samples were extracted at different times of contact (ranging from 1 min
142 upward, until equilibrium is reached) and absorbance was determined spectrophotometrically at

143 275 nm. CA quantification in the liquid phase was established using the previously obtained
 144 calibration curves in each simulant. All analyses were carried out in triplicate, using the
 145 respective simulant, in contact with CA-free fibres for the equivalent time, as blank.

146 2.3.1 CA release mathematical modelling

147 Three models were considered to describe the behaviour of the CA loaded PCL fibres in the
 148 four food simulants. Considering that the electrospun fibres fuse together at different points
 149 throughout their length to form a mat-like matrix, the CA release from the mat can be analysed
 150 as a one-dimensional mass transfer process from an infinite plane sheet (Crank, 1975) of half-
 151 thickness e , with a characteristic diffusion coefficient D (m^2/s). The diffusion coefficients for
 152 each simulant were determined by modelling the obtained data to Fick's second law for long-
 153 time diffusion (eq.1), considering that i) the distribution of the active is homogenous within the
 154 matrix, ii) the initial concentration of CA in each simulant is zero, and iii) there is no
 155 degradation of CA during the migration process.

156

$$157 \frac{M_t}{M_\infty} = \frac{8}{\pi^2} \sum_{n=0}^{\infty} \left[\frac{1}{(2n+1)^2} \exp\left\{-\frac{D(2n+1)^2\pi^2 t}{4e^2}\right\} \right] \text{ (eq. 1)}$$

158

159 where:

160 t = release process time (s),

161 M_t = mass of CA released at time t ,

162 M_∞ = mass of CA released at equilibrium,

163 e : half thickness of the fibre layer (m).

164 The boundary conditions taken into account for eq.1 are as follows:

$$165 \begin{array}{ccc} t = 0 & -e < x < e & c = c_0 \end{array}$$

$$166 \begin{array}{ccc} t > 0 & x = \pm e & c = 0 \end{array}$$

167 The mathematical solutions from Fick's equation with eleven terms ($n=0\div10$) were optimized
 168 using the Solver tool (Microsoft Excel 2016®) by minimizing the Sum of Squared Errors
 169 (SSE).

170 Peleg's model (eq. 2) (Peleg, 1988) was also applied to the data, in order to obtain the
 171 equilibrium value for the carvacrol release and the release rate for each simulant.

$$172 \frac{t}{M_t} = k_1 + k_2 t \text{ (eq. 2)}$$

173 where:

174 t = release process time,

175 M_t = mass of CA released at time t,

176 $1/k_1$ = release rate,

177 $1/k_2$ = mass of CA released at equilibrium (M_∞).

178 Lastly, the Korsmeyer-Peppas model (Siepmann & Peppas, 2011) (eq. 3) was considered in
179 order to analyse the mechanisms controlling the CA release, with the caveat that it can only be
180 applied for ($\frac{M_t}{M_\infty} \leq 0.6$), which means a progress of the mass transfer process of 60% (Hines &
181 Kaplan, 2011).

182

183
$$\frac{M_t}{M_\infty} = kt^n \text{ (eq. 3)}$$

184 where:

185 M_t/M_∞ = mass of CA released at time t with respect to the mass released at equilibrium;

186 t = release process time (h);

187 k = rate constant (h^{-n});

188 n = diffusional exponent which takes values in the 0÷1 interval (dimensionless).

189 **2.3.2. Prediction of CA antimicrobial effect based on the release study**

190 Peleg's parameters determined in the modelling step were employed to predict the amount of
191 CA released throughout time in the target product, which must reach the minimum inhibitory
192 concentration (MIC) of the target microorganism to exert the antimicrobial effect. To this end, a
193 contact surface equivalent to a Petri dish of 5.5 cm in diameter with a volume of 10 mL of agar
194 medium (pH=5.6±0.2) was considered. For this simulation, two ES deposition times of 60 and
195 90 minutes (with the subsequent total amount of CA in the mat) and the release profile of CA
196 from PCL mats in A simulant were considered. Simulant A was the most similar in water
197 content to the culture medium.

198 **2.4. Incorporation of PCL fibres in multilayer starch films**

199 PCL fibres were electrodeposited on thermoplastic starch films (S) in order to obtain multilayer
200 films (S-PCL-S). To this end, corn starch films were obtained by melt blending of starch-
201 glycerol-water (mass ratios: 100:30:50) at 160°C and 8 rpm for 30 min in a two-roll mill (Model
202 LRM-M-100, Labtech Engineering, Thailand). The obtained pellets were conditioned at 53%
203 relative humidity for 1 week and compression moulded in a press (Model LP20, Labtech
204 Engineering, Thailand), initially at 50 bars/160 °C for 2 min, then at 130 bars/160 °C for 6 min,
205 and cooled down to 50 °C for 3 min (Ortega-Toro et al., 2015). Multilayer films were prepared
206 by electrodeposition of the 15 % PCL solution in GAA with or without CA (15 g/100 g PCL).
207 Electrodeposition times were 60 and 90 min to obtain the same CA surface concentration as in
208 previously described fibre mats. Multilayer films with the ES PCL in the middle were obtained

209 by thermocompression of two S layers (one of these containing the ES PCL fibres of 60 or 90
210 min) at 130 bars and 80 °C for 4 min, and cooled down to 50 °C for 2 min.

211 Multilayer films conditioned for 1 week at 53 % RH were characterized as to their thickness
212 (measured at 5 different points by a Palmer digital micrometre from Comecta, Barcelona,
213 Spain), water content (assessed gravimetrically), water vapour permeability (determined at 25
214 °C using a modified (Gennadios, Weller & Gooding, 1994) ASTM E96-95 gravimetric method
215 (ASTM, 1995), as described by Perdonés, Chiralt & Vargas, (2016)), oxygen permeability
216 (Standard Method D3985-95 (ASTM, 2002)) and tensile properties (ASTM standard method
217 D882 (ASTM, 2001)), as previously described by Ortega-Toro et al. (2015).

218 Antimicrobial *in vitro* tests were also carried out for multilayer films using the below described
219 method (section 2.6) for fibre mats.

220 **2.5. Thermal analysis**

221 Thermal analyses were carried out in PCL fibres and multilayer assemblies in order to analyse
222 the carvacrol effect on phase transitions and thermal stability of PCL or starch. Differential
223 scanning calorimetry analyses were performed, using a DSC (1 StareSystem, Mettler-Toledo,
224 Inc., Switzerland). Samples (5-15 mg), previously conditioned in P₂O₅, were placed into
225 aluminium pans (Seiko Instruments, P/N SSC000C008) and sealed. Samples were heated from
226 25°C to 120 °C at 10 K/min. Then, they were kept at 120 °C for 5 min, cooled to -60 °C at -10
227 K/min, kept at -60 °C for 5 min and heated again to 200 °C at 10 K/min. An empty aluminium
228 pan was used as reference. Each sample was analysed in triplicate.

229 A thermo-gravimetric analyser (TGA/SDTA 851e, Mettler Toledo, Schwarzenbach,
230 Switzerland) was used to characterize the sample thermal degradation. The analysis was
231 performed from room temperature to 500 °C at 20 °C/min under a nitrogen flow (50 mL/min).
232 DTA and DGTA curves were analysed and the temperature at which the maximum degradation
233 rate occurs was determined (T_{max}). Each sample was analysed in duplicate.

234 **2.6. Antimicrobial properties**

235 Gram (+) *Escherichia coli* (CECT 101) and Gram (-) *Listeria innocua* (CECT 910) obtained
236 from Spanish Type Culture Collection (CECT, Burjassot, Spain) were used to test the
237 antimicrobial efficiency of the electrospun PCL fibre mats. The model bacterial strains, stored
238 in protective conditions (glycerol 30%) at -25 °C, were regenerated as described by Valencia-
239 Sullca et al. (2016), by incubating them at 37 °C for 24 h in tryptic soy broth (TSB) (Scharlab,
240 S.L., Barcelona, Spain) and harvested in their exponential growth phase. The revived cultures
241 were properly diluted in TSB to obtain a target inoculum of 10⁶ colony forming units (CFU)/ml
242 for *E. coli* and 10⁵ CFU/ml for *L. innocua*.

243 Circular samples (55 mm in diameter), consisting of starch multilayers or PCL fibre mats (with
244 and without CA) electrospun over thermoplastic starch film acting as support, were placed (with

245 the electrospun side face down) on inoculated plates (1 ml inoculum on the plate surface)
246 containing 10 mL of tryptic soy agar (TSA) (Scharlab, S.L., Barcelona, Spain). CA-free PCL
247 samples and uncoated inoculated TSA plates were used as controls. Immediately after the
248 inoculation and after 2, 6, 9 and 14 days of storage at 10 °C, microbial counts were performed.
249 To this end, the Petri dish content was placed aseptically in a sterile stomacher strainer bag
250 (Seward Limited, West Sussex, UK) along with 90 mL of buffered peptone water (BPW)
251 (Scharlab, S.L., Barcelona, Spain) and homogenized for 2 min by means of a Masticator Paddle
252 Blender (IUL S.A., Barcelona, Spain). Serial dilutions of this liquid were plated and covered
253 with violet red bile agar (VRBA) (Scharlab, S.L., Barcelona, Spain) for *E. coli* and palcam agar
254 base (PAB) (Scharlab, S.L., Barcelona, Spain) enriched with palcam selective supplement for
255 *Listeria* (Scharlab, S.L., Barcelona, Spain). After incubation at 37 °C for 48 h, the colonies were
256 counted. All determinations were performed in duplicate.

257 **2.7. Statistical analysis**

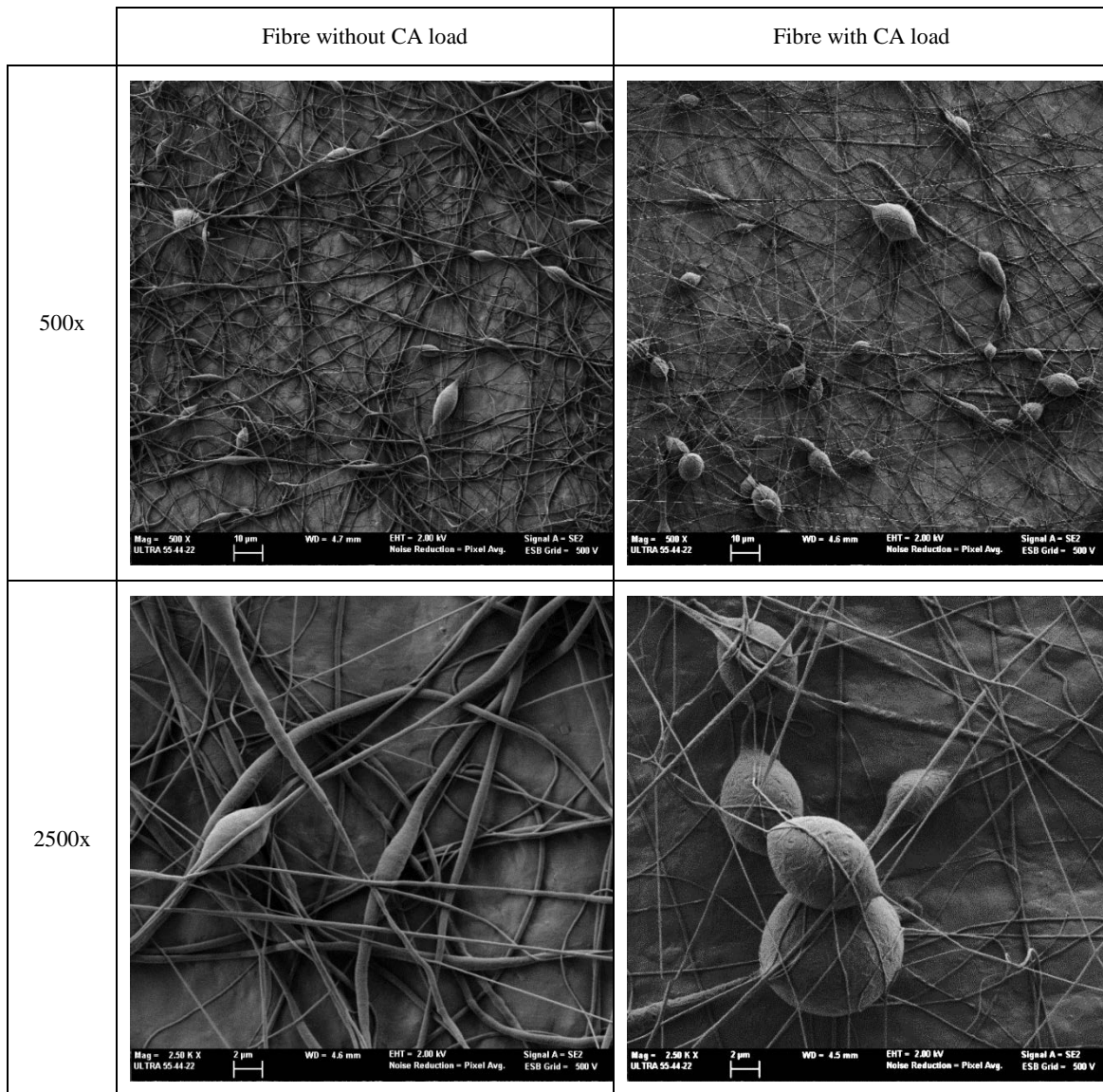
258 The experimental data was processed using one-way analysis of variance (ANOVA) with the
259 statistics program Statgraphics centurion XVI.I (StatPoint Technologies Inc. Warrenton, VA,
260 USA).

261 **3.Results and discussion**

262 **3.1. Electrospun fibre layers of CA loaded PCL**

263 The structure of the obtained electrospun matrices can be observed in Figure 1. The mat has a
264 fibrous appearance, with occasional beads of spindle-like geometry which appear more
265 frequently in the case of the CA-loaded fibres. The average diameter of active-loaded fibres was
266 200 ± 40 nm, whereas in the case of the CA-free matrix this reached higher values of 500 ± 350
267 nm. CA affected the morphology of the electrospun fibres, giving rise to thinner fibrous forms,
268 but with more beads, such as previously observed (Tampau et al., 2017). These changes must be
269 attributed to the differences in the liquid phase properties induced by CA addition and the CA
270 interactions with the polymer chains that affected the chain entanglements, responsible for fibre
271 formation (Bahrami & Gholipour Kanani, 2011).

272

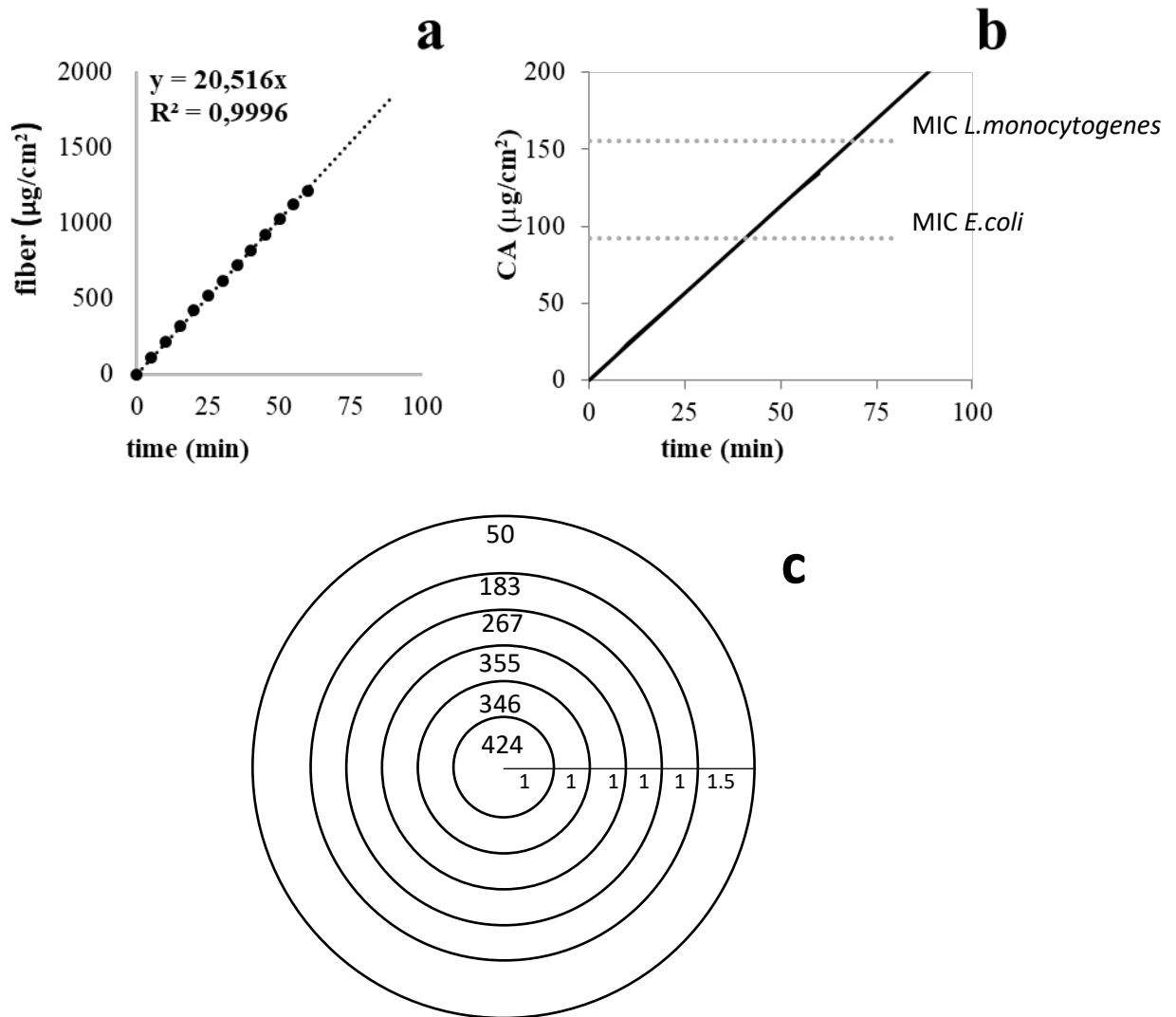


274 **Figure 1.** FESEM micrographs of the electrospun fibres for 10 minutes.

275

276 The mass fibre accumulation on the collector surface as a function of the ES time (Figure 2a)
 277 reveals the expected linear trend where the slope is coherent with the liquid flow applied in the
 278 equipment, the concentration of the compounds (PCL and CA) and the EE of CA. The
 279 quantified amount of CA in the fibre obtained for 90 min was 11.1 ± 0.1 g CA/100g fibre. This
 280 concentration represents an encapsulation efficiency of CA in the fibres of $83 \pm 1\%$ (expressed as
 281 the ratio between the final amount in the fibre and the initial content in the solution), which
 282 agrees with the values previously reported by Tampau et al. (2017). From the mean CA content
 283 in the fibre, the CA surface density in the ES layer as a function of time was predicted (Figure
 284 2b). Figure 2b also shows the values of the minimal inhibitory concentration (MIC) reported by
 285 some authors for *Escherichia coli* and *Listeria monocytogenes*, expressed as $\mu\text{g}/\text{cm}^2$. To obtain
 286 these values, the reported MIC values of CA against *E. coli* ($2.2 \cdot 10^{-4}$ g/ml) and *L.*
 287 *monocytogenes* ($3.7 \cdot 10^{-4}$ g/ml) (Burt, 2004), and the *in vitro* test conditions for the antimicrobial

288 activity of the fibres used in the present study (section 2.4), were considered. Then, the MIC
 289 ($\mu\text{g/ml}$) was assumed to be reached in 10 ml of medium with a contact surface of 23.76 cm^2 (in
 290 a plate of 5.5 cm diameter) with the fibre mat. Figure 2b is a useful tool with which to estimate
 291 the ES time required to obtain active electrospun layers of PLC-CA fibres with enough CA to
 292 exceed the MIC of different microorganisms.



293 **Figure 2.** **a)** Mass of ES fibre layer as a function of time; **b)** Mean surface density of CA
 294 ($\mu\text{g/cm}^2$) as a function of ES time and **c)** surface CA distribution ($\mu\text{g/cm}^2$ in different zones of
 295 ES layer) for 90 min of ES (radius increments shown in cm). MICs in $\mu\text{g/cm}^2$ are indicated as a
 296 dashed line, assuming a CA diffusion in 10 ml medium through 23.76 cm^2 of sample surface in
 297 contact with the fibres (equivalent to 0.42 cm of characteristic dimension for diffusion).

298

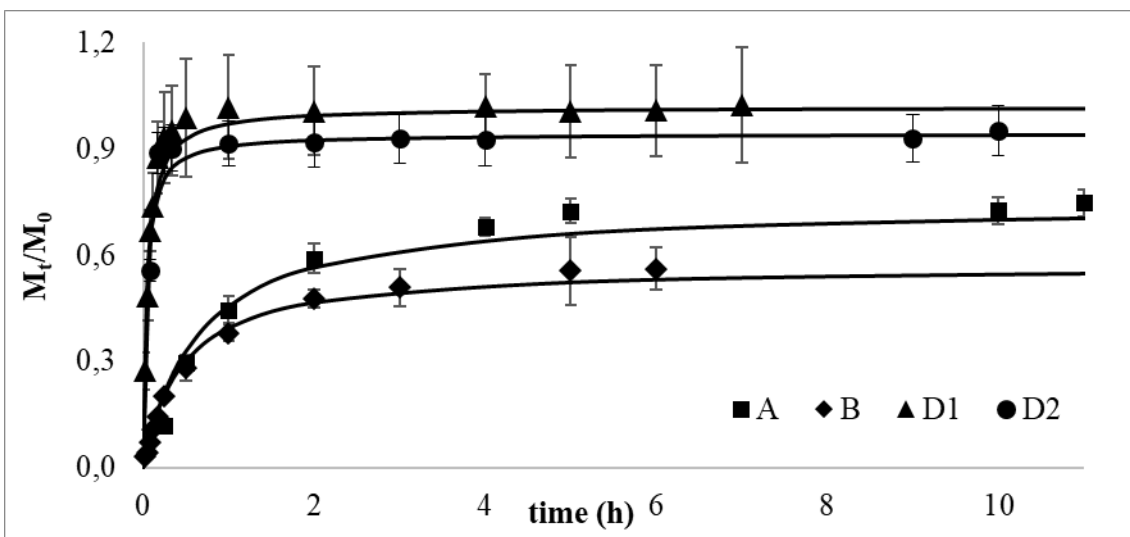
299 An ES time of 37 min and 62 min, respectively, would be required to obtain a CA mass in the
 300 fibres equivalent to the respective MICs of each bacterium. In order to ensure this quantity of
 301 active is reached, 60 and 90 min of ES time were applied to obtain potentially active layers,
 302 assuming that the complete release of the active in the culture media could not occur. By
 303 analysing the active's surface distribution, additional data was obtained to support this decision;
 304 this revealed that the ES process with the used equipment provided an uneven coating, creating

305 a concentric concentration gradient (Figure 2c), where the external area of the obtained
306 electrospun disc contained a smaller amount of fibres and so of CA. In this sense, four tangent
307 sample disks obtained from each of the four quadrants of the main circle were considered for
308 each analysis.

309 3.2. Release kinetics of CA from PCL fibres in different food simulants

310 The release kinetics of CA from the fibre mat obtained for 90 min was analysed in order to find
311 out how the active compound can be delivered in different food systems. To this end, four liquid
312 simulants of different polarity have been considered, as commented in section 2.3. The release
313 mass of CA was determined at different contact times with simulants and Figures 3 and 4 show
314 the mean values, for each food simulant.

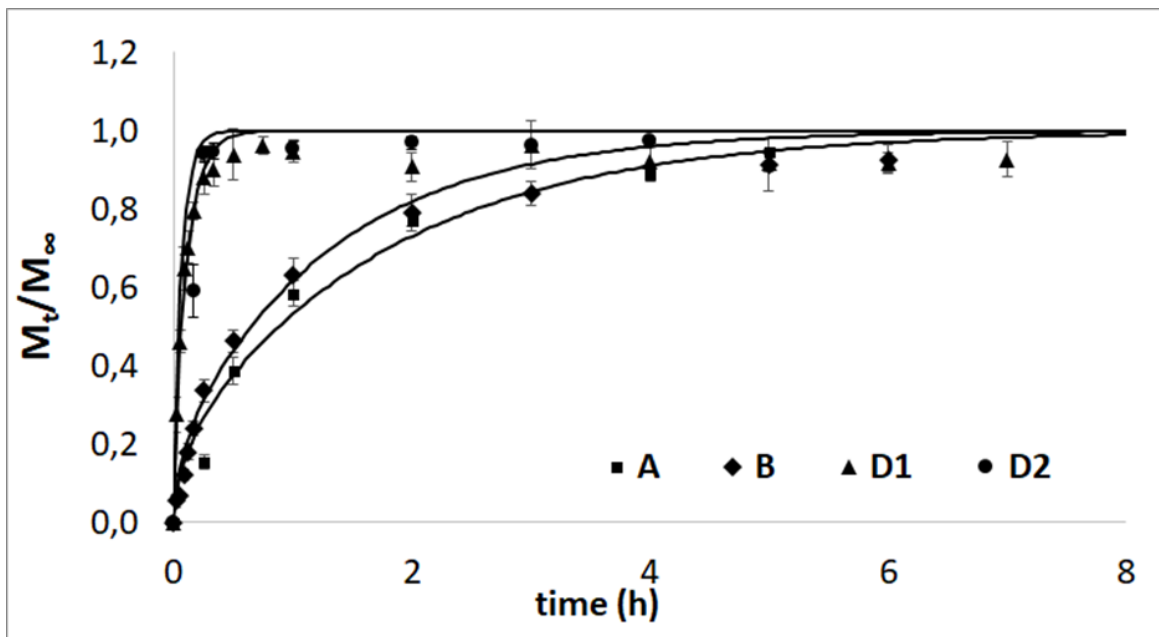
315



316

317 **Figure 3.** M_t/M_0 ratio (fraction of the active released from the fibres at each time, referred to the
318 initial content in the fibres) as a function of time for the different solvents: experimental data
319 (points) and Peleg's fitted model (continuous lines).

320



321

322 **Figure 4.** M_t/M_∞ ratio (fraction of the active released from the fibres at each time, referred to
 323 the maximum value released at equilibrium) as a function of time for the different simulants:
 324 experimental data (points) and Fick's fitted model (continuous lines).

325

326 **Table 1.** Parameters of the Peleg (M_∞ - CA mass released at equilibrium; $1/k_1$ - CA release rate)
 327 and Korsmeyer-Peppas (n - diffusional exponent; k - rate constant) models, and Fick's diffusion
 328 coefficient in the different food simulants. The M_∞/M_0 ratio represents the CA mass released in
 329 the simulant liquid at equilibrium, with respect to the total amount in the fibres determined by
 330 ethanol extraction. Different superscript letters in the same column indicate significant
 331 differences ($p < 0.05$) between simulants.

Simulant	Adjusted model								
	Peleg				Fick		Korsmeyer-Peppas		
	$1/k_1$ ($\mu\text{g CA/g film}$)	$M_\infty=1/k_2$ ($\text{g CA}/100 \text{ g film}$)*	M_∞/M_0 (%)	R^2	$D \times 10^{13}$ (m^2/s)	SSE	n	K (h^{-n})	R^2
A	$34,1 \pm 0,4^a$	$7,8 \pm 0,3^b$	75 ± 3^b	0,9993	$3,9 \pm 0,3^a$	0,03	$0,9 \pm 0,2^c$	$0,63 \pm 0,05^a$	0,9468
B	38 ± 2^a	$6,1 \pm 0,6^a$	57 ± 5^a	0,9998	$5,3 \pm 1^a$	0,12	$0,68 \pm 0,04^b$	$0,67 \pm 0,01^a$	0,9502
D1	700 ± 150^b	11 ± 1^c	100 ± 10^c	0,9981	46 ± 7^b	0,04	$0,53 \pm 0,07^{ab}$	$2,3 \pm 0,3^c$	0,9979
D2	800 ± 100^b	$11,5 \pm 0,8^c$	96 ± 7^c	0,9998	109 ± 14^c	0,05	$0,46 \pm 0,11^a$	$1,8 \pm 0,4^b$	0,8747

332 *corresponding to 100 mL volume of simulant

333 Table 1 shows the maximum values released (M_∞), referred to 100 g of the initial ES layer, and
 334 estimated by applying Peleg's model to the experimental data. Table 1 also includes the values
 335 of $1/k_1$ Peleg's parameter, related to the release rate. A good fit of the model was achieved in
 336 every case ($R^2 > 0.998$), as can be seen in Figure 3. Both the release rate and asymptotic value
 337 were greatly affected by the polarity of the food simulants. The fastest release of CA was
 338 observed in 50% ethanol (D1 simulant) and isoctane (D2 simulant), whereas the slowest
 339 delivery occurred in the more polar solvents (A: 10% ethanol and B: 3% acetic acid). Those less
 340 polar solvents also allow for the maximum delivery of CA from the PCL mats (near 100% of
 341 their content), whereas only 75 and 57 % of the content was released at equilibrium in polar

342 solvents (simulants A and B, respectively). This behaviour agrees with the differences in
343 polarity of the solvent and the corresponding chemical affinity between the polymer matrix, the
344 active compound and the solvent, as previously reported by Tehrani (2007). In fact, three steps
345 can be described for the active release from the polymeric matrix: 1) solvent diffusion into the
346 polymer network, 2) network relaxation due to solvent plasticization effects, and 3) the
347 diffusion of the compound through the relaxed polymer network, until the thermodynamic
348 equilibrium between the polymer and food system phases is reached (Requena et al., 2017).
349 These steps can be coupled, especially 2 and 3, depending on the characteristic relaxation time
350 of polymers and the diffusion times of the diffusing compound, giving rise to anomalous
351 transport behaviour (Siepmann & Peppas, 2011). At equilibrium, the relative compound affinity
352 with the solvated polymer and its solubility in the liquid food system determine its partition
353 coefficient, defined as the ratio between the mass of active released at equilibrium in the
354 simulant (M_{∞}) and its corresponding residual mass in the film ($M_0 - M_{\infty}$). The solubility of CA
355 increases in less polar solvents with respect to the more aqueous solvents (CA water solubility is
356 1250 mg/L, Yalkowsky, He & Jain, 2010), while the PCL matrix could be more relaxed in
357 contact with non-polar solvents than with the more aqueous systems, considering the non-polar
358 nature of the polymer. Both the increase in the active solubility and polymer relaxation in less
359 polar food simulants led to a faster CA release and a higher amount of CA delivered at
360 equilibrium.

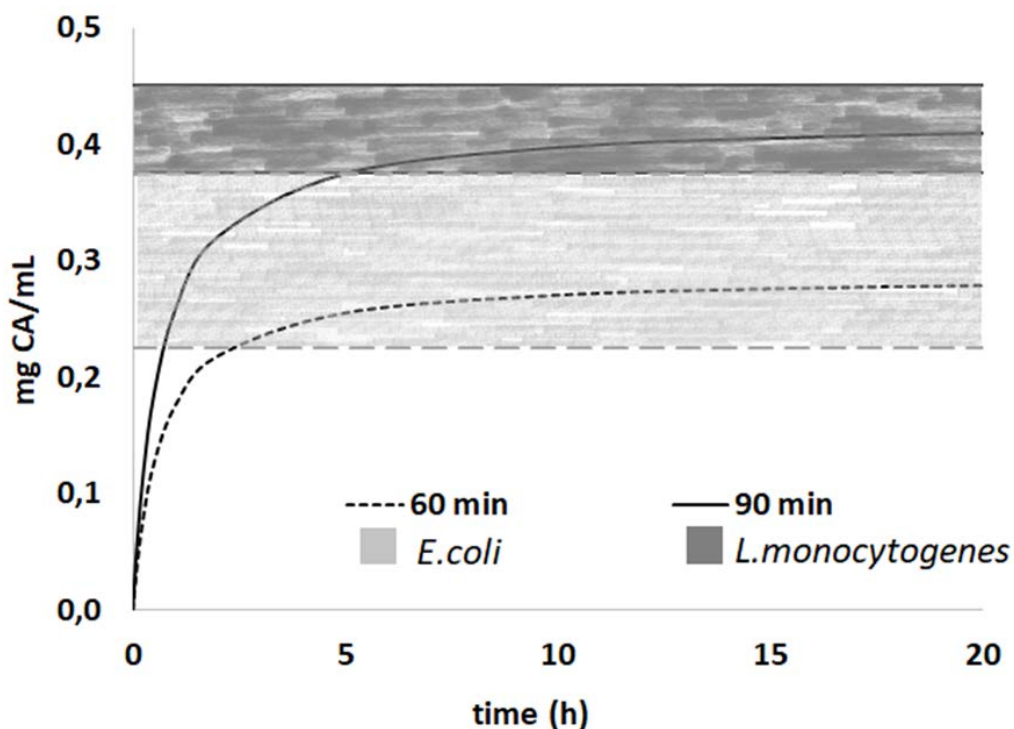
361 Table 1 also shows the parameters of the Korsmeyer and Peppas model. The values for the “n”
362 exponent reveal that a Fickian diffusion ($n \approx 0.5$) takes place when the matrix is in contact with
363 the less polar simulants, whereas an anomalous transport ($0.5 < n < 1.0$) can be assumed in the
364 aqueous media (Siepmann & Peppas, 2011), which reveals the coupling of the polymer
365 relaxation in contact with the solvent and the diffusion of CA in the matrix. This agrees with the
366 low solvent-polymer affinity, which will imply a poor capacity of the solvent to penetrate the
367 polymer network. Nevertheless, an apparent diffusion value (D) was estimated in every case
368 (Table 1) considering the electrospun layer thickness (0.16 ± 0.08 mm) and assuming a
369 homogenous medium, despite the nano-porous structure generated in the ES process (Figure 1),
370 where the solvent could also penetrate by means of capillarity. Figure 4 shows the curves fitted
371 considering the Fickian equation (eq. 1), where the good fit of the model can be appreciated,
372 ($SSE < 0.12$). The obtained D values also reflect the effect of the solvent polarity, showing the
373 highest values for simulant D (isooctane) and the lowest value for 10% ethanol in water
374 (simulant A). Other authors also report similar behaviour for the CA release from non-polar
375 polymer matrices, such as polypropylene (Ramos et al., 2014) and poly (butylene succinate)
376 (Petchwattana & Naknaen, 2015). Kuorwel, Cran, Sonneveld, Miltz & Bigger (2013) obtained
377 lower diffusion values (about 15 times lower) for the CA diffusion from starch films in
378 isooctane, which can be explained by the lower chemical affinity of starch matrix and isooctane,
379 which affects the polymer relaxation and plasticization level, and so, the compound diffusion.

380 From the obtained kinetic behaviour, no total CA release would be expected from PCL mats in
381 the more polar food systems regardless of their pH, whereas it would be fully delivered in more
382 fatty foods, due to the greater CA solubility and polymer relaxation during the food contact.
383 Likewise, less time will be required for the maximum release in fatty foods (emulated by D1
384 and D2 simulants), than in more aqueous products (emulated by A and B simulants), where
385 longer times would be needed.

386

387 3.3. Antibacterial properties of the CA loaded fibres

388 Based on the release profile, the amount of CA released in the incubation plates (55 mm in
389 diameter, with 10 mL of agar medium), used in the *in vitro* antimicrobial test, was predicted,
390 considering the behaviour of the medium to be similar to that of the simulant A (10 % ethanol in
391 water). Figure 5 shows the predicted concentration of active reached over time for two CA-
392 loaded PCL mats of different thicknesses (60 and 90 min of electrodeposition), and so, with
393 different total amounts of CA in the mat. Likewise, the previously reported range for the MIC
394 values of the tested foodborne bacteria, namely *E. coli* (0.225-5 mg·mL⁻¹) and *L. monocytogenes*
395 (0.375-5 mg·mL⁻¹) (Burt, 2004) were reflected in the plot (shadow areas). The predicted
396 equilibrium values for CA release for 60 and 90 min processed PCL mats greatly differed. In
397 the case of the 60-min process, the MIC value for *E. coli* was barely reached, while the MIC
398 value for *L. innocua* was higher than the amount of CA released. For the 90-min process, both
399 MIC values were reached, but for *L. innocua*, the amount released at equilibrium was in the
400 range limit after about 5 h of contact. This can imply a lack of antilisterial activity, since a high
401 degree of bacterial growth will occur during this period.

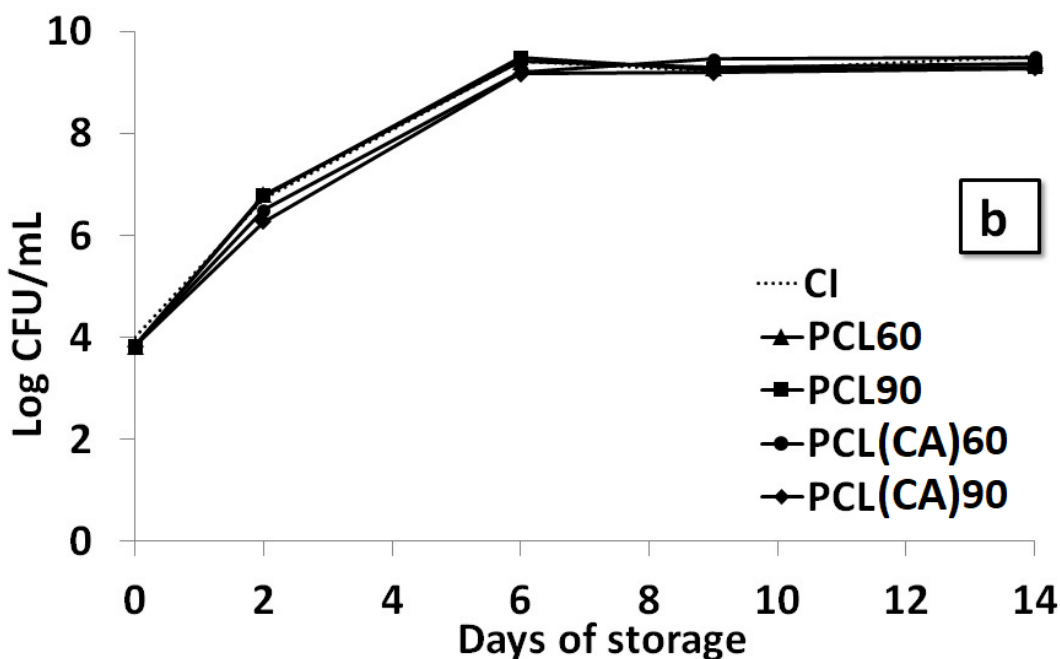
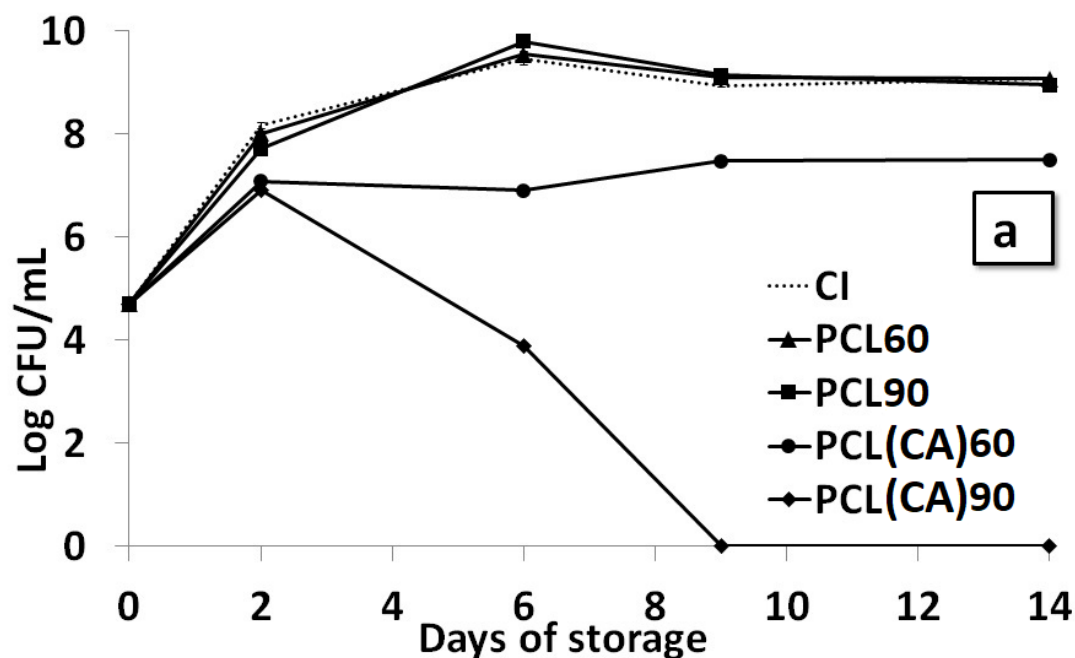


402

403 **Figure 5.** CA release prediction as a function of time in the microbial plate, assuming the
404 behaviour of A simulant (10% ethanol). MIC range for *E. coli* is from $2.25 \cdot 10^{-4}$ g/mL
405 (Cosentino et al., 1999) to $3.75 \cdot 10^{-4}$ g/mL (Du et al., 2015); MIC range for *L. monocytogenes* is
406 from $3.75 \cdot 10^{-4}$ g/mL (Pol & Smid, 1999) to $4.5 \cdot 10^{-4}$ g/mL (Cosentino et al., 1999)

407

408 Figure 6 shows the growth curves of both *E. coli* and *L. innocua* for 14 days, obtained in the *in*
409 *vitro* test, for the control samples (uncoated and CA-free PCL coated) and for the samples
410 coated with CA-loaded PCL mats electrodeposited for 60 and 90 min. No antilisterial effect was
411 observed for either CA loaded PCL mat, according to the predicted CA release which did not
412 reach or barely attained the MIC levels at longer contact times when the bacteria have grown to
413 a great extent. Limitations in the CA release from PCL mat to aqueous systems compromise the
414 antilisterial activity of the CA loaded mats, despite the enough CA load revealed by Figure 2b.
415 For *E. coli*, a bacteriostatic effect was observed from 2 incubation days onwards, revealing that
416 enough CA was released from the mat electrodeposited for 60 min at this contact time. For the
417 90-min electrodeposited mat, bacterial death progressively occurred from 2 days of contact
418 onwards, when more CA release occurred in the culture medium. The obtained antimicrobial
419 action of the CA-loaded PCL mats was coherent with the total load of CA, its release kinetics in
420 the culture medium (assuming its behaviour as simulant A) and the MIC values of the respective
421 bacteria. From this analysis, thicker CA-loaded PCL mats, with a greater surface density of CA,
422 would be required to obtain antilisterial activity, whereas 90-min electrodeposited mats were
423 able to exert an antibacterial effect against *E. coli*, taking into account the total CA load and its
424 release kinetics in the culture medium.



425

426 **Figure 6.** Growth of *E. coli* (a) and *L. innocua* (b) in the culture media at 10 °C in the uncoated
 427 inoculum control plate (CI) and plates coated with electrospun PCL without CA (PCL60;
 428 PCL90) and with electrospun PCL with CA (PCL(CA)60; PCL(CA)90). The numbers 60 and
 429 90 indicate the electrodeposition time of fibres.

430

431 3.4 Multilayer starch films with electrospun PLC fibres with and without CA

432 The efficiency with which PCL electrospun fibres improve the functional properties of starch
 433 films, either as packaging material or antimicrobial activity, was analysed through the
 434 measurement of the tensile, barrier and optical properties of the films as well as the *in vitro*

435 antibacterial capacity for *E. coli* and *L. innocua*. Table 2 shows the different film properties,
 436 where the effect of the electrospun PCL layer between the two starch sheets can be observed.

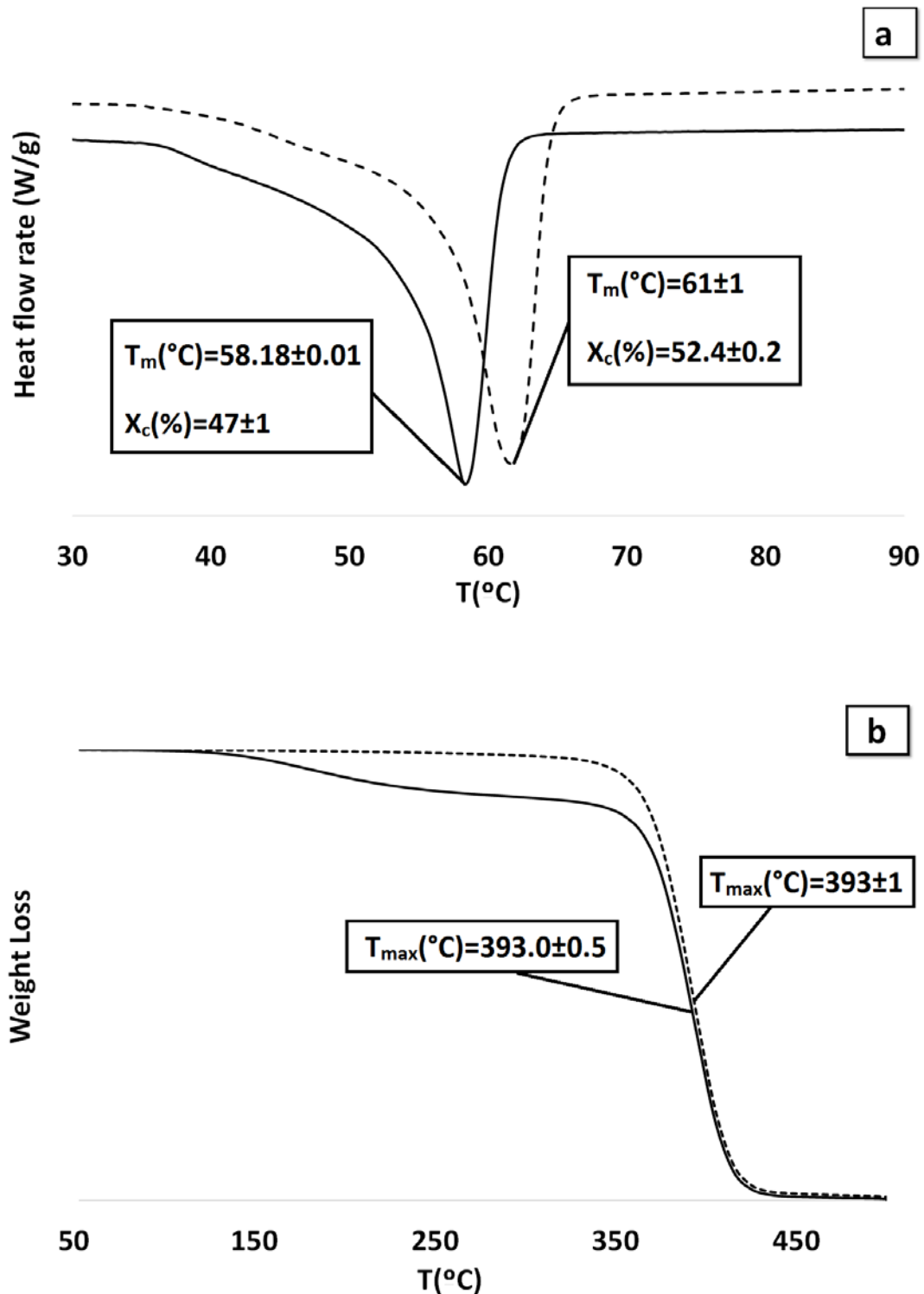
437 **Table 2.** Barrier (water vapour permeability: WVP and Oxygen permeability: OP), tensile
 438 (Elastic modulus: EM, tensile strength: TS and deformation at break: % ϵ) and optical properties
 439 (Internal transmittance: Ti, Gloss (60°), lightness: L*, chrome: C_{ab} and hue: h_{ab}) of starch
 440 multilayer films containing electrospun PCL fibres. Different superscript letters in the same row
 441 indicate significant differences (p<0.05) between multilayers.

	Multilayer				
	S-S	S-PCL60-S	S-PCL(CA)60-S	S-PCL90-S	S-PCL(CA)90-S
Thickness [mm]	0.38 ± 0.07 ^a	0.45 ± 0.01 ^b	0.45 ± 0.01 ^b	0.44 ± 0.03 ^b	0.5 ± 0.02 ^b
WVP[g/(Pa·s·m)]	570 ± 40 ^c	180 ± 60 ^b	190 ± 40 ^b	120 ± 20 ^a	94 ± 10 ^a
OP x 10¹⁴ [cm³/ Pa·s·m]	5 ± 3 ^a	3.6 ± 0.4 ^a	5 ± 1 ^a	4 ± 2 ^a	2.9 ± 0.6 ^a
EM [MPa]	206 ± 21 ^b	149 ± 23 ^a	126 ± 34 ^a	192 ± 32 ^b	126 ± 23 ^a
TS [MPa]	9.7 ± 0.7 ^d	7 ± 2 ^{bc}	6.2 ± 0.9 ^{ab}	8 ± 1 ^c	5.8 ± 0.8 ^a
%ϵ	39 ± 4 ^b	34 ± 15 ^{ab}	34 ± 10 ^{ab}	36 ± 5 ^{ab}	29 ± 6 ^a
Ti [460 nm]	0.73 ± 0.01 ^c	0.72 ± 0.01 ^{bc}	0.69 ± 0.01 ^{ab}	0.69 ± 0.04 ^{ab}	0.69 ± 0.02 ^a
Gloss [60°]	15 ± 2 ^b	17 ± 4 ^b	9 ± 2 ^a	18 ± 5 ^b	10 ± 3 ^a
L*	65.7 ± 0.6 ^b	64.8 ± 0.9 ^{ab}	64.6 ± 0.8 ^{ab}	66 ± 2 ^b	64.2 ± 0.4 ^a
C_{ab}	21.3 ± 0.3 ^{ab}	21.2 ± 0.3 ^{ab}	21.0 ± 0.6 ^{ab}	21 ± 1 ^a	21.7 ± 0.3 ^b
h_{ab}	79.8 ± 0.4 ^c	79.4 ± 0.4 ^{bc}	78.9 ± 0.6 ^{ab}	79.5 ± 0.5 ^c	78.7 ± 0.2 ^a

442 The ES layer thickness in the multilayer film can be deduced from the respective difference
 443 values as regards the SS bilayer. The resulting thickness was affected not only by the ES layer
 444 thickness but also by the relative radial flow of the PCL in the mats during the
 445 thermocompression step of multilayer. In general, no significant differences were observed in
 446 the measured thicknesses of the multilayer films, all of which were about 70-120 μ m thicker
 447 than the SS bilayer. This indicates that, regardless of the initial thickness of the ES layer, its
 448 flow and compaction during thermocompression led to non-significant differences in the
 449 multilayer film thicknesses. The equilibrium moisture content of the multilayers was not
 450 significantly affected by the presence of the PCL fibre, this being 7.3 ± 0.2 g/100 g dry film.

451 As concerns the barrier properties, the WVP was greatly reduced when PCL fibres were present
 452 between the starch layers: 65 and 80 % reductions for 60 and 90-min electrodeposited layers,
 453 respectively (Table 2). This indicates the effectiveness of the parallel assembly of the PCL
 454 sheet, with a different surface density in the multilayer film, at limiting the transport of water
 455 molecules, as previously observed in starch-PCL bilayer films by Ortega-Toro et al. (2015).
 456 Nevertheless, no changes in the oxygen permeability of the SS films were observed, due to the
 457 fact that starch layers are actually the limiting material for the gas transport. As concerns tensile
 458 behaviour, the inclusion of PCL mats in multilayer films implied a slight reduction of the film
 459 stiffness and tensile at break, which can be attributed to the discontinuity introduced at the
 460 interlayer zone in the multilayer. This reduced the film's cohesion forces at the interfacial area,
 461 where no great chemical affinity could be expected for the PCL chains in contact with the starch
 462 matrix. However, no significant changes in the film extensibility were observed, except when
 463 CA-loaded PCL fibre produced for 90 min was included. In this case, films become slightly less
 464 stretchable and resistant, which could be attributed to a certain degree of CA diffusion into the
 465 starch matrix of bilayer, giving rise to a weakening effect in the starch multilayer. This effect

466 was also observed at lower intensity in multilayers containing 60 min electrospun PCL with CA.
467 The optical properties of the multilayer films were hardly affected by the PCL mat inclusion.
468 The internal transparency (T_i) was the most affected optical property due to the greater opacity
469 of PCL than starch (Ortega-Toro et al., 2015).



470
471 **Figure 7.** a) DSC thermograms and b) TGA curves of electrospun PCL fibres carrying
472 (continuous line) or not (dashed line) CA. Melting temperature (T_m) and crystallization degree
473 (X_c) as well as temperature for the maximum degradation rate (T_{max}) of PCL are shown for each
474 sample.

475

476 DSC and TGA analyses of isolated PCL fibres and multilayers were carried out to identify
 477 changes in phase transitions or thermal stability of the polymers, either starch or PCL,
 478 associated with the presence of CA or with its internal diffusion in the multilayer. Figures 7a
 479 and b show the DSC thermograms (first heating step) for both fibres as well as the TGA curves.
 480 The melting endotherm of PCL can be observed for both mats, showing a shift of the melting
 481 peak to lower temperatures when fibres contained CA. From the melting enthalpy and the mass
 482 of PCL in each sample, the crystallization degree of PCL was estimated, by considering the
 483 enthalpy value of completely crystallized PCL (139.3 J/g, as reported by Koenig & Huang,
 484 1995). A decrease in the PCL crystallization degree was also observed in CA-loaded fibres.
 485 These results point to the effective miscibility of CA and PCL in the fibres, which reduced the
 486 melting point and crystallinity of the polymer, as previously reported for CA containing PHBV
 487 (Requena, Jiménez, Vargas & Chiralt, 2016). TGA curves (Figure 7b) show the influence of the
 488 CA-load on the thermal behaviour of the mats. Weight losses at temperatures below the PCL
 489 thermodegradation must be attributed to the CA evaporation from the mat, which did not affect
 490 the temperature of the maximum degradation rate ($393.2 \pm 0.6^\circ\text{C}$) of the polymer, but
 491 overlapped the starting degradation step.

492 In contrast, DSC thermograms of multilayer films also revealed the crystalline structure of PCL
 493 in the electrospun layer, although no effect of CA was observed on its crystallization and
 494 melting temperatures (Table 3), which coincided with those observed for pure PCL mats, these
 495 being in the range previously reported for PCL films (Ortega-Toro et al., 2015). The values of
 496 crystallization and melting enthalpy from the different scans did not show significant
 497 differences for a determined sample, which indicates that the film process formation induced
 498 similar PCL crystallization to that obtained in the DSC scan conditions. Small differences in the
 499 enthalpy values among the different samples can be attributed to the fluctuations in the
 500 electrospun layer thickness in the different small bilayer samples used in the DSC analysis. This
 501 melting behaviour of PCL in multilayers, similar to that of pure PCL fibres, suggests that a high
 502 proportion of CA migrated to the starch sheets in the multilayer assembly, since the potential
 503 remaining amount in the PCL internal sheet was not enough to affect the PCL melting
 504 behaviour. On the other hand, the glass transition temperature of the starch (Table 3) was
 505 slightly reduced in multilayers containing CA-loaded fibres, particularly when this fibre mat
 506 was thicker, with the subsequent higher load of CA. This is coherent with the afore-mentioned
 507 small changes in the tensile behaviour of multilayers containing carvacrol.

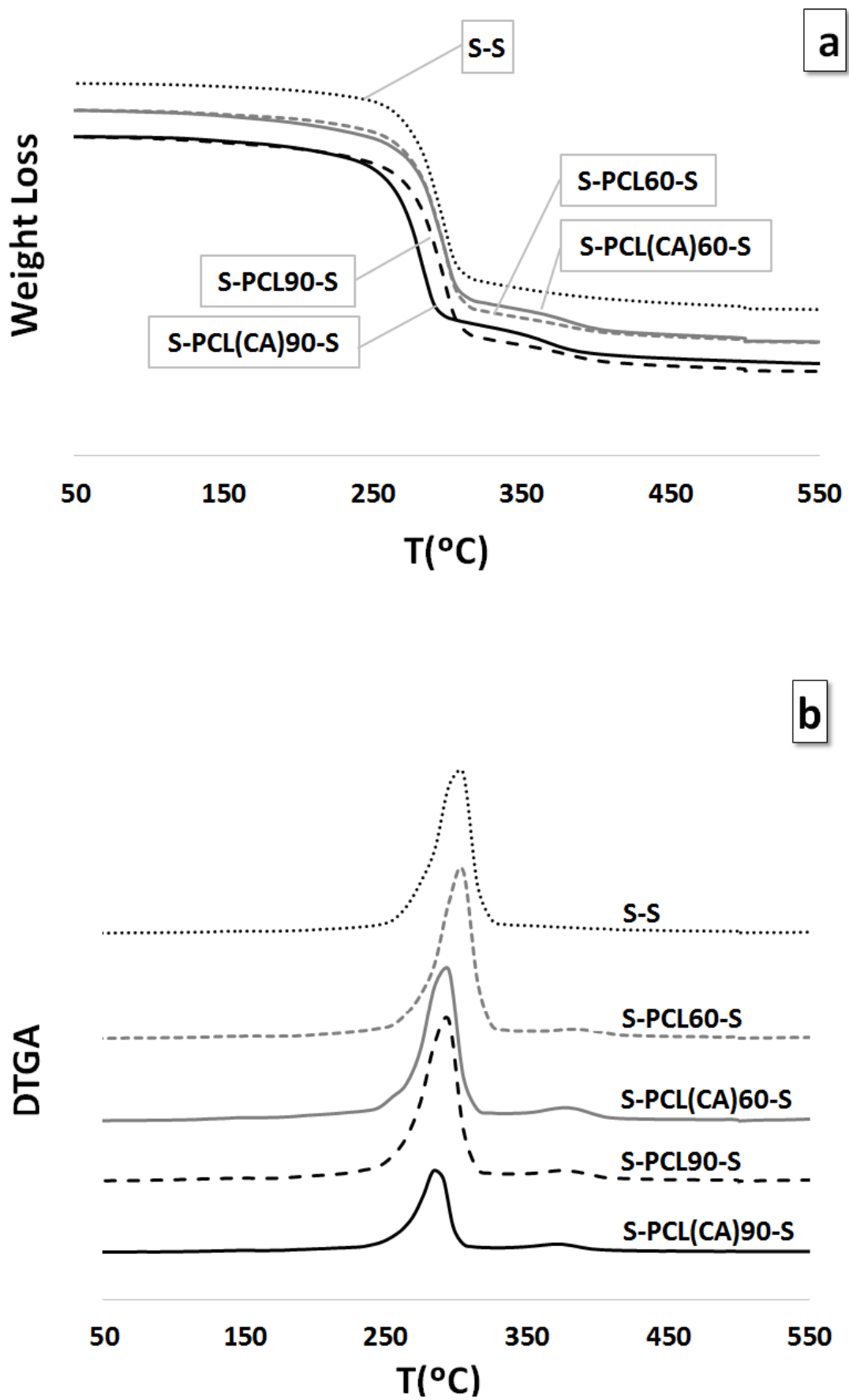
508 Table 3. Phase transitions in the starch multilayer films containing electrospun PCL fibres (glass
 509 transition temperature of starch (T_g), and melting and crystallization temperatures and enthalpy
 510 of PCL). Different superscript letters in the same row indicate significant differences ($p < 0.05$)
 511 between multilayers.

		Multilayer				
		S-S	S-PCL60-S	S-PCL(CA)60-S	S-PCL90-S	S-PCL(CA)90-S
T_g (midpoint)	1st scan	107 ± 3^{ab}	$108,0 \pm 0,6^{ab}$	108 ± 8^b	101 ± 2^{ab}	98 ± 4^a
1st heating	ΔH_m (J/g)	-	1.36 ± 0.12^a	3.45 ± 0.04^b	1.9 ± 0.2^a	3.8 ± 0.5^b

step	T_m (peak)	-	62.7 ± 0.5 ^b	62.8 ± 0.6 ^b	60.9 ± 0.4 ^a	61.6 ± 0.4 ^a
2nd heating step	ΔH_m (J/g)	-	0.90 ± 0.04 ^a	2.5 ± 0.2 ^{bc}	1.5 ± 0.2 ^{ab}	3 ± 1 ^c
	T_m (peak)	-	55.79 ± 0.09 ^a	55.6 ± 0.5 ^a	55.9 ± 0.2 ^a	56.6 ± 0.4 ^b
Cooling step	ΔH_c (J/g)	-	1.20 ± 0.04 ^a	3.1 ± 0.2 ^b	1.9 ± 0.7 ^a	3.6 ± 0.8 ^b
	T_c	-	29 ± 1 ^a	28.4 ± 0.9 ^a	28.5 ± 0.6 ^a	27.5 ± 0.4 ^a

512

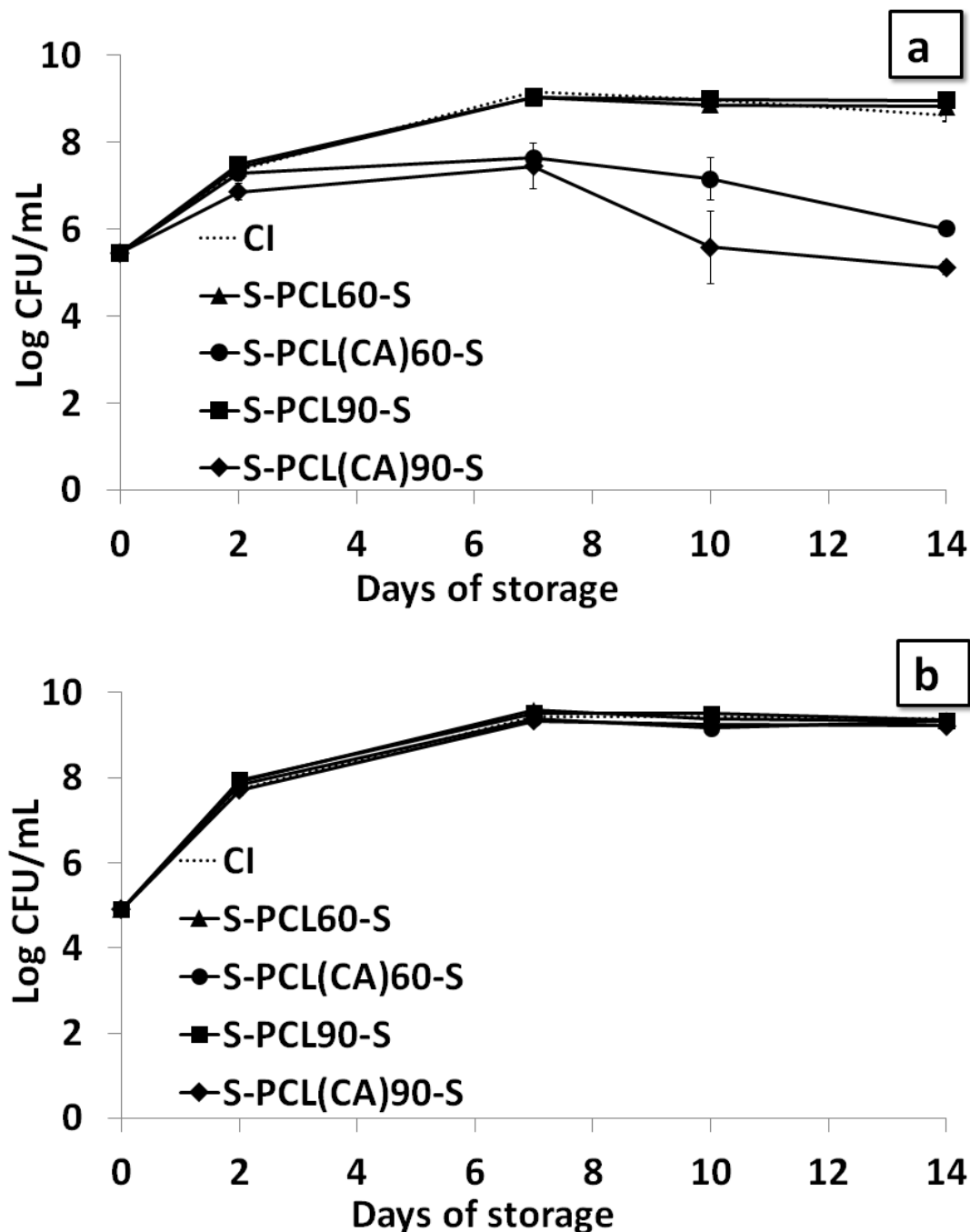
513 Likewise, the thermal degradation (Figure 8 a and b) of multilayer films exhibited the two
514 typical steps associated with starch and PCL degradation, where the influence of CA can also be
515 observed. Weight loss started earlier in 60-min electrospun CA-loaded fibres (from about
516 130°C), associated with the CA evaporation, but had no significant effect on the temperature of
517 maximum degradation rate of starch. Nevertheless, in 90-min electrospun fibres, with a greater
518 CA load, a shifting of this temperature occurred which indicated the CA-starch interactions as a
519 result of the CA diffusion into the starch layers. On the other hand, the thermodegradation of
520 PCL was also affected by CA interactions, exhibiting a lower temperature in the maximum
521 degradation rate. A higher residual mass was also observed in CA-loaded multilayers. This
522 behaviour suggests the participation of CA in both starch and PCL matrices with enough
523 interaction forces to modify their thermal stability and modify the tensile behaviour of the films,
524 as previously commented.



525

526 **Figure 8.** TGA (a) and DGTA(b) curves of starch multilayer films containing 60 (grey) or 90
 527 (black) min electrospun PCL fibres, with (continuous line) and without (dashed line) CA.

528



529

530 **Figure 9.** Growth of *E. coli* (a) and *L. innocua* (b) in the culture media at 10°C in the inoculum
 531 plates, coated with starch multilayers with 60 or 90 min electrospun CA loaded layers (S-
 532 PCL(CA)60-S and S-PCL(CA)90-S). Controls: uncoated inoculated plates (CI) and those coated
 533 with CA-free multilayers (S-PCL60-S and S-PCL90-S).

534 The CA diffusion into the starch layers could affect the antimicrobial activity of the multilayer
 535 films, with respect to that observed for CA-loaded mats, which was analysed. The antibacterial
 536 activity of the multilayer films revealed from the *in vitro* test (Figure 9), indicates a similar
 537 behaviour to that observed for the CA-loaded PCL mats. No antilisterial effect was observed,
 538 while a delayed action against *E. coli* was observed. This delay can be explained by the thicker
 539 layer (starch layer) where CA must diffuse to reach the target point in the culture medium. From
 540 2 to 7 contact days, a bacteriostatic effect could be appreciated; from 7 contact days onwards,
 541 however, a bactericidal action was observed both for 60 and 90-min CA-loaded electrospun PC.

542 This effect was more intense when CA-loaded fibres were thicker (90 min electrodeposited), in
543 agreement with the higher CA load in the multilayer. Then, CA-loaded electrospun PCL fibres
544 applied to multilayer starch films allows for effective antibacterial action against *E. coli* to a
545 similar extent to that observed in the isolated fibres, but with a slightly retarded effect.
546 Likewise, the electrospun PCL layers were greatly effective at reducing the WVP values of
547 starch films, without relevant changes in their tensile or optical properties.

548

549 **Conclusions**

550 Electrospun PCL fibre mats encapsulating CA were effective at controlling the growth of *E.*
551 *coli*, when the surface density of CA loaded fibres was 1.2 and 1.8 mg/cm², the latter being
552 more effective. In both cases, the CA released into the culture medium exceeded the MIC of the
553 bacteria. Nevertheless, these mats were not effective at controlling the growth of *L. innocua*,
554 since a greater release of CA was necessary to achieve the MIC of this bacterium. Not only did
555 the CA load in the fibres determine their antimicrobial effect, but also its release capacity into
556 the aqueous media. In this sense, the fibre showed a faster release and higher release capacity of
557 the active in fatty foodstuffs (simulants D1 and D2), where practically the total amount of CA
558 could be released from the fibres. However, in more aqueous food systems (simulants A and B),
559 such as the bacteria culture medium, a more limited CA delivery (60-75 %) occurred with a
560 slower rate, which reduced the potential effectiveness of the encapsulated active. This meant
561 that the Gram negative bacteria (*E. coli*) could be effectively inhibited, whereas no growth
562 inhibition was observed for Gram positive (*L. innocua*), which would require a greater surface
563 density of the electrospun CA-loaded PCL fibres. This behaviour was reproduced in multilayer
564 starch films containing the CA-loaded electrospun PCL fibres between two starch sheets,
565 although a delayed response was observed for the antimicrobial action. In these multilayer
566 films, a great reduction in the water vapour permeability was also observed with respect to that
567 of starch films, without any notable changes in the other packaging functions. Therefore, active
568 electrospun PCL fibre in multilayer starch films represents an interesting alternative for the
569 purposes of active food packaging.

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