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Additional Information

Nanoindentation study of the interfacial zone between cellulose fiber and cement matrix in extruded composites

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Abstract

The present study shows the application of the nanoindentation technique to evaluate the properties of the cellulose fiber-cement matrix interfacial zone in composites prepared with an auger extruder. The degree of strength of the bond between fiber and matrix is recognized as important variable that influences macro-mechanical properties, such as modulus of rupture and toughness of cement based composites. The nanoindentation measurements showed the highest hardness and elastic modulus in the part inner of the cellulosic fiber after hydration process due to precipitation and re-precipitation of cement hydration products. These results indicate that mineralization of the cellulosic fibers can affect the stress distribution and interfacial bond strength in the cement based composite.

Keywords: Vegetable fiber; Microstructure; Mechanical properties; Ordinary Portland cement; Transition zone

1. Introduction

It is widely accepted that concrete at the mesoscopic scale should be treated as a three-phase composite material constituted by bulk hardened cement paste, aggregates and an interfacial transition zone (ITZ) between these two [1]. Therefore, traditional ITZ is the region of the cement matrix around the aggregate particles, which is perturbed by the presence of the aggregate. Its origin lies in the packing of the cement grains against the much larger aggregate, which leads to a local increase in porosity and predominance of smaller cement particles in this region. The ITZ is region of gradual transition and is highly heterogeneous, nevertheless the average microstructural features may be measured [2].

The cementitious composites reinforced with cellulosic fibers are complex materials. Thus, cement based composite design must be considered for the interrelation between interfacial transition zone, fibers, matrix and processing method. Particle and fiber size distribution, particles and fibers packing and rheological behavior of the cement paste with fibers in the fresh state also interfere on the interfacial transition zone of the hardened cement based composite. Furthermore, changes in the mechanical properties over time can occur due to microstructural changes in the fiber-matrix interface and bulk, as a consequence of the continued hydration process that occurs in the fiber surroundings, i.e. if the interfacial transition zone becomes denser due to re-precipitation of cement hydration products in the porous layers [3]. Therefore, in this manuscript an interfacial transition zone (ITZ) terminology is related to a region between bulk hardened cement paste and cellulosic fiber.

Kraft pulp fiber-reinforced cement-based materials have being increasingly used in most developing countries as construction materials [4-6]. Fibers usually increase mechanical strength and the capacity of cement based composite to absorb energy through the distribution of microcracks into the material. The bonding between cellulose fiber and Portland cement plays a major role in the transference of the load from the matrix to the reinforcement, impacting the mechanical behavior of the resulting fiber-cement material. This bonding can be of physical

(mechanical interlocking) or chemical (mainly hydrogen bonds) nature, or a combination of both [7].

Extrusion process has a potential of producing fiber reinforced cementitious composites with higher performance compared to the traditional slurry dewatering processes such as Hatschek and Flow-on methods. Several advantages of extrusion molding can be pointed for manufacturing of fibrous composites: (i) low implementation cost; (ii) low capillary porosity of extruded composites; and (iii) low water/cement ratio, which plays a significant role for improving of the fiber-matrix interface zone [8-10]. As mentioned above, the control of fiber-matrix interface zone in extruded composites depends on cellulosic fibers and matrix constituent's properties, cure procedure as well as weathering conditions that the composite are exposed. Previous studies highlighted that water/cement ratio influences the interface zone thickness and that lower porosity may increase composite strength and matrix toughness [9-12]. Therefore, the optimization of the interfacial zone is not an easy task since it is necessary to satisfy the equilibrium between fiber-matrix adhesion and fiber pull-out. In addition, there is still a lack of information about the consequences of the different types of interface zone between cellulose fibers and matrix, on composite mechanical properties. Thus, an effort is required in order to understand the mechanical properties at fibermatrix transition zone.

The macroscale mechanical properties of bulk materials are dependent on the properties of their microscale/nanoscale constituents and the interface between these constituents. Microindentation technique is one of the developing tests that have been used for evaluating mechanical characteristics of microsized zones in diverse cement based composites [13-15]. Nevertheless, microindentation is not capable to detect mechanical differences at the interfaces between cement matrix and reinforced fibers, due to some limitations, such as a resolution of microindentation system in terms of load is about 5.0 mN, to analyze some structural features because of the heterogeneity of the specimen, inadequate spacing between indents or from edges, operator's visual acuity. Quasistatic nanoindentation has become the standard technique used for nanomechanical characterization of small volumes of material, such as elastic modulus, hardness at the nanoscale. It is used machines that can record small load and displacement with high accuracy and precision, during loading and unloading can be continuously monitored with high resolution in the nanonewton and nanometer ranges. In this technique is used usually the Berkovich tip is a 3-sided pyramid that have an apex appropriate for investigating the interface transition zone between fiber and matrix. Therefore, nanoindentation technique can contribute to quantitatively measure the underlying source of the bulk response at the nanoscale in cement based composites that are heterogeneous material from both a chemical composition and nanoscale structure standpoint. Being able to evaluate at this scale allows micro and nano structural "tuning" to be performed to optimize bulk properties [16,17]. The aim of this study was to evaluate the interfacial zone between fibers to cement matrix in extruded composites reinforced with eucalyptus pulp, using a nanoindentation technique allied with microscopic examinations.

2. Experimental

2.1. Materials

Unbleached unrefined eucalyptus (*Eucalyptus grandis*) Kraft pulp was provided by Fibria S/A, Brazil. The cellulose pulp was collected directly from the mill, prior to drying and pressing. It was extensively washed with water and centrifuged to remove any residual chemicals from the pulping processes.

Eucalyptus pulps were analyzed about morphology (length and width) and concentration of particles (fibers per gram and content of fines). A fine element (small particles with length lower than 200 μm that are commonly called fines, which also contain bandlike materials from both primary wall and secondary wall layers of the fibers) was detected in the pulp with dimensions lower than those of fibers. Fines do not contribute significantly to cement based composite strength but act more as filler [18]. Physical characteristics of the unbleached eucalyptus pulp are showed in Table 1.

Table 1 Physical characteristics of the unbleached eucalyptus pulp [18].

Average length (mm)	Average width (μm)	Aspect Ratio	Fibrous material (10^6 fibers/g)	Content of fines by mass (%)
0.83 ± 0.05	16.4 ± 0.2	51	18.17 ± 1.11	25.7 ± 0.6

2.2. Composite mix design

Formulation used in the composite production was based on previous study [8], and is described in Table 2.

Table 2 Mix design used in the production of the composites.

Raw material	Content (% by mass)
Portland cement (CPV-ARI) ^a	70
Limestone <i>filler</i> ^b	27
Unbleached eucalyptus pulp ^c	3

^a ASTM C150 type I [19], provided by Caue.

^b Itaú.

^c Fibria S.A.

Portland cement and limestone particles distribution was performed by laser particle size analyzer (Malvern Mastersizer S long bed, version 2.19). Cement and limestone particles showed 50% of their size below 11 μm and 16 μm respectively.

Hydroxypropylmethylcellulose (HPMC) from Aditex, is a superplasticizer which disperses cement particles, reducing the flow stress and the viscosity of the cement paste. Polyether carboxylic provided by Grace, commercially called ADVA 170, is an additive which reduces the surface tension of the water and thus causes an increase in the surface area of the liquid which improves the water absorption, forming a lubrication layer between the wall of the extruder and the surface of the fiber cement board. HPMC with 86,000 average molecular weight and 5.39 cPs viscosity (at 2% concentration in water at 20 °C) and polyether carboxylic commercially named ADVA 170, were used as rheological modifiers in fiber-cement, with dosage of 1 wt% of the total mass of the particulate inorganic materials (Portland cement and limestone) to promote the pseudo-plastic behavior of the composite, which, in turn, enabled the extrusion process.

2.3. Composite extrusion

The mass was mixed in a mechanical Eirich intensive mixer (capacity of 10 L) for 5 min at 125 rpm, 5 min at medium velocity (220 rpm) and finally 5 min at high velocity (450 rpm). The watercement (w/c) ratio was 0.33. The composite mixture was transferred to a Verdes 051 extruder (Fig. 1a-b). The speed of the extruder was 4 mm/s; die width/diameter ratio was 3.3. Composites were molded in 15 mm \times 50 mm \times 200 mm plates. The illustration of the chamber parts of the extruder is presented in Fig. 1c.

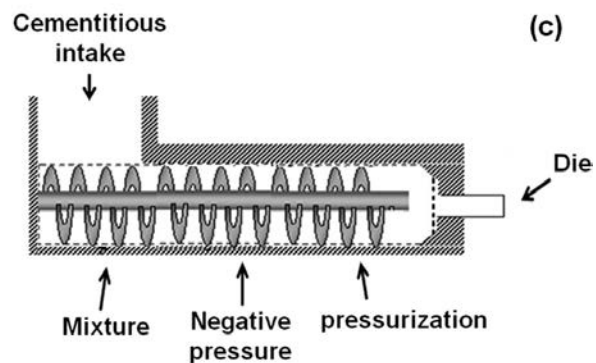
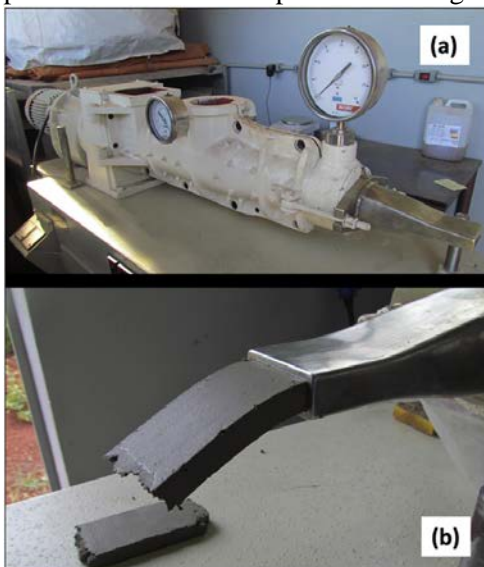


Fig. 1. (a) Extruder machine used for molding; (b) side view showing the mass exit through the die; and (c) details of the operating mechanism of the extrusion process (adapted from Ref. [20]).

The composite mixture was re-circulated into the extruder for 5 min before tailoring the samples. Pads with 15 mm x 50 mm x 200 mm were cured at water vapor saturated environment (in sealed plastic bags) at 25 ± 2 °C for two days. Subsequently, the specimens were maintained in water vapor saturated environment (in sealed plastic bags) and placed in a chamber at 45 °C for five days (thermal curing) totalizing 7 days of cure [21].

2.4. Microstructure characterization

The specimens were cut out from the middle section of a larger volume of fiber cement, with minimal deformation, by using a precision cutter, Buehler, model Isomet, with precision diamond saw, 15LC, and using an isopropyl alcohol based cutting fluid to cut. This method provides roughness smaller. The specimens were embedded in epoxy resin (MC-DUR1264FF) previously to be visualized in a scanning electron microscope (SEM). A final polishing was carried out using sequentially 8-4 mm, 4-2 mm and 1-0 mm diamond abrasives, as reported elsewhere [22,23]. The polished samples were carbon coated before being analyzed in a JEOL JSM6300 electron microscope with acceleration voltage of 20 kV. A back-scattered electron (BSE) detector was used to observe the morphological features on the cross-section and polished surfaces. BSE imaging permits the identification of cementitious phases since the electron scattering change. Dark and light areas are related to lighter and heavier elements respectively. Energy dispersive spectrometry (EDS) analysis was performed in order to identify the chemical composition in different spots on the same polished surface specimens.

2.4.1. Atomic force microscopy characterization (AFM)

In contrast to ductile material, cement based composite have regions with higher hardness, lower ductility, and a basically brittle nature. Due to the brittleness, cellulosic fibers and porosity of the composite it is quite often difficult to cut and polish. All areas of the specimens observed by nanoindentation were cut and polished with uniformity and carefully to avoid any pullout of fibers, but as the composite is heterogeneous the response of each region was different according to its resistance. Specimens for AFM characterization were cut with same system mentioned above and embedded in polyacrylamide. The semi-automatic polisher was used to adjust polishing velocity and pressure, i.e. all regions were submitted to same conditions of preparation. Then, the dried surfaces were grinded in a sequence of SiC papers (grit #500, #1000, #2000 and #4000) and polished in a cloth sequentially with 9 mm, 6 mm and 3 mm diamond sprays. Finally, the specimen surfaces were cleaned by washing in alcohol and ultrasonic bathing.

The characterization of the fiber-matrix interface zone was carried out using the tapping-mode atomic force microscopy (TMAFM) to obtain height and phase imaging data simultaneously, with Nanoscope III-Digital Instruments. Silicon cantilever with spring constant of around 70 N/m and a scan area of 15 mm x 15 mm was used. Samples were prepared in the laboratory environment with temperature around 25 °C and relative humidity between 50% and 65%. AFM images were obtained in the same laboratory environment conditions. The test locations were guided by means of an incorporated optical light microscope.

2.5. Nanoindentation tests

Specimens for nanoindentation tests were prepared the same methodology that specimens to AFM characterizations. Nanoindentation was performed in an Agilent G-200 instrument working in continuous stiffness measurement (CSM), configured at 2 nm of harmonic amplitude, 45 Hz frequency, and using a Berkovich indenter. The test locations were guided by means of an incorporated optical light microscope. To rule out the influence of surface morphology, the indentation depth should be much greater than the characteristic size of surface roughness to avoid errors and size effect in the calculations [17]. Tests were programmed to reach from 100 to 300 nm-indent depths until the maximum depth of 1000 nm. Therefore, the depth of the imprint spots used in the nanoindentation was to minimize surface roughness of the samples. Analyses were performed with 8 nm distance between the measurement points [17]. It was performed more

than 20 analyses for each region of the composite (bulk matrix, interface or transition zone, anhydrous grain and cellulose fibers).

To guarantee a significance results for nanoindentation tests were choose under a completely randomized design the ITZ areas to analyze a linear sequence with four regions (transition zone, cellulose fiber, transition zone, bulk matrix and anhydrous grain or follow sequence: anhydrous grain, bulk matrix, cellulose fiber, transition zone) and twenty replications. The effects of hardness and elastic modulus were compared via statistical contrast by student test at 5% of significance [17]. The ACTION statistical program was used. Therefore, the interfaces towards fibers and matrix were studied in greater detail.

3. Results and discussion

3.1. Chemical and microstructure characterization Fig. 2 shows SEM-BSE micrographs of the extruded composites. Energy dispersive X-ray spectroscopy (EDS) analysis (Fig. 2a) indicate that fiber-matrix interfacial zone presents high Ca/Si ratio (between 4 and 5), indicating the possible correspondence with $\text{Ca}(\text{OH})_2$ crystals with carbonated products, such as calcite, as corroborated by AFM analysis (Fig. 3) and according to Ruiz-Agudo et al. [24]. This interfacial zone rich in $\text{Ca}(\text{OH})_2$ crystals has been widely reported in literature for cementitious composites [25-29]. Bulk matrix revealed an expected lower Ca/Si ratio (approximately 2.5) as observed in Fig. 2b. The phase composition of the anhydrous grains and matrix are shown in Fig. 4a and b, respectively. Spot B presents higher concentration of SiO_2 and CaO due to the hydration products of cement in relation to the spot A (region of the anhydrous grain).

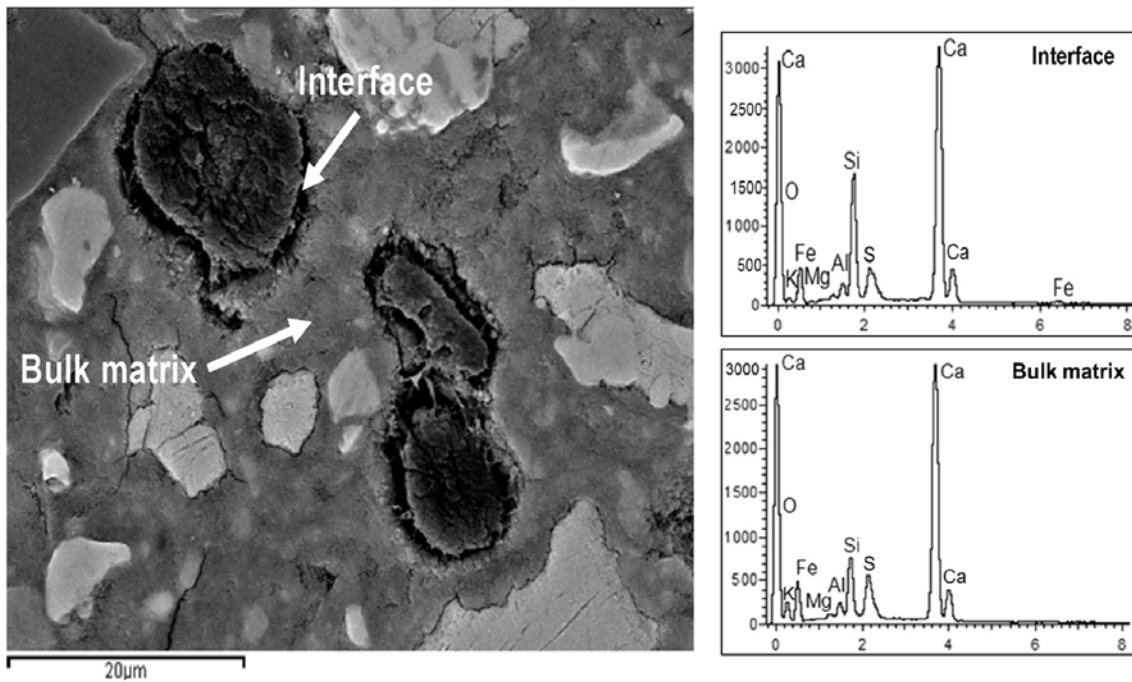


Fig. 2. Left: Scanning electron microscopy (SEM) micrographs in back scattering electron (BSE) mode of the polished cross-section of an extruded composite. Right: Energy dispersive spectra (EDS) referring to the fiber-matrix interface (upper) and to the bulk matrix (bottom).

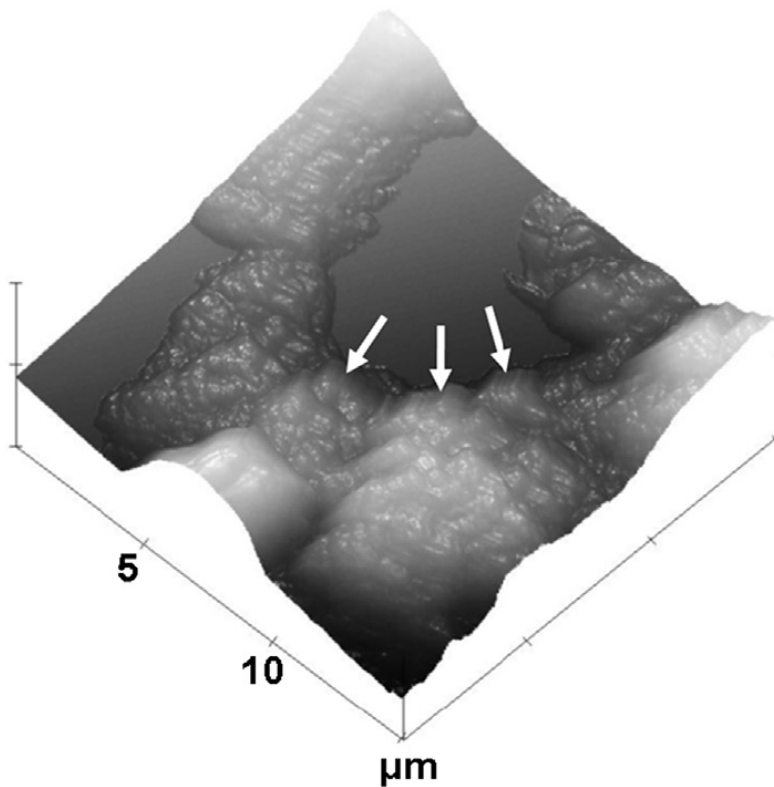


Fig. 3. AFM topography image of the interface fracture. Arrows show the possible $\text{Ca}(\text{OH})_2$ with carbonated production in the interfacial zone between the cellulose fiber and cement matrix.

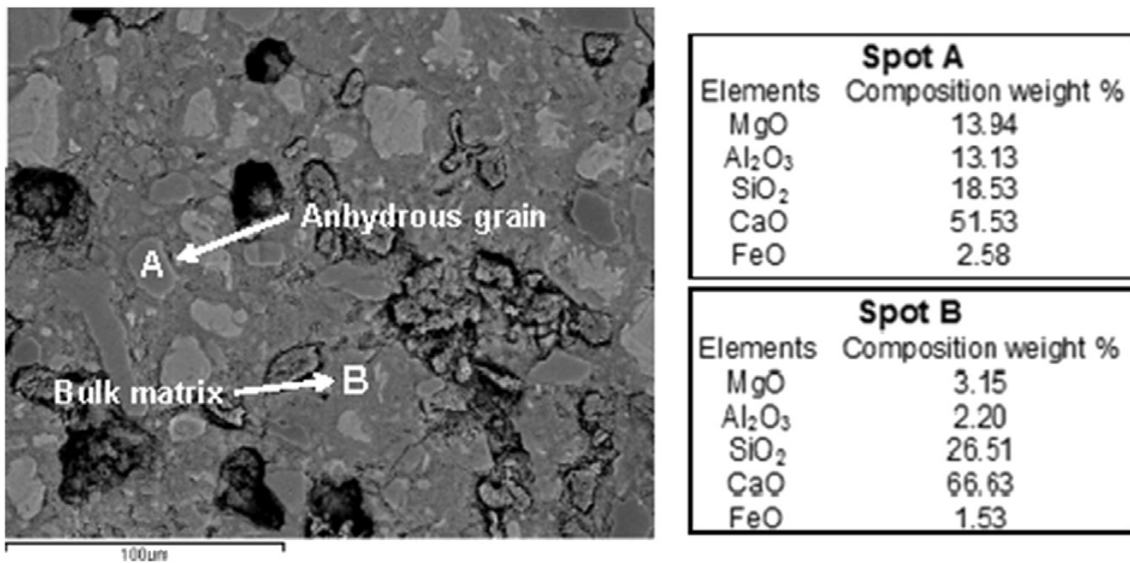
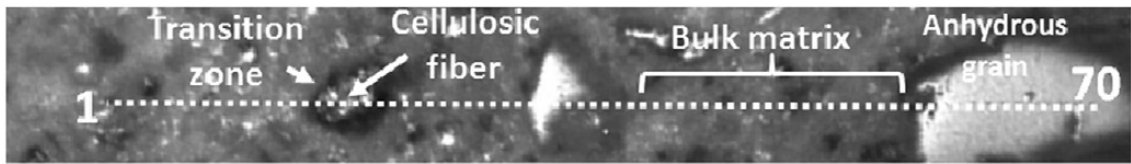


Fig. 4. Scanning electron microscopy (SEM) image in backscattering electron (BSE) mode of the polished cross-section in an extruded composite. Energy dispersive spectra (EDS) refer to the composition of the anhydrous grain (spot A) and bulk matrix region (spot B).

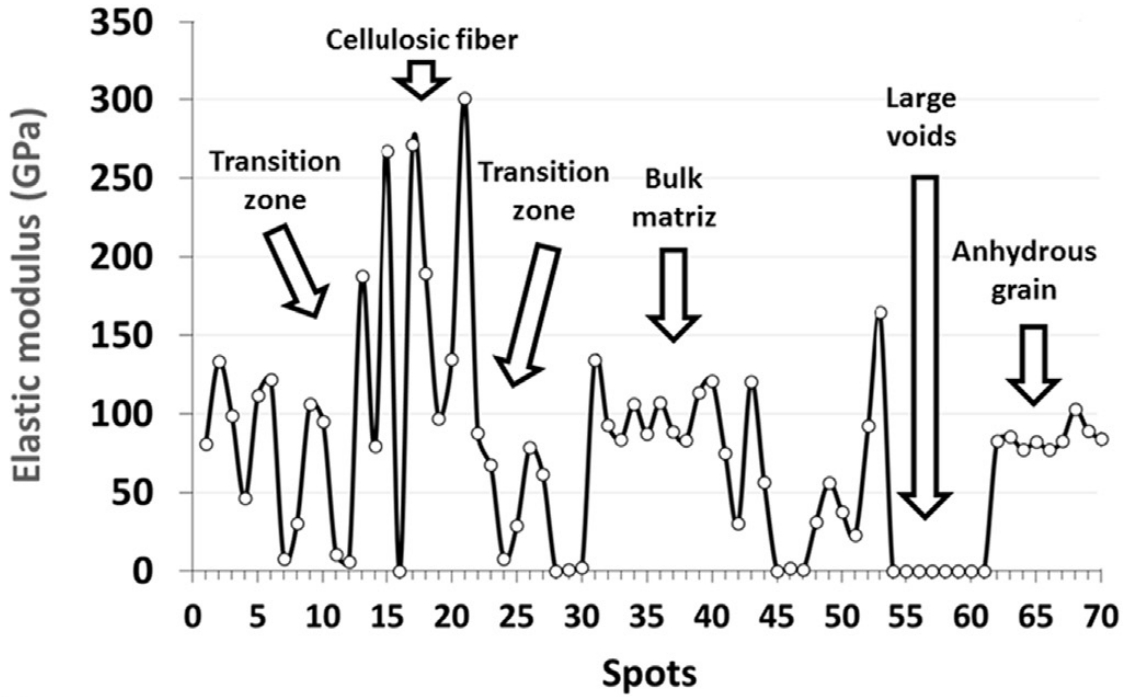
3.2. Mechanical characterization by nanoindentation

Figs. 5a and 6a present the micrographs of the nanoindentation imprints done along the bulk section of the composites. The spots numbered between 1 and 10 (Fig. 5a), 25-61 (Fig. 5a) and 81-90 (Fig. 6a) represent the bulk matrix of the composite. The spots 15-16 (Fig. 5a), 20-22 (Fig. 5a) and 101e110 (Fig. 6a) represents the interfacial zone. The spots 16-19 (Fig. 5a) and 91-100 (Fig. 6a) represents the cellulosic fiber, while 62-70 (Fig. 5a) and 81-90 (Fig. 6a) are spots related to anhydrous grains. The individual values of elastic modulus (E) acquired for each spot are

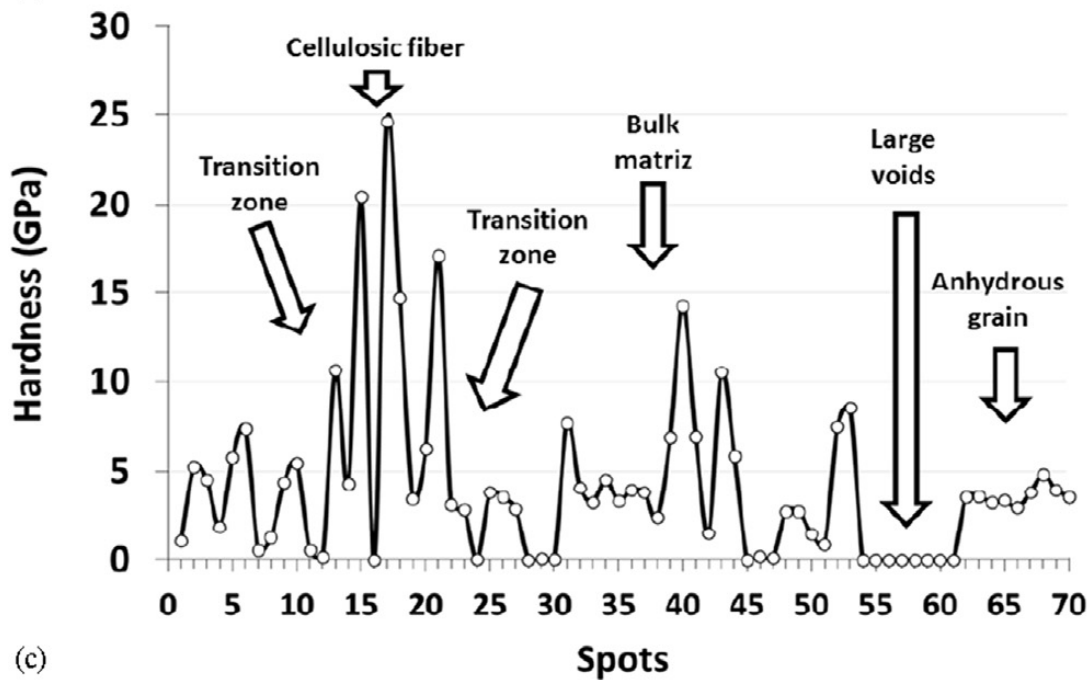
summarized in Figs. 5b and 6b; while the individual values of hardness (H) is presented in Figs. 5c and 6c.



(a)

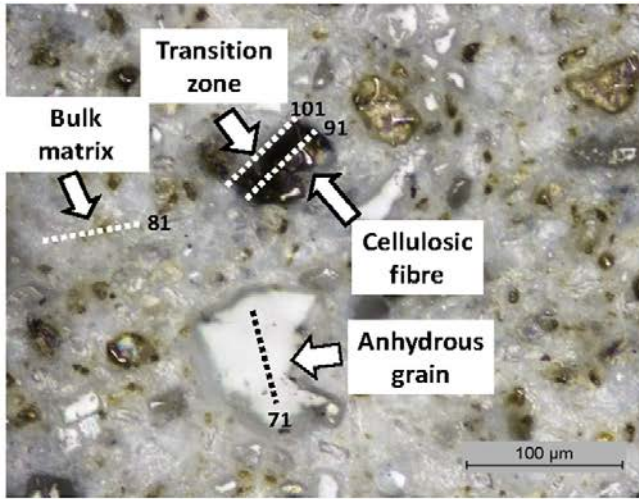


(b)

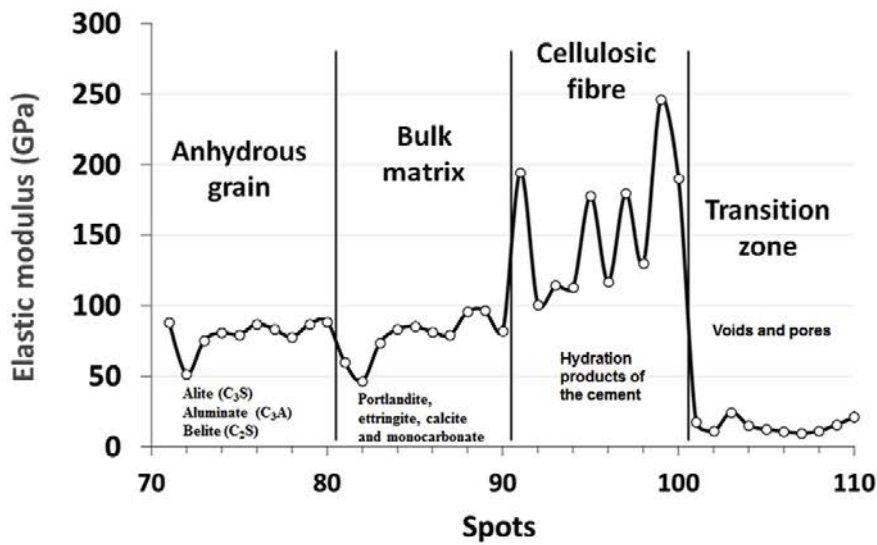


(c)

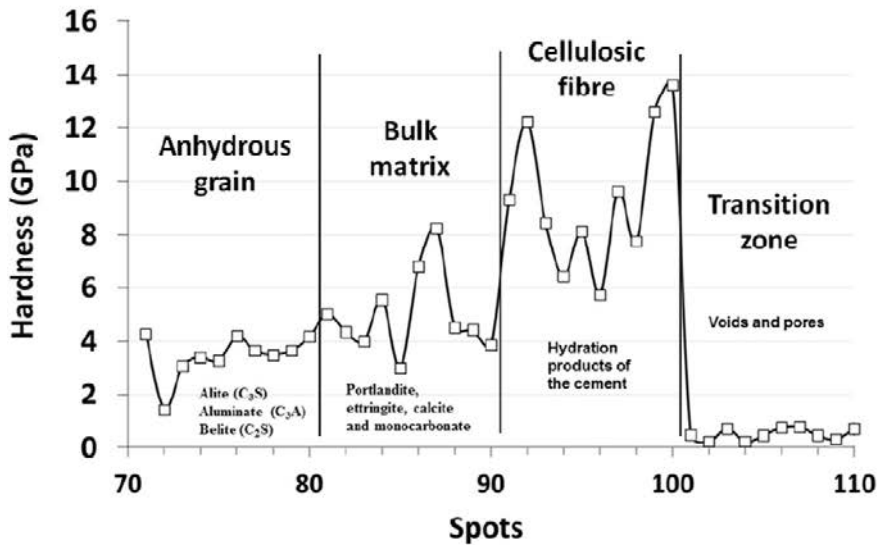
Fig. 5. (a) Light micrograph of the nanoindentation spots applied in line along the bulk section of the composite; (b) individual values of elastic modulus for each spot of the micrograph; and (c) individual values of hardness for each spot of the micrograph.



(a)



(b)



(c)

Fig. 6. (a) Light micrograph of the nanoindentation spots applied in line along the different structures (anhydrous grain, bulk density, cellulose fiber and interfacial/transition zone between fiber and matrix) of the composite; (b) individual values of elastic modulus for each spot of the micrograph; and (c) individual values of hardness for each spot of the micrograph.

Table 3 shows the average and standard deviation values of elastic modulus and hardness obtained from nanoindentation in the different regions studied. The indent spots in the bulk matrix led to elastic modulus of around 70 GPa and hardness of around 5 GPa. These results are similar to those found by Silva et al. [17], who studied nanoindentation in concrete specimens. The bulk matrix of composites based on Portland cement normally presents calcium silicate hydrates (C-S-H) and calcite phases as main components, which have higher rigidity and relative lower porosity than other phases such as Ca(OH)₂ and ettringite that are formed by poorly packed structures. This is why the spots correspondent to the interfacial/transition zone presents lower elastic modulus (~17 GPa) and hardness (~0.4 GPa). These results of the elastic modulus and hardness vary in relation to the distance of the transition zone, as showed in Fig. 5. Similar result was found by Wang et al. [30] that studied nanoindentation technique in concrete reinforced with steel fibers. The values of hardness in the 11e12, 23e26 indent spots near to the interfacial zone are smaller than those in the matrix region, which are also related to the higher porosity in the interfacial zone, resultant of the low adhesion between fiber and matrix [31,32].

Table 3 Average and standard deviation values of elastic modulus and hardness obtained for each structure of the composite by nanoindentation technique.^A

Structures	Elastic modulus (GPa)	Hardness (GPa)
Interfacial/transition zone	17.2 ± 5.6 c	0.4 ± 0.2 c
Bulk matrix	72.7 ± 14.3 b	5.3 ± 1.8 b
Anhydrous grain	89.9 ± 17.1 b	4.5 ± 1.9 b
Cellulosic fiber	144.3 ± 48.9 a	8.9 ± 2.5 a

^A Lowercase letters (a, b and c) in the same column represent comparisons between conditions of the composites. Different letters indicate significant differences amongst the properties of the composites.

Other important observation from this study in nanoscale level is related to the hydration of the cement grains in the composite. The incorporation of water during the mixing of the composite constituents leads to cement hydrate grains that have much lower average atomic numbers than cement grains that were not completely hydrated (anhydrous grains). Then, strong contrast is obtained between unreacted (anhydrous) and reacted material (hydrates) [33]. Anhydrous grain shown in Table 3 presents elastic modulus of around 90.0 GPa and hardness of around 4.5 GPa, which was not significantly different from average values of hydrated bulk matrix. Velez et al. [34] also studied nanoindentation in cement and did not find differences between hardness and elastic modulus in anhydrous phases such as alite (C₃S), calcium aluminate (C₃A) and belite (C₂S).

The highly alkaline porewater within the fiber-cement interface can induce the cellulose fibers stiffening. It occurs because cellulose fibers are hydrophilic and absorb water rich in cement hydration products, leading to the mineralization phenomenon proposed elsewhere [35-37]. This phenomenon is related to hydrophilic nature of the cellulose fibers. Therefore, this phenomenon is related to the hydrophilic nature of the cellulose fibers. This nature depends on surface nature, surface porosity and lumens the inner of the cellulose fiber. The internal humidity in the fiber can promote a continued hydration process, with precipitation and reprecipitation internally, such as C-S-H, portlandite and calcites, as suggest by Fig. 7a and b. Fig. 7a shows hydration products within the cell wall of cellulosic fibers, while the EDS spectra (Fig. 7b) in the spots 1 (upper) and 2 (bottom) proves the large amounts of Si and Ca into the cellulose fibers. It explains the higher values of elastic modulus (~144 GPa) and hardness (~9 GPa) observed for the nanoindentation spots related to the cellulosic fiber regions. Therefore, the average elastic modulus of mineralized cellulose fibers can reach higher values than cement bulk matrix. The increase of stiffness of the cellulose fibers is a drawback because can induce the loss of their reinforcement capacity. The efficiency of stress transfer from matrix to fiber that is of fundamental importance in determining the mechanical properties of short fibre reinforced composite is prejudiced. This result explain the premature loss of ductility/toughness and increase of stiffness of composites reinforced with cellulose fibers widely reported in literature [22,23,38,39].

Therefore, the fiber to matrix bonding is impaired because cellulose fiber tends to exchange water with the matrix system by capillary action, leading to high dimensional variation and detachment

of the fiber surface from the matrix, as showed in Fig. 7a, creating voids that explain the low results of elastic modulus and hardness in the transition zone. Finally, the present observations may state that an optimal situation would be the cellulose fibers protection from excessive water absorption, what can be achieved by less hydrophilic surfaces (either natural or after an adequate treatment) and without losing the fiber bridging. Fiber bridging is important to guarantee the expend of energy during fiber pullout, which leads to composite ductility.

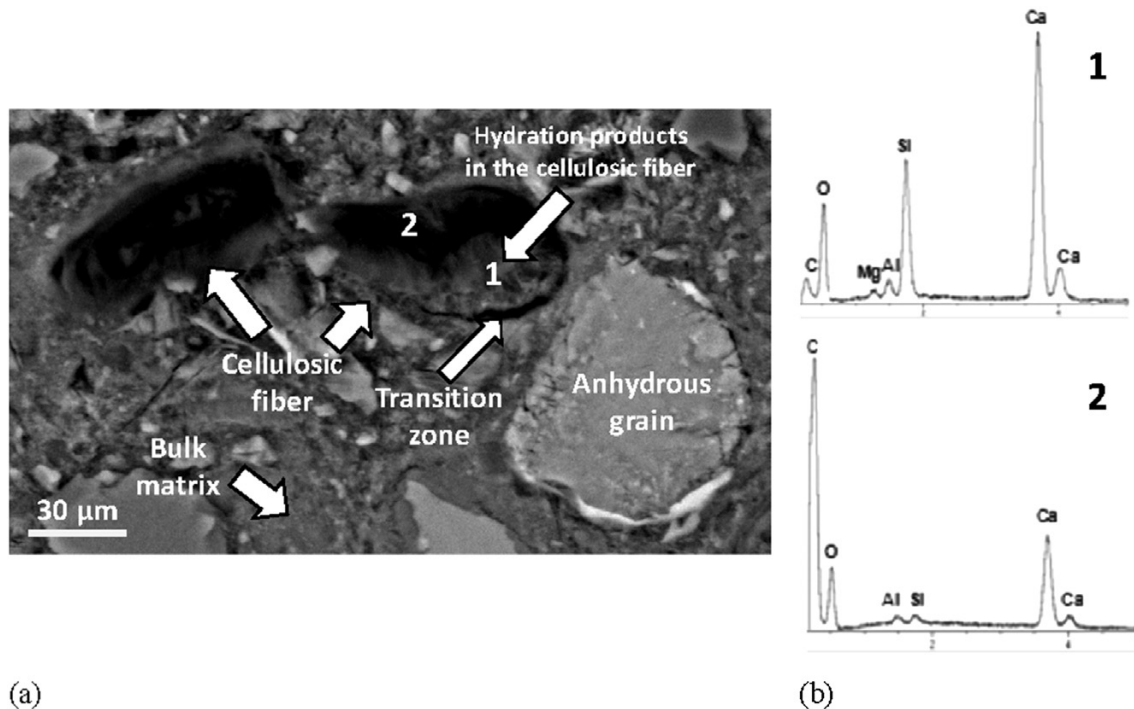


Fig. 7. (a) Scanning electron microscopy (SEM) micrographs in back scattering electron (BSE) mode of the polished cross-section of an extruded composite; and (b) energy dispersive spectra (EDS) of the spots 1 and 2 in the SEM micrographs.

4. Conclusions

High values of elastic modulus and hardness, 144 ± 48.9 GPa and 9 ± 2.5 GPa, respectively, was found by means of nanoindentation imprints in the cellulose fiber, because of the majority presence of C-S-H, portlandite and calcite phases. Then, cellulose fibers absorb water rich in cement hydration products, which induce the cellulose fiber stiffening by the precipitation and re-precipitation of hydration products into the fiber, these phenomenon lead to the so-called mineralization process. The increase of stiffness of the inner part of the cellulose fibers is a drawback because can induce the loss of their reinforcement capacity. The present finding contributes to better understanding the interface properties between fiber and matrix that affect cellulose fiber-cement materials performance, in terms of macroscopic mechanical properties. It is difficult to obtain the effective elastic modulus these composites because the low adhesion between the hydrophilic cellulose fibers and cement matrix, mineralization of the cellulose fiber and consequently a significant difference of elastic modulus between matrix, interface transition zone (ITZ), and cellulose fiber. These results indicate a clear modification of the contribution of the cellulosic fiber in the stress distribution and interfacial bond strength in the cement based composite. Therefore, it is still a challenge to modify nano and micro structures of the ITZ in order to improve mechanical properties in macro scale of the extruded fiber-cement composites.

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