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Additional Information

1 **“Plasticization effect of epoxidized cottonseed oil (ECSO) on poly(lactic acid)”**

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9
10 **ABSTRACT**

11 In this work, the use of an environmentally friendly plasticizer derived epoxidized
12 cottonseed oil (ECSO) for poly(lactic acid) (PLA) is proposed. Melt extrusion was used
13 to plasticize PLA formulations with different ECSO contents in the 0 - 10 wt.%. PLA
14 formulation with 10 wt.% shows a remarkable increase in mechanical ductile properties
15 with a percentage increase in elongation at break of more than 1100% and a noticeable
16 increase in the impact absorbed energy. Differential scanning calorimetry (DSC) and
17 dynamic mechanical thermal analysis (DMTA) revealed a clear decrease in the glass
18 transition temperature of neat PLA as the ECSO content increased. Field emission
19 scanning electron microscopy (FESEM) of fractured surfaces from impact tests showed
20 an improvement of ductility with typical rough and porous topographies. Migration tests
21 in *n*-hexane at different temperatures revealed very low migration properties thus leading
22 to new interesting plasticizers for improved PLA industrial formulations.

23
24 **Keywords:** poly(lactic acid)-PLA; epoxidized cottonseed oil (ECSO); mechanical
25 properties; thermal properties; migration.

26

27 **1.- Introduction.**

28 One of the most promising biopolymers as alternative to conventional petroleum-
29 based polymers is poly(lactic acid)-PLA with an annual production of more than 140,000
30 tons (Grande et al., 2015). PLA is an aliphatic polyester obtained by polymerization of
31 lactic acid (hydroxyl propionic acid) obtained from renewable and sustainable starch rich
32 materials such as corn and sugarcane (Al-Mulla et al., 2011; Blanco and Siracusa, 2013;
33 Sansone et al., 2012) . In the last years PLA has become an industrial alternative to some
34 petroleum-based polymers because of its relatively low price (Gordobil et al., 2015) and
35 overall balanced mechanical properties such as high tensile strength and Young's
36 modulus with similar values of those of poly(ethylene terephthalate), PET (Chieng et al.,
37 2014) which is widely used in plastic bottle manufacturing due to its excellent barrier
38 properties. PLA possesses acceptable barrier properties, and additionally it can be
39 transparent due to the low crystallization rate (Chieng et al., 2014). It is also shiny and
40 offers low flammability; all these features, together with a relatively easy processing
41 conditions, similar to many commodities, make PLA a good candidate for a wide variety
42 of products in the packaging industry, automotive parts, textile fibers, prostheses and
43 medical devices, etc. among others (Morelli et al., 2015; Murariu et al., 2008) .
44 Nevertheless, PLA is characterized by high fragility, which is drawback for some
45 technical applications in which some flexibility is required (Wang et al., 2015). For these
46 reasons, different approaches have been explored to overcome this high intrinsic fragility
47 by increasing ductile properties such as elongation at break, impact resistance, etc. while
48 maintaining its environmentally friendly nature. A typical approach has been blending
49 PLA with other ductile polymers. In this field, PLA was blended with chitosan, which is
50 able to form films but the mixtures resulted in immiscible blends due to their different

51 polarity and poly(vinyl alcohol), PVA was necessary to provide somewhat compatibility
52 (Grande et al., 2015). Thermoplastic starch (TPS) also showed immiscibility with PLA
53 and several compatibilizers such as maleic anhydride and methylene diphenyl
54 diisocyanate (MDI) were needed to improve the overall properties (Clasen et al., 2015;
55 Mittal et al., 2015; Yang et al., 2015). Blends with poly(hydroxybutyrate), PHB gave an
56 interesting improvement on barrier properties of neat PLA but resulting blends were
57 characterized by high fragility so that, different plasticizers such as acetyl(tributyl citrate),
58 ATBC (Arrieta et al., 2015) and poly(ethylene glycol), PEG (Courgneau et al., 2011)
59 were needed to overcome this drawback. PLA has also been blended with biodegradable
60 petroleum-based polymers to give fully biodegradable blends. Among these petroleum-
61 based polymers interesting results have been obtained with poly(caprolactone)-PCL
62 biodegradable polyester. Although both PLA and PCL are polyester-type polymers, they
63 show restricted miscibility but the high flexibility of PCL is enough to reduce the intrinsic
64 fragility of PLA (Carmona et al., 2015; Ferri et al., 2016a; Tabasi et al., 2015) . Other
65 petroleum-based polymers such as poly(butylene succinate)-PBS (Deng and Thomas,
66 2015) , poly(butylene succinate-*co*-adipate)-PBSA (Pivsa-Art et al., 2015), poly(butyl
67 acrylate)-PBA (Meng et al., 2012) and poly(butylene adipate-*co*-terephthalate)-PBAT
68 (Arruda et al., 2015; Kumar et al., 2010) have been successfully blended with PLA with
69 remarkable increase in flexibility.

70 Another way to improve the flexibility of neat PLA is by using plasticizers.
71 Among the wide variety of commercial plasticizers, most of them derived from
72 petroleum, epoxidized vegetable oils (EVOs) represent an interesting alternative as they
73 are cost-effective materials with high performance as plasticizers and are obtained from
74 renewable resources, mainly from vegetable oils with non-food purposes. Epoxidized oils
75 are used in the PVC industry as secondary plasticizers and thermal stabilizers due to their

76 free radical scavenging properties (Carbonell-Verdu et al., 2016; Fenollar et al., 2009;
77 Mehta et al., 2014). In the last years, the potential of different epoxidized vegetable oils
78 has been explored. Some epoxidized oils such as epoxidized soybean oil (ESBO) and
79 epoxidized linseed oil (ELO) are commercially available at a relatively low cost and,
80 consequently, they can be readily employed in industrial applications. Epoxidized
81 soybean oil (ESBO) contains an oxirane oxygen content comprised in the 6.5 – 8% range
82 and it has been reported as an interesting plasticizer for PLA. The study developed by
83 Yu-Qiong Xu et al. revealed that addition of 6 wt.% ESBO to PLA increased the
84 elongation at break from 3.98% for neat PLA to values of 6.50%. Similar findings were
85 reported by Shalini Vijayarajan et al. They evaluated the influence of ESBO content (up
86 to 20 wt.%) on ductility, measured as the ratio between the failure to yield strain. They
87 found the best overall ductile properties for ESBO contents in the 5-10 wt.% range
88 (Vijayarajan et al., 2014; Xu and Qu, 2009) . With regard to epoxidized linseed oil (ELO),
89 which contains more average number of oxirane groups per triglyceride than ESBO,
90 Javed Alam et al. reported the plasticizing effect provided by ELO with a remarkable
91 increase in elongation at break and a subsequent decrease in tensile strength. Addition of
92 carbon nanotubes contributed to balanced mechanical ductile and resistant properties
93 (Alam et al., 2014). Interesting results with low oxirane oxygen content epoxidized oils
94 have also been reported. Buong Woei Chieng et al. reported the potential of epoxidized
95 palm oil (EPO) with an oxirane oxygen content of 3.23% and a mixture of epoxidized
96 palm oil and epoxidized soybean oil with an average oxirane oxygen content of 3.58%,
97 showing that 5 wt.% of the epoxidized oil leads to a remarkable increase in ductile
98 properties. Furthermore, the mixture with higher oxirane oxygen content led to the highest
99 elongation at break values (Chieng et al., 2014).

100 Cottonseed oil (CSO) is considered a by-product of the cotton industry with a total
101 production around 5.12 million tons in 2014/15 (Trade, March 2016), being China and
102 India the major world producers with 1.396 and 1.320 million tons respectively. Although
103 some uses in the food industry are common, its generalized use is restricted because of
104 some drawbacks: on the one hand, it is worth to note that the main product of the cotton
105 industry is cotton fiber so that, cotton crops are not subjected to the severe controls and
106 regulations regarding the use of pesticides and other chemicals. On the other hand,
107 cottonseed oil contains high amounts of free gossypol which has been reported as a
108 potentially toxic substance (Gadelha et al., 2014). The particular lipid profile of cotton
109 seed oil, with approximately 75% of unsaturated fatty acids makes this oil an interesting
110 candidate for chemical modification such as epoxidation (Carbonell-Verdu et al., 2015).
111 The oxirane oxygen content of epoxidized cottonseed oil (ECSO) is located between low
112 values typical of epoxidized palm oil (EPO) and high values typical of epoxidized
113 soybean oil (ESBO) and epoxidized linseed oil (ELO). For this reason, this work aims to
114 study the potential of epoxidized cottonseed oil (ECSO) with an intermediate oxirane
115 oxygen content as environmentally friendly plasticizer for poly(lactic acid), PLA. The
116 effect of the epoxidized oil content on mechanical, thermal and migration properties of
117 PLA were tested.

118

119 **2.- Material and methods.**

120 **2.1.- Materials.**

121 Poly(lactic acid), PLA commercial grade Ingeo TM Biopolymer 6201D was
122 supplied by NatureWorks LLC (Minnetonka, USA) in pellet form with a density of 1.24
123 g cm⁻³. Its melt flow index is in the 15-30 g/10 min range measured at a temperature of
124 210 °C. The base oil for plasticizer synthesis was cottonseed oil (CSO). This was supplied

125 by Sigma Aldrich (Sigma Aldrich, Madrid, Spain). This oil is characterized by a density
126 0.92 g cm^{-3} at $25 \text{ }^\circ\text{C}$ and an iodine number of 107. This vegetable oil was subjected to an
127 epoxidation process as indicated elsewhere with hydrogen peroxide and acetic acid
128 (Carbonell-Verdu et al., 2015). After epoxidation, the oxirane number of the synthesized
129 epoxidized cottonseed oil was 5.32% and its iodine number was 1.79. **Fig. 1** shows an
130 schematic representation of the chemical structures of poly(lactic acid) and epoxidized
131 cottonseed oil.

132

133 **Figure 1.-** Schematic representation of the chemical structure of a) poly(lactic acid)-
134 PLA and b) epoxidized cottonseed oil (ECSO).

135

136 **2.2.- Manufacturing of plasticized poly(lactic acid).**

137 Firstly, PLA pellets were dried at $60 \text{ }^\circ\text{C}$ for 24 h. PLA pellets and the corresponding
138 amounts of the liquid plasticizer (see **Table 1**) were mechanically mixed in a zip bag until
139 homogenization. After this stage, the mixtures were extruded in a twin screw extruder at
140 a constant speed of 40 rpm. The temperature profile ranged from $170 \text{ }^\circ\text{C}$ (feeding zone)
141 to $180 \text{ }^\circ\text{C}$ (die). Then the extrudate was pelletized and subsequently shaped into standard
142 samples in an injection molding machine Meteor 270/75 (Mateu & Solé, Barcelona,
143 Spain) at an injection temperature of $180 \text{ }^\circ\text{C}$.

144

145 **Tabla 1.-** Composition of ECSO plasticized PLA materials and labelling.

146

147

148 **2.3.-Mechanical characterization of ECSO plasticized PLA.**

149 Mechanical characterization was conducted with standard tensile, flexural and
150 impact tests. Tensile and flexural characterization was carried out in a universal test
151 machine Ibertest Elib 30 (Ibertest S.A.E., Madrid, Spain). A minimum of five different
152 samples were tested using a 5 kN load cell. Tensile tests were carried out at a crosshead
153 speed of 10 mm min⁻¹ as recommended by the ISO 527 standard. An axial extensometer
154 from Ibertes was used to give accurate values of the Young's modulus. Regarding the
155 flexural test, the crosshead speed was set to 5 mm min⁻¹ as suggested by the ISO 178. The
156 impact absorbed energy was measured in a 6 J Charpy's pendulum (Metrotec S.A., San
157 Sebastián, Spain) following the guidelines of the ISO 197:1993.

158

159 ***2.4.- Morphology of ECSO plasticized PLA.***

160 Fractured surfaces from impact tests were observed by field emission scanning
161 electron microscopy (FESEM) in a Zeiss Ultra microscope 55 (Oxford Instruments,
162 Oxfordshire, United Kingdom) with an accelerating voltage of 2 kV. Fractured surfaces
163 were previously coated with a thin platinum layer in a sputter coater EM MED020 (Leica
164 Microsystems)

165

166 ***2.5.- Thermomechanical characterization of ECSO plasticized PLA.***

167 Thermomechanical characterization was conducted by measuring the Vicat
168 softening temperature (VST), heat deflection temperature (HDT). VST and HDT values
169 were determined in a DEFLEX 687-A2 station (Metrotec S.A., San Sebastián, Spain).
170 VST was measured according to ISO 306 (B method) at a constant heating rate of 50 °C
171 h⁻¹ and a load of 50 N. Regarding HDT values, they were obtained as recommended by
172 the ISO 75 (A method) at a fixed heating rate of 120 °C h⁻¹ and a constant load of 1.8
173 MPa. Additionally, dynamic mechanical thermal analysis (DMTA) was used to evaluate

174 changes in storage modulus (G') and damping factor. Samples sizing $40 \times 40 \times 4 \text{ mm}^3$ were
175 tested in torsion mode in an oscillatory rheometer AR G2 (TA Instruments, New Castle,
176 USA) equipped with an accessory clamp for solid samples. Samples were subjected to a
177 thermal program from $25 \text{ }^\circ\text{C}$ to $130 \text{ }^\circ\text{C}$ at a heating rate of $2 \text{ }^\circ\text{C min}^{-1}$. The frequency was
178 set to 1 Hz and the maximum deformation (γ) was 0.1%.

179

180 **2.6.- Thermal characterization of ECSO plasticized PLA.**

181 Thermal behavior of ECSO plasticized PLA samples was tested by differential
182 scanning calorimetry (DSC) and thermogravimetric analysis (TGA). Differential
183 scanning calorimetry tests were carried out in a Mettler-Toledo calorimeter 821e
184 (Mettler-Toledo Inc., Schwerzenbach, Switzerland). Samples with an average weight in
185 the 7-8 mg range were subjected to the following thermal program: initial heating from
186 $30 \text{ }^\circ\text{C}$ to $210 \text{ }^\circ\text{C}$ at a heating rate of $10 \text{ }^\circ\text{C min}^{-1}$ in nitrogen atmosphere with a nitrogen
187 flux of 66 mL min^{-1} . The glass transition temperature (T_g), cold crystallization peak (T_{cc})
188 and the melt peak temperature (T_m) were obtained for each plasticized PLA formulation.
189 The degree of crystallinity (X_c %) was calculated by Equation 1:

190

$$191 \quad X_c (\%) = 100 \times \frac{|\Delta H_{cc} + \Delta H_m|}{\Delta H_m(100\%)} \times \frac{1}{W_{PLA}} \quad (1)$$

192

193 Where ΔH_{cc} is the cold crystallization enthalpy, ΔH_m is the melt enthalpy, $\Delta H_m(100\%)$
194 is the theoretical melt enthalpy for a fully crystalline PLA structure (93 J g^{-1}) (Arrieta et
195 al., 2014) and W_{PLA} is the PLA weight fraction.

196 The initial heating step was applied to remove the previous thermal history of the
197 materials. Thermogravimetric analysis was carried out in a TGA/SDTA 851
198 thermobalance from Mettler-Toledo (Mettler-Toledo Inc., Schwerzenbach, Switzerland).

199 Samples with an average weight of 10 mg were subjected to a heating program from 30
200 °C to 700 °C at a heating rate of 20 °C min⁻¹ and constant nitrogen flux (66 mL min⁻¹).

201

202 ***2.7.- Plasticizer migration by the solvent extraction test.***

203 Plasticizer migration was studied by solvent extraction with n-hexane. Samples
204 were immersed in n-hexane and placed in an air circulating oven mod. Selecta 2001245
205 (JP Selecta S.A., Barcelona, Spain) working at different temperatures, between 30 °C and
206 60 °C during 8 hours. Finally, the samples were removed from n-hexane to measure
207 weight loss.

208

209 **3.- Results and discussion.**

210 ***3.1.- Mechanical properties of ECSO plasticized PLA formulations.***

211 The study of the effect of the epoxidized cottonseed oil (ECSO) on mechanical
212 properties of PLA-based formulations is a good method to assess the plasticization that
213 ECSO provides. PLA is a brittle polymer with relatively high tensile and flexural strength
214 values, high modulus and very low elongation at break. **Table 2** summarizes the main
215 results obtained by tensile and flexural tests. Neat PLA possesses a tensile strength of
216 63.7 MPa with very low elongation at break values around 9%. As the elastic modulus is
217 defined by the ratio strength to elongation in the linear region, high strength with very
218 low elongation give high modulus of 3.6 GPa. The plasticization effect is evident by
219 observing the changes in both mechanical resistant and ductile properties. Plasticized
220 PLA with 2.5 wt.% ECSO is characterized by slightly higher elongation at break values
221 of 13.5% and a subsequent decrease in both tensile strength and Young's modulus with
222 values of 58.5 MPa and 3.3 GPa respectively. As the total plasticizer content in PLA
223 formulations increases it is clearly detectable a remarkable improvement in ductile

224 properties by a dramatic change in elongation at break up to values of 110.5% for PLA
225 formulations containing 10 wt.% ECSO which is approximately eleven times higher than
226 the elongation at break of unplasticized PLA (9%). This represents an overall percentage
227 increase of around 1128% regarding to neat PLA, calculated as the percentage ratio of
228 the variation in elongation at break (110.5% - 9%) to the elongation at break of neat PLA
229 (9%). These results are in accordance with previous works regarding PLA plasticization
230 with epoxidized vegetable oils. Buong Woei Chieng *et al.* reported a percentage increase
231 in elongation at break of more than 2000% with the only addition of 5 wt.% epoxidized
232 palm oil (EPO) to PLA formulations. In fact, the elongation at break changed from 5.3%
233 (neat PLA) up to 114.4% for the corresponding EPO-plasticized formulation. Even better
234 results were obtained in PLA formulations plasticized with a mixture of epoxidized palm
235 and soybean oil (EPSO) with an elongation at break of 220.5%. They also reported a
236 noticeable drop on tensile strength values from values close to 60 MPa for neat PLA up
237 to values of 35 MPa for plasticized PLA formulations with an epoxidized palm oil (EPO)
238 content between 5 – 10 wt.%. This dramatic drop in tensile strength is directly related to
239 polymer-plasticizer interactions. The epoxy groups contained in the epoxidized vegetable
240 oil can interact with the hydroxyl groups located in the end chains of PLA. As they
241 suggest, these interactions are stronger as the oxirane oxygen content increases (Chieng
242 *et al.*, 2014). Their findings are in total accordance with the plasticizing effect that
243 epoxidized cottonseed oil (ECSO) provides. ECSO is characterized by a medium oxirane
244 oxygen content (5.32%) compared to epoxidized palm oil (EPO) with low values (3.23%)
245 and epoxidized linseed oil (ELO) with high values around 10%. As it can be observed in
246 **Table 2**, the tensile strength of neat PLA (63.7 MPa) is reduced to values close to 50 MPa
247 with the maximum ECSO content. This drop in tensile strength is lower than that observed
248 by Buong Woei Chieng *et al.* with epoxidized palm oil (EPO). The higher oxirane oxygen

249 content in ECSO can interact more strongly with PLA chains leading to intense polymer-
250 plasticizer interactions that provide balanced tensile strength and elongation at break
251 values. Regarding the Young's modulus, the initial value corresponding to unplasticized
252 PLA is 3.6 GPa and this is slightly reduced up to values of 3.3 GPa for plasticized
253 formulations with varying ECSO content. Although the plasticization effect of ECSO and
254 other epoxidized vegetable oils is evident through a remarkable increase in elongation at
255 break values. It has been reported that once the optimum plasticizer content has reached,
256 a plasticizer excess leads to lower elongation at break values due to a possible phase
257 separation (Ferri et al., 2016b). Although the values are not summarized in **Table 2**, a
258 plasticized PLA formulation was also prepared with 15 wt.% ECSO. The elongation at
259 break of this formulation was similar to that obtained for the plasticized PLA formulation
260 with 5 wt.% thus giving a clear evidence of plasticizer saturation which has a negative
261 effect on ductile properties due to phase separation.

262

263 **Table 2.** – Summary of mechanical properties from tensile and flexural tests of
264 plasticized PLA formulations with different weight % of epoxidized cottonseed oil
265 (ECSO).

266

267 Similar behaviour has been found for flexural tests. As it can be observed in Table
268 2 the flexural strength of neat PLA (116.3 MPa) is progressively reduced up to values of
269 78.6 MPa and 53.3 MPa for plasticized PLA formulations containing 5 wt.% and 10 wt.%
270 ECSO respectively. Although slight changes can be produced on flexural modulus, these
271 changes are in the typical error range so that, no clear tendency can be observed.

272 As above mentioned, the addition of ECSO provides a clear plasticizing effect on
273 PLA with a remarkable increase in mechanical ductile properties such as elongation at

274 break. Besides this, ECSO plasticized PLA formulations are remarkably toughened as it
275 can be seen in **Fig. 2**. The high intrinsic brittleness of neat PLA is reduced due to the
276 toughening and elastomeric effect provided by the ECSO plasticizer. An increasing
277 tendency of impact absorbed energy can be seen in **Fig. 2**. Neat PLA possesses an impact
278 energy of 30.8 kJ m^{-2} and this is progressively increased up to values of 38.42 kJ m^{-2} for
279 the plasticized PLA formulation with 10 wt.% ECSO which represents a percentage
280 increase of 25% regarding neat PLA. As shown by tensile and flexural tests, addition of
281 ECSO promotes a remarkable increase in elongation and this has a positive effect on
282 energy absorption.

283

284 **Figure 2.-** Plot evolution of the impact absorbed energy measured by the Charpy's test
285 for plasticized PLA formulations with different weight % of epoxidized cottonseed oil
286 (ECSO).

287

288 The effect of epoxidized cottonseed oil (ECSO) on the morphology of plasticized
289 PLA formulations can be observed in **Fig. 3** which shows FESEM images of fractured
290 surfaces from impact tests samples. **Fig. 3a** shows the typical fracture surface of a brittle
291 polymer with a flat and smooth surface and no evidence of plastic deformation. For low
292 ECSO plasticizer content of 2.5 wt.% (**Fig. 3b**) a homogeneous surface can be observed
293 which is representative for good compatibility between PLA and ECSO plasticizer that is
294 uniformly distributed in the PLA matrix. In addition to this, other plasticization signs can
295 be detected in the form of fibrils resulting from plastic deformation during a sudden
296 impact. This phenomenon is in total accordance with the previous mechanical properties.
297 Plasticized PLA formulations with very low amounts of ECSO plasticizer are
298 characterized by a relative low increase in elongation at break. This fact could be related

299 to strong interactions between PLA polymer chains and the epoxidized cottonseed oil.
300 These strong interactions occur because no phase separation occurs as observed in **Fig.**
301 **3b**. As the weight % of ECSO increases some spherical cavities/voids can be observed.
302 These are produced by a cavitation process caused by debonding. The empty microvoids
303 indicate presence of an epoxidized cottonseed oil rich phase dispersed in the PLA matrix
304 that becomes more evident with increasing ECSO content. Presence of fibrils is less
305 evident as the ECSO content increases. In general, FESEM reveals good miscibility
306 between PLA and ECSO plasticizer for very low plasticizer content as observed by V. S.
307 Giita Silverajah et al (Giita Silverajah et al., 2012). By increasing the plasticizer content,
308 phase separation is more evident but elongation at break is highly improved until
309 plasticizer saturation occurs. Over 10 wt.% ECSO, plasticizer saturation occurs and
310 mechanical properties are not improved.

311

312 **Figure 3.-** FESEM images of fractured surfaces from impact tests corresponding to
313 plasticized PLA formulations with different weight % of epoxidized cottonseed oil
314 (ECSO) at 2000x: a) neat PLA, b) 2.5 wt.% ECSO, c) 5.0 wt.% ECSO, d) 7.5 wt.%
315 ECSO and e) 10 wt.% ECSO.

316

317 **3.2.- Thermal properties of ECSO plasticized PLA formulations.**

318 Effective plasticization promotes important changes in thermal transitions, mainly
319 the glass transition temperature (T_g) due to increased polymer chain mobility. **Fig. 4 and**
320 **Fig. 5** show the effect of the ECSO plasticizer on dynamic mechanical response of
321 plasticized PLA formulations. As it can be seen in **Fig. 4**, the storage modulus (G') shows
322 a flat plot from room temperature up to 50 °C. Then the storage modulus undergoes a
323 drops of nearby three orders of magnitude and tends to stabilize at about 70 °C. This

324 abrupt drop is directly related to the glass transition temperature (T_g) (Chieng et al., 2014).
325 As the temperature raises and reaches values comprised in the 80 – 90 °C range, a new
326 increase in storage modulus can be detected which is related to the cold crystallization
327 process. Regarding samples with different loads of ECSO plasticizer, the storage modulus
328 curves are identical in shape but the typical values are shifted to lower temperatures thus
329 indicating a decrease in both the glass transition temperature and cold crystallization
330 process. The storage modulus of neat PLA below the glass transition temperature is close
331 to 1.4 GPa. All plasticized formulations show a slight decrease in the storage modulus in
332 this initial stage with typical values of 1.1 GPa thus indicating a clear plasticization effect.
333 Plasticizer increases chain mobility and this has a positive effect on lowering both glass
334 transition temperature and cold crystallization. The cold crystallization represents the
335 realignment of PLA chains to form a more packed structure. It is evident that the
336 plasticizer enables this rearrangement as it increases polymer chain mobility. The cold
337 crystallization temperature changes from 81.7 °C for neat PLA to values of 78 °C for
338 ECSO plasticized PLA formulations.

339

340 **Figure 4.-** Plot evolution of the storage modulus (G') as function of temperature for
341 plasticized PLA formulations with different weight % of epoxidized cottonseed oil
342 (ECSO).

343

344 By observing the evolution of the damping factor ($\tan \delta$) it is possible to detect
345 two relaxations located at 50-70 °C and 80-90 °C which are attributed to the glass
346 transition (T_g) and the cold crystallization (T_{cc}) temperatures respectively. The efficiency
347 of the ECSO plasticizer can be assessed by measuring the changes in the glass transition
348 temperature (T_g) (Liu and Zhang, 2011). Although the glass transition occurs in a

349 temperature range, maximum values of the damping factor were taken as representative
350 values (see **Fig. 5**). The T_g of neat PLA is 66.2 °C. As the ECSO content increases, a
351 decreasing tendency in T_g values can be observed up to values of 63.4 °C and 62.5 °C for
352 plasticized PLA formulations with 2.5 wt.% and 10 wt.% ECSO respectively.

353

354 **Figure 5.-** Plot evolution of the damping factor ($\tan \delta$) as function of temperature for
355 plasticized PLA formulations with different weight % of epoxidized cottonseed oil
356 (ECSO).

357

358 The plasticizing effect is also evident by observing the evolution of the Vicat
359 softening temperature (VST) and heat deflection temperature (HDT) as seen in **Table 3**.
360 Neat PLA is characterized by a HDT value of 47.6 °C and this is reduced up to values in
361 the 43 - 44 °C range for all compositions. With regard to the Vicat softening temperature
362 (VST) it changes from 52.8 °C for neat PLA up to values of 43 °C for the plasticized
363 formulation with 10 wt.% ECSO thus indicating a clear plasticization effect related to
364 increased polymer chain mobility.

365

366 **Table 3.-** Summary of thermomechanical properties, Vicat softening temperature (VST)
367 and heat deflection temperature (HDT) of plasticized PLA formulations with different
368 weight % of epoxidized cottonseed oil (ECSO).

369

370 As indicated previously, both the storage modulus (G') and the damping factor
371 ($\tan \delta$) obtained by dynamic mechanical thermal analysis revealed a decrease in the glass
372 transition temperature (T_g) of neat PLA with increasing the ECSO plasticizer content.
373 Differential scanning calorimetry (DSC) is a powerful technique that provides

374 information not only about the glass transition and cold crystallization processes but also
375 about the melt process. **Fig. 6** shows a comparative plot of the DSC heating thermograms
376 of neat PLA and plasticized PLA formulations with different ECSO content. The glass
377 transition temperature of neat PLA is 66.75 °C. The cold crystallization process is clearly
378 evident as an exothermic peak located in the 86 °C – 114 °C range with a maximum
379 crystallization rate located at 97 °C. Finally, the endothermic process between 156 °C –
380 177 °C corresponds to the melting of the crystalline phase in PLA. By observing the DSC
381 heating thermograms of the plasticized PLA formulations with different weight % of
382 ECSO plasticizer it is clearly evident a decrease in both the glass transition temperature
383 (T_g) and the cold crystallization temperature range (and its representative temperature
384 peak, T_{cc}). Thus, the plasticized formulation with low plasticizer content (2.5 wt.%
385 ECSO) possesses a glass transition and cold crystallization temperatures of 63.8 °C and
386 91.2 °C respectively. These values are still lower for PLA plasticized formulations with
387 higher plasticiser content. Therefore, PLA formulation with 10 wt.% ECSO is
388 characterized by a glass transition temperature of 60.7 °C and a cold crystallization peak
389 located at 86.95 °C. The decrease in the temperature range of the cold crystallization
390 process is directly related to the plasticizing effect that ECSO provides. The plasticizer
391 molecules, characterized by low molecular weight compared to polymer chains, diffuse
392 inside the polymer matrix and are placed between polymer chains. This situation leads to
393 an increase in the free volume that allows polymer chain motions to occur at lower
394 temperatures. Additionally, the polymer-polymer interactions are less strong because the
395 distance between polymer chains is increased by the plasticizer. Furthermore, in addition
396 to the dilution effect that ECSO provides on the cold crystallization enthalpy (ΔH_{cc}), these
397 values are lower because of the increased free volume that allows easy packing of the
398 plasticized structure. Similar findings were reported by V. S. Giita Silverajah et al. with

399 plasticized PLA with epoxidized palm oil (EPO). They concluded that good
400 miscibility/compatibility between PLA and EPO was obtained for very low EPO content
401 of 1 wt.%. They also reported a decrease in the glass transition of neat PLA for EPO
402 contents of 5 wt.% but they observed clear phase separation for these formulations (Giita
403 Silverajah et al., 2012). By taking into account the melt and cold crystallization
404 enthalpies, it is possible to assume that PLA possesses a degree of crystallinity (X_c %) of
405 20.5%. The presence of plasticizer enables chain mobility and this has a positive effect
406 on crystallization as polymer chains can move to a packed structure. The only addition of
407 2.5 wt.% ECSO leads to a degree of crystallinity of about 31.3% and similar values
408 around 32.3% are obtained for plasticized PLA formulations with 10 wt.% ECSO. Similar
409 findings were reported by Ferri et al. with plasticized PLA formulations with epoxidized
410 fatty acid esters (Ferri et al., 2016b).

411

412 **Figure 6.-** Comparative plot of the DSC heating thermograms of neat PLA and
413 plasticized PLA formulations with different weight % of epoxidized cottonseed oil
414 (ECSO).

415

416 Epoxidized vegetable oils also provide improved stability. The oxirane rings are
417 able to scavenge acid groups by catalytic degradation and this has a positive effect on
418 both thermal and light stabilization. In fact, one of the main commercial uses of
419 epoxidized vegetable oils is as secondary plasticizers and thermal and light stabilizers
420 (Mohammed et al., 2015).

421 **Table 4** shows the typical degradation temperatures for neat PLA and plasticized
422 PLA formulations with varying ECSO content. The onset degradation temperature has
423 been assessed by the temperature with a wt.% loss of 5% ($T_{5\%}$). In addition, the

424 temperature for a 50 wt.% loss has been assessed ($T_{50\%}$). The thermal stabilization effect
425 that epoxidized cottonseed oil provides can be seen by observing the evolution of the
426 onset degradation temperature. Neat PLA is characterized by a $T_{50\%}$ of 335.5 °C and this
427 is upward shifted to almost 340 °C for PLA formulations containing 10 wt.% ECSO.

428

429 **Table 4.**- Main degradation parameters obtained by thermogravimetric analysis (TGA)
430 for plasticized PLA formulations with different weight % of epoxidized cottonseed oil
431 (ECSO).

432

433 **3.3.- Plasticizer migration assessment on plasticized PLA formulations with ECSO.**

434 The potential migrants from poly(lactic acid) include lactic acid, the cyclic dimer
435 of lactic acid (lactide), the linear dimer of lactic acid (lactoyl lactic acid) and other
436 oligomers. Several studies have assessed PLA as a safe substance for food-contact and
437 lactic acid has been declared as a generally recognized as safe (GRAS) substance (Conn
438 et al., 1995). Nevertheless, most industrial PLA formulations include plasticizers and
439 other additives to tailor properties. For this reason, it is important to know the potential
440 plasticizer migration levels as it can restrict some uses in the food packaging industry.
441 The solvent extraction test is a quite aggressive test to measure to total migrated
442 plasticizer and gives interesting results about the potential use of plasticizers at industrial
443 scale. **Fig. 7** shows a comparative plot of the total plasticizer migrated in terms of
444 increasing temperature (all migrated amounts are referred at a migration time of 8 h). As
445 it can be observed, PLA does not show any relevant migration at 8 h in the temperature
446 range comprised between 30 °C and 60 °C with weight % migration values less than
447 0.02%. Regarding PLA formulations with epoxidized cottonseed oil (ECSO) the lowest
448 migration levels are obtained, as expected, at the lowest temperature considered in this

449 study (30 °C) with typical values of 0.03% and 0.06% for PLA formulations with 2.5
450 wt.% and 10 wt.% ECSO respectively. As temperature increases, the total amount of
451 migrated plasticizer increases as well but the maximum migration levels achieved are less
452 than 0.12% which indicates a very low plasticizer migration. It is important to note that
453 epoxidized vegetable oils are characterized by a molecular weight comprised between
454 850 g mol⁻¹ to 950 g mol⁻¹. Although these values are extremely lower compared to
455 poly(lactic acid) molecular weight, other monomeric plasticizers for PLA have lower
456 molecular weight, i.e. acetyl tributyl citrate, ATBC (402.5 g mol⁻¹) (Dobircau et al.,
457 2015), tributyl citrate, TbC (360.44 g mol⁻¹) (Tanrattanakul and Bunkaew, 2014), di-2-
458 ethylhexyladipate, DOA (371 g mol⁻¹) (Martino et al., 2009). For this reason, the migration
459 levels with epoxidized vegetable oils is extremely low compared to other monomeric
460 plasticizers. As expected, migrated plasticizer amounts increase with temperature with
461 similar values for all plasticized PLA formulations. These relative low migration levels
462 indicate good plasticizer compatibility as indicated previously. Interaction of epoxy
463 groups present in epoxidized cottonseed oil (ECSO) with hydroxyl groups in poly(lactic
464 acid) leads to a plasticization effect and potential chain extension phenomenon. These
465 strong interactions positively contribute to low migration levels thus indicating the
466 feasibility of using these plasticized formulations at industrial scale.

467

468 **Figure 7.-** Plasticizer migration levels by the solvent extraction test on plasticized PLA
469 formulations with different weight % of epoxidized cottonseed oil (ECSO).

470

471 **4.- Conclusions.**

472 New environmentally friendly plasticized PLA formulations were obtained with
473 epoxidized cottonseed oil (ECSO). The low intrinsic elongation at break of PLA (9%)

474 was improved up to values of 110.5% for plasticized formulations with 10 wt% ECSO.
475 Field emission scanning microscopy (FESEM) revealed good miscibility at low ECSO
476 concentrations (lower than 2.5 wt%) while slight phase separation occurred over this
477 composition. Despite this, a remarkable increase in toughness was observed for
478 compositions over 2.5 wt.% due to the particular morphology defined by a PLA matrix
479 with finely dispersed spherical ECSO domains. The glass transition temperature was
480 reduced by around 5-6 °C for plasticized formulations with 7.5 – 10.0 wt.% ECSO. This
481 decrease in T_g gave evidences of increased free volume with a positive effect on chain
482 mobility, so that both the glass transition and the cold crystallization temperature were
483 moved to lower values. Very low migration levels were assessed by the solvent extraction
484 test with *n*-hexane with a maximum migration of 0.12% at 50-60 °C while very low
485 plasticizer migration (<0.06%) was observed at 30 °C. The results obtained in this work
486 suggest that high environmentally friendly toughened PLA formulations can be obtained
487 by using epoxidized cottonseed oil (ECSO) in the 5 – 10 wt% range. Over 10 wt% ECSO,
488 plasticizer saturation occurs and this has a negative effect on overall properties.

489

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498

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615

616

617

618 **Figure captions**

619 **Figure 1.-** Schematic representation of the chemical structure of a) poly(lactic acid)-PLA
620 and b) epoxidized cottonseed oil (ECSO).

621 **Figure 2.-** Plot evolution of the impact absorbed energy measured by the Charpy's test
622 for plasticized PLA formulations with different weight % of epoxidized cottonseed oil
623 (ECSO).

624 **Figure 3.-** FESEM images of fractured surfaces from impact tests corresponding to
625 plasticized PLA formulations with different weight % of epoxidized cottonseed oil
626 (ECSO) at 2000x: a) neat PLA, b) 2.5 wt.% ECSO, c) 5.0 wt.% ECSO, d) 7.5 wt.% ECSO
627 and e) 10 wt.% ECSO.

628 **Figure 4.-** Plot evolution of the storage modulus (G') as function of temperature for
629 plasticized PLA formulations with different weight % of epoxidized cottonseed oil
630 (ECSO).

631 **Figure 5.-** Plot evolution of the damping factor ($\tan \delta$) as function of temperature for
632 plasticized PLA formulations with different weight % of epoxidized cottonseed oil
633 (ECSO).

634 **Figure 6.-** Comparative plot of the DSC heating thermograms of neat PLA and plasticized
635 PLA formulations with different weight % of epoxidized cottonseed oil (ECSO).

636 **Figure 7.-** Plasticizer migration levels by the solvent extraction test on plasticized PLA
637 formulations with different weight % of epoxidized cottonseed oil (ECSO).