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Additional Information

1 **Effect of the CuO addition on a Sb-doped SnO<sub>2</sub> ceramic electrode applied**  
2 **to the removal of Norfloxacin in chloride media by electro-oxidation**

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9  
10 **Abstract**

11 Norfloxacin is employed as in veterinary and human medicine against gram-positive and gram-  
12 negative bacteria. Due to the ineffective treatment at the wastewater treatment plants it  
13 becomes an emergent pollutant. Electro-oxidation appears as an alternative to its effective  
14 mineralization. This work compares Norfloxacin electro-oxidation on different anodic materials:  
15 two ceramic electrodes (both based on SnO<sub>2</sub> + Sb<sub>2</sub>O<sub>3</sub> with and without CuO, named as CuO and  
16 BCE, respectively) and a boron doped diamond (BDD). First, the anodes were characterized by  
17 cyclic voltammetry, revealing that NOR direct oxidation occurred at 1.30 V vs. Ag/AgCl. The  
18 higher the scan rate the higher both the current density and the anodic potential of the peak.  
19 This behavior was analyzed using the Randles–Sevcik equation to calculate the Norfloxacin  
20 diffusion coefficient in aqueous media, giving a value of  $D = 7.80 \times 10^{-6} \text{ cm}^2 \text{ s}^{-1}$  at 25 °C), which is  
21 close to the predicted value obtained using the Wilke-Chang correlation.

22 The electrolysis experiments showed that both NOR and TOC decay increased with the applied  
23 current density, presenting a pseudo-first order kinetic. All the anodes tested achieved more

24 than 90% NOR degradation at each current density. The CuO is not a good alternative to BCE  
25 because although it acts as a catalyst during the first use, it is lost from the anode surface in the  
26 subsequent uses. According to their oxidizing power, the anodes employed are ordered as  
27 follows: BDD>BCE>CuO.

28

29 **Keywords:** Boron-doped diamond (BDD) anode, Ceramic anodes, Electro-oxidation, Norfloxacin  
30 (NOR), Voltammetric study.

31

## 32 1- Introduction

33 Norfloxacin (NOR), whose molecular structure is presented in Fig. SM-1, is widely used in  
34 veterinary and human medicine against gram-positive and gram-negative bacteria (da Silva et  
35 al., 2018; Huang et al., 2008; Mora-Gómez et al., 2019). The presence of antibiotics in the  
36 different water bodies, represents a public health risk as they not only develop and increase  
37 bacteria resistance to the antibiotic action, but also may cause specific damage to human health  
38 (Jojoa-Sierra et al., 2017; Ma et al., 2018; Mora-Gómez et al., 2019; Wohlmuth da Silva et al.,  
39 2018).

40 The main antibiotic discharge points into the wastewater network are pharmaceutical  
41 industries, hospitals and populated areas (Gogoi et al., 2018; Sim et al., 2011; Wohlmuth da Silva  
42 et al., 2018). This compound has a n-octanol/water distribution coefficient (log D) of -0.65 (Wang  
43 et al., 2015), so it is highly hydrophilic and its removal of wastewater presents greater difficulty  
44 compared to other compounds (Tadkaew et al., 2011). Other authors (Wells, 2006) studied this  
45 same property for different pH values, and they demonstrated that for the entire pH range the  
46 Log D of NOR was less than 1, indicating its facility of evasion from conventional methods in  
47 wastewater treatments. Consequently, NOR and other antibiotics have become emerging  
48 pollutants (EPs) as their presence has been increased up to  $\text{ng L}^{-1}$  and, in some cases, to  $\mu\text{g L}^{-1}$  in  
49 worldwide water bodies (Al Aukidy et al., 2012; Osorio et al., 2016; Watkinson et al., 2009; Zou  
50 et al., 2011).

51 Electrochemical advanced oxidation processes appear as an interesting alternative to treat  
52 these effluents. Their main advantages over other techniques are their high versatility, no  
53 addition of chemical products is needed and low energy consumption (Comninellis and Chen,  
54 2010; Mora-Gómez et al., 2018). These technologies consist of degrading the organic compound  
55 by means of direct oxidation on the anode surface and/or the generation of strong oxidizing  
56 species. The anodes employed can be classified into two main categories: active anodes, those

57 that mainly oxidize the compound on the anode surface, and inactive anodes, which causes the  
58 degradation of the organic compounds by the electrogenerated oxidizing species (Bejan et al.,  
59 2012; Cañizares et al., 2004; Kapařka et al., 2009). In order to achieve the complete oxidation of  
60 the organic substances the inactive anodes are more suitable than the active ones (Bejan et al.,  
61 2012). There are many different inactive anodes, among which boron doped diamond (BDD) is  
62 one of the most usually employed.

63 However, although BDD presents a high chemical stability, high oxygen overpotential and a wide  
64 working anode potential (Comninellis and Chen, 2010; Kapařka et al., 2009), it also has several  
65 drawbacks such as its high manufacture cost, which range from 8000 to 18000 € m<sup>-2</sup> (Cañizares  
66 et al., 2009; Sabatino et al., 2019; Soriano et al., 2019), and its own oxidation during its use. The  
67 C-H groups on BDD surface may be slowly oxidized to C-OH and C=O groups, that can be further  
68 oxidized to COOH and finally to CO<sub>2</sub>. This process is studied at some works and would finally  
69 cause the failure of the BDD anode (Chaplin et al., 2013, 2011; Tryk et al., 2001). Consequently,  
70 new anodic materials able to avoid these drawbacks should be manufactured.

71 In this sense, the present work will study the electrochemical behaviour of ceramic electrodes  
72 based on tin oxide doped with Sb<sub>2</sub>O<sub>3</sub>. This material is selected due to its high conductivity and  
73 corrosion resistance. These low-cost electrodes have been previously tested showing good  
74 results during NOR degradation in sulfate media. In these studies, the NOR degradation attained  
75 using BCE varied from 85 to 100% as the applied current increased, and the BDD reached the  
76 complete removal of NOR for the same experimental conditions tested. (Mora-Gómez et al.,  
77 2019, 2018). However, tin oxide presents low sintering capacity, which can be improved by  
78 adding sintering aids, as CuO (Mihaiu et al., 2001; Popescu et al., 2002; Zaharescu et al., 1991).  
79 This work is focused on the study of the electrochemical oxidation of NOR in a common  
80 electrolyte, NaCl, using three different electrodes: a commercial BDD, a ceramic electrode based

81 on SnO<sub>2</sub> doped with Sb<sub>2</sub>O<sub>3</sub> (basic ceramic electrode, BCE) and another ceramic electrode with  
82 similar characteristics in which CuO is added as sintering aid (CuO).

83

## 84 **2- Experimental**

### 85 2.1- Reagents and solutions preparation

86 The supporting electrolytes were prepared using analytical grade reagents and distilled water.  
87 NaCl and the Norfloxacin employed to make standard solutions were purchased from Sigma-  
88 Aldrich. All the standard solutions were made using ultrapure water HPLC gradient grade from  
89 Fisher Chemical. The solutions employed at the experiments consisted of 1.64 g L<sup>-1</sup> of NaCl  
90 (supporting electrolyte) in the presence or absence of 100 ppm of NOR.

91

### 92 2.2- Cyclic Voltammetry experiments

93 All these experiments were performed in a conventional three-electrode cell employing 50 mL  
94 of solution. An Ag/AgCl (saturated KCl) electrode was employed as reference electrode (RE), and  
95 all the potentials reported herein are referred to it. The counter electrode was a Pt foil with 1  
96 cm<sup>2</sup> area. In the cyclic voltammetry experiments, the working electrode (WE) was one of the  
97 three different materials tested: a BDD with a 2500 ppm doping level (from NEOCOAT SA<sup>®</sup>,  
98 Switzerland) and the two new microporous SnO<sub>2</sub> ceramic electrodes (BCE and CuO). All the  
99 anodes presented a contact surface of 1 cm<sup>2</sup>. The synthesis of the ceramic electrodes is well  
100 explained in previous works (Mora-Gómez et al., 2019, 2018). They are manufactured using SnO<sub>2</sub>  
101 as main component (purity 99.85%, Quimialmel<sup>®</sup> S.A., Spain), Sb<sub>2</sub>O<sub>3</sub> as a dopant (purity 99%,  
102 Alfa-Aesar<sup>®</sup>, Germany), and CuO (when present) as a sintering aid (purity 97%, Panreac<sup>®</sup> S.A.,  
103 Spain). Their composition is summarized in Table SM-1, at supplementary information.

104 The cyclic voltammetry, CV, experiments were carried out using a computer controlled by an  
105 Autolab PGSTAT302N (Methrom) potentiostat/galvanostat. Previously to each CV, the WE was  
106 polarized at 2 V vs. Ag/AgCl for 30 s in order to clean the anode surface and assure the complete  
107 oxidation of the metal oxides present at the ceramic electrodes. The scan rate ranged from 10  
108 to 100 mV s<sup>-1</sup>. The initial potential was the open circuit potential, and polarization was first  
109 applied in the positive direction within an upper and lower potential limit of 3 and -1 V vs.  
110 Ag/AgCl. All the cyclic voltammetry experiments were carried out at room temperature.

111

### 112 2.3- Electrolysis experiments

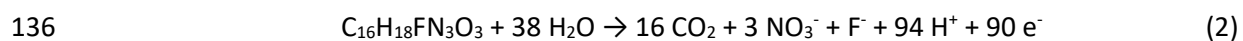
113 The electrochemical reactor was a Pyrex glass of 250 cm<sup>3</sup> filled with a 100 ppm of NOR in 1.64 g  
114 L<sup>-1</sup> of NaCl. All the experiments were performed under stirring conditions. The cathode was an  
115 AISI 304 stainless steel sheet of 20 cm<sup>2</sup> while the RE was the same aforementioned. Regarding  
116 the anode, the same electrode materials as those employed at the CV experiments were tested.  
117 All the anodes were submerged in the solution in order to obtain a contact surface of 5.28 cm<sup>2</sup>.

118 Electro-oxidation experiments were carried out under galvanostatic conditions at three different  
119 current densities (76, 114, 189 mA cm<sup>-2</sup>) by the use of a power supply. The electrode potential,  
120 cell voltage and current were recorded during the electrolysis while samples were taken from  
121 the reactor each 30 min. From these samples, the NOR degradation was calculated by the  
122 measure of its absorbance wavelength at 275 nm using an Unicam UV4-200 UV/vis  
123 spectrometer. The NOR mineralization was followed by both the total organic carbon (TOC) and  
124 the inorganic ions (NH<sub>4</sub><sup>+</sup>, NO<sub>3</sub><sup>-</sup> and F<sup>-</sup>) using a Shimadzu TNM-L ROHS TOC analyser and a  
125 Metrohm Ionic Chromatograph 883 Basic IC Plus, respectively. All the experiments were  
126 performed at room temperature for 4 h.

127 The mineralization current efficiency, was determined as follows (Guinea et al., 2010):

128 
$$MCE = \frac{nFV\Delta(TOC)_t}{7.2 \times 10^5 mIt} \quad (1)$$

129 where n is the number of electrons transferred in the mineralization process (90), F is the  
130 Faraday constant, V is the volume of the reactor (in L), I is the applied current (A),  $\Delta(TOC)_t$  is the  
131 removed organic carbon ( $\text{mg L}^{-1}$ ) at a given time, m is the number of C atoms in a NOR molecule  
132 (16) and  $7.2 \times 10^5$  is the conversion factor for the units homogenization ( $60 \text{ s min}^{-1} \times 12000 \text{ mg}$   
133  $\text{mol}^{-1}$ ). n was assumed as 90 supposing the complete mineralization of NOR according to reaction  
134 (2). This assumption was corroborated experimentally as shown at the results and discussion  
135 section, and was also observed by da Silva et al (da Silva et al., 2018):



137

138



### 139 3- Results and discussion

#### 140 3.1- Voltammetric analysis

141 Fig. 1a presents the cyclic voltammetry obtained for a solution containing just the supporting  
142 electrolyte (NaCl) and both NaCl and NOR for all the electrodes under study. Comparing the  
143 supporting electrolytes CVs of the BDD and the BCE anodes, the electrochemical window is wider  
144 for the BDD anode (2.05 V vs. Ag/AgCl) than for the BCE (1.85 V vs. Ag/AgCl for the BDD) as  
145 expected (Lobato et al., 2005; Panizza et al., 2008; Polcaro et al., 2006). The supporting  
146 electrolyte voltammogram obtained for the CuO anode practically overlaps the one obtained for  
147 the BCE (not shown).

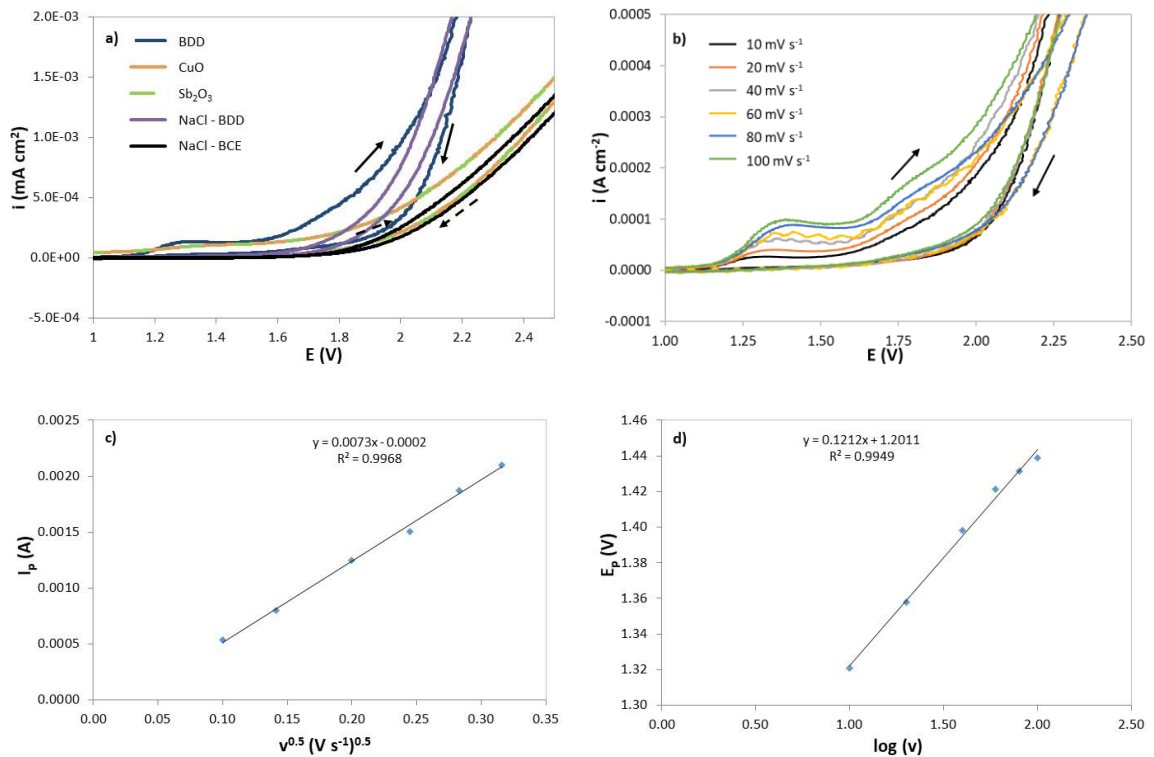
148 When the CVs obtained in presence of NOR for the different electrode materials are compared  
149 with those obtained with only the supporting electrolyte, Fig. 1a, an anodic peak at 1.30 V vs.  
150 Ag/AgCl is observed, which only appears in the presence of NOR in solution, and is in  
151 concordance with the results of da Silva et al. (da Silva et al., 2018). Although other authors place  
152 the NOR oxidation peak between 0.9 and 1.1 V vs. Ag/AgCl (da Silva et al., 2015; Devaraj et al.,  
153 2013; Goyal et al., 2012; Huang et al., 2008; Liu et al., 2018), this difference in the peak potential  
154 can be associated to the variation of the different working conditions such as, the supporting  
155 electrolyte, NOR concentration, pH and, specially, electrode material, etc. since these works use  
156 electrodes as sensors. Additionally, another anodic wave at 1.7 V vs. Ag/AgCl is also detected.  
157 This can be related to the contribution of two processes: the oxidation of an intermediate  
158 compound generated during NOR oxidation (Coledam et al., 2016; da Silva et al., 2018) together  
159 with the onset of the oxygen electrochemical reaction.

160 From Fig. 1b, which represents the effect of scan rate on the CVs, it is clearly observed that the  
161 NOR oxidation peak increases and shifts to more anodic potentials as the scan rate increases  
162 and, additionally, it does not show a reversal peak which is a typical behaviour of an  
163 electrochemical irreversible process (Nicholson and Shain, 1964). This fact, which was also

164 observed by other authors working in NOR detection (da Silva et al., 2015; Devaraj et al., 2013;  
 165 Goyal et al., 2012; Huang et al., 2008; Liu et al., 2018), is commonly observed in the electro-  
 166 oxidation of organic compounds (Carrillo-Abad et al., 2018).

167 In order to determine the limiting step of the direct oxidation of NOR, its current peak intensity  
 168 ( $I_p$ ) was plotted as a function of the square root of the scan rate ( $v^{1/2}$ ), Fig. 1c. The  $I_p$  values  
 169 showed a linear trend with  $v^{1/2}$ , indicating the presence of a diffusion-controlled oxidation  
 170 process. Nevertheless, the linear fitting does not intercept the origin, which is related to an  
 171 additional process other than diffusion (García-Gabaldón et al., 2011; Trejo et al., 1998) taking  
 172 place during the NOR direct oxidation on the anode surface. This process is probably associated  
 173 with the adsorption of a NOR oxidation product on the anode surface, which also explains the  
 174 absence of a NOR reduction counter peak at Fig. 1b (Aoki and Chen, 2016).

175



176 **Fig. 1:** Voltammetric study of NOR in a NaCl solution: a) anodic response as a function of the  
 177 electrode material at 10 mV s<sup>-1</sup>, b) CV zoom at NOR peak for BDD at different scan rates, c)

178 peak current as a function of the square root of the scan rate and, d) peak potential variation  
179 with the scan rate logarithm. Anodes surface = 1 cm<sup>2</sup>, [NOR]<sub>0</sub>= 100 ppm, [NaCl]<sub>0</sub>= 1.64 g L<sup>-1</sup>

180

181 The relationship between  $I_p$  and  $v$  for an irreversible process can be studied by the Randles-  
182 Sevcik equation:

$$183 \quad I_p = 0.496 nFCAD^{\frac{1}{2}} \left( \frac{(\alpha \cdot n_{\alpha})Fv}{RT} \right)^{\frac{1}{2}} \quad (3)$$

184

185 where A is the electrode area in cm<sup>2</sup>, D is the diffusion coefficient of the oxidized species in cm<sup>2</sup>  
186 s<sup>-1</sup>, C is the NOR concentration in mol cm<sup>-3</sup>, v is the scan rate in V s<sup>-1</sup>, F is the Faraday constant in  
187 C, R is the gas constant in J mol<sup>-1</sup> K<sup>-1</sup>, T is the absolute temperature in K and n is the number of  
188 transferred electrons which is assumed as 2 according to the mechanism suggested for the NOR  
189 oxidation in previous works (Devaraj et al., 2013; Liu et al., 2018) and is shown at supplementary  
190 information (Fig. SM-2). Otherwise, Fig. 1d represents the increase of  $E_p$  with the logarithm scan  
191 rate, displaying a linear relationship that permits the  $\alpha \cdot n_{\alpha}$  determination by the Laviron's theory  
192 (Laviron, 1979). On the basis of the previous discussion a  $\alpha \cdot n_{\alpha}$  of 0.489 is determined.

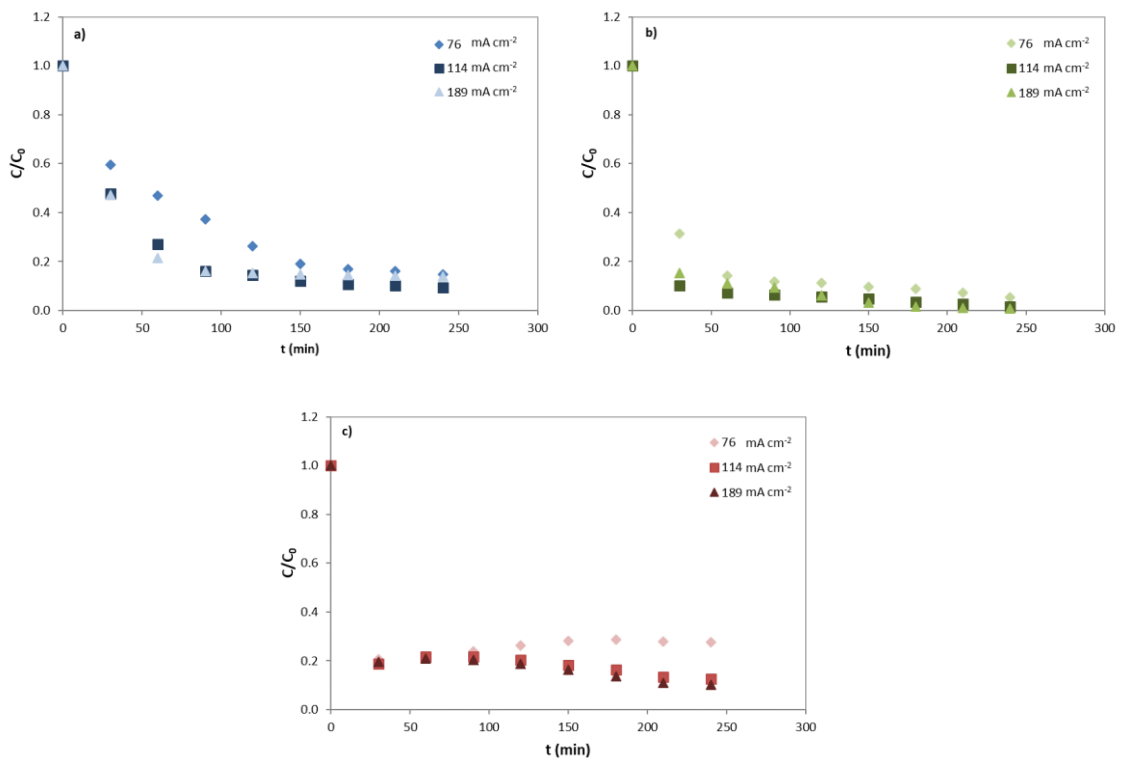
193 Therefore, the NOR diffusion coefficient in aqueous media was calculated using the linear fitting  
194 of the experimental data presented in Fig. 1c and the Randles-Sevcik equation, giving a value of  
195  $7.80 \times 10^{-6}$  cm<sup>2</sup> s<sup>-1</sup>. This result is very similar to that calculated by the semi-empirical Wilke-Chang  
196 correlation ( $7.26 \times 10^{-6}$  cm<sup>2</sup> s<sup>-1</sup>), which is generally employed to estimate this parameter for  
197 organic compounds in aqueous media (Perry and Green, 2008). The Wilke-Chang correlation  
198 (Wilke and Chang, 1955) together with the values employed for its determination is shown at  
199 the Supplementary information (as Eq. SM1).

200

201 3.2- Effect of current density

202 Figs. 2a-c show the evolution of the NOR decay profile with time for all the applied current  
203 densities using the electrochemical reactor of 250 cm<sup>3</sup> with the different electrodes under study.  
204 Although the first increase in the current density (from 76 to 114 mA cm<sup>-2</sup>) caused a growth in  
205 the degradation rate, this growth did not appear when the highest current was applied. This fact  
206 together with the exponential trend showed in all the figures is characteristic of a process under  
207 mass transport control (Urtiaga et al., 2018). It is worth to note that in both ceramic electrodes  
208 a maximum 90% NOR degradation is observed.

209



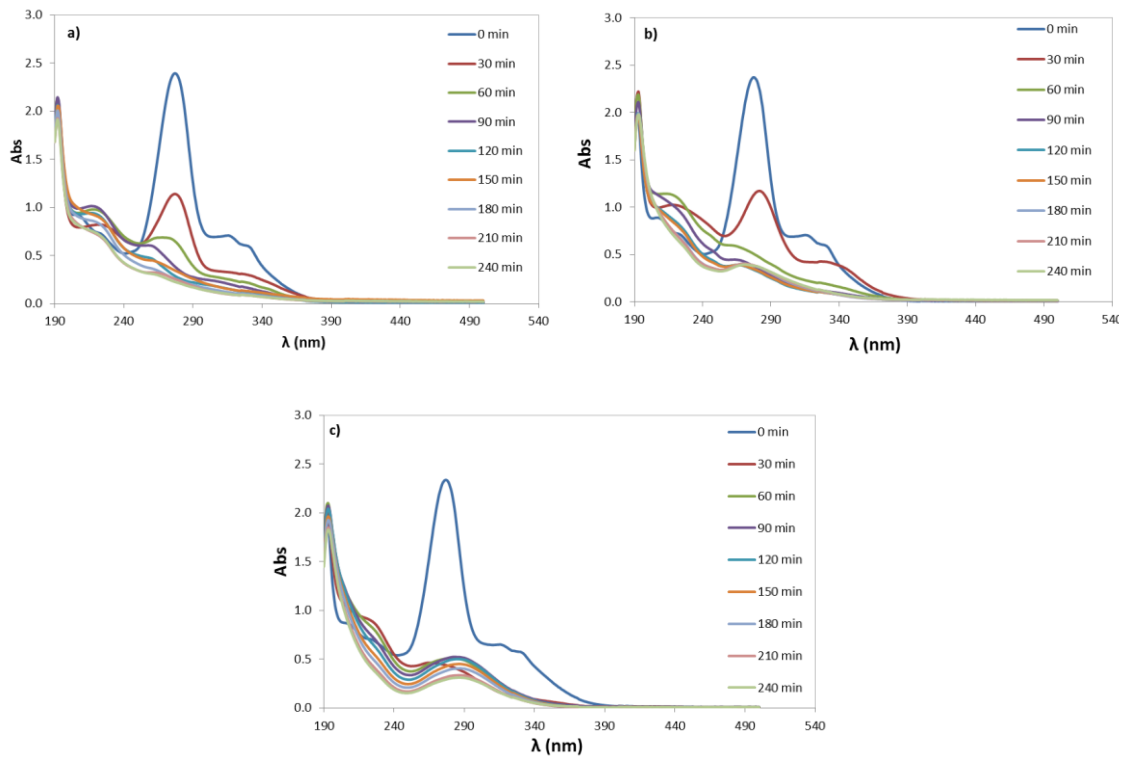
210 **Fig. 2:**  $C/C_0$  decay profile for all the currents density applied for the different anode materials:

211 a) CuO, b) BDD and c) BCE .  $[NOR]_0 = 100$  ppm,  $[NaCl]_0 = 1.64$  g L<sup>-1</sup>

212

213 A deeper sigh into the UV diagram (Figs. 3a and 3b) for these electrodes show a change in the  
214 main peak, which is related to the intermediate degradation compounds, that modify the values  
215 obtained at  $\lambda = 276$  nm, causing the misreading of the values for NOR concentration. Concerning  
216 the evolution of the UV-visible spectra, previously to the electrolysis start, the same spectrum  
217 is obtained for all the electrodes tested. Two main peaks are observed for the original sample,  
218 one at 276 nm and another at 330 nm, which are associated with the aromatic ring absorption  
219 and the electronic transition of the quinolones nitrogen atom from  $n \rightarrow \pi^*$  (HOMO-LUMO)  
220 respectively (Neugebauer et al., 2005). Once the electrolysis began, both peaks decreased,  
221 which indicates the NOR degradation. However, the spectrum significantly changes from one  
222 electrode to another. While the quinolone group absorption remained during the first  
223 electrolysis time for both CuO and BDD (Figs. 3a and 3b) electrodes, it completely disappeared  
224 when the BCE was used (Fig. 3b). Furthermore, the main peak (276 nm) was displaced to higher  
225 wavelengths during the BCE experiment (Fig. 3c) while it was moved to lower  $\lambda$  values for the  
226 CuO and BDD electrodes (Figs. 3a and 3b). These differences can be attributed to the generation  
227 of by products with less substituent for the CuO and BDD electrodes, while intermediates with  
228 a more chromophore substituent were generated when the BCE was used (Woodward, 1942a,  
229 1942b, 1941; Woodward and Clifford, 1941). Therefore, two different degradation mechanisms  
230 may be expected as function of the electrode material used. On the other hand, a new  
231 absorption peak between 200 and 220 nm appears in all the electrolysis performed, which is  
232 related to the generation of short chain carboxylic acids (Coledam et al., 2016; Mora-Gómez et  
233 al., 2019) as corroborated in the ionic analysis section.

234



235 **Fig. 3:** UV-vis spectrum at the sampling times for the different anode materials: a) CuO, b) BDD  
 236 and c) BCE. These spectra were obtained at  $114 \text{ mA cm}^{-2}$  in all cases.  $[\text{NOR}]_0 = 100 \text{ ppm}$ ,  
 237  $[\text{NaCl}]_0 = 1.64 \text{ g L}^{-1}$

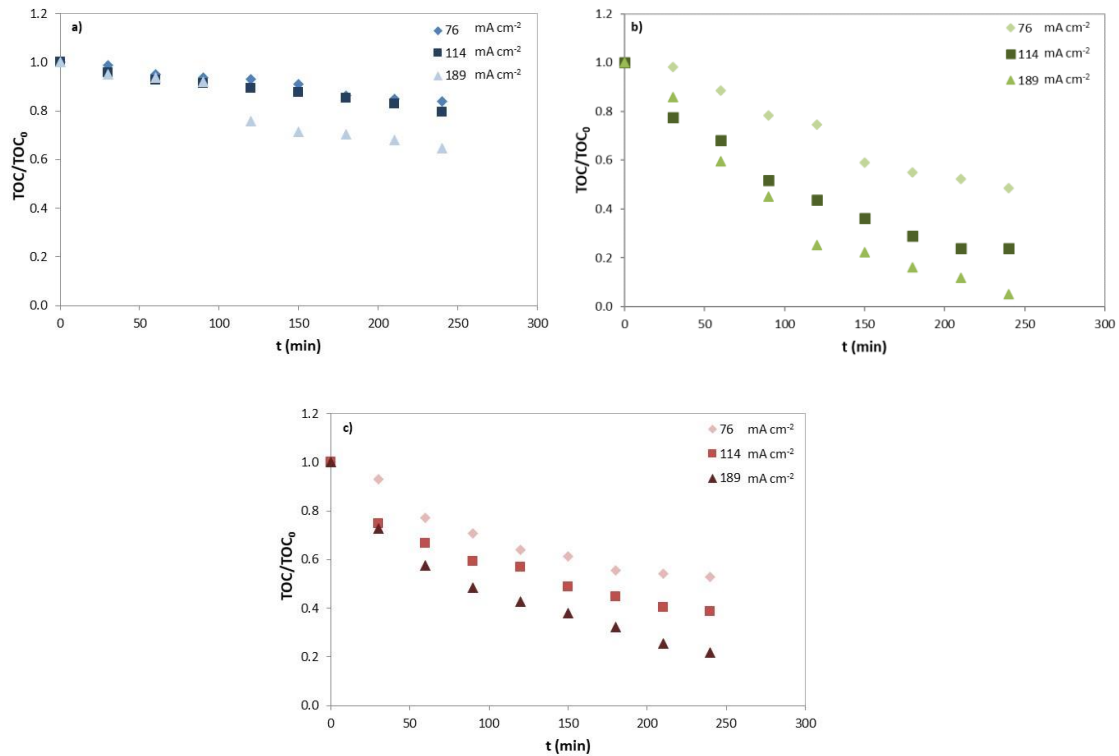
238

239 Regarding TOC decay evolution with time for all the applied current densities and electrodes  
 240 (Figs. 4a-c), the higher the applied current density the higher the velocity of NOR mineralization  
 241 for all the anodes studied, which is related to the higher amount of oxidizing species generated  
 242 during the electrolysis at higher values of applied current (Diban, 2018; Pipi et al., 2014) that  
 243 also contribute to the NOR mineralization.

244 From the comparison of Figs. 4a-c it is inferred that the higher mineralization degree is obtained  
 245 for the BDD, followed by the BCE and, finally, using the CuO anode. At the lowest applied current  
 246 density the results obtained with the BDD and the BCE are very similar but as the applied density  
 247 current increases, the differences between the mineralization rates obtained by using both  
 248 anodes also increase. This can be attributed to the higher oxidising power of BDD anodes, since

249 the it has a higher oxygen overpotential, and presents a weaker hydroxyl radical adsorption than  
250 the ceramic one (Chen et al., 2005; Mora-Gómez et al., 2019).

251



252 **Fig. 4:** TOC/TOC₀ decay profile for all the currents density applied for the different anode  
253 materials: a) CuO, b) BDD and c) BCE. [NOR]₀= 100 ppm, [NaCl]₀= 1.64 g L⁻¹

254

255 The mineralization current efficiency (Fig. SM-3) showed a common behaviour for all the anodes  
256 under study. For any applied current, the MCE decreases with time, which is characteristic of a  
257 process under mass transfer control. Nevertheless, MCE decreases with the highest applied  
258 current density due to the higher amount of energy wasted in parallel reactions together with  
259 the generation of an excess of oxidants that reacts among them. The results obtained for the  
260 BCE are very similar to those obtained by the BDD anode (not shown).

261

262 3.2- Effect of anode material

263 Table 1 summarizes the apparent kinetic constants for NOR removal as a function of the anode  
 264 material for all the applied current densities, which are calculated from Fig. 2 assuming a  
 265 pseudo-first order kinetics reaction for NOR electro-oxidation. The higher the applied current  
 266 the higher the apparent kinetic constant. This fact may be related to the higher generation of  
 267 oxidizing species as the applied current increases.

268 **Table 1:** Apparent first order kinetic constant values ( $\text{min}^{-1}$ ) for NOR degradation at each anode  
 269 tested as a function of the applied current density.  $[\text{NOR}]_0 = 100 \text{ ppm}$ ,  $[\text{NaCl}]_0 = 1.64 \text{ g L}^{-1}$

$i(\text{mA cm}^{-2})$	CuO	BDD	BCE
76	0.0105	0.0337	0.026
114	0.036 first use 0.021	0.0421	0.0315
189	0.0222	0.0502	0.0318

270

271 From Table 1, it is inferred that NOR electro-oxidation is faster on the BDD than in the ceramic  
 272 anodes, and that the basic ceramic anode (BCE) presents higher values than those obtained for  
 273 the CuO, except when this latter is used for the first time. It is also noticeable that CuO electrode  
 274 presents a considerable different oxidizing power from its first use in relation to the following  
 275 ones as can be seen in Figs. SM-4a and SM-4b at supplementary information. The oxidizing  
 276 power of the CuO electrode during its first use is comparable to that obtained using the BDD  
 277 anode (Figs. 2b and 4b) because the CuO present at the anode is dissolved into the solution  
 278 acting as a catalyst for the oxidation of the organic compounds. This catalytic behaviour of  $\text{Cu}^{+2}$   
 279 has also been observed by other authors during the oxidation of phenolic compounds (Gözmen  
 280 et al., 2003; Santos et al., 2002).



281 The presence of copper in solution has been experimentally confirmed by atomic absorption  
282 spectroscopy. During the successive uses of this anode, the results are getting worse, reaching  
283 lower TOC values than those obtained using the basic ceramic electrode of SnO<sub>2</sub> (BCE) as  
284 observed in Fig. 4. This is due to the fact that when the electrode loses the Cu<sup>+2</sup> of the surface,  
285 its composition becomes similar to that of the BCE. The lower active surface of the CuO electrode  
286 in relation to the BCE is responsible for this mechanism, since the copper electrode is denser  
287 (apparent density of 6.696 g cm<sup>-3</sup>) than the BCE (apparent density of 3.3533 g cm<sup>-3</sup>).

288 Summarizing the data obtained previously in terms of degree of mineralization and degradation,  
289 it is concluded that the BDD presents the best percentages of both parameters, and the CuO  
290 does not represent a good alternative to the basic ceramic anode as the loss of copper worsens  
291 its behaviour. Although the ceramic electrodes presented lower values of NOR removal than the  
292 BDD, the values obtained are similar or higher than those obtained by another anodes also  
293 formed by metallic oxides, as observed in the Table presented at the supplementary material  
294 (Table SM-2), where the comparison of the NOR removal rate obtained using different anodic  
295 material is presented.

296

### 297 *3.3- Ionic analysis*

298 The generation of fluoride, together with nitrogen ionic compounds and short-chain organic  
299 acids are expected from NOR oxidation. In addition, the chloride ions provided by the supporting  
300 electrolyte were also measured as they can be oxidized to some active species, which may also  
301 be involved in NOR oxidation.

302 Concerning the main nitrogen ionic compound generated during NOR electro-oxidation, a high  
303 controversy exists since some authors present the ammonium as main compound (Mora-Gómez  
304 et al., 2019; Özcan et al., 2016), while others only detects NO<sub>3</sub><sup>-</sup> (da Silva et al., 2018). In the

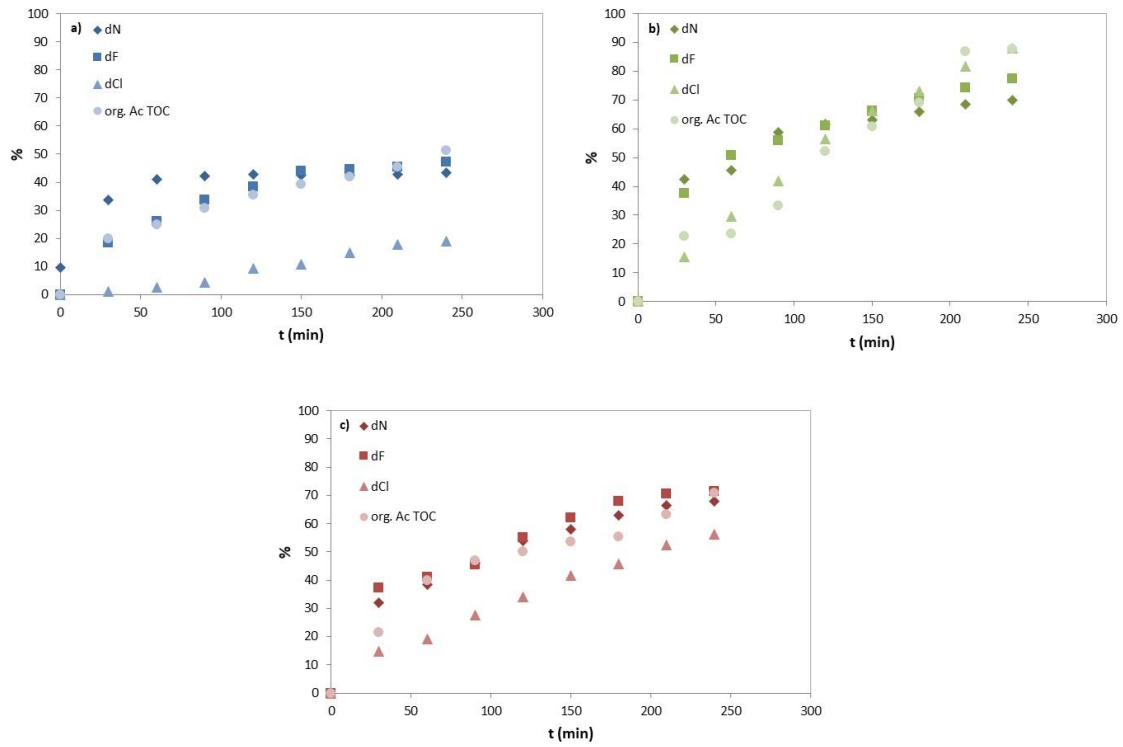
305 present work, nitrite and nitrate ions were detected, while any ammonium was observed in any  
306 experiment. This fact may be related to the presence of chlorine oxidizing species that can  
307 oxidize ammonium to both nitrite and nitrate ions.

308 Regarding the organic ions detected (supplementary Fig. SM-5), they ranged from the small ones  
309 (formate, acetate and oxalate ions), medium sized (such as tartrate ions) and aromatic (like  
310 phthalate ions). For BDD and BCE, Figs. SM-5b and SM-5c, initially, the organic ions detected  
311 were mainly the aromatic, although the BDD presented significant formate and acetate ions  
312 from the very beginning. As the electrolysis drew on, phthalate ions disappeared while tartrate  
313 and oxalate increased but they became the main organic ions only for a discrete instant as  
314 formate and acetate concentrations continuously grown up to become the main organic ions  
315 detected. On the other hand, the CuO anode, Fig. SM-5a presented a high amounts of phthalate  
316 and tartrate for the whole electrolysis time, maybe due to its lower oxidation power, while in  
317 terms of oxalate, formate and acetate, it showed a similar behaviour to that observed for the  
318 other materials tested.

319 Fig. 5 presents the denitrogenation (dN) and defluorination (dF) from NOR, dechlorination (dCl)  
320 from electrolyte and the % of TOC measured associated with the organic ions measured for all  
321 the materials tested at  $114 \text{ mA cm}^{-2}$ . Each of these parameters increased, independently of the  
322 anodic material, as the experiment drew on due to both NOR and NaCl oxidation. However,  
323 these rates are directly related to the oxidizing power discussed above, that is, the BDD presents  
324 the higher dCl rate, almost a 90%, associated with the higher generation of oxidative species,  
325 which helps to obtain greater dN and dF rates. Additionally, BDD also shows the higher amount  
326 of TOC related to the acid organic ions, achieving more than the 85% of the removal of the  
327 remaining TOC. After the BDD, the BCE anode presents good values as dN, dF and the TOC  
328 removal related with organic acids reached a 70% which is a significant value considering that  
329 the dCl rate only achieved a 50%. It is worth to note that the CuO anode showed the lowest

330 values but it is also remarkable that dN, dF and the TOC removal from organic acids are close to  
331 the 50% while the dCl rate is close to the 15%.

332



333 **Fig. 5:** Defluorination, denitrogenation, dechlorination and fractional TOC of organic acid ions  
334 profile for all the anodic materials tested: a) CuO, b) BDD and c) BCE.  $[NOR]_0 = 100$  ppm.

335  $[NaCl]_0 = 1.64g L^{-1}$  and  $i = 114 mA cm^{-2}$

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#### 342 4- Conclusions

343 This work is focused on the study of the electrochemical oxidation of NOR in a common  
344 electrolyte, NaCl, using three different electrodes: a commercial BDD, a ceramic electrode based  
345 on SnO<sub>2</sub> doped with Sb<sub>2</sub>O<sub>3</sub> (basic ceramic electrode, BCE) and another ceramic electrode with  
346 similar characteristics in which CuO is added as sintering aid (CuO).

347 From the voltammetric analysis obtained for the different electrode materials an anodic peak at  
348 1.30 V vs. Ag/AgCl was observed, which was attributed to NOR oxidation. Additionally, another  
349 anodic wave at 1.7 V vs. Ag/AgCl was also detected, related to the contribution of two processes:  
350 the oxidation of an intermediate compound generated during NOR oxidation together with the  
351 onset of the oxygen electrochemical reaction. NOR oxidation was classified as an irreversible  
352 process and controlled not only by diffusion. The diffusion coefficient for NOR in aqueous media  
353 was calculated as  $7.80 \times 10^{-6} \text{ cm}^2 \text{ s}^{-1}$  at 25 °C.

354 The higher mineralization and degradation degrees were obtained for the BDD, followed by the  
355 BCE and, finally, using the CuO anode. The apparent kinetic constants for NOR removal assuming  
356 a pseudo-first order kinetics reaction were calculated as a function of the anode material,  
357 presenting the following tendency: BDD>BCE>CuO. The CuO electrode presented a considerable  
358 different oxidizing power from its first use in relation to the following ones because the copper  
359 present at the anode is dissolved into the solution acting as a catalyst for the oxidation of the  
360 organic compounds. After this process, although the composition of the CuO anode becomes  
361 similar to that of the BCE, its lower active surface worsens the process.

362 From the results obtained in this study, it is inferred that the BDD presents the best results. The  
363 CuO does not represent a good alternative to the basic ceramic anode as the loss of copper  
364 worsens its behaviour. Although the ceramic electrodes presented lower values of NOR removal  
365 than the BDD, the values obtained are similar or higher than those obtained by other anodes  
366 also formed by metallic oxides.

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