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Additional Information

1 **Cesium–induced ionic conduction through a single**
2 **nanofluidic pore modified with calixcrown moieties**

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16

17 **Abstract**

18 We demonstrate experimentally and theoretically a nanofluidic device for the
19 selective recognition of cesium ion by exploiting host–guest interactions inside
20 confined geometry. For this purpose, a host molecule, *i.e.*, the amine-terminated *p*-
21 *tert*-butylcalix[4]arene-crown (*t*-BuC[4]C–NH₂), is successfully synthesized and
22 functionalized on the surface of a single conical nanopore fabricated in polyethylene
23 terephthalate (PET) membrane through carbodiimide coupling chemistry. On
24 exposure to cesium cation, the *t*-BuC[4]C–Cs⁺ complex is formed through host–guest
25 interaction, leading to the generation of positive fixed charges on the pore surface.
26 The asymmetrical distribution of these groups along the conical nanopore leads to the

1 electrical rectification observed in the current–voltage (I – V) curve. On the contrary,
2 other alkali cations are not able to induce any significant change in the rectification
3 characteristics of the nanopore. The success of the chemical modification is monitored
4 from the changes in the electrical readout of the nanopore. Theoretical results based
5 on the Nernst-Planck and Poisson equations demonstrate further the validity of the
6 experimental approach to the cesium-induced ionic conduction of the nanopore.

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11 *Keywords:* synthetic nanopores; host–guest interactions; calixcrown; current
12 rectification; surface modification; track-etching.

13

14

1 **1. Introduction**

2 Metal ions in biological systems take part in a variety of essential processes and some
3 of these ions can cause adverse effects. In particular, cesium ion is linked to negative
4 health effects in its radioactive form.^{1,2} Cesium has some similarities with potassium
5 and then it can be easily imported in the cellular matrix through membrane transport
6 mechanisms, causing serious health problems.³⁻⁵ Previously, cesium detection in
7 environmental and biological samples has been achieved through electrochemical and
8 optical sensors,^{6,7} ion exchange,⁸ atomic absorption spectroscopy⁹ and mass
9 spectrometry techniques.¹⁰

10 Recently, much attention has been paid to the synthesis of a variety of macrocyclic
11 molecules such as crown ethers, cryptophanes and calixarenes for the detection of
12 metal ions based on host-guest complexation.^{6,7,11-14} Among the various
13 supramolecular compounds, calixarene are frequently used in ionic recognition and
14 separation processes. The phenolic moieties in calixarene can be functionalized with a
15 variety of chemical groups to miniaturise ion selective receptors. Particularly,
16 calix[4]arene-crown derivatives in 1,3-alternate conformation are considered as highly
17 efficient cesium complexing agents.¹¹⁻¹⁴ In this conformation, the cesium ion is
18 coordinated with the oxygen atoms of the crown ether and interact also with the π -
19 electrons of the two aromatic rings in the calixarene unit.¹⁴ This property of
20 calixcrown derivatives finds application in the selective removal of radio-active
21 cesium (¹³⁷Cs) from nuclear wastes.^{15,16}

22 For practical applications, calixcrown moieties should be properly organized on
23 suitable surfaces.¹⁷⁻¹⁹ To this end, Zhang and Echegoyen have arranged the self-
24 assembled layer (SAM) of two conformational isomers of thiol-terminated *p-tert*-
25 butylcalix[4]crown on gold surface. They have demonstrated that the 1,3-alternate

1 isomer exhibits remarkable binding capability towards Cs⁺ ion.¹⁹ Similarly, Britt and
2 co-workers have also demonstrated a SAM coated microcantilever functionalized with
3 1,3-alternate calix[4]benzocrown for Cs⁺ ion recognition.¹⁷ Recently, Yi *et al.* studied
4 stimuli-responsive microspheres having calixcrown moieties for the selective removal
5 of Cs⁺ ion from seawater.¹⁸ Mori and co-workers have also studied the Cs⁺ ion
6 selectivity of various conformations of calixcrown moieties using liquid membrane
7 electrodes.¹² However, the design and development of a nanofluidic pore allowing for
8 Cs⁺ sensing is still a challenge.

9 In living organisms, ion channels regulate the flow of ions across the cell
10 membrane to facilitate the electrochemical communication with the extracellular
11 environment.²⁰ Ion channels with defined interfacial chemistry display gating and
12 current rectification phenomena because of their ability to selectively transport a
13 specific ion.²⁰⁻²² Ionic conduction is also regulated through environmental stimuli.^{21,22}
14 The ion channels can be in the open or closed states in response to external stimuli.
15 Moreover, these channels have proved useful for a variety of applications in
16 nanobiotechnology such as sensing and manipulation of single molecules.²³⁻²⁵
17 Inspired from the functionality of ion channels, various routes have been proposed to
18 fabricate synthetic nanopores in insulating solid-state materials.²⁶⁻²⁸ Alternatively, the
19 nanopore fabrication using ion track technology allows also to control the shape, size,
20 and surface properties on demand.²⁹⁻³⁶ These nanopore have already been
21 functionalized with metal chelating ligands to miniaturise electrochemical devices for
22 the detection of different metal cations.^{30,37-43} Moreover, the nanopore surface has also
23 been decorated with a variety of host molecules for the binding of guest analytes
24 through host-guest interactions.⁴⁴⁻⁵¹ To the best of our knowledge, however, Cs⁺ ion
25 detection using a nanofluidic pore has not yet been explored.

1 We demonstrate here cesium-induced ionic conduction using a nanofluidic diode
2 based on a single asymmetric nanopore decorated with calixcrown moieties. To this
3 end, we have designed and synthesized the amine-terminated *p-tert-*
4 *butylcalix[4]arene-crown* (*t*-BuC[4]C–NH₂) molecule (Figure 1). The calixcrown
5 molecules are then functionalized on the nanopore walls by using carbodiimide
6 coupling chemistry. On exposure to the cesium cation in the external solution, the *t-*
7 BuC[4]C–Cs⁺ complex is formed which imparts positive charge to the pore surface.
8 This fact leads to selective transport of ions through the nanopore. On the contrary,
9 other alkali cations do not induce any significant change in the rectification
10 characteristics of the nanopore. The confirmation of the chemical modification is
11 achieved via the changes in the rectified ion current observed in the current–voltage
12 (*I–V*) characteristics prior to and after the nanopore functionalization.

13

14 **2. Materials and methods**

15 **2.1 Materials**

16 All the chemicals and solvents, including *N*-(3-dimethylaminopropyl)-*N'*-
17 ethylcarbodiimide·HCl (EDC·HCl), pentafluorophenol (PFP), 5(6)-
18 carboxyfluorescein (Fcn), *N*-Boc-1,6-hexanediamine, *N,N*-diisopropylethylamine
19 (DIPEA), *p-tert*-butylcalix[4]arene (*t*BuC[4]), pentaethylene glycol ditosylate,
20 potassium *t*-butoxide (*t*-BuOK), potassium carbonate, chloroacetyl chloride, cesium
21 carbonate (Cs₂CO₃), potassium iodide (KI), trifluoroacetic acid (TFA), lithium
22 chloride (LiCl), sodium chloride, (NaCl), potassium chloride (KCl), rubidium chloride
23 (RbCl) and cesium chloride (CsCl) were purchased from Sigma-Aldrich, Taufkirchen,
24 Germany, and used without further purification.

1 ^1H and ^{13}C NMR spectra were recorded at 500 and 125 MHz in CDCl_3 ,
2 respectively. High-resolution mass spectra were measured using a Finnigan MAT90
3 mass spectrometer. Analytical TLC (silica gel, 60F-54, Merck) and spots were
4 visualized under UV light and/or phosphomolybdic acid-ethanol. Flash column
5 chromatography was performed with silica gel 60 (70-230 mesh, Merck) and basic
6 aluminum oxide (activated, basic, ~ 150 mesh, 58 \AA , Aldrich).

7 **2.2 Fabrication of single asymmetric nanopores**

8 Asymmetric track-etching technique was employed to fabricate single conical
9 nanopores in polyethylene terephthalate (PET) membranes.²⁹ The first step was the
10 irradiation of $12 \mu\text{m}$ thick PET foils (Hostaphan RN 12, Hoechst) with single swift
11 heavy ions (Au) of kinetic energy 11.4 MeV/nucleon at the linear accelerator
12 UNILAC (GSI Helmholtz Centre for Heavy Ion Research, Darmstadt, Germany).
13 Secondly, the latent ion tracks of the swift heavy ion in polymer membranes were
14 sensitized with soft UV light. Finally, the chemical track-etching process was
15 performed in a custom-made conductivity cell having three chambers. This
16 conductivity cell was employed for the simultaneous fabrication of the single-pore
17 and multipore membranes. To achieve this goal, a single-shot (1 ion hitting the foil)
18 membrane and a membrane irradiated with 10^7 ions per cm^2 were placed on the sides
19 of the middle compartment of the conductivity cell and clamped tightly. An etching
20 solution (9 M NaOH) was filled in the middle compartment having apertures on both
21 sides. In this way, the chemical etchant was in direct contact with both membranes.
22 The two compartments on either side of the middle chamber were filled with a
23 stopping solution (1 M KCl + 1 M HCOOH). To monitor the etching process, gold
24 electrodes were inserted on both sides of the single-ion irradiated membrane and a
25 potential of -1 V was applied across the membrane. The etching process was carried

1 out at room temperature. The current remained zero as long as the etchant had not
2 permeated the whole length of the membrane. After the breakthrough (a point at
3 which the etchant pierced the membrane), an increase in the ionic current through the
4 nascent pore was observed. The etching process was stopped when the current
5 reached a certain defined value. Then, the membranes were thoroughly washed with
6 stopping solution in order to neutralise the etchant, followed by deionized water. The
7 etched membranes were then dipped in deionized water overnight in order to remove
8 the residual salts. This process resulted in polymer samples containing approximately
9 conical single pores with the carboxylic groups (COOH) generated on the inner pore
10 walls due to the hydrolysis of ester bonds in the back-bone of polymer chains.

11 **2.3 Synthesis of *t*-BuC[4]C–NH₂ (6)**

12 Figure 1 depicts the preparation of the different calixcrown ether derivatives
13 involved in the synthesis of amine-terminated *p*-tert-butylcalix[4]arene-crown (**6**) by
14 following previously reported reaction procedures with slight modifications.^{19,52,53}
15 Details of the synthesis are given in the Electronic Supplementary Information (ESI)
16 file.

17 Briefly, we have synthesized amine-terminated *p*-tert-butylcalix[4]arene-crown (*t*-
18 BuC[4]C–NH₂) to modulate the nanopore surface chemistry and transport properties
19 in response to cesium cation. The reaction scheme for the synthesis of *t*-BuC[4]C–
20 NH₂ is shown in Figure 1. Commercially available *p*-tert-butylcalix[4]arene (**1**) and
21 pentaethylene glycol ditosylate (**2**) were treated with potassium *tert*-butoxide in dry
22 benzene to afford calixcrown ether derivative (**3**) according to an established
23 method.⁵² Chloroacetamide (**4**) was synthesized by adopting a previously reported
24 procedure.⁵³ Then, chloroacetamide (**4**) and calixcrown (**3**) were refluxed in the
25 presence of cesium carbonate with the a catalytic amount of potassium iodide to

1 afford Boc-protected amine, *i.e.*, *t*-BuC[4]C–NHBoc (**5**).¹⁹ Deprotection was carried
2 out using trifluoroacetic acid to afford *t*-BuC[4]C–NH₂ (**6**) which was used without
3 further purification in the nanopore modification reaction. The chemical structures of
4 calixcrown ether derivatives were characterized through ¹H NMR, ¹³CNMR and
5 HRMS-FAB techniques.

6

7 **2.4 Chemical functionalization of nanopore surface**

8 The chemical etching led to the generation of the carboxylic acid (–COOH) groups
9 on the pore surface. These groups exposed were first converted into amine-reactive
10 esters through carbodiimide coupling chemistry. To this end, the track-etched single-
11 pore membrane was immersed in an ethanol solution containing EDC (100 mM) and
12 PFP (200 mM) at room temperature. The activation process was carried out for 1 h.
13 The membrane was washed with ethanol several times. Then, the activated pore was
14 dipped in *t*-BuC[4]C–NH₂ (12 mM) solution prepared in anhydrous ethanol for 15 h.
15 During this reaction period, amine-reactive PFP-esters were covalently coupled with
16 terminal amine group of the *t*-BuC[4]C–NH₂. Subsequently, the modified pore was
17 washed thoroughly with ethanol followed by careful rinsing with deionized water.

18 **2.5 Contact angle measurements**

19 For contact angle analysis, native PET foil without heavy ion irradiation was used.
20 To prepare the samples for the contact angle (CA) measurements, the PET foil was
21 exposed to an etchant solution for 55 min. After washing with water, the PET foil was
22 dipped in the stopping solution for 2 h. The sample was then exposed to deionized
23 water overnight. The wetting properties (contact angle) of unmodified and *t*-BuC[4]C-
24 modified surfaces were measured using a contact angle measurement setup equipped

1 with a side camera (IDS uEye camera) and a goniometer. To measure the respective
2 contact angle, 5 microliter of water was dropped on the foil using a micropipette.

3 **2.6 Current-voltage measurements**

4 The unmodified and modified pores were characterized by measuring the current–
5 voltage (I – V) curves before and after functionalization. The electrolyte solutions were
6 prepared in 10 mM tris-buffer (pH 6.5). The measurement of I – V curve was
7 performed using a picoammeter/voltage source (Keithley 6487, Keithley Instruments,
8 Cleveland, Ohio, USA) using LabVIEW 6.1 (National Instruments). For this purpose,
9 the single-pore (as-prepared) membrane was fixed between the two compartments of
10 the conductivity cell. An aqueous electrolyte was filled in both halves of the cell. The
11 electrodes having a Ag wire coated with AgCl were inserted into each half-cell
12 solution to establish a transmembrane potential difference (voltage) and the ionic
13 current through the pore was then measured. For the case of the conical nanopore, the
14 ground and working electrodes were placed on the base and tip side of the pore,
15 respectively. In order to record the I – V curves, a scanning triangle voltage signal from
16 –2 to +2 V was used.

17 **2.7 Modeling**

18 The experimental I – V curves were described using a theoretical model based on the
19 Poisson and Nernst-Planck (PNP) equations.⁵⁴⁻⁵⁷ This simplified, one dimensional
20 model assumes a conical geometry for the nanopore and gives the I – V curves in terms
21 of the tip (d) and base (D) diameters of the pore and the surface concentration σ of
22 fixed charges. For the sake of simplicity, this surface concentration was assumed to be
23 constant along the pore length (see further details in the supporting information).

24

25 **3. Results and discussion**

1

2 Figure 2 shows I - V curves of the as-prepared (before modification) single
3 nanopore measured in 100 mM KCl solution at neutral and acidic conditions. To
4 achieve this goal, the single-pore membrane was fixed between the two chambers of
5 the conductivity cell. The electrolyte solution was filled on both sides of the single
6 pore membrane and the electrodes on each side of the nanopore were arranged in such
7 a way that high currents at positive voltages and low currents at negative voltages
8 were obtained. Experimental and theoretical⁵⁴⁻⁵⁹ studies have proved that the as-
9 prepared single conical pores exhibit cation-selectivity and rectify the ion current (*i.e.*,
10 cations preferentially flow from the tip towards the base opening) due to presence of
11 ionized $-\text{COO}^-$ groups on the pore surface at neutral pH conditions.

12 The base opening (D) diameter of the conical pore was directly calculated from the
13 field emission scanning electron microscopy images of the multipore membrane
14 simultaneously etched with the single-pore foil (Figure 3C). The pore tip (d) opening
15 was calculated from the linear fitting of the I - V curve for the uncharged nanopore ($\sigma =$
16 0) at pH 3.5 (Figure 2). In our case, $D = 270$ nm, $d = 20$ nm and the membrane
17 thickness $L = 12$ μm . Once the pore diameters were determined, the only free
18 parameter of the model was the surface charge density that was calculated from the
19 fitting of the I - V curve of the as-synthesized pore at pH 6.5 to the theoretical model.
20 The best fitting was obtained for $\sigma = -0.2 e/\text{nm}^2$, where e is the elementary charge
21 (Figure 2).

22 Figure 3A shows the immobilization of $t\text{-BuC}[4]\text{C-NH}_2$ molecules on the
23 nanopore surface. For this purpose, the ($-\text{COOH}$ groups were first activated by
24 exposing the single-pore membrane to a solution of EDC and pentafluorophenol
25 (PFP). This step resulted in the formation of an amine-reactive PFP reactive-ester

1 on the pore surface. Then, the activated single-pore membrane was exposed to a *t*-
2 BuC[4]C–NH₂ solution. During this reaction step, the PFP-reactive intermediate
3 was covalently coupled with the terminal-amine groups of *t*-BuC[4]C molecules.

4 To verify the success of the chemical reaction, the wetting properties of PET
5 foil were studied by CA measurements before and after modification. The PET foil
6 was treated using the same experimental protocol as the single-pore membrane.
7 Figure 3B shows the contact angle of the PET foil having carboxylic acid groups
8 ($54 \pm 2^\circ$). After the functionalization of *t*-BuC[4]C moieties, the contact angle
9 increases to $83 \pm 2^\circ$. This result suggests that the modified PET surface becomes
10 more hydrophobic due to the presence of tertiarybutyl group *t*-BuC[4]C moieties.

11 Previous studies have demonstrated that calixcrowns (host) in 1,3-alternate
12 conformation exhibit remarkable affinity towards Cs⁺ ion (guest).¹⁷⁻¹⁹ The host-
13 guest complexation occurring inside the confined environment is used to tune the
14 pore surface charge density and ionic conduction. Figure 4A represents the Cs⁺ ion
15 complexation with *t*-BuC[4]C moieties on the nanopore surface via coordination
16 with crown ether oxygen atoms and cation– π interactions originated from the
17 aromatic rings of 1,3-alternate conformation.¹⁴

18 Figure 4B shows the *I*–*V* characteristics of the single conical pore decorated
19 with *t*-BuC[4]C moieties in the presence of various cesium concentrations
20 prepared in 10 mM Tris-buffer (pH 6.5). At low Cs⁺ concentration (≤ 1 mM), the
21 *t*-BuC[4]C-modified pore was in a nonconducting “*off*” state, as observed from the
22 very low current (~ 0.2 nA) flowing across the membrane. When the modified
23 pore was exposed to 10 mM and 25 mM Cs⁺ concentrations, the rectified current at
24 -2 V increases from 0.2 to 0.4 nA and 0.7 nA, respectively. This fact suggests that
25 the Cs⁺ ion makes the pore selective to anions due to the formation of a positively

1 charged complex between the *t*-BuC[4]C moieties and the metal cation on the pore
2 surface. By further increasing the Cs⁺ ion concentration from 25 mM to 100 mM,
3 the rectified ion current increases from 0.7 nA to 1.66 nA at -2 V. No significant
4 change in the positive current at +2 V was observed on exposing the modified pore
5 to various Cs⁺ concentrations. This fact clearly shows that *t*-BuC[4]C-Cs⁺
6 complexes switched the pore transport behaviour from the nonconducting “*off*”
7 state to conducting “*on*” state. Moreover, the Cs⁺ concentrations affect the extent
8 of anion conduction across the modified single-pore membrane. The continuous
9 curves of Figure 4B show the model results assuming that the complexation of Cs⁺
10 ions produces a positive fixed charge density on the pore surface. These curves
11 have been calculated using the same charge density $\sigma = 0.05 e/\text{nm}^2$ for all the Cs⁺
12 concentrations. The presence of the Cs⁺ ions can reverse the pore charge even at
13 low concentrations. The qualitative agreement between theory and experiment for
14 $V < 0$ suggests that the substitution of the carboxylic acid groups of the as-
15 prepared pore with the tert-butylcalix[4]arene-crown (*t*-BuC[4]C) moieties is
16 significant. For $V > 0$ the model overestimates the currents, probably due to a
17 spatial accumulation of positive fixed charges close to the pore tip (see reference
18 54 for details).

19 We proceeded now to study the specificity of the *t*-BuC[4]C-modified pore
20 towards various alkali cations. To this end, the same modified pore was exposed to
21 electrolyte solutions having different alkali ions at 100 mM and pH 6.5. The I - V
22 curves of Figure 4C did not show any significant change in the rectified current in
23 response to Li⁺, Na⁺, K⁺, and Rb⁺ cations. On the contrary, upon exposure to Cs⁺
24 ion, the pore exhibited high ionic current rectification (Figure 4C). This fact
25 suggests that the functional calixcrown moieties have the ability to specifically bind

1 the Cs⁺ cation. The *t*-BuC[4]C–Cs⁺ complex imparts positive charge to the pore
2 surface, leading to current rectification. This fact shows that the inner pore was
3 switched from a hydrophobic and uncharged non-conductive state to a hydrophilic
4 and charged conductive state upon exposure to Cs⁺ ion. This result is in agreement
5 with the model calculations shown in Figure 4C, where the curve corresponding to
6 CsCl was calculated using $\sigma = 0.05 \text{ e/nm}^2$. In the case of the other alkali cations,
7 the estimated σ value was very low ($\sim 0.005 \text{ e/nm}^2$). Again, the deviations between
8 theory and experiments for $V > 0$ should be ascribed to an accumulation of
9 positive fixed charges close to the pore tip.⁵⁴

10

11 Figure 5A shows the pore conductance as a function of the Cs⁺ concentration and
12 the type of alkali cation. The data is obtained from the respective I – V curves of Figure
13 4B and 4C at -2 V . Increasing the Cs⁺ concentration led to a gradual increase in the
14 pore conductance because of the positive charges originated by the binding of Cs⁺
15 cations to calixcrown moieties. For the case of other alkali cations, the small increases
16 in the conductance were noticed on exposure to Li⁺ and Na⁺ cation solutions. While
17 the pore exhibits some conductance on exposing to K⁺ and Rb⁺ solutions, suggesting
18 the partial complexation of these cations with the calixcrown moieties, the sharp
19 increase in the Cs⁺ conductance with respect to the case of the other alkali cations
20 confirms the specific binding of this cation. Figure 5B shows the variation of current
21 over different cycles of reversible binding and unbinding of the Cs⁺ ions in the
22 absence and presence of this ion. The reversible changes observed in the ion current
23 suggest again that the surface charge density dictates the transport properties,
24 switching the nanopore from a nonconducting state with calixcrown moieties to a
25 conducting state with *t*-BuC[4]C–Cs⁺ chelates.

1 Finally, we have also examined the reproducibility of the nanofluidic device. To
2 this end, we have fabricated another single conical nanopore. In this case, we have
3 also performed the control measurements using a non-functionalized nanopore. Figure
4 6A shows $I-V$ curves of the as-prepared single nanopore before modification
5 measured in 100 mM alkali electrolyte solutions. Then the conical nanopore was
6 functionalized with calixcrown moieties under the same set of experimental
7 conditions. Figure 6B shows that addition of Li^+ , Na^+ , K^+ and Rb^+ cations in the
8 electrolyte solution caused relatively small changes in the current. On the contrary,
9 exposure to Cs^+ ion led to a significant change in the $I-V$ behavior: current
10 rectification was observed due to the formation of $t\text{-BuC}[4]\text{C}-\text{Cs}^+$ complexes,
11 enabling anion conduction through the modified nanopore.

12

13 **4. Conclusions**

14 In summary, we have described experimentally and theoretically Cs^+ ion induced
15 ion conduction through a nanofluidic pore functionalized with calixcrown moieties
16 via carbodiimide coupling chemistry. On exposure to Cs^+ ion, the asymmetrical
17 distribution of positively charged groups along the conical nanopore led to the
18 selective conduction of anions, resulting in the inversion of the rectification observed
19 in the $I-V$ curve. On the contrary, other alkali cations did not show drastic changes in
20 the current rectification. Theoretical modelling of the experimental results based on
21 the Nernst-Planck and Poisson equations confirmed further the validity of the
22 experimental approach to cesium-induced ion conduction through single nanofluidic
23 pores.

24

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11

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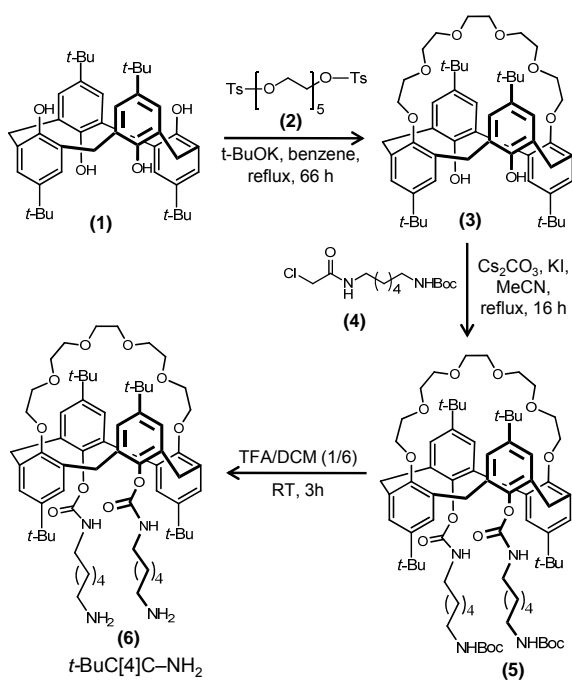
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1 **Figures and Legends**

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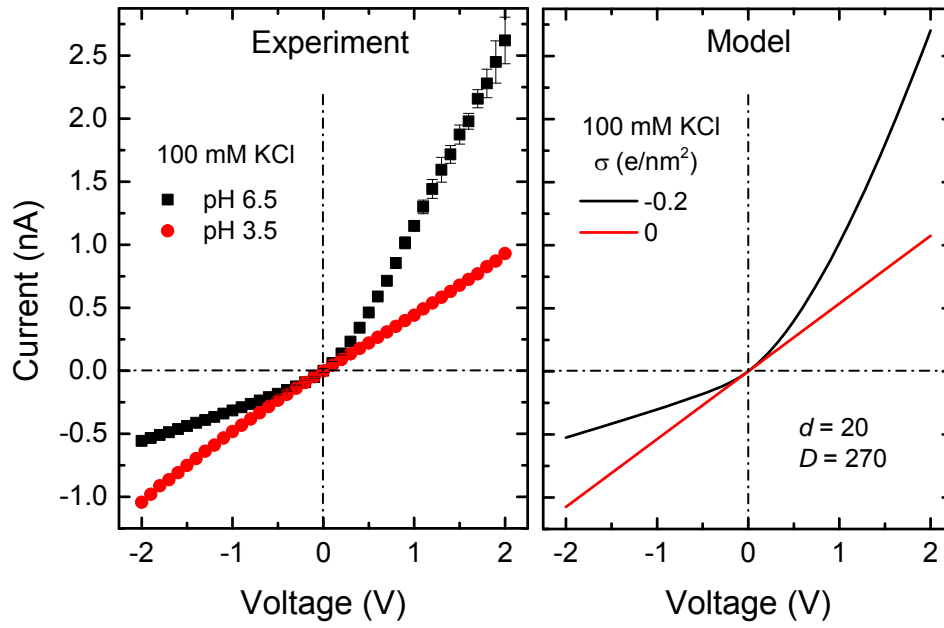


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4 **Figure 1.** Synthesis of the amine terminated *tert*-butylcalix[4]arene crown (*t*-BuC[4]C-NH₂).

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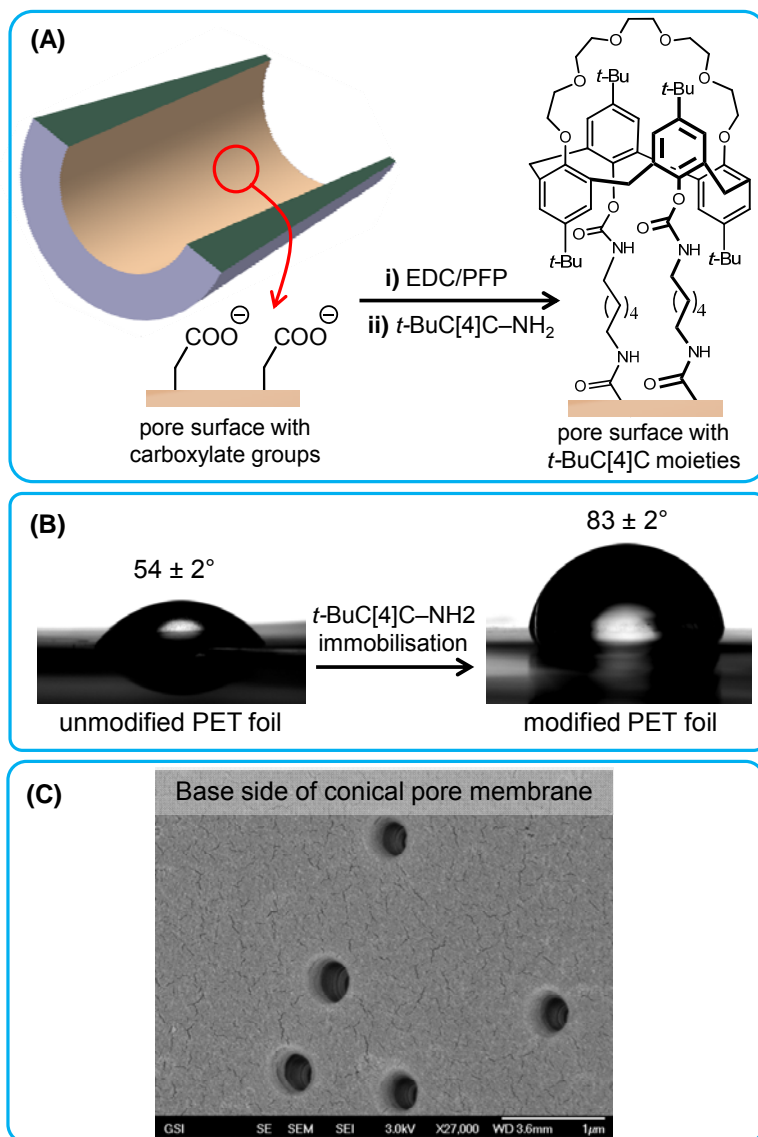
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2 **Figure 2.** Experimental and theoretical I - V characteristics of the as-prepared single
 3 conical pore recorded in 100 mM KCl solution at pH 6.5 and 3.5, separately.

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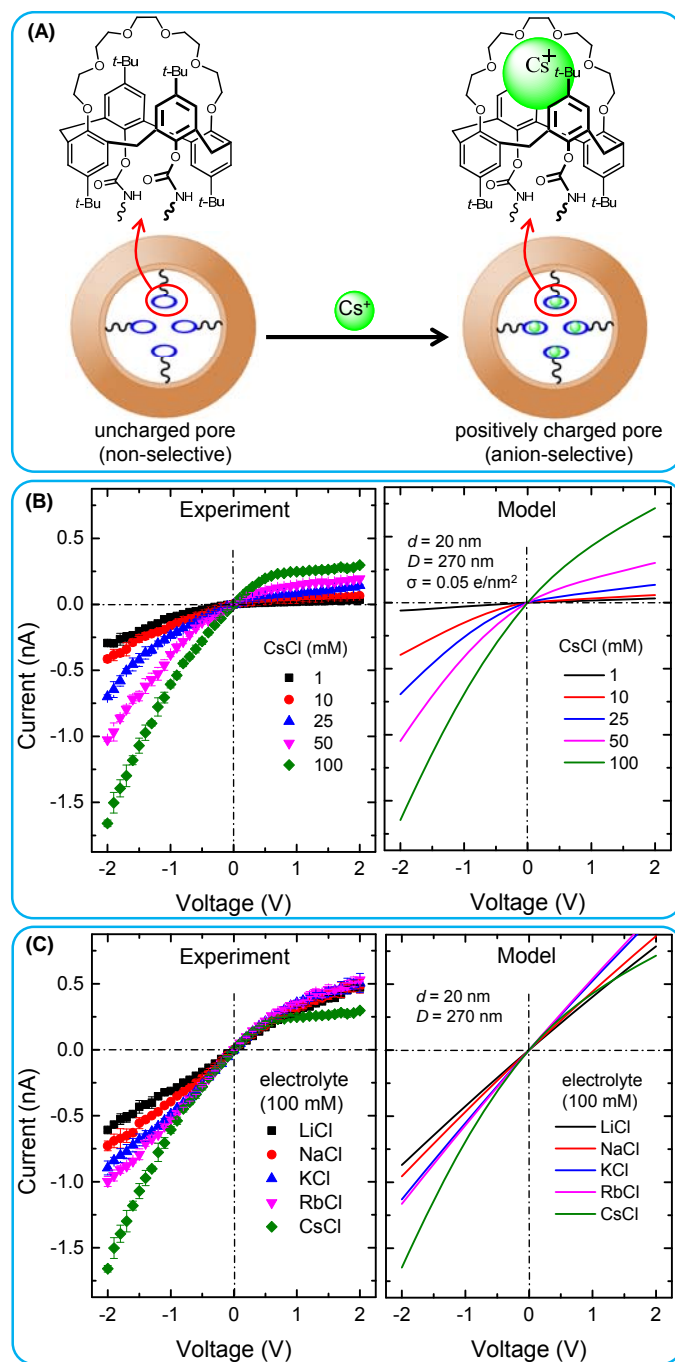


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2 **Figure 3.** (A) The functionalization of the carboxylic acid groups on the surface of
 3 asymmetric nanopore with *tert*-butylcalix[4]arene–crown (*t*-BuC[4]C) moieties via
 4 carbodiimide coupling chemistry. (B) The contact angles of an unmodified PET
 5 membrane with carboxylic acid groups and a functionalized membrane having *t*-
 6 BuC[4]C moieties. (C) FESEM image of the base opening side of the conical pores in
 7 a PET membrane having 10^7 pores cm^{-2} .

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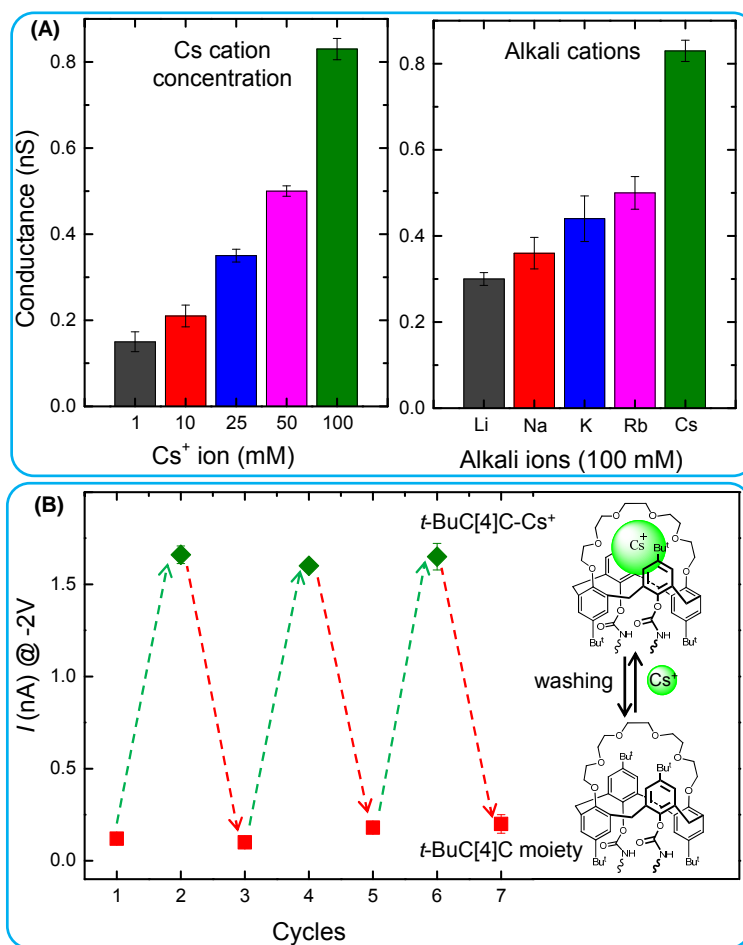


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2 **Figure 4.** (A) Scheme representing the Cs^+ ion complexation with *t*-BuC[4]C
 3 moieties on the pore surface (B) Experimental and theoretical $I-V$ characteristics of
 4 the modified pore exposed to various concentrations of cesium cation. (C)
 5 Experimental and theoretical $I-V$ characteristics of the modified pore after the
 6 addition of various alkali cations (100 mM, chloride salts) in the electrolyte solution,
 7 separately. For CsCl, the estimated value of σ is 0.05 is e/nm^2 , while σ is ~ 0.005
 8 e/nm^2 for the other alkali cations.

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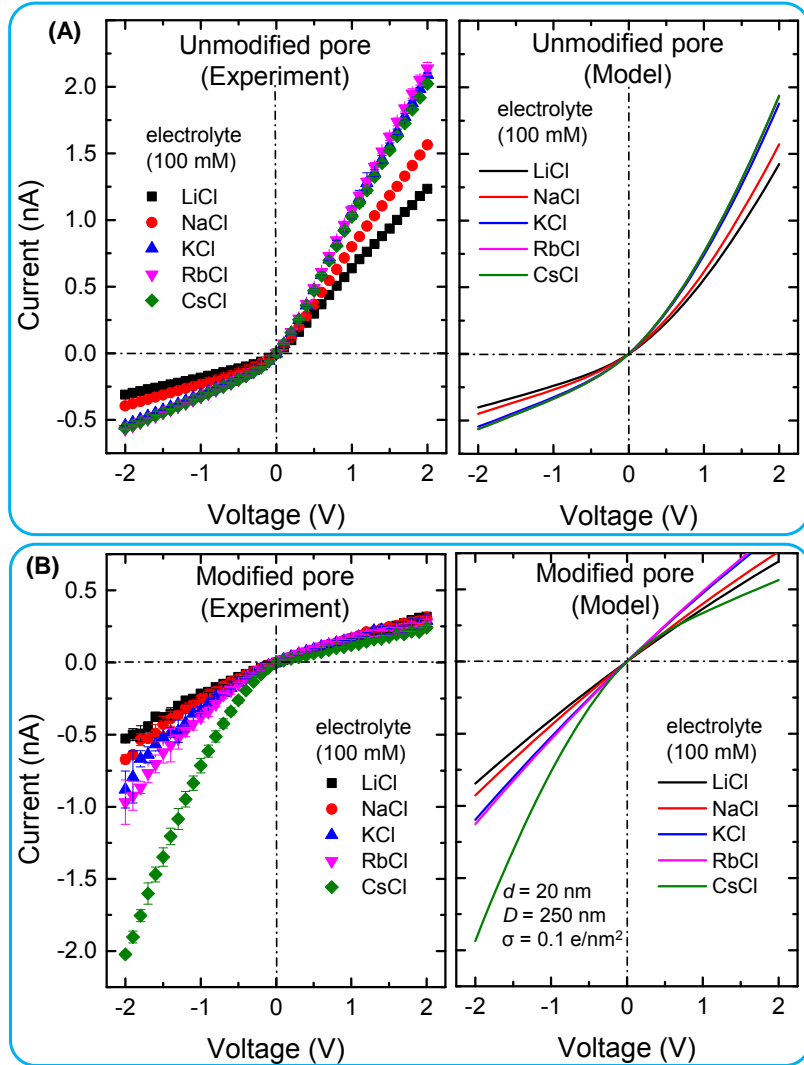


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3 **Figure 5.** (A) The nanopore conductance measured upon exposure to different Cs
 4 cation concentrations and alkali ions obtained from the experimental $I-V$ curves
 5 shown in Figure 4B and C, respectively. (B) Different cycles representing the changes
 6 in the current because of the reversible complexation/decomplexation of Cs⁺ ion with
 7 the t-BuC[4]C moieties. The current values shown are obtained at a potential of -2 V
 8 by exposing the modified pore to electrolyte solutions with no CsCl (red) and 100
 9 mM CsCl (green).

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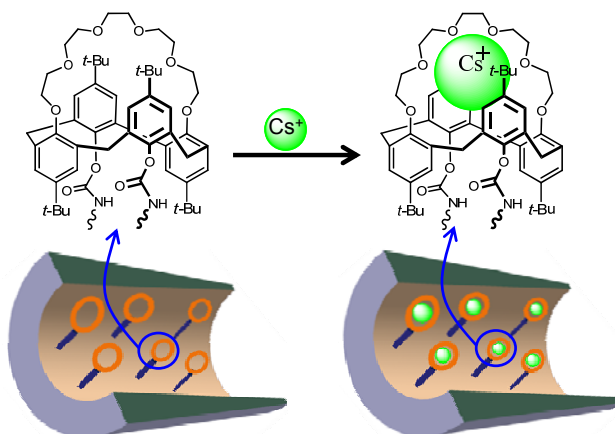
3 **Figure 6.** (A) Experimental and theoretical I - V characteristics of the unmodified pore
 4 on exposure to various alkali cation (100 mM, pH 6.5) solutions, separately. (B)
 5 Experimental and theoretical I - V characteristics of the modified pore after the
 6 addition of various alkali cations (100 mM, chloride salts) in the electrolyte
 7 solution, separately. For CsCl, the estimated value of σ is 0.1 is e/nm^2 , while σ is
 8 $\sim 0.01 \text{ e/nm}^2$ for the other alkali cations.

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TOC graphics



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