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Additional Information

Impact of *Eucalyptus nitens* and *Pinus radiata* fiber properties on the production process of lignocellulose nanofibrils

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Abstract

Lignocellulose nanofibrils (LCNFs) are nano-objects produced from different sources of lignocellulosic biomass. The physical and chemical properties of the plant cells that make up the raw material are diverse and affect the qualities of the LCNFs. Due to their chemical differences, understanding the effect that the physical properties of the raw material confer on LCNFs is complex. This study aims at comprehending the impact that the physical properties and the chemical composition of the raw materials from *Pinus radiata* and *Eucalyptus nitens* have on the mechanical processes of fibrillation and the final properties of LCNFs. The anatomical, physical, and component differences of the fibers in the different fractions and species showed a non-effect on the longitudinal disintegration during the mechanical processes of fibrillation. In addition, the LCNFs produced from smaller fibers showed a smaller average width, with *Eucalyptus nitens* being the species that had smaller and more homogeneous nanofibrils. Nevertheless, the organization of the components (hemicellulose) from *Pinus radiata* on the surface of the fibrils, generates very hydrated and large fibrils and flocs, increasing the viscosities of the LCNF suspensions. A model was established between the intrinsic viscosity and aspect ratio of LCNFs ($\rho[\eta] = 0.15p^{1.68}$) and delignified CNFs ($\rho[\eta] = 0.031p^{1.94}$) that are independent of the pretreatments and the flexibility of the fibrils. Finally, the glass transition temperatures of lignin were not affected by physical changes in the raw material.

Keywords: *Pinus radiata*, *Eucalyptus nitens*, lignocellulose nanofibers, morphology, rheology

1. Introduction

The anatomical differences between the *Pinus radiata* D. Don and *Eucalyptus nitens* Maiden wood are well known (Sixta, 2006; Ek, 2009). Such differences originate mainly from the biological nature of the different species and their genetic and growth conditions (Ek, 2009). The types of plant cells that make up pine wood (tracheids, parenchyma, and epithelial cells) are less and more uniform than the plant cells of eucalyptus wood (vessels, libriform fiber, fiber tracheids, longitudinal and ray parenchyma). In addition, the plant cells of both species have differentiated morphology, e.g., Pine fibers (tracheids) are longer, thicker-walled, less flexible, and more resistant than eucalyptus fibers, which are smaller, thinner, and have smaller fibrillar angles (Melo, 1995; Donaldson, 2008; de Assis, 2019). The anatomical and morphological differences of plant cells can influence the production and characteristics of lignocellulosic nanomaterials.

Lignocellulose nanofibrils (LCNFs) are nano-objects with widths lower than 100 nm, produced from the cell walls of plant cells of wood and plants by mechanical fibrillation processes in an aqueous medium (Chirayil, 2014; Oliaei, 2021). LCNFs have lengths of several micrometers, so

43 their aspect ratio is in the order of hundreds (Albornoz-Palma, 2020b). LCNFs have great potential
44 for uses; they are more ecological and cheaper to produce than delignified CNFs (Delgado-Aguilar,
45 2015; Rojo, 2015; Zhang, 2022).

46 LCNFs are semi-flexible fibrils that generate viscous suspensions due to the hygroscopic
47 nature of the cellulose and hemicellulose, the high aspect ratio, and the high specific surface area of
48 the nanofibrils, causing strong interconnections at low concentrations (Nechyporchuk, 2016). At low
49 concentrations of fibrils (dilute region), the interaction effects between them are insignificant, and the
50 suspensions behave as Newtonian fluids (Iwamoto, 2014). However, above the critical concentration
51 (Krishnan, 2010), the fibrils begin to interact, forming flocs and entangles, and suspensions behave
52 as Pseudoplastic fluids (Saarrinen, 2009; Lê, 2018).

53 The differences in the viscosities of the fibrils depend on the physical and chemical
54 characteristics of the fibrils, whose features are given by the raw material and the production
55 processes of LCNFs.

56 The role of the physical properties of the plant cells of raw material in the production of
57 LCNFs is still unclear, as there is some controversy regarding its importance and the variables that
58 affect the production processes of LCNFs (Espinosa, 2017a; Espinosa, 2017b). In several studies, it
59 has been shown that the morphology of the fibers of the raw material, crystallinity index, and the
60 degree of polymerization have a degree of influence on the fibrillation process (Stelte, 2009; Besbes,
61 2011; Wang, 2015; Qin, 2016; Henríquez-Gallegos, 2021), the characteristics of the LCNFs, the
62 mechanical properties, and reinforcement capability (Zimermann, 2004; Henriksson, 2008; Stelte,
63 2009). For this reason, this study seeks to understand the effect of the physical properties of fibers
64 from *Pinus radiata* and *Eucalyptus nitens*, and their components, on the mechanical production
65 process and the morphological and rheological characteristics of LCNFs.

67 2. Materials and methods

68
69 **Materials:** Wood chips from *Pinus radiata* and *Eucalyptus nitens* were used. The wood chips
70 were fibrillated in a Bauer Bros Co. Model 148-2 disc refiner. The pulp of each species was
71 fractionated by size in water using a Bauer-McNett classifier with 28, 48, and 100 mesh. The
72 classification was carried out for 30 min, with constant water flow and agitation, obtaining four pulps
73 with different fiber sizes. The pulps were named indicating the species and the fraction of the fibers
74 retained in the mesh (Pinus-28, Pinus-48, Pinus-100, Pinus-under 100, Eucalyptus-28, Eucalyptus-
75 48, Eucalyptus-100, and Eucalyptus-under 100).

76 **Chemical compositions of fibers:** The polymeric components of the fibers were quantified
77 according to the methodology mentioned in Andrade et al. (2021). The extractives and ash were
78 quantified according to T 204 cm-97 (with acetone as solvent) and T 211 cm-9, respectively.

79 **Morphology of fibers:** An L&W fiber analyzer was used to quantify the morphological
80 parameters of the fibers. Fiber suspensions of 0.4% (w/w) consistency were used for the
81 measurements. All measurements were performed in triplicate.

82 **Degree of polymerization (DP):** Bleached pulp with acid chlorite was used for
83 measurements of the degree of polymerization (DP) of the cellulose. Measurements were made
84 according to the methodology shown in Chakraborty et al. (2006) with cupriethylenediamine (CED)
85 as solvent.

86 **The crystallinity index (CrI):** The pulps were freeze-dried. 50 mg freeze-dried pulp was
87 pressed to form pellets. The pellets were analyzed by X-ray Diffraction (XRD) in a D4 Endeavor X-
88 ray diffractometer (Bruker AXS GmbH, Karlsruhe, Germany) with monochromatic Cu K α radiation
89 ($\lambda = 0.154$ nm) at 40 kV and 20 mA, in a 2θ range between 5° and 45° , with scan steps of 0.02° . The
90 crystallinity index (CrI) was calculated according to Segal et al. (1959), using the Equation 1:

$$91 \text{CrI} = \frac{I_{200} - I_{am}}{I_{200}} \quad (1)$$

93 where I_{200} is the maximum intensity of the (200) reflection ($22.3^\circ 2\theta$), and I_{am} is the minimum in
94 the diffraction intensity associated with the intensity of the amorphous fraction ($18.5^\circ 2\theta$).

95 **Production of LCNF suspensions:** For the production of LCNFs, the pulps were refined
96 with 50,000 revs in a PFI No 205 mill at a consistency of 10% (w/w) and homogenized in a Panda
97 Plus 2000 homogenizer (GEA Niro Soavi) at a pressure of 700 bar, 0.5% (w/w), and 0, 5, 10, and 15
98 passes.

99 **Morphological characteristics of the LCNFs:** Apparent length distributions of the fibrils in
100 the LCNF suspensions were measured in an S3500 Laser Diffraction Particle Size Analyzer
101 (Microtrac Inc. USA) with a lignin refractive index of 1.61 (Li et al. 2018).

102 The average width of the LCNFs with 0, 5, and 10 passes through the homogenizer was
103 determined by micrographs of Transmission Electron Microscopy (TEM) (JEM 1200EX-II JEOL)
104 quantifying 240 fibrils. For LCNFs with 15 passes, width distributions were determined by
105 quantifying 1000 fibril widths using TEM micrographs (Hitachi TEM). ImageJ software was used for
106 processing the TEM images.

107 The aspect ratio of the LCNFs was calculated by dividing the average apparent length and
108 the average width of fibrils.

109 The radius of gyration and the flexibility parameter (ϵ) of LCNFs were determined using the
110 wormlike chain model (Mansfield, 2008) with the experimental values of average apparent lengths
111 and widths, and intrinsic viscosity.

112 **The viscosity of LCNF suspensions:** The viscosities of the LCNF suspensions were
113 measured in a rotational viscometer (Fungilab Premium-L, USA) in the concentration range from
114 0.02 to 0.5% (g/ml). Measurements were made at 25°C and 73.38 s^{-1} . The intrinsic viscosity was
115 determined using the viscosity values below the critical concentration according to the methodology
116 of Albornoz-Palma et al. (2020a).

117 **Lignin nanoparticles:** The determination of the quantity and chemical characterization of
118 the lignin nanoparticles was carried out according to Albornoz-Palma et al. (2022b).

119 **ζ -potential of LCNF suspension:** Particle Metrix equipment (Stabino) was used to measure
120 the ζ -potential of LCNFs in suspensions of 0.05% (w/w) consistency. The method used was the
121 Transmission Potential method.

122 **Formation of nanopapers:** The LCNF suspensions at 0.5% (w/w) were concentrated to 15%
123 (w/w) by vacuum filtration, using $0.45\ \mu\text{m}$ pore diameter membranes. Subsequently, these
124 suspensions were filtered in a Buchner funnel with a 6 cm diameter fritted disk, producing nanopapers
125 with a grammage of $800\ \text{g/m}^2$. The nanopapers were pressed and dried in a vacuum oven for 7 days.

126 **Fourier-Transform Infrared Spectroscopy (FTIR):** FTIR spectroscopy measurements
127 were performed on the dried nanopapers. An Agilent Cary 630 FTIR spectrometer (Mettler Toledo,
128 USA) in the attenuated total reflectance mode (ATR) was used. The spectra were measured in the
129 wavelength range from 4000 to $500\ \text{cm}^{-1}$ with a $4\ \text{cm}^{-1}$ resolution and an average of 64 scans. Spectra
130 were area normalized and compared (principal component analysis) in The Unscrambler X program.

131 **Thermogravimetric analysis (TGA):** To determine the mass losses due to the thermal
132 decomposition of the fibrils, a TGA 851 analyzer (Mettler Toledo, USA) was used. Samples of 3 to
133 5 mg were introduced into $70\ \mu\text{L}$ alumina capsules. The temperature range was from 25°C to 600°C
134 at a rate of $10^\circ\text{C}/\text{min}$ in an oxidative atmosphere ($50\ \text{mL}/\text{min}$).

135 **Differential Scanning Calorimetry (DSC):** Calorimetry analysis in a DSC 822e (Mettler
136 Toledo, USA) was performed on the nanopapers to determine the glass transition temperatures of
137 lignin and the difference in heat capacities. Samples of 3 and 5 mg were placed in $40\ \mu\text{L}$ aluminum
138 crucibles. The analysis method consisted of different consecutive heating/cooling/heating segments
139 between -10°C and 200°C with a heating/cooling rate of $10^\circ\text{C}/\text{min}$. The tests were carried out under
140 an inert atmosphere of N_2 at $50\ \text{mL}/\text{min}$ flow.

141 **Data analysis:** The statistical analysis of the data was performed with the Statgraphics
142 Centurion XVIII program.

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145 **3. Results and discussion**

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Pulp characterization: The fractionation by the size of the mechanical pulps of *Pinus radiata* and *Eucalyptus nitens* generated raw materials with different physical characteristics.

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Table 1 shows that the plant cells of both *Pinus radiata* and *Eucalyptus nitens* presented lower average lengths and average diameters for the fractions obtained with finer meshes. The mechanical refining process used to produce raw materials generates the disintegration of wood chips through shear forces, producing a heterogeneous material in size made up of aggregates of fibers, fibers, and fines. The smallest fibers in the higher mesh fractions were significantly damaged in their structure, showing a lower degree of polymerization and crystallinity index. The latter parameter only had significant changes (with 95% confidence) between the fractions 100 and under 100 for *Eucalyptus nitens* and the fraction under 100 for *Pinus radiata*.

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The percentage of fines had higher in the smaller size fraction due to their reduced dimensions, mainly fiber fragments and smaller plant cells. However, in 28 mesh fractions, the percentage of fines increases because the larger fibers tend to become entangled, favoring the retention of smaller structures.

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Table 1. Morphological characteristics of *Eucalyptus nitens* and *Pinus radiata* pulps.

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Properties	Average length (mm)	Variation of length ^(a) (%)	Average diameter (μm)	Variation of diameter ^(a) (%)	Crystallinity index ^(b) (%)	Variation of crystallinity index ^(a) (%)	Fines (%)	Degree of polymerization	Variation of degree of polymerization ^(a) (%)	Coarseness (mg/100m)
Pinus-28	2.03±0.06	-	44.1±0.2	-	67.4±0.6	-	32.7±0.6	1720	-	1.322±0.008
Pinus-48	1.11±0.04	45	41.7±0.3	5%	66.1±0.7	2%	19.1±3.2	1337	22%	0.724±0.007
Pinus-100	0.53±0.02	74	33.1±0.5	25%	62.0±1.5	8%	28.1±3.9	1302	24%	0.335±0.018
Pinus-under 100	0.27±0.01	87	29.9±0.6	32%	53.5±1.0	21%	69.5±0.8	1102	36%	0.148±0.009
Eucalyptus-28	1.37±0.05	-	41.2±0.1	-	64.9±0.2	-	48.3±0.8	1643	-	9.705±0.010
Eucalyptus-48	0.89±0.05	35	35.9±0.1	13%	64.3±0.3	1%	28.4±0.1	1470	11%	1.543±0.055
Eucalyptus-100	0.51±0.03	63	29.1±0.1	29%	63.6±0.8	2%	30.2±1.8	1354	18%	0.464±0.02
Eucalyptus-under 100	0.24±0.01	82	22.9±0.1	44%	57.0±0.7	12%	69.0±0.7	1187	28%	0.118±0.017

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^(a)Normalized respect to 28 mesh

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^(b)XRD diffractograms are shown in [Figure S1](#) and [Figure S2](#) (Supplementary material)

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Until now, there has been talked about the physical changes of the fibers that component the different fractions. However, anatomical aspects of plant cells and their effects have not been considered. The plant cells that make up pine pulp are simple than those of eucalyptus because they have a limited and more uniform number of cell types. The xylem of softwoods (e.g., *Pinus radiata*) is mainly made up of tracheids (2-3 mm in length and 30 μm in diameter approximately ([Donaldson, 2016](#))) and, to a lesser extent by, parenchyma and epithelial cells. On the other hand, the xylem of hardwoods (e.g., *Eucalyptus nitens*) is mainly made up of vessels, fibers (libriform fibers and fiber tracheids) (0.7-1.6 mm in length and 20 μm in diameter approximately ([Elmas, 2018](#); [Morais, 2019](#))), and to a lesser extent by parenchyma (longitudinal and ray cells) ([Ek, 2009](#)). The anatomical differences between the different structures produce differences in the refining process, with the parenchyma being the easiest to fibrillate, due to its thin-walled and smaller size ([Ek, 2009](#); [Wang, 2015](#)), predominantly found in the smaller size fractions.

179

When comparing the variation in diameter between both species (**Table 1**), it is observed that it is higher for *Eucalyptus nitens*. As mentioned above, the plant cells in the eucalyptus pulps are more diverse, with the fibers found in higher proportions in the larger fractions. This is observed in

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182 the high coarseness values for fraction 28, which is associated with density and cell wall thickness,
183 being higher in libriform and tracheid fibers (Ek, 2009; Carrillo, 2015; Morais, 2019).

184 The variations of the length, crystallinity index, and degree of polymerization were higher
185 for pine pulp, due to the more aggressive refining process required to produce the pulps. For
186 simplicity, the term fibers is used to refer to all anatomical structures of the pulps (plant cells),
187 regardless of species.

188 The characterization of the main components in the pulps of *Pinus radiata* is shown in Figure
189 1a. The percentage of cellulose, hemicellulose, and lignin does not change significantly in the 28, 48,
190 and 100 mesh fractions. However, in the under 100 mesh fraction, the percentage of lignin increases,
191 and cellulose decreases. The extractives were $0.97\pm 0.01\%$ for all fractions. The ashes were 0.25 ± 0.01 ,
192 0.51 ± 0.01 , 0.71 ± 0.01 , and $2.1\pm 0.1\%$ for the 28, 48, 100, and under 100 mesh fractions respectively.
193 On the other hand, *Eucalyptus nitens* pulps (Figure 1b) showed an increase in the percentage of lignin
194 between the 28 and 48 mesh fractions (with no significant differences in the smaller size fractions)
195 and a decrease in the percentage of cellulose. The 100 and under 100 fractions showed a reduction in
196 the percentage of hemicellulose. The extractives were $0.96\pm 0.02\%$ for all fractions. The ashes were
197 0.28 ± 0.01 , 0.20 ± 0.01 , 0.19 ± 0.01 , and $1.2\pm 0.1\%$ for the 28, 48, 100, and under 100 mesh fractions
198 respectively.
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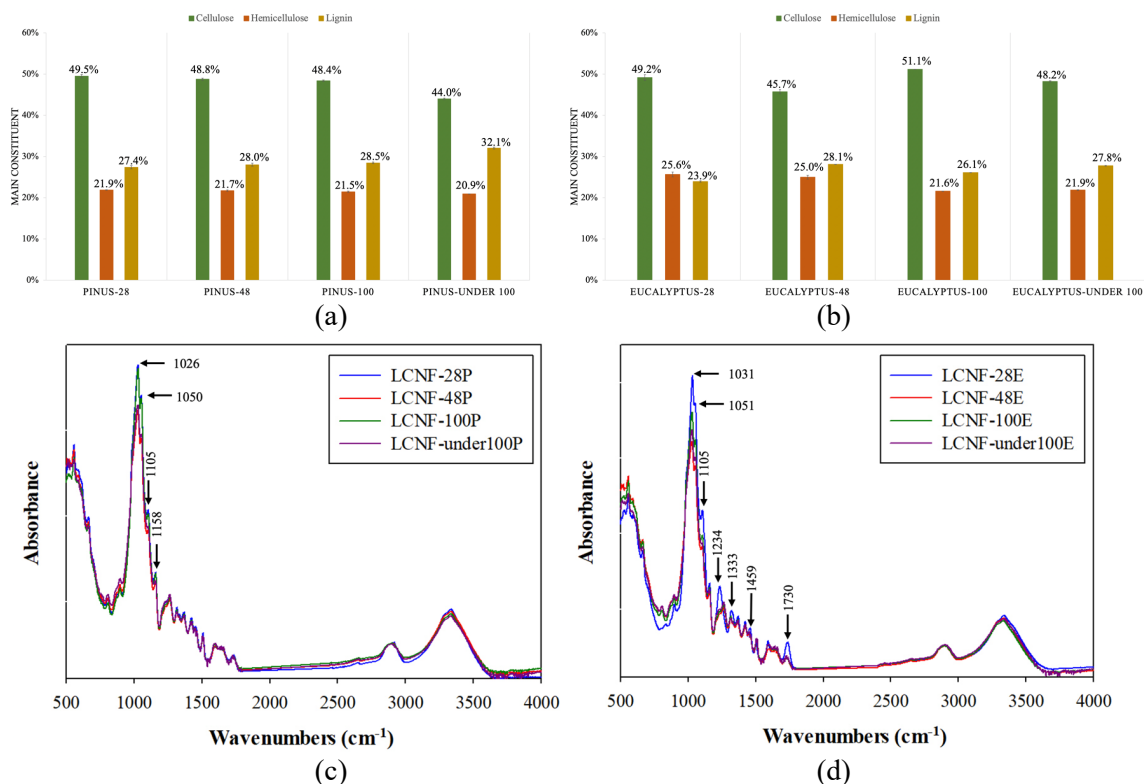
200 **LCNF characterization:** The production of LCNFs is affected by the characteristics of the
201 raw material used (Stelte, 2009; Besbes, 2011; Wang, 2015). The surface chemical changes of the
202 LCNFs with 15 passes through the homogenizer were studied using FTIR-ATR absorption spectra
203 shown in Figure 1c-d. The LCNFs produced from pine pulp (Figure 1c) show slight differences in
204 the absorbance of specific peaks between the different fractions. Peaks at 1158, 1105, 1050, and 1026
205 cm^{-1} showed a lower absorbance for 48 and under 100 fractions. These peaks are associated with the
206 C–O stretching of the ether group and vibrations of C–O bonds in primary and secondary alcohol
207 groups (Bykov, 2008; Lehto, 2018). This decrease in the peaks indicates a lower presence of the β -
208 O-4 ether bonds in the lignin, generated by the severe mechanical process applied to the fibers that
209 produced its rupture.

210 For the LCNFs from *Eucalyptus nitens* pulps, a higher number of differences in the absorption
211 peaks were observed between the fibrils of the 28 mesh fraction and the other fractions. Peaks 1730
212 (C=O stretching of the unconjugated ketone, carboxyl, and ester group (Stark, 2016), 1459 (CH₂
213 deformation stretching in lignin and xylan (Lehto, 2018)), 1333 (aryl ring breathing with C–O stretch
214 (Stark, 2016)), 1234 (C–O stretching of the ether group (Lehto, 2018)), 1105, 1051, and 1031 cm^{-1}
215 (C–O bonds in primary and secondary alcohol) show a lower absorbance for the LCNFs from 48,
216 100, and under 100 mesh fractions. This indicates that the lignin in the LCNFs of these fractions has
217 a greater presence on the surface of the fibrils, a lower degree of oxidation, and a lower amount of β -
218 O-4 ether bonds.

219 To understand the changes in fibrils during the production process of LCNFs, due to the
220 physical characteristics of the fibers in the raw materials, their characteristics of the fibrils have been
221 determined at different stages of the production process, the results of which are shown in Table 2.

222 The mechanical refining pretreatment generates a decrease in the average widths and apparent
223 lengths of the fibrils (0 passes) produced from raw materials with a smaller fiber morphology.
224 However, when analyzing the variations of width, no changes are observed between the fractions of
225 the different species or between the species. The decrease in the average diameters and the constant
226 values in the variation of width of the fibrils indicate that the anatomical, physical, and chemical
227 differences of the fibers in the raw materials do not have an effect on the longitudinal disintegration
228 during the refining process, but they generate differences in the characteristics of LCNFs.
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235 **Figure 1.** (a) Chemical composition of mechanical *Pinus radiata* pulps, (b) chemical composition of
236 mechanical *Eucalyptus nitens* pulps, (c) FTIR-ATR spectrum of LCNFs from *Pinus radiata* pulp, and
237 (d) LCNFs from *Eucalyptus nitens* pulp.

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Average lengths don't show an important difference between the 28, 48, and 100 mesh fractions, and decrease for the under 100 mesh fraction, in both species (Table 2). The variation of apparent length decreases with the decrease in the size of the fibrils in the raw material. Refining is a disintegration process of the fibers using shear forces, which affects larger fibers in greater proportion, generating their rupture.

To analyze the effect that the mechanical process of homogenization has on the average width, and apparent length of the fibrils with different numbers of passes (5, 10, and 15 passes, Table 2), a decrease in these parameters is observed, whose variation increases and is higher for the width. This is because the homogenization process produces the disintegration of the fibrils using the mechanism of impact and shear force mainly (surface mechanisms) (Lee, 2014; Costa, 2018), which favors longitudinal disintegration, increasing the aspect ratio.

However, when comparing the variation of apparent lengths and widths of the different LCNFs by species, no important differences are observed, except the LCNFs of *Pinus radiata* of the under 100 mesh fraction with 10 and 15 passes, whose variations of apparent length decreased. This non-change in values of variations indicates that the morphological, chemical, and anatomical differences of the fibers of the raw materials don't have a significant effect on fibrillation by homogenization. LCNFs of *Pinus radiata* show a higher variation of apparent lengths, caused by the lower coarseness, which favors their rupture.

The aspect ratio increased with the number of passes through the homogenizer (Table 2) because the homogenization process predominantly generates longitudinal disintegration over transverse disintegration. The decrease in the size of the fibrils due to homogenization produces an increase in the fibrillar population per unit mass, causing an increase in the hydrodynamic volume of

261 the fibrils in the suspensions, which is reflected in an increase in the resistance to flow and intrinsic
 262 viscosity.

263 Chemical differences in the raw materials can generate differences in the surface charges of
 264 the fibrils, affecting the production process and the characteristics of the LCNFs. Although the FTIR-
 265 ATR spectra showed an increase in the oxidized groups in the LCNF from *Eucalyptus nitens* and the
 266 28 mesh fraction, ζ -potentials of fibrils do not change significantly between the LCNFs of different
 267 fractions and the number of passes through the homogenizer (Figure 1d and Table 2). The LCNFs of
 268 *Eucalyptus nitens* showed a slightly higher ζ -potential (absolute value) caused by the higher presence
 269 of hemicellulose (uronic acids) (Nechporchuk, 2016).

270

271 **Table 2.** Morphological characterization of LCNFs with different numbers of passes through the
 272 homogenizer and lignin content.
 273

Raw materials	Passes	Average Width (nm)	Variation of width ^(a) (%)	Apparent average length (μm)	Variation of apparent length ^(c) (%)	Aspect ratio	Intrinsic viscosity (ml/g)	ζ -potential ^(d)	Lignin nanoparticle ^(e) (% g/g)
Pinus-28	0	197.3	99.6 ^(b)	12.6	99.4 ^(b)	64	96	-21.0	1.6
	5	120.6	39	8.9	29	74	151	-19.8	1.6
	10	88.1	55	7.3	42	83	173	-19.3	1.7
	15	51.8	74	6.2	51	120	281	-19.1	1.9
Pinus-48	0	166.2	99.6 ^(b)	12.4	98.9 ^(b)	75	145	-20.5	1.5
	5	93.5	44	8.9	28	95	206	-20.3	1.5
	10	76.7	54	7.3	41	95	220	-20.1	1.5
	15	50.4	70	6.2	50	123	300	-20.0	1.6
Pinus-100	0	140.3	99.6 ^(b)	12.5	97.6 ^(b)	89	183	-20.9	1.2
	5	89.5	36	8.8	30	98	219	-20.4	1.5
	10	61.8	56	7.3	42	118	290	-20.2	1.5
	15	48.5	66	6.2	50	128	326	-20.1	2.4
Pinus-under 100	0	110.2	99.6 ^(b)	9.1	96.6 ^(b)	83	167	-20.8	1.3
	5	66.1	40	7.4	19	112	279	-19.7	1.4
	10	53.1	52	6.3	31	119	296	-19.4	1.5
	15	38.3	65	5.3	42	138	349	-19.4	2.3
Eucalyptus-28	0	145.8	99.6 ^(b)	8.8	99.4 ^(b)	60	109	-23.3	2.0
	5	92.4	37	7.6	14	82	156	-22.6	2.1
	10	67.1	54	6.4	27	95	194	-22.1	2.1
	15	48.5	67	5.5	38	113	273	-24.0	2.6
Eucalyptus-48	0	132.1	99.6 ^(b)	8.5	99.0 ^(b)	64	122	-23.6	1.8
	5	88.7	33	7.6	11	86	175	-22.6	2.3
	10	58.1	56	6.4	25	110	266	-22.6	2.3
	15	46.2	65	5.5	35	119	293	-22.9	2.6
Eucalyptus-100	0	114.7	99.6 ^(b)	8.4	98.4 ^(b)	73	134	-23.0	1.6
	5	81.5	29	7.6	10	93	203	-22.8	1.8
	10	52.7	54	6.5	23	123	297	-22.3	3.0
	15	39.1	65	5.3	37	136	333	-23.0	3.1
Eucalyptus-under 100	0	88.9	99.6 ^(b)	7.5	96.9 ^(b)	84	173	-23.7	1.9
	5	54.4	39	6.3	16	116	295	-23.0	2.1
	10	41.3	54	5.4	28	131	338	-22.6	3.2
	15	32.8	63	4.7	37	143	380	-22.2	3.5

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^(a) Normalized to the average width of 0 passes through the homogenizer for each raw material.

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^(b) Normalized to the parameters of raw materials

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^(c) Normalized to the apparent average length of 0 passes through the homogenizer for each raw material

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^(d) pH of 4.9 ± 0.3

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^(e) Based on total lignin.

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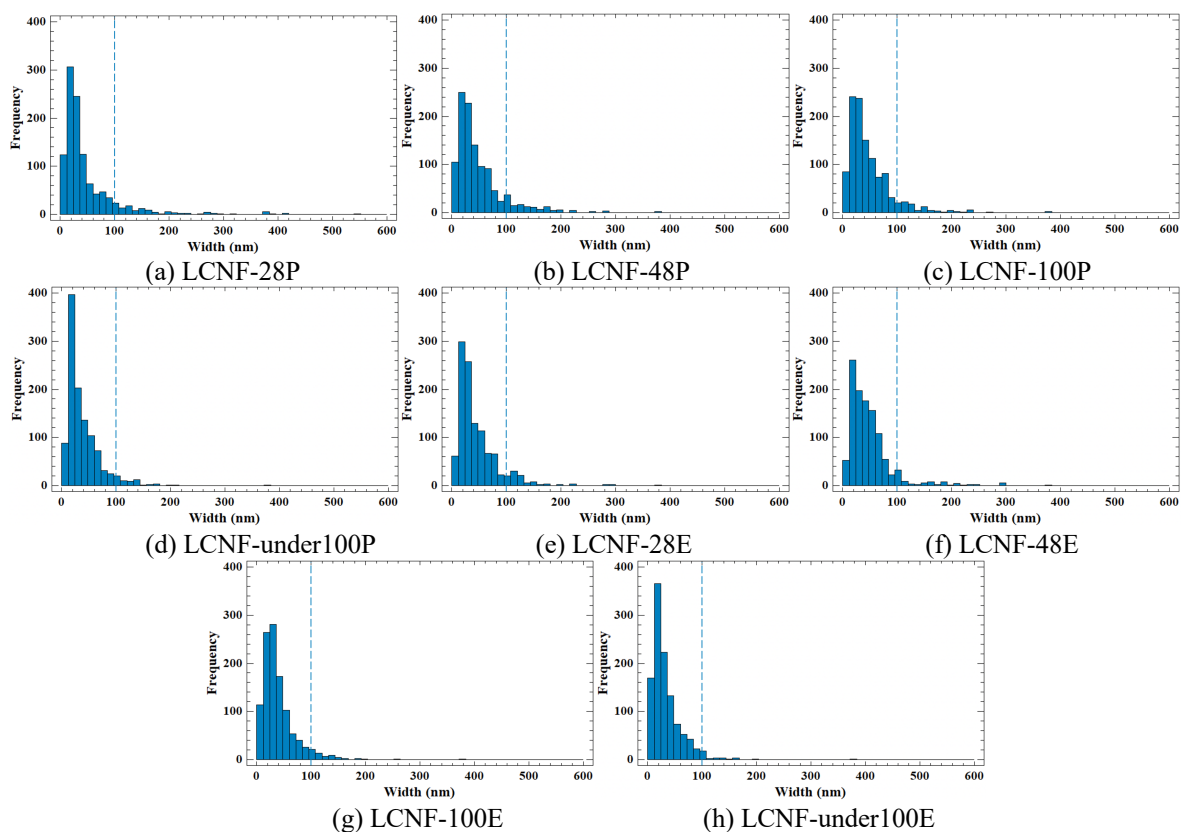
The mechanical processes of fibrillation produce removing part of the lignin from the fibrils, forming lignin nanoparticles (Jiang, 2018). Lignin nanoparticles are spherical form structures found in suspension or between the fibrils (ζ -potential = -0.1 ± 0.1 mV). Table 2 shows that the amount of lignin nanoparticles formed is low (lower than 3.5% of the total lignin of the fibrils), being slightly higher for the LCNFs from *Eucalyptus nitens*. The lignin in eucalyptus fibrils has a higher proportion of S-type lignin (Rencoret, 2007; Antes, 2015). S-type lignin is more reactive and less crosslinked than G-type lignin (Muraleedharan, 2018), which facilitates its removal.

287 Finally, it can be established that both the mechanical processes of refining and
 288 homogenization are not affected by the morphology, structure, and anatomy of the fibers of the raw
 289 materials of each species. However, slight differences between species were observed, where
 290 transverse disintegration and ζ -potential were lower for LCNFs from *Eucalyptus nitens*.
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292 **LCNF characteristics with 15 passes through the homogenizer:** To analyze the effect of
 293 the physical characteristics of the raw materials on the features of the LCNFs, more detailed
 294 measurements of the morphology of the LCNFs with 15 passes through the homogenizer were made,
 295 whose distributions are shown in Figure 2 and Figure 3. The quantification of the distribution
 296 parameters is shown in Table 3.

297 Width distributions shown in Figure 2 for the LCNFs from *Pinus radiata* and *Eucalyptus nitens*
 298 show a displacement in the distributions to smaller width values for the LCNFs produced from
 299 the smallest size fibers. This displacement is reflected in the increase in nanofibrillar yield (fibrils
 300 with diameters lower than 100 nm). For LCNFs from *Pinus radiata*, the nanofibrillar yield increased
 301 from 93% for LCNF-28P to 94, 96, and 98% for LCNF-48P, LCNF-100P, and LCNF-under100P,
 302 respectively. For LCNFs from *Eucalyptus nitens*, the nanofibrillar yield increased from 90% for
 303 LCNF-28E to 91, 92, and 95% for LCNF-48E, LCNF-100E, and LCNF-under100E, respectively.
 304 Table 3 shows that the average widths decrease for the LCNFs produced from the smaller size fibers
 305 (for both species), without significant changes in the coefficient of variation, except for LCNF-28P.
 306 These results indicate that the LCNFs produced from smaller fibers are thinner but without changes
 307 in homogeneity. *Eucalyptus nitens* is the species that has a more homogeneous nanomaterial in width
 308 for the same production process.
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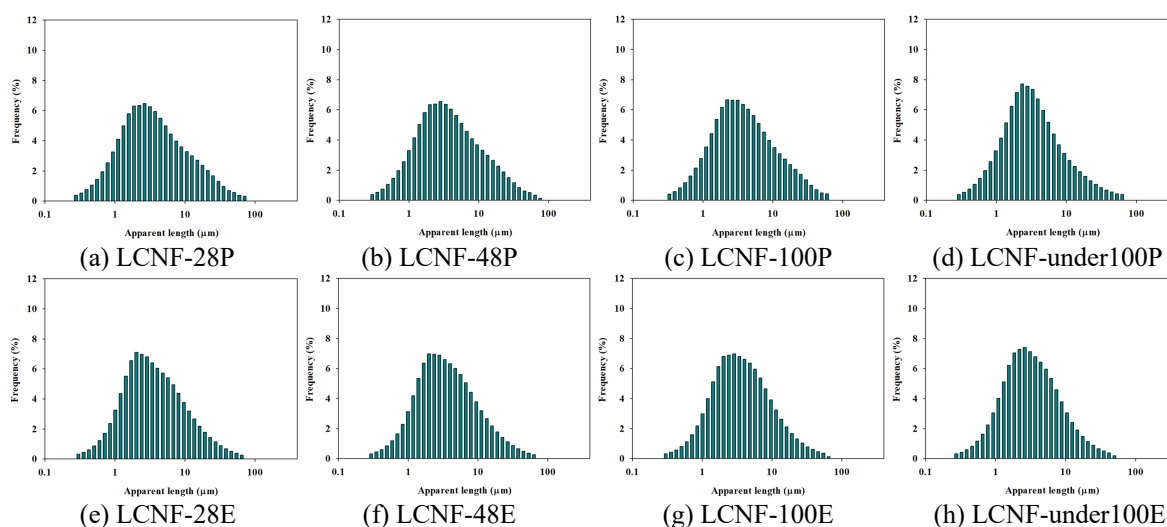
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Figure 2. Width distributions of LCNFs from *Pinus radiata* and *Eucalyptus nitens* pulps.

319 The average width of the LCNFs of the *Eucalyptus nitens* pulps is directly related (linear
 320 relationship) to the average diameter of the fibers of its raw material, with a correlation coefficient
 321 (r) of 0.99. For the LCNFs of *Pinus radiata*, the average width is linearly related to the crystallinity
 322 index of the raw material, with an $r = 0.98$ (correlation coefficient). The other variables (degree of
 323 polymerization and coarseness) of the raw materials are not directly related to the average width of
 324 the LCNFs.

325 **Figure 3** shows the apparent length distributions. These distributions show non-change
 326 between fractions, for each species, except for LCNF-under100E. When quantifying the parameters
 327 that characterize the distributions (**Table 3**), it is observed that the average apparent lengths and the
 328 coefficient of variation of the distributions non-change significantly between the 28, 48, and 100 mesh
 329 fractions for LCNFs from *Pinus radiata*, and the 28 and 48 mesh fractions for LCNFs from
 330 *Eucalyptus nitens*. These results show that the production process of LCNFs affects to a greater extent
 331 the transverse disintegration of the larger fibers, producing nanomaterials of equal or smaller lengths
 332 but without changes in the homogeneity of the apparent lengths, regardless of the specie.
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339 **Figure 3.** Apparent length distributions of LCNFs from *Pinus radiata* and *Eucalyptus nitens* pulps.

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341 Viscosity data as a function of concentration is shown in **Figure 4**. For LCNFs from *Pinus*
 342 *radiata*, it is observed that the viscosity does not change between the 28, 48, and 100 mesh
 343 fractions, and increases slightly for the under 100 mesh fraction. This increase is observed in the semi-dilute
 344 region, above the critical concentration (**Table 3**). The interactions between the fibrils are significant,
 345 and flocs have been generated that increase the resistance to flow (**Saarrinen, 2009**). The flocs depend
 346 on the size of the fibrils and their surface charge. **Table 2** and **Table 3** show that the ζ -potential of the
 347 LCNFs does not change between the fractions and the radius of gyration of the fibrils decreases only
 348 for the under 100 mesh fraction. This decrease indicates an increase in the fibrillar population per unit
 349 mass, generating larger flocs, which increase resistance to flow. For LCNFs from *Eucalyptus nitens*,
 350 a progressive increase in viscosity is observed for the LCNFs produced from the smaller size
 351 fractions. This trend is due to the decrease in the size of the fibrils, observed in a reduction in the
 352 radius of gyration (**Table 3**), which produces an increase in the fibrillar population, generating larger
 353 flocs.

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359 **Table 3.** Morphological and rheological characterization of LCNFs from *Pinus radiata* and
 360 *Eucalyptus nitens* with 15 passes through the homogenizer.
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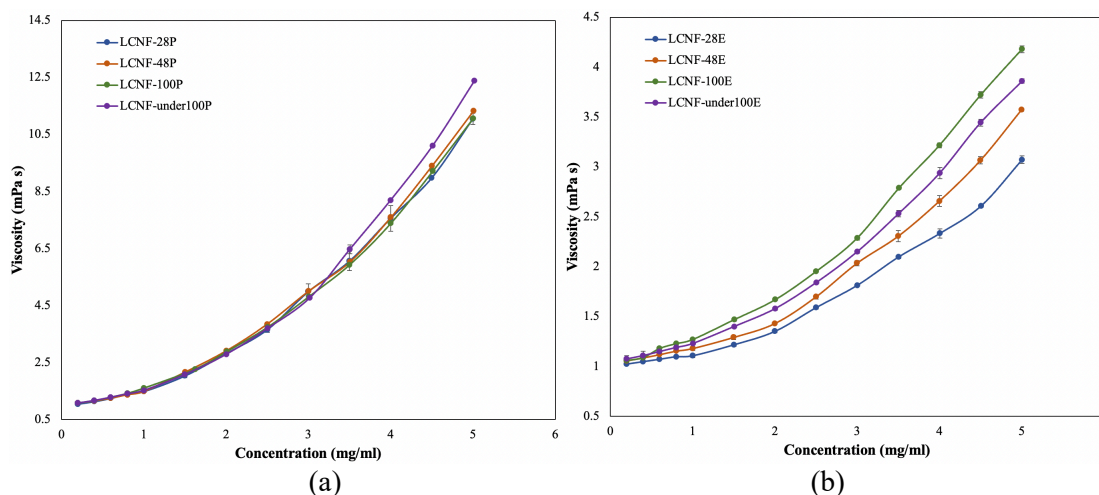
Sample	Average width (nm)	S.D. of the sample ^(a) (nm)/Coefficient of variation (%)	Average apparent length (μm)	S.D. of the sample (μm)/Coefficient of variation (%)	Radius of gyration ^(b) (μm)	Critical concentration (mg/ml)
LCNF-28P	51.8±0.8	21.8/42	6.2±0.1	2.83/46	1.53	3.33
LCNF-48P	50.4±0.3	16.4/33	6.2±0.2	2.86/46	1.53	2.68
LCNF-100P	48.5±0.7	14.5/30	6.2±0.1	2.81/45	1.53	2.50
LCNF-under100P	38.3±0.6	11.4/30	5.3±0.3	2.44/46	1.30	2.13
LCNF-28E	48.5±0.3	13.8/28	5.5±0.1	2.76/50	1.36	3.66
LCNF-48E	46.2±0.7	12.5/27	5.5±0.4	2.76/50	1.35	2.58
LCNF-100E	39.1±0.5	10.4/27	5.3±0.2	2.76/52	1.30	2.14
LCNF-under100E	32.8±0.6	8.7/27	4.7±0.2	2.70/57	1.15	2.08

^(a)Normal distribution by the central limit theorem.

^(b)Wormlike chain model (Mansfield, 2008)

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Figure 4 shows that the viscosities of suspensions of LCNFs from *Pinus radiata* are higher than those of LCNFs from *Eucalyptus nitens*. The difference between the viscosities cannot be explained by the size of the fibrils, because the LCNF-under100P and LCNF-100E have the same average apparent lengths and widths. This difference could be due to the organization of the components on the surface of the LCNFs, where pine fibrils would be surrounded by hemicellulose, unlike eucalyptus fibrils, where glucomannans would be partially exposed (Kumagai, 2018; Tarasov, 2018). LCNFs from *Eucalyptus nitens* have a slightly higher ζ -potential than LCNFs from *Pinus radiata* (Table 2). This could be due to the higher amount of exposed hydroxyl groups of the cellulose on its surface. In addition, hemicellulose is a polymer more hygroscopic than cellulose, so it attracts a greater amount of water molecules due to its branched structure (Correia, 2017), which produces highly hydrated and large fibrils and flocs, generating an increase in resistance to flow.



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Figure 4. Viscosities of LCNFs from (a) *Pinus radiata* and (b) *Eucalyptus nitens* pulps.

There is a power law relationship between morphological characteristics, expressed by aspect ratio (p), and rheological properties, by intrinsic viscosity ($[\eta]$), for cellulose nanomaterials (Tanaka, 2015; Albornoz-Palma, 2020a; Bastida, 2022). When studying this relationship for LCNFs with different production processes and raw materials (mechanical LCNFs, shown in this work; chemical

386 LCNFs (oxidized with acid chlorite, Table S1 Supplementary material); and enzymatic LCNFs
 387 (Albornoz-Palma, 2022b) from pine and eucalyptus pulps) the following expression was obtained:

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$$\rho[\eta] = 0.15p^{1.68} \quad (1)$$

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 391 While when studying the relationship between delignified CNFs with different mechanical,
 392 enzymatic, and chemical treatments (Albornoz-Palma, 2020a; Albornoz-Palma, 2022a), the
 393 following expression was obtained:

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$$\rho[\eta] = 0.031p^{1.94} \quad (2)$$

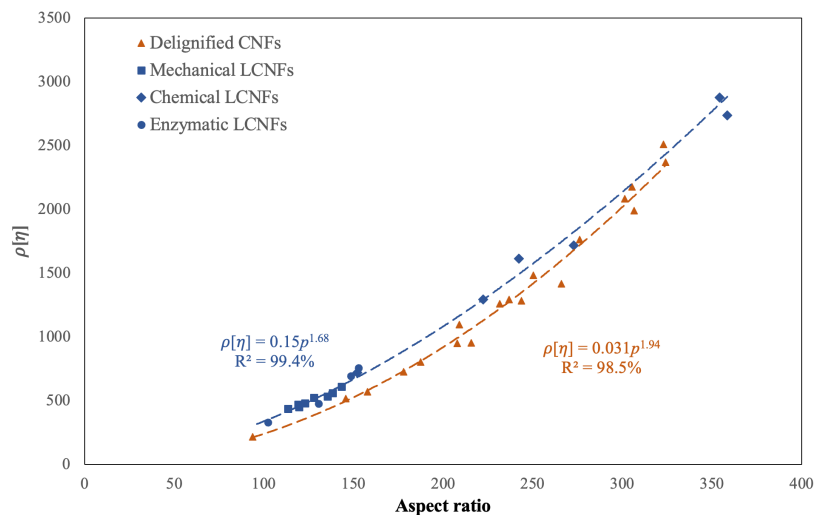
395
 396 where ρ is the density of the LCNFs and delignified CNFs ($\rho = 1.6 \text{ g/cm}^3$ (Daicho, 2019)).

397 Figure 5 shows the relationship between the experimental data and the models. The
 398 determination coefficients (R^2) are 99.4 and 98.5% for the LCNFs and delignified CNFs model,
 399 respectively, which shows that the models are suitable. These models allow related the dependence
 400 of the intrinsic viscosity to a single morphological variable, independent of the degree of flexibility
 401 of the fibrils and the species of the raw materials. In addition, they allow the determination of the
 402 average width of the fibrils (a complex variable) through two variables of quick and simple
 403 measurement (apparent length and intrinsic viscosity).

404 To compare the models of Albornoz-Palma et al. (2020a) ($\rho[\eta] = 0.051p^{1.85}$) and Equation 2,
 405 differences of lower than 2.6% and 4.9% in the prediction of the aspect ratio and intrinsic viscosity
 406 are obtained in the ranges studied ($\rho[\eta]$ entre 517 y 2144; y p entre 145 y 306). This slight difference
 407 is mainly due to incorporating new experimental data that allow the prediction range to be extended.

408 Figure 5 shows that the LCNFs have a higher intrinsic viscosity than the delignified CNFs,
 409 because the LCNFs are less flexible due to the cementing capacity of the lignin (Pandey, 1999;
 410 Albornoz-Palma, 2022b). This causes the fibrils to occupy a larger hydrodynamic volume per unit
 411 mass. As the aspect ratio increases, the differences between the intrinsic viscosities of delignified
 412 CNFs and LCNFs decrease, due to the elongated shape of the fibrils that allows higher flexibility and,
 413 therefore, a lower intrinsic viscosity. The changes in flexibility between high and low aspect ratio
 414 fibrils are much higher in LCNFs ($\epsilon = 0.35 - 0.54$) than in delignified CNFs ($\epsilon = 0.32$), resulting
 415 in a lower power index. The ϵ parameter is associated with the flexibility of the fibrils (Mansfield,
 416 2008).

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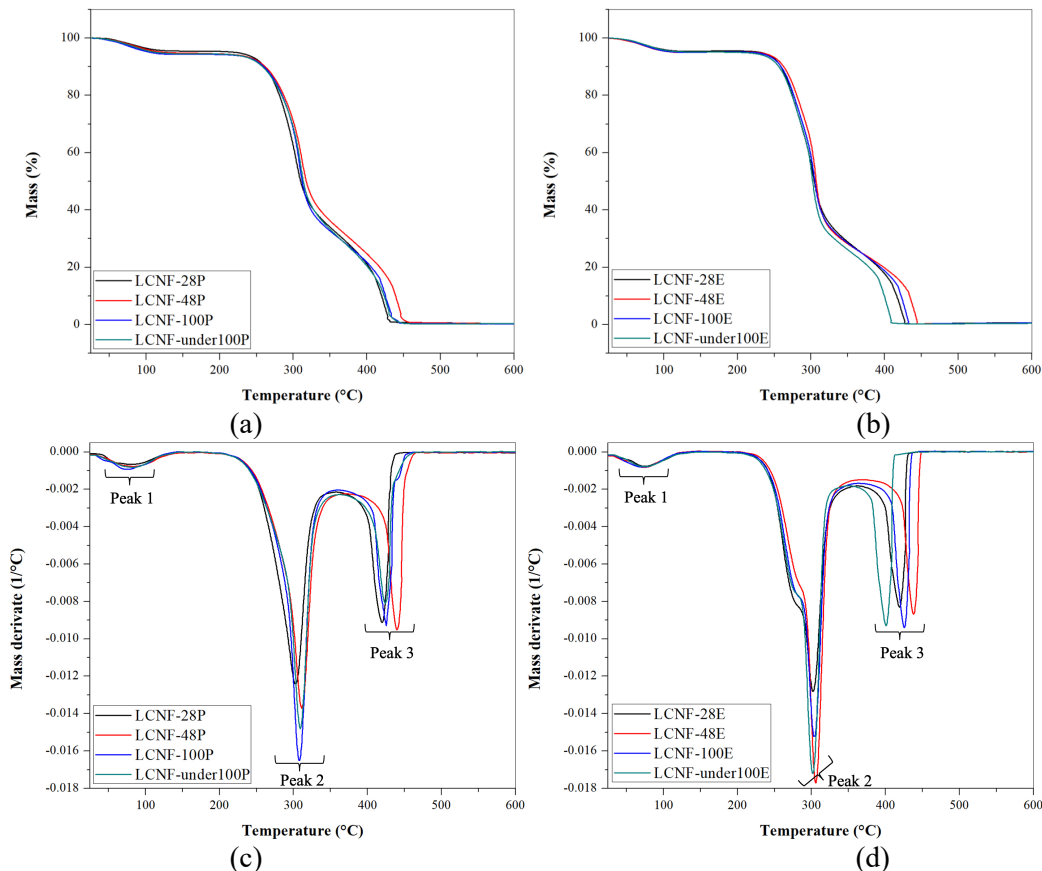
Figure 5. Relationship between intrinsic viscosity and aspect ratio of LCNFs and delignified CNFs.

422 **Thermogravimetric and differential scanning calorimetry analysis of LCNFs.** The
 423 differences in the components and their distribution in the LCNFs can generate differences in the
 424 water removal and thermal stability of these nanomaterials. Cellulose and hemicellulose have higher
 425 degradation rates than lignin. Cellulose begins its degradation around 315°C, hemicellulose at 190°C,
 426 and lignin at approximately 210°C (Yang, 2007; Dorez, 2014; Cao, 2019).

427 Figure 6 shows the TGA and DTG (Derivative Thermo Gravimetry) spectra of the different
 428 LCNFs. Three stages of mass loss are observed in the LCNFs of both species. The first stage (Peak
 429 1) corresponds to the moisture loss of the LCNFs nanopapers. Despite the extensive drying process
 430 to which the nanopapers were subjected, they maintained a moisture percentage close to 5% due to
 431 the high hygroscopic capacity of cellulose and hemicellulose. The moisture in nanopapers of LCNFs
 432 from *Pinus radiata* is slightly higher than in nanopapers of LCNFs from *Eucalyptus nitens*. This is
 433 because glucomannans have a somewhat higher moisture adsorption capacity than xylans
 434 (Kulasinski, 2016), which generates higher water retention, increasing the onset and endset
 435 temperatures of evaporation (T_{onset} and T_{endset}).

436 In the second stage, the degradation onset temperature (T_{onset}) and the maximum degradation
 437 temperature ($T_{peak 2}$), mainly associated with the thermal decomposition of hemicellulose and lignin,
 438 do not show significant differences, regardless of the fraction and species.
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Figure 6. (a-b) TGA and (c-d) DTG spectra of LCNFs from *Pinus radiata* y *Eucalyptus nitens*.

The third stage (Peak 3) is mainly associated with cellulose and lignin degradation. The maximum degradation temperature ($T_{peak 3}$) is higher for the 48 mesh fraction and lower for the smaller size fractions in both species. This decrease could indicate a reduction in the degree of polymerization and crystallinity index in the fibrils produced from fibers with a lower value of these variables and

451 the same mechanical fibrillation process (Andrea Andrade, 2021; Albornoz-Palma, 2020a), which
 452 facilitates the thermal degradation of cellulose.

453 The glass transition temperature of lignin (T_g) was studied by DSC analysis in the
 454 temperature range from -10 to 200°C. Table 4 shows that the T_g values don't change between the
 455 LCNFs of different fractions, for each species, except for the LCNF-28E, whose T_g value is lower.
 456 This decrease in T_g is explained by the higher degree of oxidation of lignin (Figure 1d), which
 457 facilitates its movement.

458 Finally, the increase in the difference in heat capacity (ΔC_p) at T_g in the LCNFs-100P,
 459 LCNFs-under100P, and LCNFs-under100E indicates an increase in the crosslinking density of lignin
 460 (Heitner, 2019).

461 **Table 4.** Moisture and peaks of the thermal decomposition of LCNF nanopapers, and glass transition
 462 temperature and heat capacity difference of lignin in LCNFs from *Pinus radiata* and *Eucalyptus*
 463 *nitens*.
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 465

Peak	Moisture			Degradation			T _g (°C)	ΔC _p (J/gK)
	Peak 1			-	Peak 2	Peak 3		
Sample	T _{onset} (°C)	T _{endset} (°C)	Moisture (%)	T _{onset} (°C) ^(a)	T _{peak 2} (°C)	T _{peak 3} (°C)		
LCNF-28P	52	109	4.6	259±1	303±1	419±1	98±1	0.165
LCNF-48P	54	107	5.0	263±1	311±1	437±1	99±1	0.112
LCNF-100P	50	107	4.9	263±1	308±2	428±2	98±1	0.298
LCNF-under100P	51	110	5.0	261±1	309±1	425±1	95±3	0.263
LCNF-28E	49	99	4.3	258±1	302±1	419±2	80±3	0.240
LCNF-48E	45	99	4.4	262±1	306±2	438±1	91±2	0.241
LCNF-100E	45	98	4.3	258±1	307±1	424±1	89±1	0.215
LCNF-under100E	47	104	4.2	257±1	304±2	402±1	89±2	0.271

^(a)Corresponding to a degradation of 5%

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468 4. Conclusions

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470 The mechanical refining process produced a heterogeneous material in size, formed by
 471 aggregates of fibers, fibers, and fines. The smaller fibers in size showed greater damage in their
 472 structure, presenting a lower degree of polymerization, crystallinity index, and coarseness.

473 The anatomical, physical, and component differences of the fibers of the different size
 474 fractions did not show an effect on the longitudinal disintegration during the mechanical processes of
 475 refining and homogenization, in both species. However, an increase in transverse disintegration was
 476 shown during the mechanical refining process, being higher for LCNFs from *Eucalyptus nitens*. The
 477 homogenization process was not affected in the transverse disintegration of the fibers by the
 478 characteristics of the raw materials, being higher for the LCNFs of *Pinus radiata*.

479 The LCNFs produced from smaller fibers were thinner (lower width), but without changes in
 480 homogeneity, with *Eucalyptus nitens* being the species that had more homogeneous nanofibrils. On
 481 the other hand, the apparent length of the LCNFs was the same for the smaller mesh fractions (28 and
 482 48 mesh), which showed that the mechanical processes affected larger fibers in greater proportion,
 483 without changes in homogeneity and regardless of the species. In addition, the LCNFs produced from
 484 *Eucalyptus nitens* pulp presented the fibrils with smaller sizes.

485 The viscosities of LCNF suspensions from *Pinus radiata* were higher than the viscosities of
 486 the LCNF suspensions from *Eucalyptus nitens*, which could be due to the organization of the

487 components on the surface of LCNFs, where the glucomannans are partially exposed on the surface
488 of the fibrils of eucalyptus, generating very hydrated and large fibrils and flocs.

489 A power law model was established between a rheological parameter, intrinsic viscosity, and
490 a morphological parameter, aspect ratio, for lignocellulose nanofibrils, independent of the
491 pretreatments used during their production and of the flexibility of the fibrils.

492 Calorimetric analysis showed that the glass transition temperature of lignin was not affected
493 by physical changes in the raw material and mechanical processes. Finally, the lignins in the fibrils
494 produced from the higher mesh fractions showed increased crosslinks.

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496 **Ethics approval and consent to participate**

497

498 **Not applicable**

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500 **Consent for publication**

501

502 **Not applicable**

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504 **Availability of data and materials**

505

506 **Not applicable**

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508 **Competing interests**

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517

518 **Authors' contributions**

519

520 **Gregory Albornoz-Palma:** Conceptualization, Methodology, Validation, Formal analysis,
521 Investigation, Writing – Original Draft; **Isidora Ortega-Sanhueza:** Investigation; **Roberto Teruel-**
522 **Juanes:** Methodology, Validation, Investigation; **Sergio Henríquez-Gallegos:** Investigation,
523 Writing – Review & Editing; **Amparo Ribes-Greus:** Conceptualization, Validation, Resources,
524 Writing – Review & Editing, Supervision; **Miguel Pereira:** Conceptualization, Validation,
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526

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528

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539 **Reference**

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