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Additional Information

# Numerical modelling of cancellous bone damage using an orthotropic failure criterion and tissue elastic properties as a function of the mineral content and microporosity

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## Abstract

*Background and Objective: Elastic and strength properties of lamellar tissue are essential to analyze the mechanical behavior of bone at the meso- or macro-scale. Although many efforts have been made to model the architecture of cancellous bone, in general, isotropic elastic constants are assumed for tissue modelling, neglecting its non-isotropic behavior. Therefore, isotropic damage laws are often used to estimate the bone failure. The main goals of this work are: (1) to present a new model for the estimation of the elastic properties of lamellar tissue which includes the bone mineral density (BMD) and the microporosity, (2) to address the numerical modelling of cancellous bone damage using an orthotropic failure criterion and a discrete damage mechanics analysis, including the novel approach for the tissue elastic properties aforementioned.*

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*Methods:* Numerical homogenization has been used to estimate the elastic properties of lamellar bone considering BMD and microporosity. Microcomputed Tomography ( $\mu$ -CT) scans have been performed to obtain the micro-finite element ( $\mu$ -FE) model of cancellous bone from a vertebra of swine. In this model, lamellar tissue is orientated by considering a unidirectional layer pattern being the mineralized collagen fibrils aligned with the most representative geometrical feature of the trabeculae network. We have considered the Hashin's failure criterion and the Material Property Degradation (MPDG) method for simulating the onset and evolution of bone damage.

*Results:* The terms of the stiffness matrix for lamellar tissue are derived as functions of the BMD and microporosity at tissue scale. Results obtained for the apparent yield strain values agree with experimental values found in the literature. The influence of the damage parameters on the bone mechanics behavior is also presented.

*Conclusions:* Stiffness matrix of lamellar tissue depends on both BMD and microporosity. The new approach presented in this work enables to analyze the influence of the BMD and porosity on the mechanical response of bone. Lamellar tissue orientation has to be considered in the mechanical analysis of the cancellous bone. An orthotropic failure criterion can be used to analyze the bone failure onset instead of isotropic criteria. The elastic property degradation method is an efficient procedure to analyze the failure propagation in a 3D numerical model.

*Keywords:* Lamellar bone porosity, cancellous bone numerical modelling, finite element method, damage initiation, material property degradation, orthotropic failure criterion

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## 1. INTRODUCTION

Cancellous bone is a highly porous and heterogeneous material with varying material properties (Nazarian A, 2006), mainly found at the epiphysis and metaphysis of long bones and in the vertebral bodies. Adult bone tissues, both cortical and cancellous, are laminated at the microscale (Parfitt AM, 1987). The tissue arranged at these layers is the so-called lamellar bone tissue, being the mineralized collagen fibrils its main constituent. Consequently, strength and stiffness properties of lamellar tissue are essential to analyze the mechanical behavior of bone at the meso and macro-scale. In the lamellar tissue, mineralized fibre bundles, embedded in the extra-fibrillar matrix, confer a predominant orthotropic symmetry (Reisinger et al., 2010; Martínez-Reina et al., 2011; Vercher-Martínez et al., 2018) leading to an anisotropic behavior under generic multi-axial loading.

On the other hand, the mineral content and the porosity at lamellar tissue level (microporosity) are two essential parameters related with the bone mechanics behavior. It is well known that an increase in the volumetric bone mineral density (BMD) has a direct implication on the rise of the stiffness and, if it is excessive (i.e. due to the absence of bone resorption in the bone turnover process) the tissue will become more brittle (Currey, 1986, 1988; Schaffler et al., 1988; Tommasini and Landis, 2008).

Regarding the microporosity, it also contributes to decline the mechanical response of bone tissue. Several agents contribute to the microporosity increase (Manolagas et al., 2012): old age, estrogen deficiency in postmenopausal women, glucocorticoids and immobilization. In addition, when a rapid bone loss is prevalent (commonly after menopause), depth cavities may occur due to an excessive osteoclastic resorption leading to the trabecular bone perforation of structural elements causing the loss of the structural continuity (Parfitt AM, 1984). Osteoclastic perforation was also observed by Mosekilde L (1990) in a scanning electron microscope study of the remodelling process of vertebral trabecular bone. In Gentsch et al. (2003) two types of resorption lacunae in trabecular bone were observed. Moreover, lacunar and tunnelling perforation are distinguished denoting microstructural changes, related with disturbed bone turnover. Advances in computer tomography techniques have shown that porosity is responsible of a substantial amount of bone loss and consequently, the resultant higher bone fragility and mechanical competence deterioration.

The non-isotropic nature of lamellar tissue is also a relevant aspect to be



38 included in the quantification of bone mechanical properties. In the review of  
39 biomechanics and mechanobiology of trabecular bone presented in Oftadeh  
40 et al. (2015), it is stated that at the microstructural scale, trabeculae consist  
41 of groups of parallel lamellae bounded by cement lines primarily oriented  
42 also parallel to the trabecular surfaces. In addition, the three-dimensional  
43 ultrastructure bone arrangement in relation to the local trabecular direction  
44 is analyzed in Georgiadis et al. (2016). They state that bone ultrastruc-  
45 ture is mostly aligned to trabecular microstructure near trabecular surface.  
46 However, when going towards trabecular core, the ultrastructure alignment  
47 decreases to around 40%.

48 Cancellous tissue was characterized as a microstructure consisting of lay-  
49 ers interspaced with transition zones where the proportions of collagen and  
50 mineral vary (Donnelly et al., 2006). Hosaka-Takamiya et al. (2016) observed  
51 that collagen bundles in trabecular bone run along the long axis of the tra-  
52 becula. In Hammond et al. (2018), the fibril orientation is addressed in a  
53 numerical model of trabecular bone.

54 In the study presented by Rami et al. (2017), a three dimensional mul-  
55 tiscala micromechanical model, where the lamellar tissue is modelled as a  
56 multilayered laminate, is suggested. The mineralized collagen fibrils follow  
57 a determined angular orientation pattern. That work deals with the linear  
58 anisotropic mechanical properties of the cancellous bone, no strength analysis  
59 is performed.

60 In the literature we find several references that reveal the importance of  
61 considering the tissue properties in the trabecular bone numerical models,  
62 when the mechanical competence of bone is under study. Hammond et al.  
63 (2019) state that, only when the tissue anisotropy is considered in their nu-  
64 merical models, the shape and distributed microcracking typically observed  
65 in trabecular bone are reproduced. In Hammond et al. (2018), the effect of  
66 tissue properties on predicted stresses and strains is observed. This improves  
67 the correlation between the solution from numerical models with experimen-  
68 tal data. For example, material heterogeneity seems to play an important  
69 role in resisting bone damage under cyclic loads with long service lives (Torres  
70 et al, 2016). In addition, in Renders et al. (2008), the authors demonstrate  
71 that the no consideration of the mineralization heterogeneity overestimates  
72 the apparent Young's moduli.

73 With the recent advances in acquisition techniques of high-resolution  
74 medical image and postprocessing software, the numerical analysis of the  
75 strength of trabecular bone through  $\mu$ -FE models has become an interesting

76 option. Non-linear  $\mu$ -FE models were used to simulate pre- and post-yielding  
77 cancellous bone behavior (Hambli, 2013a; Nagaraja S, 2005; Schwiedrzik et  
78 al., 2013; García, 2019; Hambli, 2013b; Belda et al., 2019).

79 In this numerical context, an interesting option to simulate the bone fail-  
80 ure is the Continuum Damage Mechanics (CDM) approach. In this approach,  
81 the initiation and propagation of cracks is based on a smeared crack approach  
82 (Lemaitre, 1985; Hambli, 2011a,b, 2013a,b). In a quasi-static loading case,  
83 isotropic damage laws are often used to represent the non-linear behavior  
84 of cancellous bone (Nagaraja S, 2005; Hambli, 2013a). In addition, finite  
85 element deletion technique is also considered to model the complete fracture  
86 of the trabeculae (Hambli, 2013b).

87 Concerning the strength of trabecular bone, a detailed review of several  
88 failure criteria applied in material science is reported in Oftadeh et al. (2015),  
89 that have been considered for the study of the bone mechanics under multi-  
90 axial stresses. In that work, it is also highlighted that bone mechanical  
91 behavior is highly dependent on tissue properties.

92 In order to implement a damage evolution law in a three-dimensional nu-  
93 merical analysis, the Material Property Degradation (MPDG) is a procedure  
94 very efficient computationally. This method can simulate the post-damage  
95 degradation of brittle anisotropic materials. The MPDG results in a non-  
96 linear evolution where the damage variable,  $d$ , takes predefined discrete val-  
97 ues depending on the dominant failure mode, and assumes an instant stiffness  
98 reduction of the material. In contrast, in the Continuum Damage Mechanics  
99 (CDM) approach, the damage variable gradually increases with the amount  
100 of fracture energy dissipated. The discrete damage method is also applied to  
101 the study of progressive failure in laminate structures.

102 The main contributions of the approach presented in this work can be  
103 summarized in: (1) it addresses the numerical modelling of the cancellous  
104 bone mechanical behaviour, not only considering the microstructure, as usu-  
105 ally done in literature, but also the non-isotropic elastic tissue properties,  
106 as a function of BMD and microporosity. (2) It includes the orientation of  
107 the mineralized collagen fibrils in the main geometrical feature of the tra-  
108 beculae network. (3) It also proposes to use an interactive failure criterion  
109 based on the inferred orthotropic lamellar strength limits. (4) The post-yield  
110 behaviour is also addressed in a sensitivity analysis by means of an efficient  
111 technique, based on the elastic property degradation.

112 The objective of the work becomes even more important when character-  
113 izing the mechanical competence of bone that exhibits certain pathologies.

114 In osteoporosis, for example, the presence of “non-natural” micro pores by  
115 the osteoclastic perforation as a consequence of the bone turnover unbal-  
116 anced process (Parfitt AM, 1984; Mosekilde L, 1990; Gentsch et al., 2003)  
117 is scarcely addressed in literature from a mechanical point of view. These  
118 pores, as in any structural material, will undermine the mechanical func-  
119 tionality of bone. In addition, in this pathology, the increase of the mineral  
120 content in certain regions causes more heterogeneity and fragile behaviour  
121 of bone. In order to deal with these phenomena from a numerical point of  
122 view, non-isotropic detailed constitutive models are needed.

123 The scheme shown in Fig. 1 depicts the work flow of the numerical study  
124 carried out in this work. As a starting point, we consider the equations for the  
125 estimation of the non-isotropic elastic constants of lamellar tissue as a func-  
126 tion of the BMD, developed in a previous authors’ work (Vercher-Martínez  
127 et al., 2018). Subsequently, the influence of the microporosity on the elastic  
128 constants is also included explicitly in the numerical models and, therefore,  
129 the new homogenization stiffness matrix is derived as a function of BMD and  
130 porosity,  $p$ . On the other hand, the tissue strength limits have been inferred  
131 from the literature (Ascenzi and Bonucci, 1968; Giner, 2014). Hence, the  
132 resulting mechanical properties of lamellar tissue are applied into the  $\mu$ -FE  
133 model of a representative volume of trabecular bone from swine lumbar ver-  
134 tebra. As a first approximation, in the numerical model, the bundles of fibres  
135 are oriented following the predominant direction of the trabeculae network.  
136 Then, Hashin’s orthotropic three-dimensional quadratic failure criterion for  
137 fibre composites (Hashin, 1980) has been implemented to estimate the onset  
138 of the failure in quasi-static displacement-controlled tension and compres-  
139 sion numerical simulations. Finally, the damage evolution law follows the  
140 MPDG method. A study of the influence of the damage parameters is also  
141 performed.

142

## 143 2. Methods

### 144 2.1. Modelling porosity at lamellar tissue

145 Porosity induces a strong influence on strength and stiffness of bone.  
146 These mechanical properties vary inversely with increasing porosity (Schaffler  
147 et al., 1988; Currey, 1988). In the following equation, we summarize the three  
148 main sources that contribute to the formation of tissue porosity in cortical  
149 bone (Martínez-Reina et al., 2011):

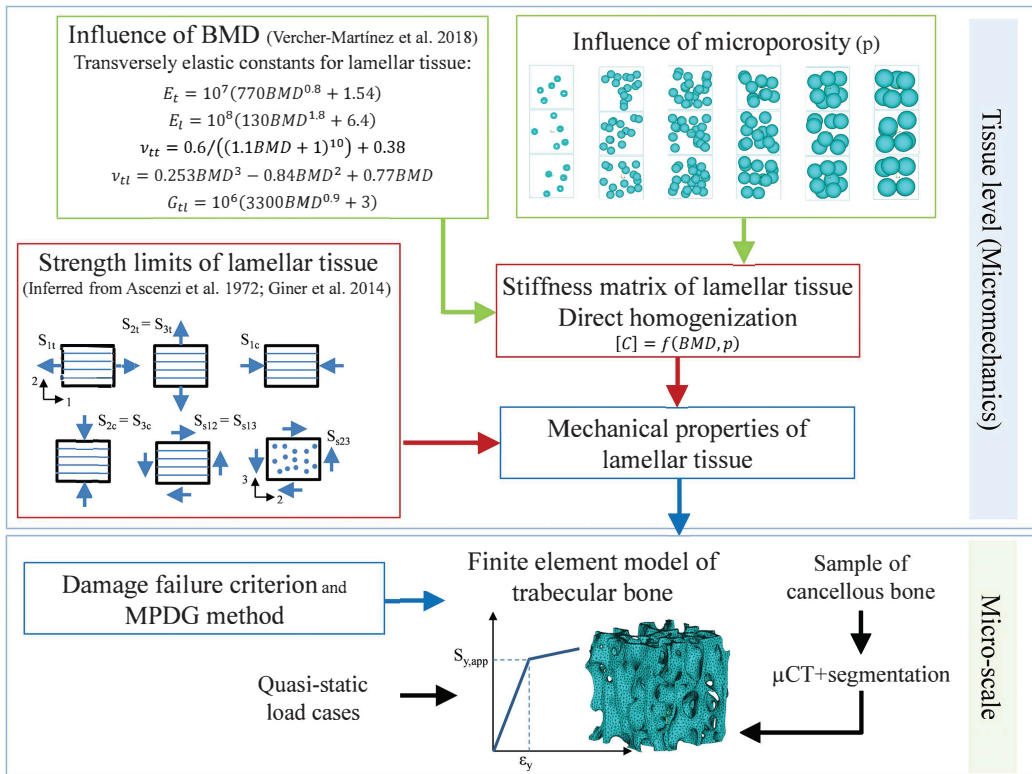


Fig. 1: Work flow of the analysis performed in this work, from tissue micromechanics characterization to the  $\mu$ -FE numerical model

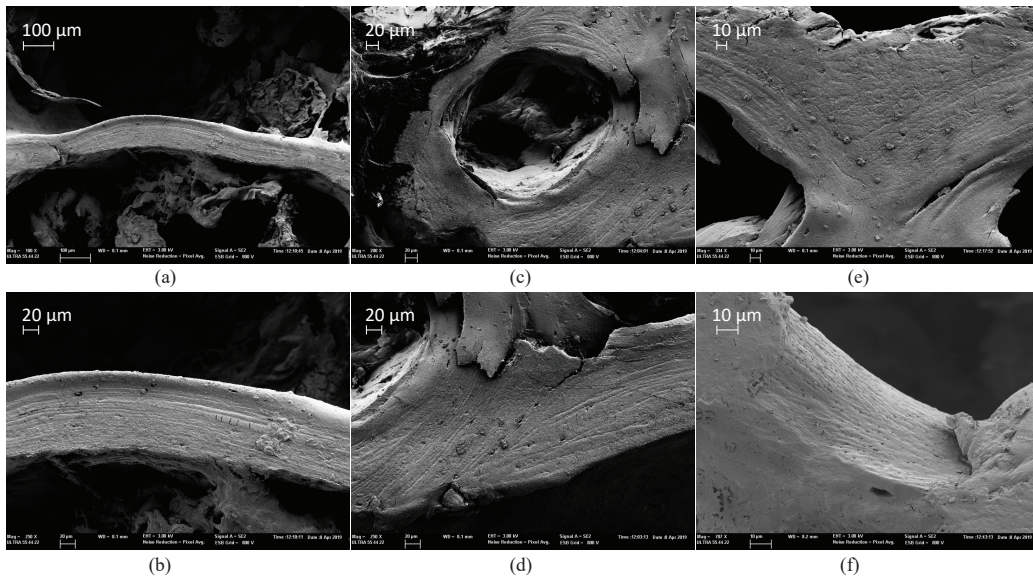
$$P_{tissue} = P_{lac} + P_{can} + P_{vas} \quad (1)$$

150 where  $P_{lac}$  is the porosity due to the lacunae, that are small ellipsoids with  
 151 approximate diameters  $4 \times 9 \times 22\mu\text{m}$  (Marotti, 1979) that contain bone cells  
 152 (osteocytes).  $P_{can}$  represents the porosity due to the canaliculi, they are very  
 153 fine channels radiating from the lacunae. They both constitute the lacuno-  
 154 canalicular system that produces a porosity of about 5% (Cowin, 1999).  $P_{vas}$   
 155 denotes the vascular porosity that is mainly due to Havers' canals that run  
 156 the length of osteons together with Volkmann canals and its evaluation de-  
 157 pends on the bone turnover activity. Following the work of Martínez-Reina et  
 158 al. (2011), vascular porosity could vary between 1 and 20%. Consequently,  
 159 the total porosity for lamellar tissue in cortical bone varies between 6 and  
 160 25%. In Eq. 1, the collagen-apatite porosity has been neglected.

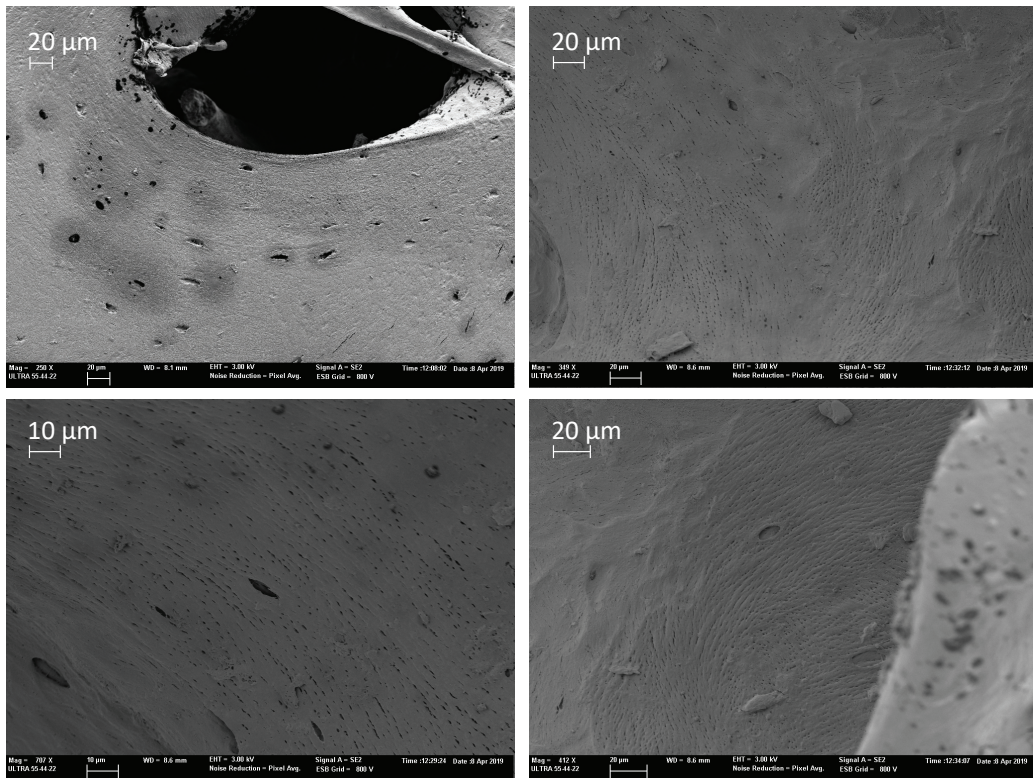
161 Regarding the porosity of lamellar tissue in trabecular bone, lacuno-  
 162 canalicular system is also present in the trabecular packets or hemiosteons of  
 163 cancellous bone. For this term, the same porosity value than in cortical bone  
 164 is assumed (up to 5%). With respect to the variable  $P_{vas}$ , marrow cavity  
 165 harbours the vascularization in trabecular bone, instead of Havers' canals,  
 166 however, this term is also associated with the bone turnover activity. For  
 167 trabecular bone, no reference value has been found in literature in order to  
 168 quantify the microporosity due to the osteoclastic perforation. In this work  
 169 a value up to 20% will be assumed.

170 In Fig. 2, the mineralized skeleton of a swine vertebral trabecular bone  
 171 sample is observed using the Field Scanning Electron Microscope (FESEM)  
 172 of the Microscope Service at the Polytechnical University of Valencia. These  
 173 images show that lamellar tissue deposition exhibits a predominant multi-  
 174 layer arrangement. The lamellar nature of the cancellous tissue is clearly  
 175 manifested at certain regions (Fig. 2, a-d). In contrast, some regions show  
 176 a more homogeneous appearance (Fig. 2, e). The lacunocanalicular porosity  
 177 at tissue level is also observed (see Fig. 2, f). Furthermore, in Fig. 3 empty  
 178 lacunae are clearly distinguished.

179 In the present work, the porosity at tissue level is explicitly modelled  
 180 by subtracting non-overlapping spheres randomly distributed from a repre-  
 181 sentative elementary volume of lamellar tissue model. The different values  
 182 of porosity considered are 1, 5, 10, 15, 20 and 25% (Martínez-Reina et al.,  
 183 2011). In order to obtain averaged properties, 10 models with random distri-  
 184 bution of spheres have been analyzed for each value of porosity. In Fig. 4, a,



*Fig. 2: Mineralized skeleton of the lamellar tissue deposition in a swine vertebral trabecular bone sample (Field Emission Scanning Electron Microscope - FESEM) (a) Cross section of a plate structure showing a prone planar multilayer lamellar tissue deposition. (b) A magnification of (a). In (c) lamellar arrangement exhibits a circumferential pattern around a cavity. (d) A magnification of (c). In (e) a branched region is localized showing a more homogeneous tissue arrangement. In (f), the surface of a strut is observed. Microporosity due to lacunocanaliculi system is clearly identified*



*Fig. 3: Porosity due to the lacunocanicular system in cancellous bone from swine vertebral sample (Field Emission Scanning Electron Microscope - FESEM)*

Table 1: Values of BMD ( $g/cm^3$ ) and porosity (%) at tissue level considered for estimating the elastic constants of lamellar tissue.

BMD	0.653	0.75	0.85	0.95	1.05	1.16	1.24	1.32	1.39	1.44	1.48	1.50
Porosity	1, 5, 10, 15, 20, 25											

185 three models out of the ten random distributions of voids are shown for each  
186 porosity. The numerical model of a representative volume of porous lamellar  
187 tissue is depicted in Fig. 4, b, where the elastic properties for the non-porous  
188 part of the lamellar tissue are estimated as a function of BMD, using the  
189 equations developed in Vercher-Martínez et al. (2018):

$$E_t^{lam} = 10^7 (770 \text{ BMD}^{0.8} + 1.54) \quad (2)$$

$$E_l^{lam} = 10^8 (130 \text{ BMD}^{1.2} + 6.4) \quad (3)$$

$$\nu_{tt}^{lam} = \frac{0.6}{(1.1 \text{ BMD} + 1)^{10}} + 0.38 \quad (4)$$

$$\nu_{tl}^{lam} = 0.253 \text{ BMD}^3 - 0.84 \text{ BMD}^2 + 0.77 \text{ BMD} + 0.01 \quad (5)$$

$$G_{tl}^{lam} = 10^6 (3300 \text{ BMD}^{0.9} + 3) \quad (6)$$

190 where  $E$  is the Young's modulus,  $\nu$  represents the Poisson's ratio and  $G$  is the  
191 shear modulus. In addition, subscripts  $l$  and  $t$  indicate the longitudinal and  
192 transverse directions of the fibre bundles and  $lam$  indicate that the properties  
193 are estimated for non-porous lamellar tissue.

194 In Fig. 4 -c, the reference system (1, 2, 3) corresponds to an orientated ref-  
195 erence system where 1 indicates the longitudinal direction of the mineralized  
196 collagen fibrils ( $l$ ), 2 and 3 are two orthogonal or transverse directions ( $t$ )  
197 of the fibril array. This reference system is only used to set the transversely  
198 isotropic elastic properties and strength limits for lamellar tissue.

199 Cancellous tissue is less mineralized than cortical bone, mainly due to a  
200 higher activity of the bone turnover in the trabeculae network. Therefore,  
201 considering the work of Koller et al. (2007), the minimum value for the BMD  
202 at tissue level is assumed  $0.653 \text{ g/cm}^3$  and the maximum is derived from the  
203 work of Yu et al. (1998) being  $1.5 \text{ g/cm}^3$ . Table 1 summarizes the numerical  
204 values of porosity and BMD at tissue level that have been analyzed in the  
205 present work.



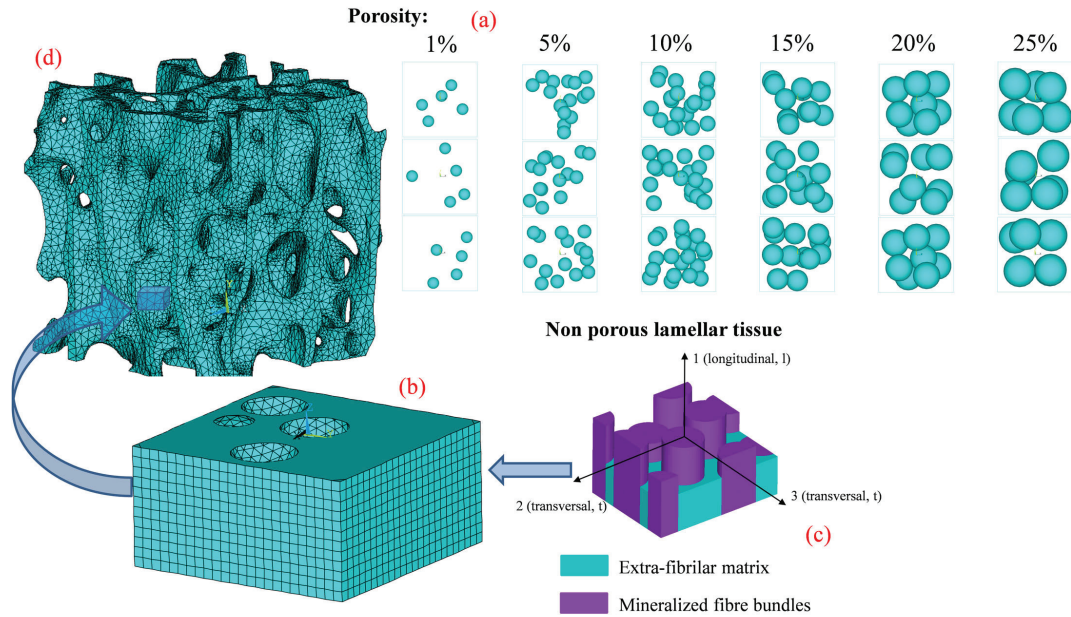


Fig. 4: (a) Random distribution of non-overlapping spheres representing the voids. Three examples of the ten models for each level of porosity (plane view projections). (b) Numerical model of the representative elementary volume of porous lamellar tissue. (c) Transversely isotropic elastic properties of lamellar tissue as a function of BMD at tissue level (Vercher-Martínez et al., 2018). (d)  $\mu$ -FE model of trabecular bone with homogenized tissue elastic properties. Note that reference system (1,2,3) corresponds to an orientated reference system where 1 indicates the longitudinal direction of the fibrils, 2 and 3 are two transverse directions.

206 In order to estimate the averaged apparent stiffness of the porous lamellar  
 207 tissue the following procedure has been carried out. First, a direct homoge-  
 208 nization technique has been applied by means of the finite element method.  
 209 Periodic boundary conditions are enforced guaranteeing that the hexahedron  
 210 analyzed behaves as a continuum domain. The displacement gradients along  
 211 the corresponding external surfaces must be equal, and, for this purpose,  
 212 the equations established in Hohe (2003) are employed. Assuming the linear  
 213 elastic Hooke’s law (Eq. 7)

$$\sigma_{ij} = C_{ijkl} \epsilon_{kl} \quad (7)$$

214 where  $\sigma_{ij}$  and  $\epsilon_{kl}$  are the stress and strain tensors, the elements of the consti-  
 215 tutive elastic tensor  $C_{ijkl}$  are derived applying six independent unitary strain  
 216 fields.

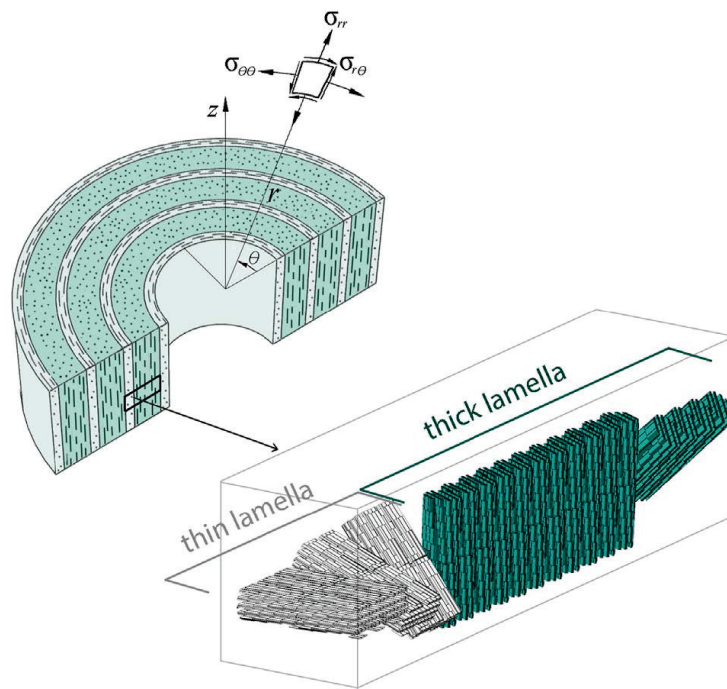
217 Lastly, the elastic constants are explicitly expressed as a function of BMD  
 218 and porosity,  $p$ , using non-linear multi-variable regressions. These equations  
 219 will be provided in Sec. 3.1, and applied to define the elastic properties of  
 220 lamellar tissue for the  $\mu$ -FE model (see Fig. 4, d).

## 221 2.2. Inferring strength limits of lamellar tissue

222 In the secondary osteons, lamellae arrange circumferentially around the  
 223 Havers canal. Within a lamella, mineralized collagen fibrils maintain their  
 224 orientation constant and change it across the radial direction of the osteon  
 225 in successive lamellae building the so-called rotated plywood pattern. In the  
 226 work of Giner (2014) the lamellar structure observed in a secondary osteon,  
 227 was condensed in two equivalent layers: the thin and thick lamellae. In the  
 228 thin lamella, fibrils are mainly aligned with the circumferential direction of  
 229 the osteon and in the thick lamella, fibrils are roughly parallel to the long  
 230 axis of the osteon (see Fig. 5). Thin and thick lamellae do not have any  
 231 direct application on the trabecular bone numerical model, we use them only  
 232 to estimate the strength limits of lamellar tissue.

233 The in-plane strength properties for lamellar tissue were derived relating  
 234 the results from several works of Ascenzi and Bonucci in which tensile and  
 235 shear loading were applied on different types of isolated osteons (Ascenzi and  
 236 Bonucci, 1967, 1972), with the circumferential  $\sigma_{\theta\theta}$ , radial  $\sigma_{rr}$  and shear  $\sigma_{r\theta}$   
 237 stresses (see Fig. 5) and their respective strength limits.

238 In the current work, the strength limits in an orthotropic material  $S_{1t}$ ,  
 239  $S_{2t}$ ,  $S_{3t}$ ,  $S_{s23}$ ,  $S_{s13}$  and  $S_{s12}$  depicted in Fig. 6 (following the customary termi-  
 240 nology in structural composites materials), are inferred: the circumferential



*Fig. 5: Model of an osteon showing the cylindrical reference system  $(r, \theta, z)$ . The thin and thick lamellae are defined bunching the layers where mineralized collagen fibrils are orientated in a predominant direction (Giner, 2014)*

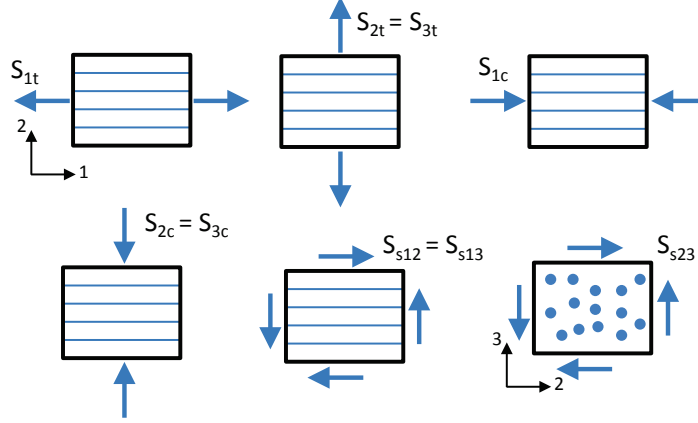


Fig. 6: Schematic representation of the strength limits in an orthotropic material following customary terminology in structural composites materials. Reference system  $(1,2,3)$  corresponds to the orientated reference system where 1 indicates the longitudinal direction of the fibrils, 2 and 3 are two orthogonal directions (see Fig. 4).

241 tensile strength  $S_{\theta\theta,t}$  for thin lamellae derived in Giner (2014) is correspond-  
 242 ing to  $S_{1t}$  and  $S_{\theta\theta,t}$  for thick lamellae corresponds to  $S_{2t}$  and  $S_{3t}$ . Additionally,  
 243 the shear strength  $S_{r\theta,s}$  for thick lamellae is equivalent to  $S_{s23}$  and for thin  
 244 lamellae represents  $S_{s12}$  and  $S_{s13}$  (see Figs. 5-6).

245 In order to infer the strength limits under compressive loading, the work of  
 246 Ascenzi and Bonucci (1968) has been considered. In their work, the stress-  
 247 strain curves for compressive loading tests in the longitudinal direction of  
 248 the osteon were obtained for different degrees of calcification and ages. The  
 249 experimental analysis were developed for different types of osteons classified,  
 250 according to the predominant orientation of the mineralized collagen fibrils,  
 251 in osteons of type I, II and III. In type I osteons, fibrils are mainly orientated  
 252 in the circumferential direction of the osteon, consequently, the strength limit  
 253 inferred from the stress-strain curve corresponds to  $S_{2c} = S_{3c}$ . Additionally,  
 254 in type III osteons, fibrils are mainly aligned with the longitudinal direction  
 255 of the osteon, hence, the strength limit  $S_{1c}$  can be estimated. The numerical  
 256 values considered in the current work are averaged from the full calcified  
 257 samples and are summarized in Table 2.

### 258 2.3. $\mu$ -FE of trabecular vertebral specimen

Table 2: *Strength limits for full calcified lamellar tissue. Approximated from Ascenzi and Bonucci (1968) and Giner (2014)*

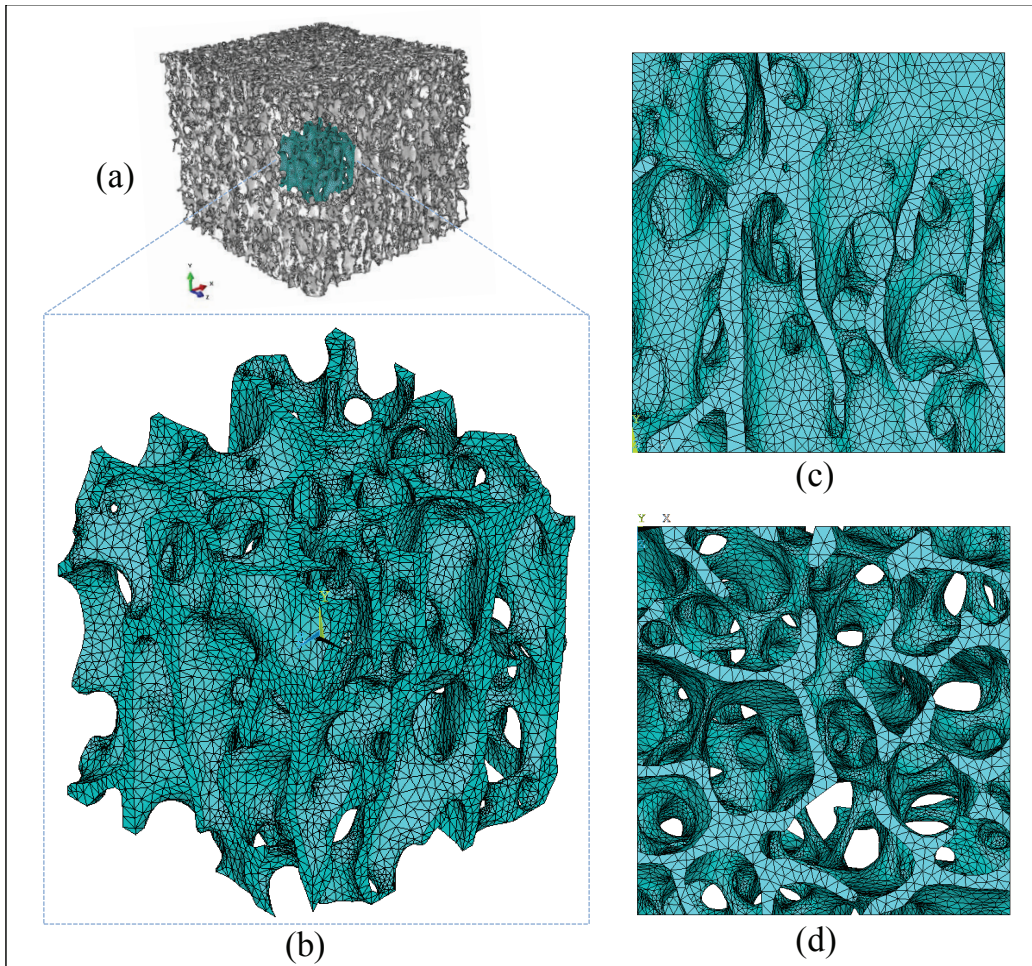
$S_{1t} = 120\text{MPa}$
$S_{1c} = -115\text{MPa}$
$S_{2t} = S_{3t} = 50\text{MPa}$
$S_{2c} = S_{3c} = -160\text{MPa}$
$S_{s12} = S_{s13} = 46\text{MPa}$
$S_{s23} = 38\text{MPa}$

259 The trabecular bone sample was prepared in Instituto de Biomecánica de  
 260 Valencia (IBV) from lumbar vertebrae of one skeletally mature swine recently  
 261 euthanised. The parallelepiped-shaped sample, was at least 10 mm side.

262 The specimen was scanned by  $\mu$ -CT (Skyscan 1172, Bruker, Kontig,  
 263 Bégica) at the Estación de Biología Mariña de A Graña (Universidad de  
 264 Santiago de Compostela, Spain)  $\mu$ -CT service, with an isotropic voxel res-  
 265 olution of 13.58  $\mu\text{m}$  (voltage 100kV, intensity 100  $\mu\text{A}$ , Al/Cu filter).  $\mu$ -CT  
 266 images were segmented using ScanIp software (Simpleware, UK). Before gen-  
 267 erating the mesh, the geometrical model was cut leading to a cube-shaped  
 268 volume with approximately  $2 \times 2 \times 2$  mm side.

269  $\mu$ -FE mesh was generated using ScanIp Software (Simpleware, UK), lead-  
 270 ing to a mesh of 3D linear elements. The finite element model of the specimen  
 271 was able to reproduce with good accuracy the heterogeneous microstructure  
 272 of cancellous bone (see Fig. 7). The numerical model is built in a global  
 273 reference system (x,y,z) where loads and constraints will be applied. In this  
 274 mesoscale finite element model, y direction denotes the principal bone di-  
 275 rection where plates predominate, instead, x and z directions show a higher  
 276 porosity and a foremost strut-like structure.

277 The stiffness matrix for tissue has been estimated considering the ap-  
 278 proach developed in this work, assuming uniform reference values for BMD  
 279 and porosity at tissue level: **BMD = 0,85 g/cm<sup>3</sup>**,  $p = 5\%$ . Regarding the  
 280 strength limits, values summarized in Table 2 are specified in the numerical  
 281 model. As a first approximation, we have assumed that the fibril bundles are  
 282 unidirectionally orientated following the predominant direction of the tra-  
 283 becule (Hosaka-Takamiya et al., 2016), consequently, local reference systems  
 284 are defined with the purpose of considering the non-isotropic lamellar tissue  
 285 properties, both elastic and strength features, in the  $\mu$ -FE model.



*Fig. 7:  $\mu$ -FE model of a lumbar vertebra trabecular bone. (a) Geometrical model from segmentation of  $\mu$ -CT images, (b) isometric (c) front and (d) top view. The numerical model has been analyzed with Ansys® APDL Software*

286 *2.4. Bone failure modelling*

287 *2.4.1. Orthotropic failure criteria for damage initiation*

288 Considering lamellar tissue as a laminate structure, the Hashin criterion  
 289 (Hashin, 1980), which is widely used in analysis of structural composite ma-  
 290 terials to predict intralaminar failure, should be an interesting option to  
 291 analyze the failure initiation at tissue level. The failure Hashin criterion is  
 292 specially formulated to account for different damage mechanisms. Matrix  
 293 failure is associated with intralaminar transverse and shear loads, whereas  
 294 fibre failure is related to longitudinal tension.

295 The three-dimensional formulation of this orthotropic damage criterion  
 296 is given by the following equations:

$$f_f = \left( \frac{\sigma_{11}}{X_t} \right)^2 + \frac{(\tau_{12}^2 + \tau_{13}^2)}{S^2}; \quad \sigma_{11} > 0 \quad (8)$$

$$f_f = \frac{\sigma_{11}}{X_c}; \quad \sigma_{11} < 0 \quad (9)$$

$$f_m = \frac{(\sigma_{22} + \sigma_{33})^2}{Y_t^2} + \frac{(\tau_{23}^2 - \sigma_{22}\sigma_{33})}{Q^2} + \frac{(\tau_{12}^2 + \tau_{13}^2)}{S^2}; \quad \sigma_{22} + \sigma_{33} > 0 \quad (10)$$

$$f_m = \frac{(\sigma_{22} + \sigma_{33})}{Y_c} \left[ \left( \frac{Y_c}{2Q} \right)^2 - 1 \right] + \frac{(\sigma_{22} + \sigma_{33})^2}{4Q^2} + \frac{(\tau_{23}^2 - \sigma_{22}\sigma_{33})}{Q^2} + \frac{(\tau_{12}^2 + \tau_{13}^2)}{S^2}; \quad \sigma_{22} + \sigma_{33} < 0 \quad (11)$$

297 where  $X_t = S_{1t}$ ,  $X_c = S_{1c}$ ,  $Y_t = S_{2t}$ ,  $Y_c = S_{2c}$ ,  $S = S_{s12}$  and  $Q = S_{s23}$  are the  
 298 strength limits for lamellar tissue detailed in Sec. 2.2.

299 The most critical of the failure modes is selected by means of:

$$f = \max(f_f, f_m) \quad (12)$$

300 Note that in the above equations,  $f$  denotes the inverse of reserve factor,  
 301 hence, critical values are greater or equal to one.

302 *2.5. Material Property Degradation MPDG for damage evolution law*

303 Once the failure has initiated, the damage evolution law based on the ma-  
 304 terial property degradation MPDG method is considered. In this smeared



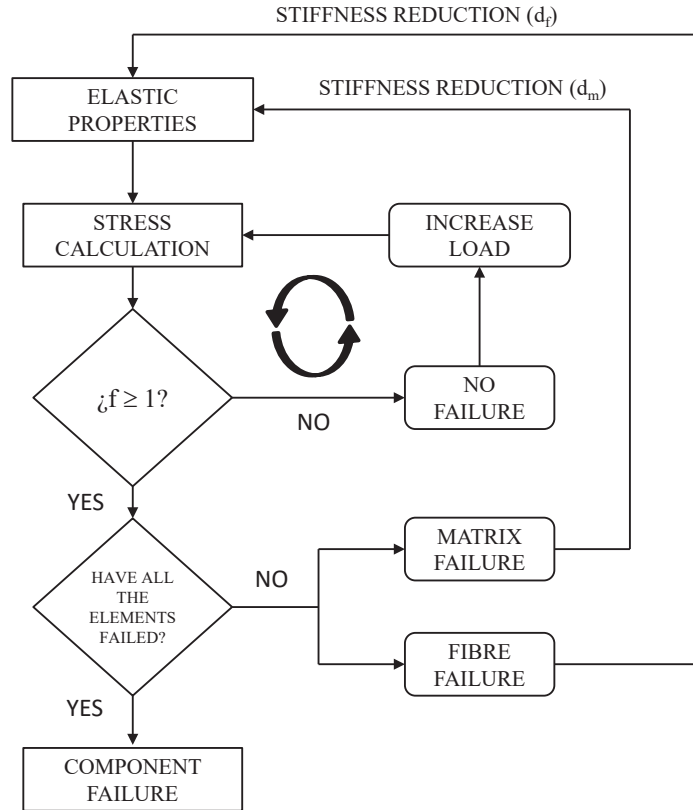


Fig. 8: Scheme of the material property degradation MPDG method

305 crack approach, the discrete domain is as a continuum mesh where the con-  
 306 tinuity in the displacement field is preserved. In order to reproduce the pres-  
 307 ence of cracks, the material stiffness is reduced once the failure is achieved  
 308 accomplishing a certain failure criterion. A scheme of the method imple-  
 309 mented is shown in Fig. 8.

310 This progressive damage model is used to analyze the post-damage degra-  
 311 dation of brittle anisotropic materials. The instant stiffness reduction is  
 312 applied by means of the degradation parameter  $d$  that affects the element stiff-  
 313 ness matrix. **In the damage model, no tissue properties that could influence**  
 314 **on the bone fracture toughness are considered.**



315 Assuming a linear elastic behavior,  $\tilde{\sigma} = \mathbf{C}\epsilon$  is verified, where  $\tilde{\sigma}$  is the  
 316 effective Cauchy stress (stress measured in the undamaged domain) and  $\mathbf{C}$   
 317 the undamaged constitutive matrix. Hence, the relationship for a damaged  
 318 material is given by the following equation:

$$\sigma = \mathbf{C}_d \epsilon \quad (13)$$

where  $\sigma$  is the nominal stress (effective stress averaged over the entire domain, including both damaged and undamaged domains),  $\epsilon$  is the strain and  $\mathbf{C}_d$  is the damaged constitutive matrix. The relationship between the effective stress  $\tilde{\sigma}$  and the nominal can be found in Barbero and Cabrera (2018).  $\mathbf{C}_d$  can be written in terms of the damage variables as follows:

$$\mathbf{C}_d = \begin{pmatrix} \frac{S_{11}}{(1-d_f)} & S_{12} & S_{13} & 0 & 0 & 0 \\ S_{21} & \frac{S_{22}}{(1-d_m)} & S_{23} & 0 & 0 & 0 \\ S_{31} & S_{32} & \frac{S_{33}}{(1-d_m)} & 0 & 0 & 0 \\ 0 & 0 & 0 & \frac{S_{44}}{(1-d_s)} & 0 & 0 \\ 0 & 0 & 0 & 0 & \frac{S_{55}}{(1-d_s)} & 0 \\ 0 & 0 & 0 & 0 & 0 & \frac{S_{66}}{(1-d_s)} \end{pmatrix}^{-1} \quad (14)$$

319 where  $S_{ij}$  represent the terms of the the compliance matrix of the undam-  
 320 aged material  $\mathbf{S}$  and  $d_f$ ,  $d_m$  and  $d_s$  are the fiber, matrix and shear damage  
 321 variables, respectively. Eq. 14 represents the three-dimensional approach of  
 322 the stiffness matrix for a damaged unidirectional lamina formulated under  
 323 the plane-stress assumption in Matzenmiller et al. (1995). Valid values for  
 324 the damage variables are between 0 and 1, where 0 implies no damage and  
 325 1 complete loss of stiffness in the affected mode. This method assumes four  
 326 damage modes:

$$d_f = \begin{cases} d_f^t & \text{if } \sigma_{11} \geq 0 \\ d_f^c & \text{if } \sigma_{11} < 0 \end{cases} \quad (15)$$

$$d_m = \begin{cases} d_m^t & \text{if } \sigma_{22} + \sigma_{33} \geq 0 \\ d_m^c & \text{if } \sigma_{22} + \sigma_{33} < 0 \end{cases} \quad (16)$$

$$d_s = 1 - (1 - d_f^t) (1 - d_f^c) (1 - d_m^t) (1 - d_m^c) \quad (17)$$

327 Note that the shear damage variable  $d_s$  is not an independent variable  
 328 being determined by  $d_f$  and  $d_m$  by means of Eq. 17. Degradation parameters

329 are scalar user-specified quantities. In this work, both the initiation failure  
 330 criterion and the MPDG method have been implemented in the finite element  
 331 code using scripts in Ansys APDL. Following a usual procedure in structural  
 332 composite materials, the nonlinear analyses are performed assuming certain  
 333 parameter values. A parametric study on the post-yield behavior has been  
 334 performed in the current work, considering different values for the damage  
 335 parameters  $d_f$  and  $d_m$ . For the damage variable associated with the failure  
 336 due to loads acting on the longitudinal direction of the fibrils, two values have  
 337 been considered:  $d_f = 0.9$  and  $d_f = 0.9999$ . In laminate strength analysis,  
 338 fibre failure is usually associated with a severe failure mode, hence, a high  
 339 value is usually assumed. That means a very important reduction of element  
 340 stiffness. In fact, the large value of the degradation parameter,  $d_f = 0.9999$ ,  
 341 entails the elimination of the element, producing an overload on the neigh-  
 342 bour elements that will not be supported. This situation usually prompts a  
 343 catastrophic failure. Regarding the damage variable  $d_m$ , this failure mode  
 344 occurs mainly when loads are acting on the transverse direction of the fibrils  
 345 or shearing. The numerical values here considered are:  $d_m = 0.5, 0.95$  in the  
 346 light of the possibility of redistributing the loads when matrix fails, being  
 347 still able to bear certain level of load, and, finally, the ply discount approach  
 348 is also considered, being  $d_m = 0.9999$  (Barbero and Cosso, 2014; Barbero  
 349 and Cabrera, 2018). In this later approach, for the sake of completeness, the  
 350 stiffness of the element that reaches the damage onset as a consequence of  
 351 tranverse loading, is reduces almost to zero. This technique addresses with  
 352 the matrix total damage.

### 353 3. RESULTS

#### 354 3.1. Stiffness of lamellar tissue as a function of BMD and microporosity

355 In this section, subscript 1 denotes the longitudinal direction of the fibrils,  
 356 subscripts 2 and 3 represent two orthogonal directions in the tranverse plane  
 357 of lamellar tissue as shown in Fig. 4. In Figs. 9-11, the orthotropic terms of  
 358 the symmetric stiffness matrix of lamellar tissue are depicted, as a function  
 359 of the variables BMD and porosity. The anisotropic terms are negligible.  
 360 The markers correspond to the averaged results obtained from ten numerical  
 361 homogenized random models. Nonlinear regressions are shown as solid lines.

362 As summarized in Table 1, the analysis have been performed for six values  
 363 of porosity and, for each one, twelve values of BMD. Additionally, as detailed  
 364 in Sec. 2.1, for each level of porosity, ten geometrical configurations with

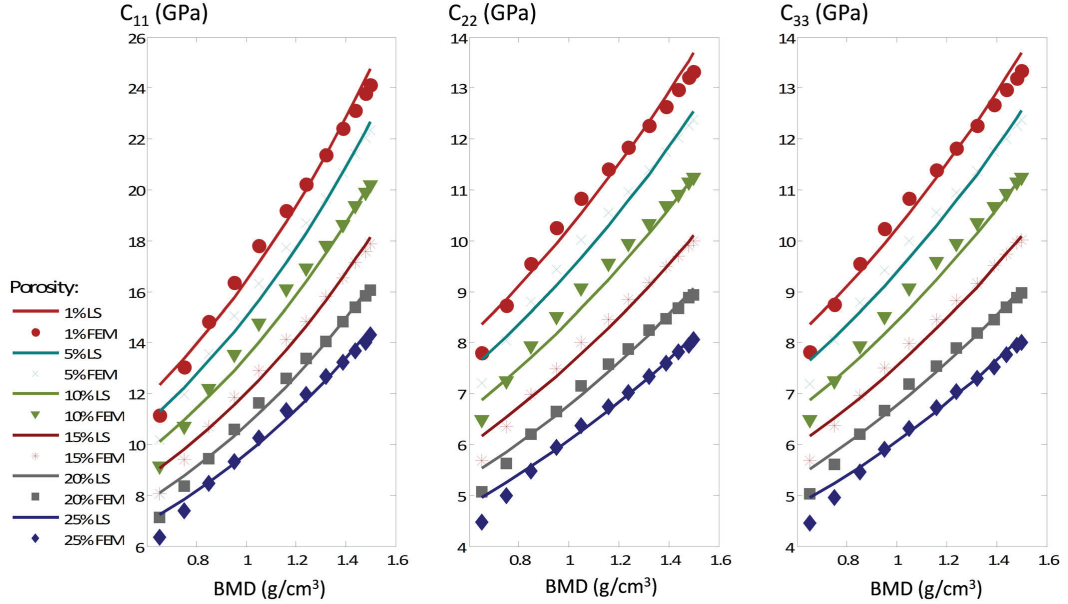


Fig. 9: Evolution of the terms  $C_{11}$ ,  $C_{22}$  and  $C_{33}$  of the stiffness matrix for lamellar tissue as a function of BMD and porosity. Markers denote the averaged results from ten numerical homogenized random FE models. Solid lines represent the least square fitting by an exponential function.

365 randomly distributed voids have been analyzed. For this propose, scripts in  
 366 *Matlab*<sup>©</sup> and *Ansys*<sup>©</sup> APDL have been programmed.

367 A coherent trend is observed in these results. The higher level of BMD  
 368 makes the bone stiffer for all porosity values. Likewise, for a given value  
 369 of BMD, the bone loses stiffness as the porosity increases. It should be  
 370 highlighted the uniform trend of the principal diagonal terms of the stiffness  
 371 matrix with both variables, BMD and porosity. A transverse isotropic be-  
 372 havior is observed, being the stiffest direction coincident with the mineralized  
 373 collagen fibrils orientation (values of  $C_{11}$  are the highest, whereas  $C_{22}$  and  
 374  $C_{33}$  are very similar).

375 For the sake of clarity, in Fig. 12 the multivariable regressions for the  
 376 orthotropic terms of the stiffness matrix of lamellar tissue are depicted in  
 377 three-dimensional plots. Numerical results are represented by blue markers

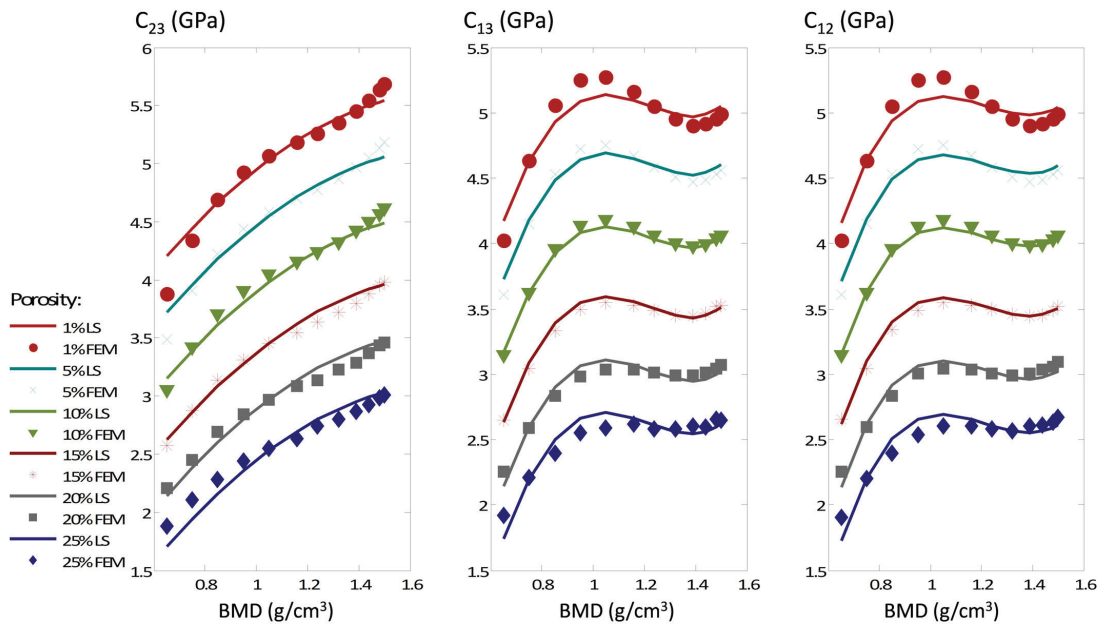


Fig. 10: Evolution of the terms  $C_{23}$ ,  $C_{13}$  and  $C_{12}$  of the stiffness matrix for lamellar tissue as a function of BMD and porosity. Markers denote the numerical averaged results from homogenization by FE. Solid lines represent the least square fitting.

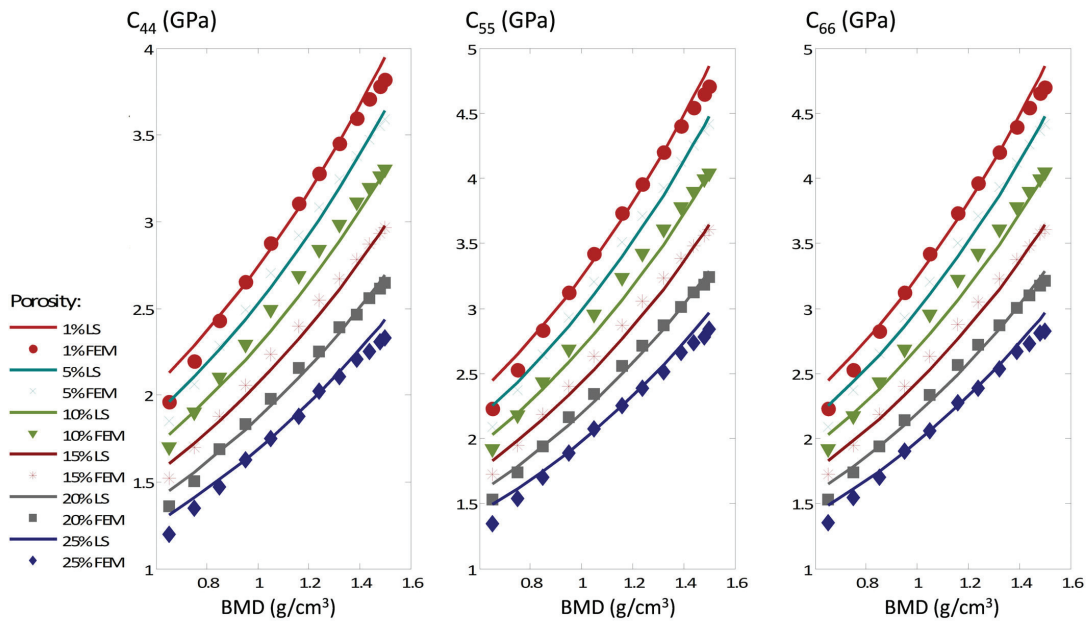


Fig. 11: Evolution of the terms  $C_{44}$ ,  $C_{55}$  and  $C_{66}$  of the stiffness matrix for lamellar tissue as a function of BMD and porosity. Markers denote the numerical averaged results from homogenization by FE. Solid lines represent the least square fitting by an exponential function.

378 and the regression fitted as a grey surface.

379 Subsequently, in order to apply these results in a more general context,  
 380 the equations that explicitly relate the stiffness terms with the variable BMD  
 381 and porosity, are derived from the least square fitting (note that results  
 382 from Eqs. 18 to 23 are expressed in GPa and results from Eqs. 24 to 26 are  
 383 expressed in Pa):

$$C_{11} = 7.3876 e^{-0.022229 p} e^{0.82134 \text{ BMD}} \quad R^2 = 0.99 \quad (18)$$

$$C_{22} = 5.4868 e^{-0.021726 p} e^{0.58165 \text{ BMD}} \quad R^2 = 0.99 \quad (19)$$

$$C_{33} = 5.8386 e^{-0.021805 p} e^{0.58304 \text{ BMD}} \quad R^2 = 0.99 \quad (20)$$

$$C_{44} = 1.3475 e^{-0.02013 p} e^{0.72977 \text{ BMD}} \quad R^2 = 0.99 \quad (21)$$

$$C_{55} = 1.4673 e^{-0.02058 p} e^{0.81231 \text{ BMD}} \quad R^2 = 0.991 \quad (22)$$

$$C_{66} = 1.4682 e^{-0.02060 p} e^{0.81189 \text{ BMD}} \quad R^2 = 0.991 \quad (23)$$

$$C_{23} = 2.1878 \times 10^9 - 1.2627 \times 10^8 p + 8.4022 \times 10^5 p^2 + 4.0292 \times 10^9 \text{ BMD} \\ - 1.1405 \times 10^9 \text{ BMD}^2 \quad R^2 = 0.992 \quad (24)$$

$$C_{13} = -6.6623 \times 10^9 - 1.1082 \times 10^8 p - 3.9345 \times 10^5 p^2 + 30227 p^3 + 3.0459 \times 10^{10} \text{ BMD} \\ - 2.5596 \times 10^{10} \text{ BMD}^2 + 7.0279 \times 10^9 \text{ BMD}^3 \quad R^2 = 0.995 \quad (25)$$

$$C_{12} = -3.6721 \times 10^9 - 1.0889 \times 10^8 p - 6.1566 \times 10^5 p^2 + 36350 p^3 + 1.9131 \times 10^{10} \text{ BMD} \\ - 1.0812 \times 10^{10} \text{ BMD}^2 + 5.8818 \times 10^8 \text{ BMD}^3 \quad R^2 = 0.995 \quad (26)$$

### 384 3.2. Numerical modelling of the trabecular bone strength

385 In this section, the finite element model of a representative volume of  
 386 trabecular vertebral bone from a swine specimen (see details in Sec. 2.3)  
 387 is analyzed under displacement controlled tension and compression loading.  
 388 Assuming quasi-static conditions, the bone strength assessment in longitu-  
 389 dinal and transversal directions is under scope. Through Eqs. 18-26 we esti-  
 390 mate the stiffness properties assuming uniform values of BMD and porosity:  
 391 **BMD = 0,85 g/cm<sup>3</sup>**,  $p = 5\%$ . Strength properties for lamellar tissue are  
 392 summarized in Table 2. As mentioned in Sec. 2.3, customary reference sys-  
 393 tems are defined to align the element coordinate systems considering that the

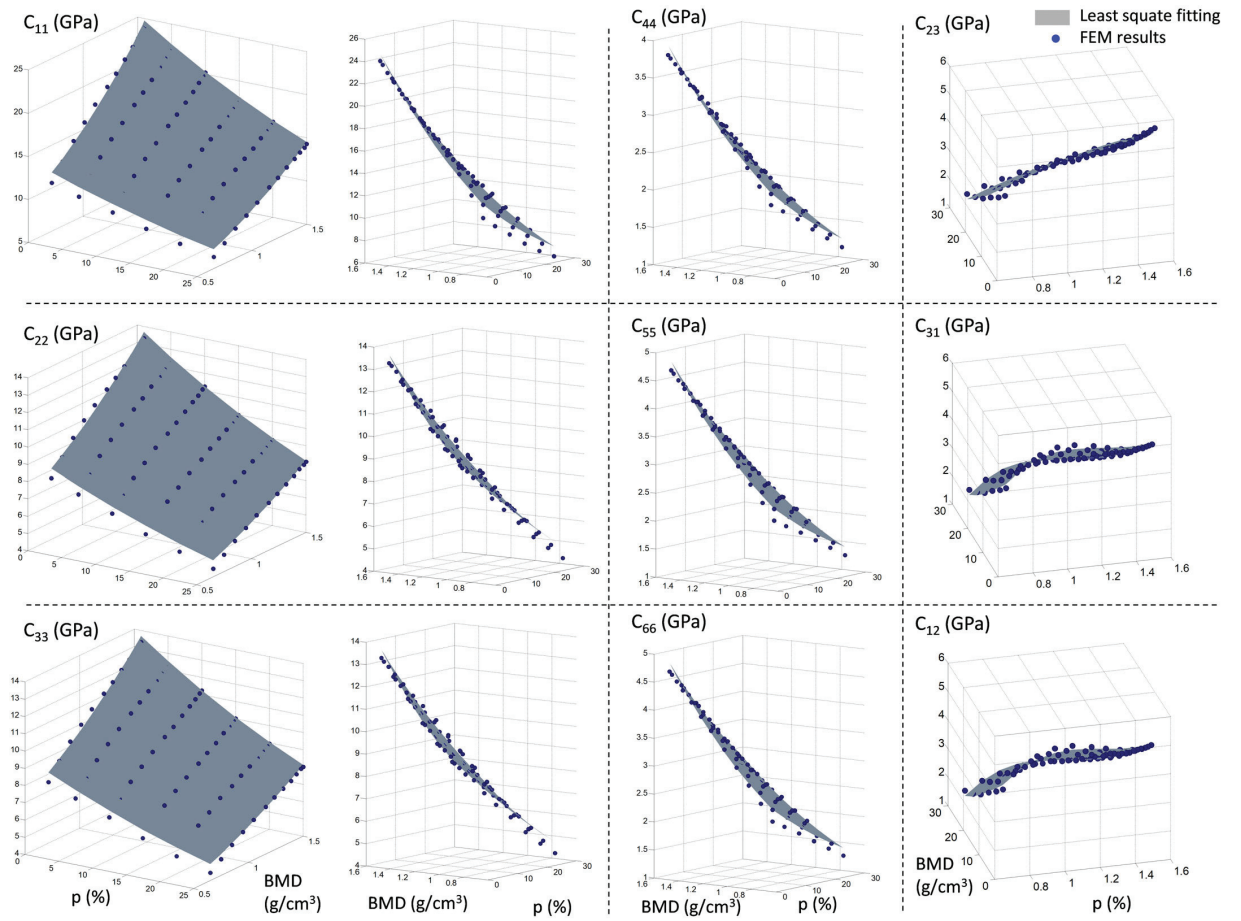


Fig. 12: Three-dimensional representation of the orthotropic stiffness matrix terms for lamellar tissue as a function of BMD and porosity

394 mineralized collagen bundles are orientated parallel to the main geometrical  
395 feature of the trabecula. Therefore, transversely isotropic elastic properties  
396 and strength limits of lamellar bone, defined in an orientated reference sys-  
397 tem (1,2,3) (see Fig. 4), are currently oriented in the mesoscale model. The  
398 bone failure onset and the post-yield regime are analyzed through the Hashin  
399 criterion and Material Property Degradation method respectively. Besides,  
400 the influence of the damage variables  $d_f$  and  $d_m$  (see details in Sec. 2.5) is  
401 also analyzed. The numerical values considered are  $d_f = 0.9, 0.9999$  and  
402  $d_m = 0.5, 0.95, 0.9999$ .

403 The graphs shown in Figs. 13-15 outline the stress-strain relationships for  
404 the tensile and compressive loading, where the apparent stress is estimated  
405 from the resultant force on the supported area in the same direction of the  
406 applied displacement. It is remarkable the anisotropic mechanical behavior  
407 of trabecular bone being able to identify approximately an orthotropic trend.

408 If we compare the results shown in Fig. 13-(a) and (b), the damage pa-  
409 rameter  $d_f$  presents an important influence on the longitudinal tensile and  
410 compressive mechanical behavior (y direction of the model). These results  
411 are expected because the mineralized bundles of fibres are mainly orientated  
412 in that direction. In general, the graphs of Fig. 13-(a) show an elastic regime  
413 followed by a damage zone where the elements progressively fail simulating  
414 the presence of diffuse microcracks. Afterwards, a more generalized element  
415 failure is observed suggesting that bone fracture initiates. Further material  
416 softening and densification is observed for  $d_m = 0.95$  and  $0.9999$ , whereas  
417 strain hardening behavior is noticeable for  $d_m = 0.5$ . The yield strain can  
418 be estimated as  $\epsilon_y^+ = |\epsilon_y^-| = [0.0058 - 0.0071]$ , resulting similar for the dif-  
419 ferent values of  $d_m$  analyzed. The compression yield stress is slightly higher  
420 than the tension yield stress, for  $d_m = 0, 5$ :  $S_y^+ = [6.41 - 7.08]$  MPa and  
421  $|S_y^-| = [6.41 - 7.22]$  MPa.

422 In Fig. 13-(b) a quasi-brittle response is observed for tensile and com-  
423 pressive loading. This situation is promoted by the damage parameter value  
424  $d_f = 0.9999$  that gives rise to the elimination of elements just after failure  
425 initiation. The elastic linear zone is followed by a small damage mechanics  
426 regime. In this case, we observe that the resistance of the sample in the y  
427 direction is fully conditioned by the strength of the bundles in longitudinal  
428 direction, leading abruptly to a catastrophic failure when elements begin to  
429 fail. In tensile load, and considering  $d_m = 0.5$ , the yield strain is estimated as  
430  $\epsilon_y^+ = [0.0052 - 0.0068]$ , and for compressive load as  $|\epsilon_y^-| = [0.0056 - 0.0068]$ .



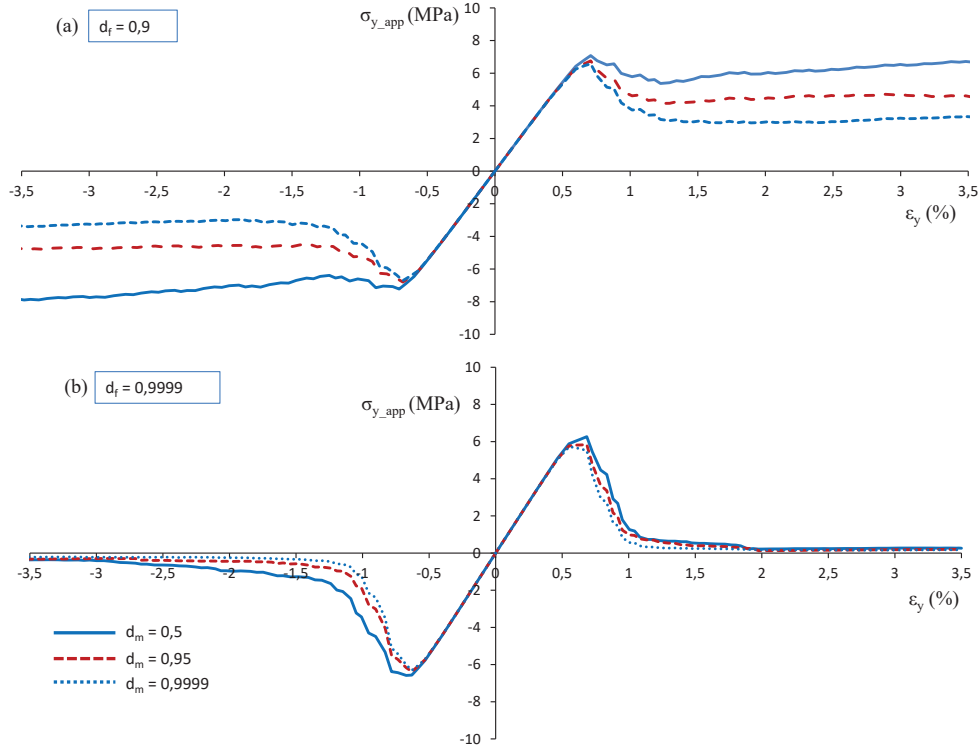


Fig. 13: Tensile and compressive stress-strain relationships under displacement control in  $y$  direction of the model for (a)  $d_f = 0.9$  and (b)  $d_f = 0.9999$ . Results for different values of the degradation parameter  $d_m = 0.5, 0.95$  and  $0.9999$ .

431 The apparent elastic moduli in the longitudinal direction results equal  
 432 for both tension and compression loading cases, being estimated in  $E_{y,app} =$   
 433  $1.104$  MPa.

434 In Figs. 14-15 the results for tensile and compressive loading in  $x$  and  $z$   
 435 transverse directions are shown. A high influence of  $d_m$  on the post-yielding is  
 436 observed independently of  $d_f$ . A quasi-brittle behavior is obtained only when  
 437 the ply discount is assumed, i.e. removing the element when the transverse  
 438 failure mode occurs,  $d_m = 0.9999$ . In the other two situations, an important  
 439 damage mechanism regime is observed. The post-yielding behavior changes  
 440 with  $d_m$ . For  $d_m = 0.5$  a strain hardening behavior is exhibited and for  
 441  $d_m = 0.9$  the relationship indicates an increment of elongation at an almost  
 442 constant stress value without strain hardening regime.

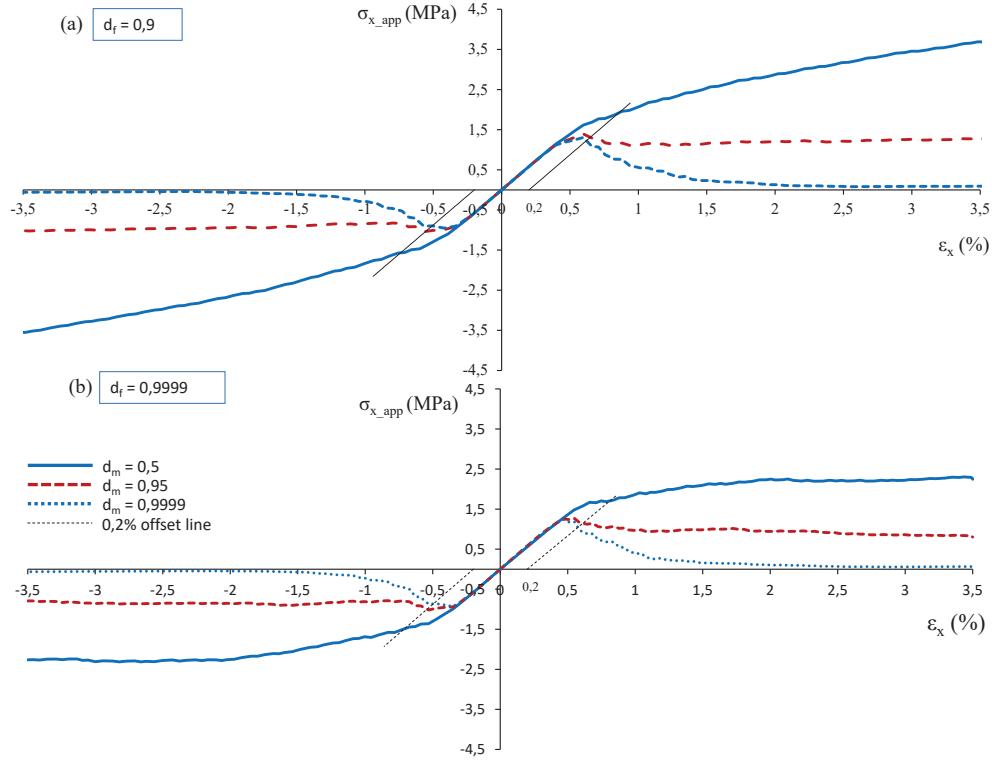


Fig. 14: Tensile and compressive stress-strain relationships under displacement control in  $x$  direction of the model for (a)  $d_f = 0,9$  and (b)  $d_f = 0,9999$ . Results for different values of the degradation parameter  $d_m = 0,5, 0,95$  and  $0,9999$ .

443 The maximum yield strain is reached for  $d_m = 0,5$ . For this case, it can  
 444 be defined the elastic limit at the 0,2% of strain. The averaged transverse  
 445 yield strain is  $\epsilon_y^+ = [0,0079 - 0,0084]$  in tension and  $|\epsilon_y^-| = [0,0075 - 0,0085]$   
 446 in compression.

447 The apparent elastic modulus in the transversal  $x$  and  $z$  directions are  
 448 estimated as  $E_{x,app} = 292$  MPa and  $E_{z,app} = 252$  MPa. These values are the  
 449 same both for tension and compression.

450 In our results, it is observed the quasi-brittle stress-strain behavior of  
 451 bone described in Zioupos (1998) assuming the damage parameter  $d_f =$   
 452  $0,9999$  in case of longitudinal load and  $d_m = 0,9999$  for transverse load.  
 453 When an element fails, such a large stiffness reduction is equivalent to the  
 454 elimination of the element, hence the remaining elements are not able to

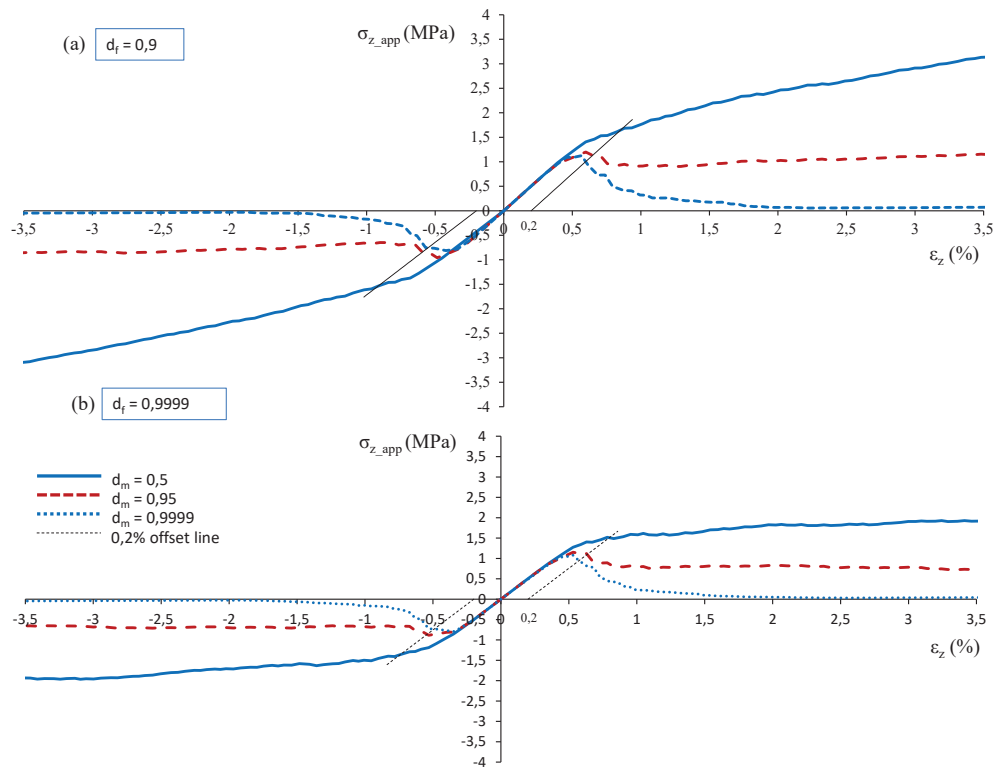


Fig. 15: Tensile and compressive stress-strain relationships under displacement control in  $z$  direction of the model for (a)  $d_f = 0.9$  and (b)  $d_f = 0.9999$ . Results for different values of the degradation parameter  $d_m = 0.5, 0.95$  and  $0.9999$

455 stand the overload, which causes a generalized fail.

456 On the other hand, regarding the elastic regime, results shown in Table 3  
457 highlight the influence of the non-isotropic elastic properties of lamellar tissue  
458 on the elastic behavior of cancellous bone at mesoscale. Results shown in  
459 series with markers are obtained considering Eqs. 18-26 for  $BMD = 0.653,$   
460  $0.85, 1.05, 1.24, 1.39$  and  $1.48 \text{ g/cm}^3$  and natural values of porosity  $p = 1,$   
461  $2.5, 5, 7.5$  and  $10\%$ .

462 When isotropic properties for tissue are defined in the numerical model,  
463  $E = 10 \text{ GPa}$  and  $\nu = 0.3$  (Wili et al., 2017), the apparent moduli result  
464  $E_{x,app}^{iso} = 428.7 \text{ GPa}, E_{z,app}^{iso} = 367.9 \text{ GPa}$  and  $E_{y,app}^{iso} = 1164.6 \text{ GPa}$ . As it can  
465 be appreciated, stiffness in the transverse directions of the sample ( $x$  and  
466  $z$  directions) can be easily overestimated, particularly for increasing levels  
467 of microporosity. In the longitudinal direction ( $y$  direction), the apparent  
468 modulus can be reasonably estimated for regular values of BMD and natu-  
469 ral microporosity considering isotropic properties for tissue. However, when  
470 bone is highly mineralized, the differences increase following a potential law  
471 in both variables, bone mineral density and porosity.

472 Convergence analyses have been performed in order to guarantee the re-  
473 sults accuracy. The energy norm of the estimated solution,  $\|U\|$ , has been  
474 obtained as a function of the total number of degrees of freedom (DOF) in  
475 the numerical model (where  $\|U\| = \sqrt{2\Pi}$ , being  $\Pi$  the computed total strain  
476 energy expressed in mJ). The influence of the discretization has been anal-  
477 ysed applying a quasi-static compression load in the three orthogonal direc-  
478 tions of the sample. To this aim, isotropic properties have been defined in the  
479 model. Note that the total DOF may vary slightly for each load case because  
480 the number of constraints is different. The strategy is based on a uniform  
481 mesh refinement. Values summarized in the Table 4 show that discretization  
482 assumed in this work provides accurate results without compromising the  
483 computational cost.

484 In Table 5 the results presented in the current work are summarized,  
485 together with reference values found in the literature.

## 486 4. Discussion

487 Bone fracture risk assessment is nowadays a prominent topic of interest  
488 in an increasingly aged population. In this sense, many enhancements in  
489 high-resolution image acquisition and its treatment have been made to cap-  
490 ture the patient specific real architecture of bone. This enables to perform

Table 3: Evolution of the apparent moduli in  $x$ ,  $y$  and  $z$  directions of the cancellous bone numerical model considering two approaches to define the lamellar tissue elastic properties. Results shown in series with markers are obtained considering the Eqs. 18-26 presented in this work which estimate the stiffness matrix of lamellar tissue as a function of BMD and porosity at tissue level. The dotted black line shows the numerical solution when isotropic elastic properties are defined for lamellar tissue:  $E = 10$  GPa and  $\nu = 0.3$ , (Wili et al., 2017).

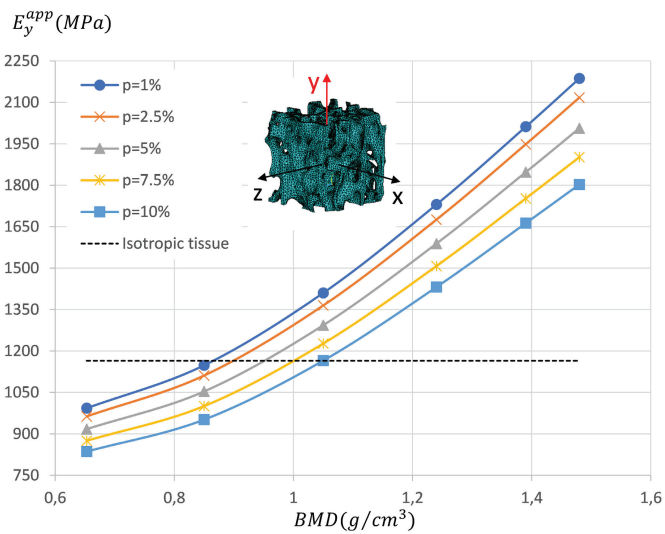
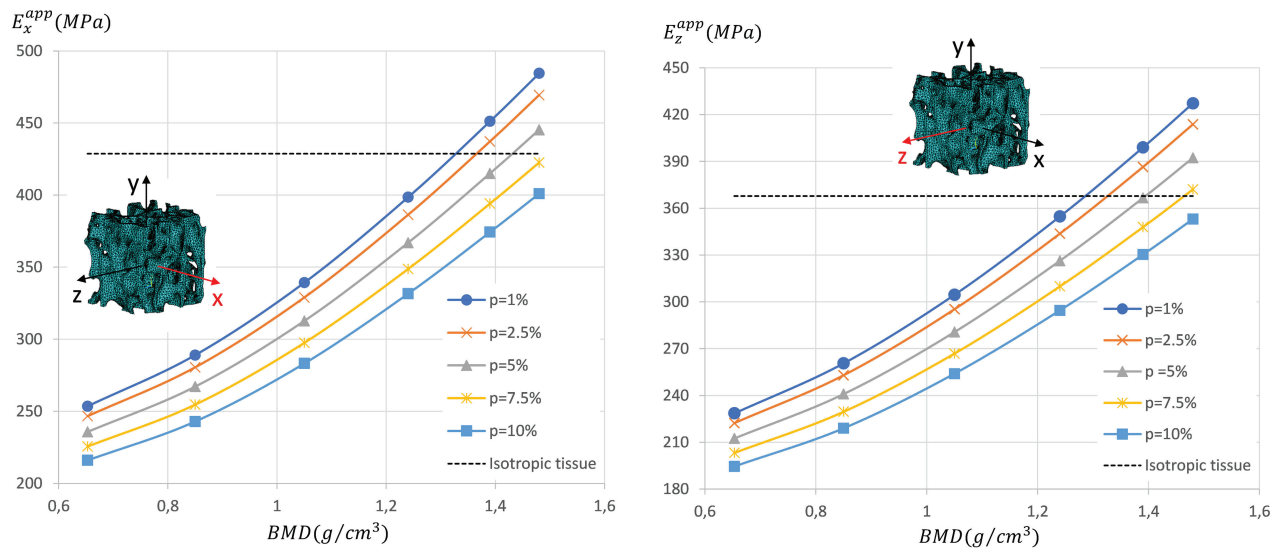


Table 4: *Estimated solution in energy norm,  $\|U\|$ , for different discretizations in the  $x, y, z$  directions. DOF represents the number of degrees of freedom in the numerical model*

x-direction		y-direction		z-direction	
DOF	$\ U\ $	DOF	$\ U\ $	DOF	$\ U\ $
$5.10 \times 10^4$	1.8	$5.10 \times 10^4$	3.0	$5.12 \times 10^4$	1.7
$6.46 \times 10^4$	1.8	$6.39 \times 10^4$	3.0	$6.42 \times 10^4$	1.7
$1.01 \times 10^5$	1.7*	$1.00 \times 10^5$	2.9*	$1.01 \times 10^5$	1.7*
$6.11 \times 10^5$	1.6	$6.11 \times 10^5$	2.9	$6.12 \times 10^5$	1.6
$1.85 \times 10^6$	1.6	$1.85 \times 10^6$	2.9	$1.86 \times 10^6$	1.5

\* *Results obtained for the mesh refinement used in this work*

491 numerical analysis of detailed micro-finite element ( $\mu$ -FE) models. At this  
 492 point, in most cases, isotropic elastic constants are assumed for tissue mod-  
 493 elling, neglecting its anisotropic behavior, mineral content and the porosity  
 494 influence on its mechanical response. In fact, isotropic damage continuum  
 495 approaches are often used to estimate bone failure through numerical mod-  
 496 elling (Lemaitre, 1985; Hambli, 2013a,b; Schwiedrzik et al., 2013; Wili et al.,  
 497 2017).

498 On the other hand, slow bone loss is associated with an incomplete os-  
 499 teoblastic deposition and leads to thinner structural elements. This is one  
 500 characteristic indicator of an age-related or senile osteoporotic bone. This  
 501 bone feature is revealed at micro scale length and can be included in a micro-  
 502 numerical model obtained from processing  $\mu$ -CT images. Essential morpho-  
 503 metric parameters can be then captured. However, a very important impact  
 504 on the porosity at tissue level is observed when a rapid bone loss accounts  
 505 as a result of a deeper osteoclastic perforation that can generate discontinu-  
 506 ities in the bone structure. This occurs most commonly in postmenopausal  
 507 women, induced by the abrupt reduction of estrogens (Parfitt AM, 1984). To  
 508 the authors' knowledge, the influence of the porosity at tissue level on the  
 509 elastic behaviour of bone has not been addressed in the literature. It is essen-  
 510 tial to characterize the mechanical properties of bone tissue with prevalence  
 511 of osteoclastic perforation (Gentzsch et al., 2003) in the post-menopausal  
 512 women with osteoporosis, where BMD and microporosity values are altered  
 513 as a consequence of an unbalanced bone turnover process.

514 The transversely isotropic model for the elastic tissue properties presented  
 515 in this work addresses the anisotropic behaviour due to mineralized collagen

Table 5: *Results summary. Estimation of yield strain and Young’s moduli (apparent values) for trabecular bone and comparison with some reference values from literature.*

Longitudinal loading case ( <i>y</i> -direction)				
	Tension		Compression	
	This work	Literature	This work	Literature
Yield strain*	[0.0052-0.0071]	$0.0078 \pm 0.0004$ (Kopperdahl et al., 1998) <sup>(1,m)</sup> $0.0078 \pm 0.0004$ (Kopperdahl et al., 1998) <sup>(3,m)</sup> $0.0072$ (Wolfram et al., 2011) <sup>(1,m)</sup> $0.0078 \pm 0.00041$ (Wili et al., 2017) <sup>(1,c)</sup>	[0.0052-0.0071]	$0.0124 \pm 0.00197$ (Turner et al, 1989) <sup>(4,m)</sup> $0.0109 \pm 0.0012$ (Kopperdahl et al., 1998) <sup>(3,m)</sup> $0.0084 \pm 0.0006$ (Kopperdahl et al., 1998) <sup>(1,m)</sup> $[0.0046-0.0063]$ (Nagaraja S, 2005) <sup>(3,m)</sup> $0.0081$ (Wolfram et al., 2011) <sup>(1,m)</sup> $0.00951 \pm 0.00125$ (Wili et al., 2017) <sup>(1,c)</sup> $0.00119 - 0.0168$ (Belda et al., 2019; Belda R., 2020) <sup>(2,m)</sup>
Young’s modulus (MPa)*	1104	$384.1 \pm 155.1$ (Wolfram et al., 2011) <sup>(1,m)</sup> $1017 \pm 0.088$ (Rami et al., 2017) <sup>(5a,c)</sup> $1800 \pm 0.058$ (Rami et al., 2017) <sup>(4b,c)</sup> $908.2$ (Belda et al., 2019) <sup>(2,c)</sup>	1104	$309 \pm 109$ (Kopperdahl et al., 1998) <sup>(1,m)</sup> $384.4 \pm 162.9$ (Wolfram et al., 2011) <sup>(1,m)</sup> $1017 \pm 0.088$ (Rami et al., 2017) <sup>(5a,c)</sup> $1800 \pm 0.058$ (Rami et al., 2017) <sup>(5b,c)</sup> $1265.2$ (Belda R., 2020) <sup>(2,m)</sup> $908.2$ (Belda et al., 2019) <sup>(2,c)</sup> $1022.9$ (Belda et al., 2019) <sup>(2,m)</sup>
Transversal loading cases (averaged values for <i>x</i> and <i>z</i> -directions)				
	Tension		Compression	
	This work	Literature	This work	Literature
Yield strain*	[0.0079-0.0084]	$\approx 0.007$ (Wolfram et al., 2011) <sup>(1,m)</sup> $0.00899 \pm 0.00181$ (Wili et al., 2017) <sup>(1,c)</sup>	[0.0075-0.0085]	$\approx 0.0082$ (Wolfram et al., 2011) <sup>(1,m)</sup> $0.0105 \pm 0.00115$ (Wili et al., 2017) <sup>(1,c)</sup>
Young’s modulus (MPa)*	252 – 292	$129.7 \pm 54.7$ (Wolfram et al., 2011) <sup>(1,m)</sup> $196 \pm 58 - 306 \pm 14$ (Rami et al., 2017) <sup>(5a,c)</sup> $499 \pm 63 - 538 \pm 70$ (Rami et al., 2017) <sup>(5b,c)</sup> $361.04 - 382.66$ (Belda et al., 2019) <sup>(2,c)</sup>	252 – 292	$119.5 \pm 74.2$ (Wolfram et al., 2011) <sup>(1,m)</sup> $196 \pm 58 - 306 \pm 14$ (Rami et al., 2017) <sup>(5a,c)</sup> $499 \pm 63 - 538 \pm 70$ (Rami et al., 2017) <sup>(5b,c)</sup> $361.04 - 382.66$ (Belda et al., 2019) <sup>(2,c)</sup> $328.1 - 346.0$ (Belda et al., 2019) <sup>(2,m)</sup>

<sup>(1)</sup> Human vertebral trabecular bone

<sup>(3)</sup> Bovine proximal tibia trabecular bone

<sup>(5a)</sup> Human vertebral bone T11/woman/60 years

\* Apparent values

<sup>(2)</sup> Swine vertebral trabecular bone

<sup>(4)</sup> Bovine distal femora trabecular bone

<sup>(5b)</sup> Human vertebral bone T12/man/56 years

*m*: Mechanical test; *c*: Computational analysis

516 fibrils orientation, the bone mineral density and microporosity. In addition,  
517 in a trabecular bone numerical model at mesoscale, the main fibrils orien-  
518 tation is also considered. In Belda R. (2020), an isotropic Young’s modulus  
519 for tissue was calibrated from experimental compression tests. Results of  
520 that work highlighted that different Young’s moduli for tissue in the three  
521 orthogonal directions of the sample were to be inferred in order to reproduce  
522 the experimental results. A plausible explanation is that the fibrils orien-  
523 tation varies with the direction and this is in consonance with the building  
524 substructures of cancellous bone (plates and rods). Composition, distribu-  
525 tion and architecture of lamellar tissue are very important to reproduce the  
526 micromechanics failure mode of trabecular bone (Hammond et al., 2019).

527 We highlight different novelties of this work: (1) New explicit equations  
528 for the estimation of the elastic constants of lamellar tissue are provided.  
529 These equations have been obtained as a function of BMD and microporos-  
530 ity in a multiscale analysis, which enable to study the influence of these  
531 characterizing parameters on the mechanical behavior of bone particularly  
532 with certain pathologies, like osteoporosis. (2) The strength limits for fully  
533 calcified lamellar tissue have been inferred from literature. (3) The approach  
534 herein presented considers the orientation of the mineralized fiber bundles in  
535 the trabeculae network, which is essential for the implementation of elastic  
536 and strength tissue properties in the numerical model. (4) In accordance  
537 to the non-isotropic elastic and strength properties of lamellar tissue, an  
538 orthotropic failure criterion is proposed to analyze the damage onset of can-  
539 cellous bone. The Hashin’s interactive failure criterion is considered. (5) The  
540 Material Property Degradation (MPDG) method is used to model numeri-  
541 cally the damage evolution law at tissue level. A detailed study about the  
542 influence of the damage parameters on the mechanical post-yielding response  
543 of trabecular bone is also presented.

544 In Sec. 3.1, Eqs. 18-26 provide the terms of the stiffness matrix of lamellar  
545 tissue as a function of two essential tissue parameters, the BMD and the micro-  
546 porosity. In the main terms, a power regression in BMD is observed. There  
547 are previous studies that show a similar tendency for the Young’s moduli of  
548 lamellar tissue (Currey, 1986; Vercher-Martínez et al., 2018). Additionally,  
549 we observe an inverse power relationship for the microporosity.

550 Results summarized in Table 3 show the influence of the non-isotropic  
551 elastic properties of lamellar tissue on the apparent moduli of the cancellous  
552 bone at mesoscale. When tissue isotropic properties are assumed, stiffness  
553 can be frequently overestimated in the transverse directions. In the longi-



554 tudinal direction, stiffness equally exhibits a high dependence on BMD and  
555 microporosity although tissue isotropic properties can be reasonable for val-  
556 ues of BMD and microporosity within a natural range.

557 On the other hand, in Sec. 3.2, numerical results show that small differ-  
558 ences are found for apparent yield strain between tension and compression  
559 for the sample analyzed, in agreement with Kopperdahl et al. (1998) and  
560 Wolfram et al. (2011). It is known that, although yield strain represents a  
561 pretty uniform failure property, it is more influenced by the apparent density  
562 in compression than in tension, especially in less dense bone (Kopperdahl  
563 et al., 1998). This can be the main reason for some discrepancies observed  
564 in the literature for the apparent yield strain in compression (Turner et al,  
565 1989; Belda et al., 2019). In tension, the apparent yield strain estimated  
566 in this work is also in good agreement with values found in the literature  
567 (see Table 5), and confirms the tendency to a more uniform value, being less  
568 sensitive to the apparent density and anatomical site (Belda R., 2020).

569 In addition, in accordance with Wolfram et al. (2011), no relevant differ-  
570 ences between apparent moduli in tension and compression were observed,  
571 for both longitudinal and transverse direction of the trabecular bone sam-  
572 ple. The apparent elastic moduli estimated in this work are, in general, in  
573 good agreement with the values reported in literature (Belda et al., 2019;  
574 Rami et al., 2017). However, our results differ from others (Wolfram et al.,  
575 2011; Kopperdahl et al., 1998). This can be motivated by differences in the  
576 mineral content of the samples, anatomic site, bone volume fraction, shape  
577 complexity of the structure or experimental conditions.

578 Nevertheless, this study presents some limitations. Bone surfaces present  
579 a high activity of bone remodelling. Consequently, lamellar tissue is often  
580 renewed at surfaces, leading to a lower mineral content than through in the  
581 core. Hence, the tissue elastic constants change through a cross section of a  
582 trabecula. For example, a higher elastic modulus (between 5 to 12 %) was  
583 found at the core than at the cortex of a trabecula. Additionally, values for  
584 the strength limits of lamellar tissue summarized in Table 2 correspond to  
585 fully calcified tissue and assume healthy bone porosity. However, strength  
586 limits are strongly dependent on the degree of calcification (Ascenzi and  
587 Bonucci, 1968) and on the degree and shape of porosity. Further investiga-  
588 tions will be necessary to quantify the influence of the presence of micropore  
589 on the strength limits for lamellar tissue. These considerations should be  
590 addressed in a more general context. As aforementioned, orientation of the  
591 mineralized collagen fibrils in the trabeculae network has been estimated

592 based on the main geometrical feature. A more refined strategy would be  
593 necessary to automatize the orientation of the elementary coordiante system,  
594 considering the predominant geometrical definition (Hammond et al., 2018),  
595 but also including information based on the main pattern that osteocyte  
596 long axis follows in the microstructure, what effectively will provide informa-  
597 tion about how mineralized collagen fibrils are aligned. Besides, BMD and  
598 porosity vary within the trabeculae architecture and, in contrast, these val-  
599 ues have been assumed homogeneous in our numerical mesoscale trabecular  
600 bone model. Lastly, the influence of the penalization parameters on mod-  
601 elling the post-yield behavior has been addressed. However, a more refined  
602 mesh should be necessary for a more detailed analysis. Future works are  
603 needed to overcome the described limitations.

604 This work proposes a new methodology to address the mechanical re-  
605 sponse of trabecular tissue considering orthotropic elastic and strength tissue  
606 properties. The quantification of BMD and porosity at tissue level as charac-  
607 terizing tissue parameters needs of future studies to validate this approach.

608

## 609 **5. Conclusions**

610 The present work addresses the numerical analysis of the mechanical  
611 response of cancellous bone including a new approach for the elastic and  
612 strength lamellar tissue properties. The non-isotropic elastic behavior of  
613 lamellar tissue deals with the influence of the bone mineral density and, as a  
614 novelty, the microporosity or porosity at tissue level. In addition, according  
615 to the strength limits inferred from literature, the failure onset is modeled  
616 by means of the Hashin failure criterion in combination with the Material  
617 Property Degradation (MPDG) method. The value of the degradation pa-  
618 rameters can simulate different post yielding scenarios compatible with the  
619 bone damage mechanisms observed in literature, as a quasi-brittle failure or  
620 significant loss of stiffness, due to smeared crack regions where the presence  
621 of multiple microcracks reduces the load transmission capability.

622 Results show that, when isotropic elastic tissue properties are considered,  
623 the anisotropic ratio of the apparent moduli of cancellous bone is, in gen-  
624 eral, undervalued, particularly when microporosity increases. Not only the  
625 microstructure but tissue properties govern the elastic response of bone at  
626 the mesoscale.

627 On the other hand, the yield strain for tension and compression quasistatic  
628 loadings has been estimated in the three orthogonal directions of the bone  
629 sample. Results indicate that, for the cancellous bone analyzed, there are no  
630 significant differences between tension and compression behavior for each di-  
631 rection. These results are in agreement with several works found in literature  
632 (see Table 5) and they are also compatible with the evidence that a higher  
633 apparent density in bone results in a higher yield strain in compression than  
634 in tension.

635 To conclude, the approaches presented in the current work enable to deal  
636 with a scarcely treated topic from the mechanical point of view: the under-  
637 mined capabilities of osteoporotic bone due to severe alterations in param-  
638 eters as BMD and porosity at tissue level. Adaptation of the morphometric  
639 parameters at micro scale level is commonly investigated under this pathol-  
640 ogy, but neglecting the underlying changes at the tissue level.

#### 641 **Declaration of Competing Interest**

642 The authors declare that they have no known conflict of interests or per-  
643 sonal relationships that could have appeared to influence the present work.

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650 interest

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